

Review

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Review

Direct Solar Thermal Water Splitting Using Iron Electrodes at High Temperatures. A Review

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Abstract: Green hydrogen is a fuel that in the coming years will play a very important role in the energy transition process in developed countries, especially those seeking to reach a level of zero emissions, for which reason its demand is expected to increase significantly. This article analyzes the fundamentals to produce hydrogen, with a focus on the oxidation reaction of a thermochemical solar cycle for the dissociation of water vapor. Solar thermochemical cycles have been extensively investigated, but it is still in the development stage by groups of researchers who have focused on finding the right materials and conditions to improve the efficiency of the process, especially at high temperatures. The theoretical foundations are analyzed, compiled from exhaustive scientific investigations related to the oxidation of iron in water vapor, the relationship with the activation energy of the corrosive process, thermodynamic aspects and the kinetic model according to a heterogeneous reaction. In addition, some high temperature oxidation mechanisms, pH effects, reactors and materials (fluidized beds) used are presented. The scientific review indicates that the production of hydrogen through a thermochemical cycle is more efficient than electrochemical processes (electrolysis), if the limitations of the reduction stage of the cycle are overcome.

Keywords: solar thermochemical water splitting; iron corrosion; solar reactors

1. Introduction

The greatest amounts of insolation are generally found at lower latitudes and in arid climates. The Atacama Desert in northern Chile has exceptionally ideal solar radiation conditions, and the potential for producing green hydrogen is widely acknowledged. As a result, this study aims to lay the groundwork for the synthesis of hydrogen using a thermo-chemical solar process while considering an environmentally favourable, inexpensive, and abundant catalyst material in the region, such as iron oxide. The goal of the research is to discover the theoretical foundations of hydrogen production by iron oxidation at high temperatures, considering the thermodynamic, thermochemical, and electrochemical foundations required to comprehend the heterogeneous reaction. The kinetic modeling of the fluidized bed material inside a thermochemical solar reactor requires this understanding.

The unique solar radiation conditions in the Atacama Desert offer significant potential for various solar technologies, including photovoltaic (PV) systems and concentrated solar power (CSP) plants. PV systems harness the sun's energy directly using solar panels, while CSP plants concentrate sunlight to generate heat and subsequently produce electricity. The desert's high levels of solar radiation allow for enhanced energy capture and conversion efficiency, making it an attractive site for solar power installations.

In addition, the region boasts a stable and predictable climate, with little rainfall and a lack of extreme temperature variations. This climatic consistency is crucial for the long-term performance and reliability of solar energy systems. Furthermore, the desert's vast expanses of land provide ample space for large-scale solar installations, facilitating the deployment of cutting-edge solar technologies and enabling the generation of significant amounts of clean energy. The knowledge gained from studying solar resources in this unique environment not only contributes to advancements in solar technology but also serves as a valuable blueprint for harnessing solar power in other regions with similar radiation characteristics.

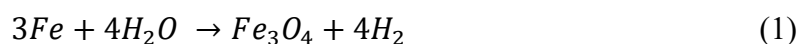
Although thermochemical solar cycles using ceria have 25% annual efficiency compared to 14% for the electrolysis process (Alkaline eletrolysis) [1,2], they still present a low level of technological preparation to produce low emission hydrogen [3]. Regarding production costs, it is estimated that hydrogen from solar photovoltaic and wind energy is between 4.0 – 9.0 USD/Kg H₂, and could be below 1.5 USD/Kg H₂, with photovoltaic solar electricity costs of 14 USD/MWh for the year 2030[3].

The essential thing about this review is that it provides the fundamental theoretical information to understand the difficulty that arises when using pure iron or iron oxides as a thermocatalyst material in a thermochemical cycle to produce hydrogen. For this reason, it is important to understand the reaction mechanisms of the material in the presence of water steam from thermodynamics, its identification as a non-catalytic surface heterogeneous reaction from electrochemistry and aspects of the oxidation mechanisms at high temperature considering water vapor as alkaline reactant and the effects of pH on the reaction. In addition, some solar reactors that have used fluidized bed materials are presented. This technical information is intended to provide the reader with data to replicate these reactors, especially for researchers who are just beginning the topic of thermolysis using iron. But at the same time, it shows the potential of its application in a high temperature thermolysis process considering only the oxidation stage.

The scarce information related to the oxidation of iron in the presence of water vapor, a product of the reaction with hydroxyl (OH), generates space for future research with iron and other materials such as copper slag that have iron oxides. Furthermore, because the scientific information collected is mainly based on studies of the reaction of iron with oxygen (O₂).

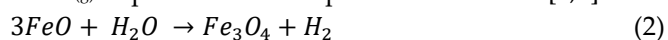
2. Theoretical Background

In specific thermodynamic conditions, the interaction between water vapor (H₂O(g)) and Iron metal (Fe) can lead to the production of Hydrogen (H₂) through the oxidation of Fe in a range of 200 to 1000 °C. This oxidation reaction occurs in an environment abundant in H₂O(g) and is expressed as follows [4]:



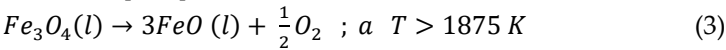
In 1926, Dunn et al. [5] proposed that the oxidation process at high temperatures is controlled only by the diffusion of O₂ through the oxide layer. The presence of H₂O(g) accelerates the oxidation process of the mild Fe and two grades of pure iron [6].

The H₂ production from the use of Magnetite (Fe₃O₄) was proposed by Nakamura et al. in 1977 [7,8]. However, the production of H₂ by oxidation with H₂O(g) used the concentrated solar radiation was proposed by Lede et al. [9] in 1983. Recent studies have presented notable innovations in the application of solar thermochemical reactors for converting concentrated solar energy into chemical fuels. These advancements involve a two-step thermochemical process specifically designed to produce H₂ from H₂O(g). By harnessing the immense potential of solar energy, this innovative approach offers a sustainable and highly efficient method for generating H₂ fuel. [8,10]. The exothermic reaction from H₂O(g) to produce H₂ is expressed as follows [7,8].



The Fe₃O₄ produced in this reaction is transformed in a solar oven to FeO through an endothermic sub-process [8,10]. However, a significant barrier arises in this sub-process as a result of the premature fusion of FeO occurring at a lower temperature than the required threshold for the complete thermodynamic cycle. This inherent challenge presents difficulties in effectively employing

a thermochemical solar reactor for the production of H₂, limiting its efficiency and feasibility. The reaction is expressed as follows [7,11].



High-temperature corrosion is influenced by (i) temperature, (ii) gas composition, (iii) exposure time, and (iv) system pressure, and can be characterized through the thickness reduction (penetration) and the thickness growth rate. The oxidation rate increases significantly during increasing temperature [12], providing a homogeneous medium and improving the thermodynamic equilibrium [13]. The Gibbs free energy is expressed as follows.

$$\Delta G^o = -RT \ln (K) \tag{4}$$

When a metal is exposed to corrosion at high temperatures in the presence of water vapor, it yields a value of ΔG^o, as determined by equation 5, based on the reaction $xM + yH_2O \rightarrow M_xO_y + yH_2$ [12]. The expression for the Gibbs free energy is as follows:

$$\Delta G^o = -yRT \ln \left(\frac{P_{H_2}}{P_{H_2O}} \right) \tag{5}$$

Meredig & Wolverson et al. [14] indicate that during the corrosion process, the contact of a metal (M) in contact with steam can produce H₂ by the following expression [14].



Where the expression for the Gibbs free energy is as follows

$$\Delta G^o = \Delta H^{MO_x} - \Delta H^{MO_{(x-1)}} - \Delta H^{H_2O} - T(S^{H_2} - S^{H_2O}) \leq 0 \tag{7}$$

Where MO_x is the oxidized metal and MO_{x-1} is the reduced oxide.

The water-splitting reaction using FeO as electrode is favorable only at temperature under 800 °C, when ΔG^o < 0. The theoretical FeO conversion decreases with increasing temperature [15] and Kodama & Gokon et al.[16] indicate that the oxidation of FeO in the presence of water vapor exhibited a spontaneous reaction (ΔG<0) at temperatures below 1000 K.

Young et al. [17] suggest that various metal oxides can form volatile compounds by direct reaction with water vapor and that hydroxides and oxy-hydroxides can be produced by hydration [18]. On the other hand, Belton & Richardson et al. [19] demonstrate the existence of a volatile iron hydroxide at temperatures exceeding 1300°C. Compared to non-volatile metal oxides, volatile metal oxides exhibit a higher oxygen exchange capacity, which directly correlates with their H₂ production capacity [20].

Table 1 shows the thermodynamic parameters (ΔG^o, ΔH^o, ΔS^o, and T) used to determine the activation energy of the reaction of water vapor with iron, the products of the vapor-dissociation into molecules, ions, and iron oxidation products (oxides and hydroxides). Table 2 shows the typical thermodynamic parameters generated during the reactions oxidation/reduction process of metals in the presence of H₂O_(g) at high temperatures.

Table 1. Thermodynamic parameters of Water, Oxides, and Hydroxides.

Chemical species	Gibbs free energy ΔG (kJ/mol)	Enthalpy ΔH (kJ/mol)	Entropy S (kJ/mol)	Temp. K (1 atm)	Ref.
H ₂ O (l)	-237,1	-285,8	70,0	298	[21,22]
	-228,6	-241,8	188,8	298	[21,22]
H ₂ O (g)	-214,1	-244,7	213,8	600	[22]
	-198,2	-247,1	228,6	900	
H ₂ (g)	0	0	130,6	298	[21,22]
	0	0	151,0	600	[22]
	0	0	163,1	900	
OH- (aqueous ion)	-157,3	-230,0	-10,7	298	[21]
	34,7	38,9	183,7	298	[22]
OH (g)	29,5	38,9	204,4	600	
	24,9	38,4	216,5	900	
Fe	0	0	27,09	298	[21]

Fe ²⁺	-90,0	-91,1	-107,1	298	[21]
Fe ³⁺	-16,7	-49,9	-280,0	298	[21]
Fe. ₉₄₇ O	-244,9	-266,3	56,6	298	[21]
	-245,1	-266,3	57,5	298	[22]
	-224,8	-263,7	93,0	600	
	-205,8	-262,7	115,1	900	
	-251,4	-272,0	60,6	298	[21,22]
FeO	-231,7	-269,4	97,3	600	[22]
	-213,1	-268,4	120,2	900	[21,22]
	-774,4	-826,2	87,4	298	
Fe ₂ O ₃	-742,3	-824,2	87,4	298	[22]
	-661,4	-817,6	173,3	600	
	-585,3	-808,4	235,4	900	
	-731,4	-811,6	93,0	298	[23]
	-1012,7	-1115,7	146,1	298	[21]
Fe ₃ O ₄	-1015,2	-1118,4	146,1	298	[22]
	-914,4	-1107,4	272,1	600	
	-821,8	-1088,6	368,3	900	
FeO (OH)	-491,8	-562,6	60,4	298	[21]
	-486,9	-568,9	87,9	298	[22]
Fe (OH) ₂	-405,7	-564,1	160,4	600	
	-327,6	-558,9	207,7	900	

Table 2. Thermodynamic Parameters of the Reaction of Iron or Oxides in Water Vapor.

Reaction	Gibbs free energy ΔG (kJ/mol)	Enthalpy ΔH(kJ/mol)	Temp. K (pres. atm)	Ref.
0.75 Fe + H ₂ O → 0.25Fe ₃ O ₄ + 4H ₂	-14,52	-32,09	600	[24]
Fe + 2H ₂ O → FeO + H ₂	-199,3	-	1000	[25]
Fe + 2H ₂ O → Fe(OH) ₂ (g) + H ₂	38,9 + 5*80T (cal)	-	-	[19]
2Fe + 3H ₂ O → Fe ₂ O ₃ + 3H ₂	48,9	-	973 - 1123	[26]
3Fe + 4H ₂ O → Fe ₃ O ₄ + 4H ₂	11,0	-	973 - 1123	[26,27]
3FeO + H ₂ O → Fe ₃ O ₄ + H ₂	-	-33,6	873	[15,16,28,29]
Fe ₂ O ₃ + H ₂ O → 2FeO(OH)	10,39	-47,54	400	[24]
Fe ₃ O ₄ + 2H ₂ O → Fe(OH) ₂ + 2FeO(OH)	41,02	-79,23	400	[24]
Fe ₃ O ₄ + 4H ₂ O → Fe(OH) ₂ + 2Fe(OH) ₃	228,22	-120,48	600	[24]
2Fe ₃ O ₄ + H ₂ O → 3Fe ₂ O ₃ + H ₂	40,5	-	-	[30]
2Fe ₃ O ₄ + 4H ₂ O → 6FeOOH + H ₂	44,4	-	-	[30]

3. Kinetic Model

The mechanisms in which a pure metal or alloy is oxidized at high temperatures require a series of successive steps [18,31]. The steps are:

- Step a) Adsorption of a gaseous component,
- Step b) Dissociation of the gaseous molecule and transfer of electrons,
- Step c) Nucleation and growth of crystals,
- Step d) Diffusion and transport of cations, anions, and electrons through the oxide layer.

However, the reaction between a solid (Fe) and a gas (H₂O_(g)) corresponds to a heterogeneous reaction where the kinetics is totally or partially controlled for the mass transport process. the product species generate by convection and diffusion process and can be gaseous, or insoluble solid [32]. Diffusion and heat transfer are phenomena that intertwine with the chemical kinetics of non-catalytic

heterogeneous reactions, with their respective diffusion equations extending to fluidized solids [33]. The reduction of iron oxide with hydrogen or the oxidation of iron with steam serve as two typical examples of this type of reaction. However, these reactions occur under the conditions of a highly compacted unreacted solid with limited porosity, where the chemical reaction proceeds rapidly while diffusion proceeds relatively slowly [34].

Wen et al. [34] consider that a superficial non-catalytic heterogeneous reaction consists of the following steps (Figure 1):

Step a) diffusion of fluid reactants through the fluid film surrounding the solid.

Step b) diffusion of the fluid reagents through the porous solid layer.

Step c) adsorption of the fluid reagents on the surface of the solid reagent.

Step d) chemical reaction with the solid surface,

Step e) desorption of the fluid products from the solid reaction surface.

Step f) diffusion of the product far from the reaction surface through the porous surface, solid media, and through the fluid film surrounding the solid.

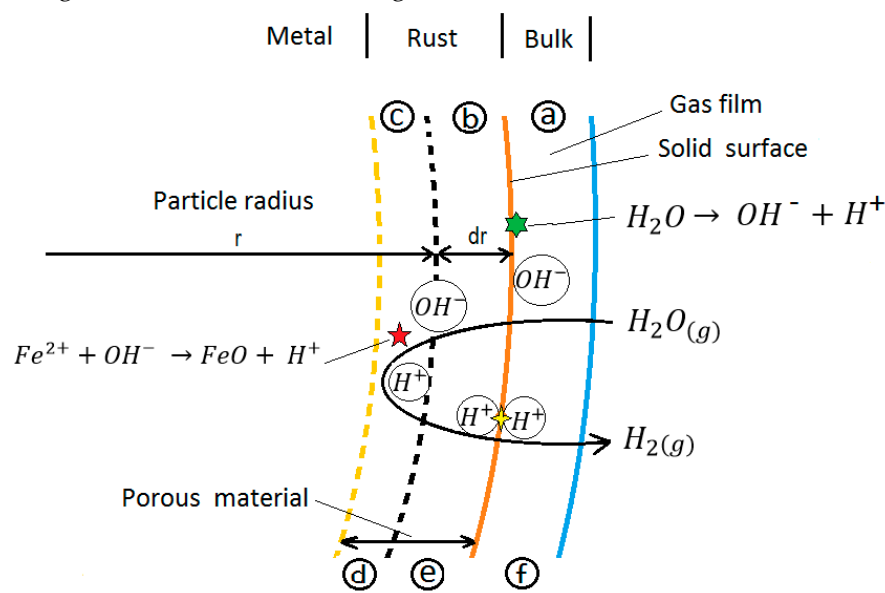


Figure 1. Step diagram of the non-catalytic superficial heterogeneous reaction of the electrochemical dissolution of iron in water vapor.

The study involved to Surman et al. [35] accomplishing the oxidation of pure Fe at 500°C using hydrogen, water vapor or various mixtures of hydrogen-vapor. The oxidation kinetics were compared with models of chemical diffusion and solid-state reactions. The diffusion of volatile iron in the form of $\text{Fe}(\text{OH})_2$ played a crucial role as the rate-controlling step. This diffusion occurred from the metal-oxide interface through the porous oxide of the inner layer, ultimately depositing Fe_3O_4 on the crystals of the outer layer, which acted as sinks.

The oxidation kinetics was approximated to the parabolic law, and the vapor phase transport hypothesis led to the expression of the parabolic rate constant (K_w), represented by the following expression.

$$K_w = 4.8 \times 10^4 \times F_p \times T^{1/2} \times \left(\frac{P_{\text{H}_2\text{O}}}{P_{\text{H}_2}} \right) \times e^{-38900/RT} \quad (8)$$

Where, K_w is the parabolic rate constant in ($\text{g}^2/\text{m}^4\text{s}$) and F_p is the porosity factor of the inner layer, $P_{\text{H}_2\text{O}}$, and P_{H_2} are the partial pressures of H_2O and H_2 respectively.

Park et al. [36] studied the hydrogen production used iron oxide and inert silica with $\text{H}_2\text{O}_{(g)}$ under temperatures between 460-700°C and vapor partial pressure between 0.002-0.02 MPa. Whence, the Fe is oxidized to Fe_3O_4 (first step), then Fe_3O_4 is gradually oxidized to Fe_2O_3 in the second step, researchers have if the $\text{H}_2\text{O}_{(g)}$ is in the equilibrium state of adsorption and that the reaction rate-determining step is the desorption of hydrogen from the active sites.

The proposed initial oxidation rate when the partial pressure of water is fixed and is represented by the following expression.

$$\left(\frac{dX_B}{dt}\right)_{t=0} = \left(\frac{bk}{1+K_{e,ads}P_{H_2O}}\right)P_{H_2O} - \frac{\left(\frac{bk}{1+K_{e,ads}P_{H_2O}}\right)}{K_e}P_{H_2} \tag{9}$$

Where, b is the stoichiometric coefficient of the oxidation reaction of Fe to Fe₂O₃, k is the overall rate constant based on unit solid volume, K_e is the chemical equilibrium constant, K_{e,ads} is the equilibrium adsorption constant, P_{H₂O}, and P_{H₂} are the partial pressures of water and hydrogen respectively.

The heterogeneous kinetic models are explained in detail by Donovan & Berra et al. [37], Levenspiel et al. [38], and Klaewkla et al. [39]. However, Wen & Wang et al. [40] propose a kinetic model of the non-catalytic heterogeneous reaction in a solid-gas reaction system considering the heat and mass transfer as combined effects.

Valipour & Saboohi et al. [41] analyzed multiple mathematical models and developed a comprehensive model for a multi-reactant system utilizing porous granules within a sample of a mobile packed bed. Our model takes into account various factors, including external mass transfer, internal diffusion through the pores, chemical reactions, heat generation or consumption due to reactions, and heat transfer via effective conduction throughout the solid matrix.

The three main types of growth laws that have been established experimentally, linear, parabolic and the logarithmic law are illustrated in Table 3, this equations represent the oxidation reaction rate of a metal at high temperatures published in popular science magazines.

Table 3. Reaction rate equations, applicable to high-temperature oxidation.

Equation	Observation	Ref.
	Logarithmic reaction rate:	
$X = K_{reaction} \ln(t + t_o) + A$	It represents the initial oxidation states at low temperatures.	[31]
$X = K_{reaction} \ln(Bt + 1)$	-	[31]
	Lineal reaction rate:	
$\frac{dx}{dt} = K_{Lineal}$	It represents a constant rate of oxide growth applicable at very high temperatures.	[31]
$X = K_{Lineal}t + C_{Lineal}$	-	[31]
$\Delta W/A = K_L t$	-	[42]
	Parabolic reaction rate:	
$\frac{dX}{dt} = \frac{K_{Parab}}{x}$	It adjusts to the processes controlled by the diffusion of species.	[31]
$X^2 = K_{parab}t + C_{parab}$	-	[31,43]
$(\Delta W/A)^2 = K_{parab} * t$	-	[42]
$\frac{dX}{dt} = k f(X)$	The function f(X) depends on the reaction mechanism for diffusion in one, two, or three dimensions.	[44]
$\frac{dX}{dt} = k(T) * f(X)$	The reaction rate of a sample of iron slag in water vapor.	[45]
$k(T) = k_o * e^{\left(\frac{-E_a}{RT}\right)}$		
Where:		
X: the thickness of oxide consumed per surface unit or the weight gain per area unit.		
$\Delta W/A$: the mass gain per unit area (g/cm ²)		
t: is the time.		
k(T): the reaction rate constant.		

$f(X)$: function that represents the reaction mechanism.

k_0 , K_{reaction} , K_{Lineal} , K_{Parab} : reaction constants.

E_a : is the activation energy.

R is the gas constant.

T is the absolute temperature.

A , B : constants.

C_{Lineal} , C_{Parab} : constants of integration.

The Fe oxidation kinetic follows a linear-parabolic behavior in the presence of $\text{H}_2\text{O}_{(\text{g})}$, suggesting a transition process, where the diffusion of the hydroxyl ion is a determining factor in the kinetic during the oxidation process [42]. Fujii & Meussner et al. [46] indicated that the kinetic oxidation of iron at 1100°C can adjusted to a parabolic law during the first 3 hours and later a linear behavior is established, where increase the weight per unit area of the all specimens (rate of weight gain was $6,2 \text{ mg/cm}^2 \text{ hr.}$). The effect of temperature on the reaction rate is obtained by applying the Arrhenius and is represented by the following expression [44,45].

$$K = K_0 e^{\frac{-\Delta G^0}{RT}} \quad (10)$$

Where K is the equilibrium constant, K_0 is the reaction constant, ΔG^0 is the activation energy, R is the gas constant, and T is the absolute temperature, respectively.

If the reaction rate is governed by diffusion in the solid state through the oxide layer. During the diffusion process, the oxide thickness (rust) increase, but at the same time the kinetic decreases with time [31]. In the cathodic subprocess occur the reaction $2\text{H}_2\text{O}_{(\text{g})} + 2e^- \rightarrow 2\text{OH}^- + \text{H}_2$ product of the high temperature dissociation process. The scientific literature indicates that the formation of OH^- from $\text{H}_2\text{O}_{(\text{g})}$ is proportional to the concentration of $\text{H}_2\text{O}_{(\text{g})}$ and that the appearance of OH^- cannot be interpreted in terms of the simple dissociation of $\text{H}_2\text{O}_{(\text{g})}$. It is necessary to consider the pathway that involved in the formation of intermediate products generated during ORR such as (i) H_2O_2 , and (ii) HO_2^- [47].

4. High-Temperature Oxidation Mechanisms

Despite the extensive scientific information on the oxidation of iron at environmental temperature and at high temperature due to the presence of the oxygen ion and in order to reduce or avoid iron corrosion, it is still not clear enough what the mechanism of electrochemistry reaction that explains the presence of oxides such as FeO , Fe_3O_4 and Fe_2O_3 , especially when iron interacts with the hydroxyl ion (OH^-).

Shrinivasan et al. [48] developed a multi-step optical system that enables water decomposition for OH^- ion detection, demonstrating that rate constants can be accurately measured at lower temperatures such as 500 K. This stands in contrast to alternative methods that necessitated temperatures exceeding 2570 K [49].

Lede et al. [9] propose a kinetic mechanism of thermal dissociation of H_2O between 2000 – 3000K represented by the following expressions.



Where, G is the extinguishing gas (argon at room temperature or steam at 400 – 450K and between 2.5 - 9.4 bar, H is the H^+ and OH is the OH^- ions, respectively.

Fe corrosion is an electrochemical dissolution process that produces an electron transfer to an intermediate species formed by the interaction between ferrous ions and water. In the reduction of Fe , the intermediate species is found on the iron surface and must continue to react in a second reaction to form Fe [50].

The initial rate of oxidation of a newly exposed surface of Fe or steel recently exposed to $\text{H}_2\text{O}_{(\text{g})}$ is always lower, if we compare it to the rate of corrosion generated by O_2 . Since the trend is for H_2 to dissolve in solid metals in the atomic form, rather than the molecular form, according to researchers

Tuck et al. [6,42] it would seem unlikely that H₂ would dissolve in an oxide network, as it is too large to diffuse, so they suggest that the cathodic reaction could occur within the scales.

Equation 15 shows the general reaction for the production of H₂ gas.



The mechanism of evolution of H₂ (HER) would occur on the surface of the pores and micropores where the hydroxyls (OH⁻) formed by the dissolution of the vapor are discharged through a cathodic reaction according to Figure 2.

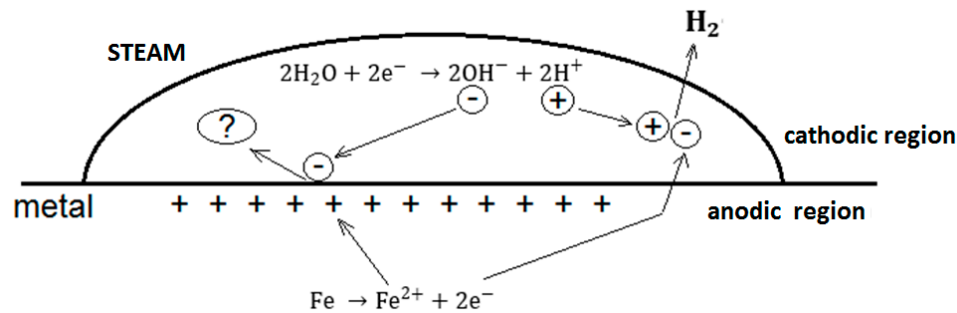


Figure 2. Diagram of the electrochemical dissolution of iron.

Rahmel & Tobolski et al. [51] proposed that the oxide layer is still thin and malleable in the first step the of oxidation process. In the presence of H₂O_(g), the transport mechanisms are specifically through cavity or pore formation is probably to occur. The mechanism is represented by Figure 3.

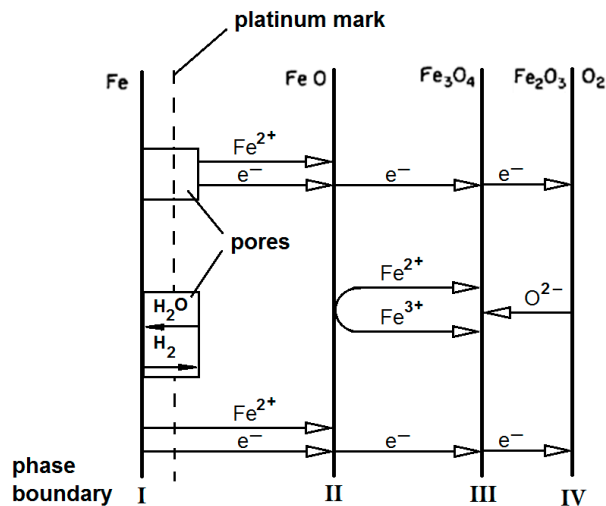


Figure 3. The mechanism in O₂-H₂O vapor mixtures after long periods (adapted from [51].)

Schuetze et al. [27] proposes that the H₂O_(g) is involved in the transport processes that lead to scale growth in the oxide layer. The researchers indicate, the Fe cations (Fe²⁺/Fe³⁺) can simultaneously diffuse towards the outer surface in this process.

In particular, in the large pores in the inner part of the scale of FeO, a circulation mechanism is assumed that consists of the oxidation of Fe in contact with H₂O_(g), thus releasing H₂ that can move back in the pore towards the outer part of the scale, where it reduces the oxide thus forming a H₂O_(g) molecule again (Figure 3 and Table 4).

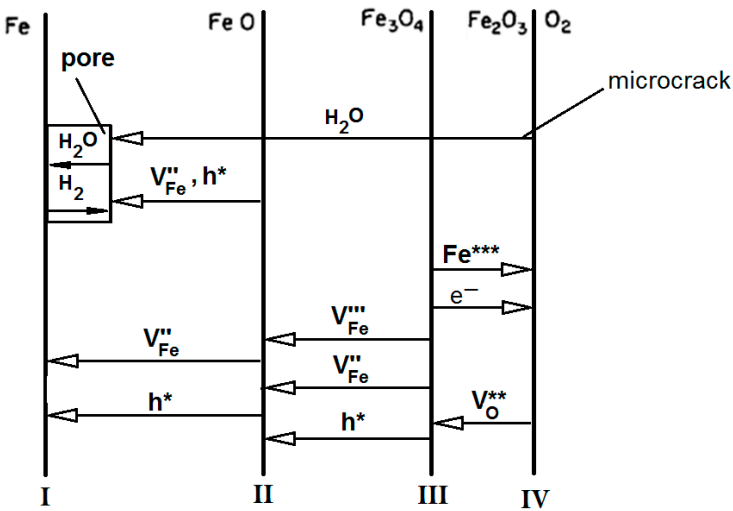


Figure 4. Schematic diagram of the transport processes in the growth of oxide scales on iron for oxygen atmospheres containing H₂O (adapted from [27]).

Table 4. The nomenclature used in the transportation process was proposed by Schuetze [27].

Nomenclature	Meaning
V'''_{Fe}	Fe ³⁺ vacancy on a cation position
V''_{Fe}	Fe ²⁺ vacancy on a cation position
h^*	h ⁺ electron-hole
V^{**}_O	O vacancy on an anion position
Fe^{***}	Fe ³⁺ ion on an interstitial position

Based on the knowledge of the structure and diffusion of iron oxides, the oxidation mechanism of pure iron above 570 °C in an iron-oxygen systems are shown in Figure 5 [52], Figure 6, and Table 5 [42].

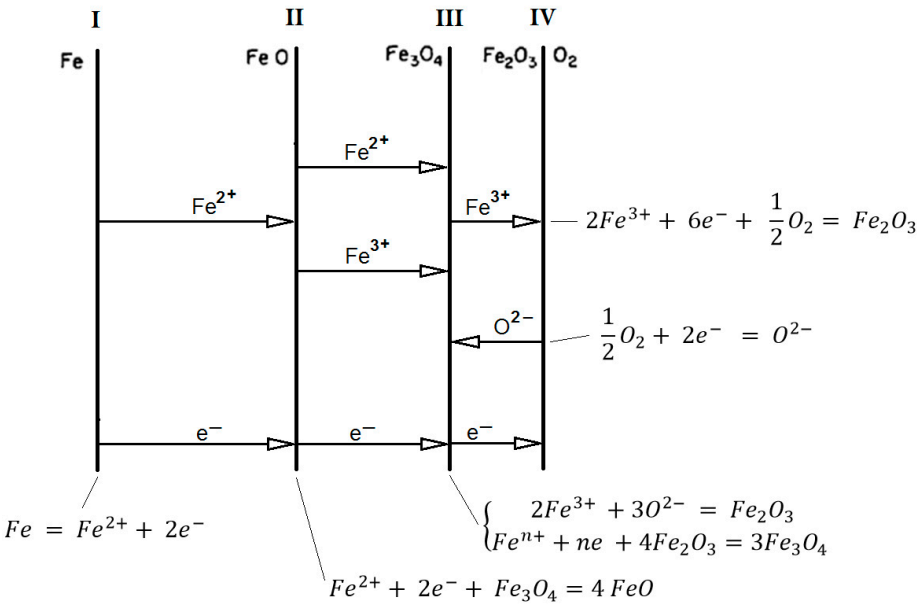


Figure 5. Iron oxidation mechanism showing diffusion of various ionic species (adapted from [52,53]).

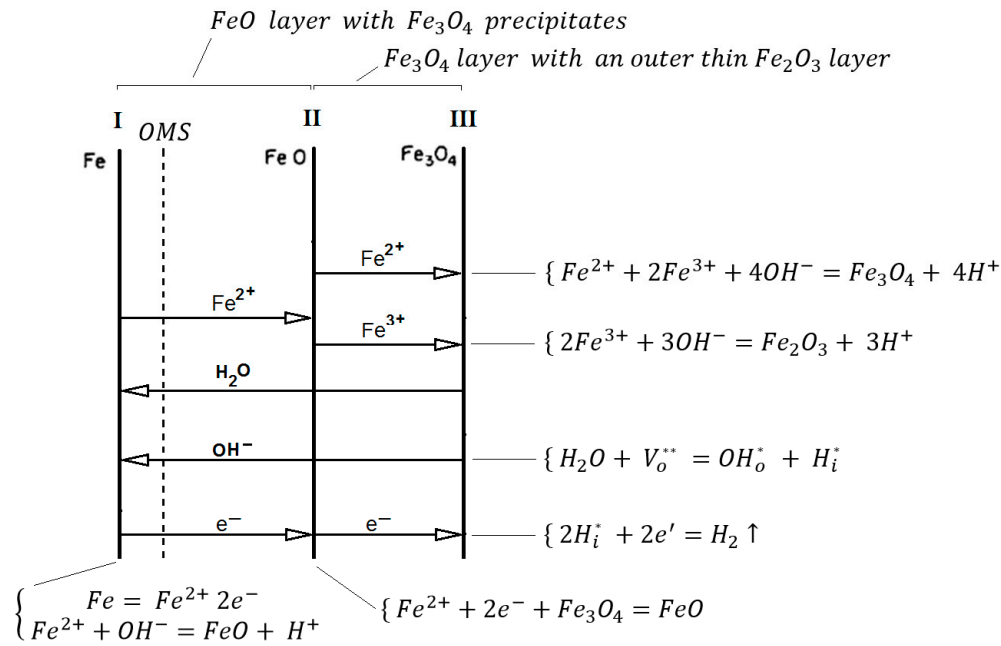
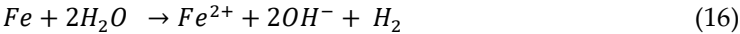


Figure 6. Mechanism of Fe oxidation processes in steam (adapted from [42].)

Table 5. Transport process nomenclature proposed by Yuan et al. [42].

Nomenclature	Meaning
V_o^{**}	Fully ionized oxygen vacancy
OH_o^*	Substitutional hydroxide ion
H_i^*	h^+ electron-hole
e'	Electron
OMS	Original Metal Surface

The first reaction that occurs is from the formation of OH⁻ ions, which would increase cation vacancies (eq.16) and would also be the main diffusion species [5,6,42].



Yuan et al. [42] suggest the formation of ferrous oxide (FeO) (eq.17, Figures 7 and 8), that is, iron in the presence of water vapor at high temperatures will precipitate FeO and Fe₃O₄ [42] or Fe₂O₃ [26].



However, FeO in contact with Fe(OH)₂ [54] is generated at temperatures above 1300°C [19] and can form Fe₂O₃ in conjunction with the release of H₂ gas [42].

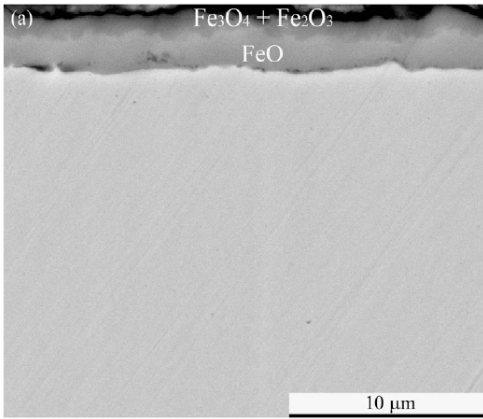


Figure 7. Cross-sectional SEM image of oxides formed on iron after 1 h of oxidation at 650 °C [42].

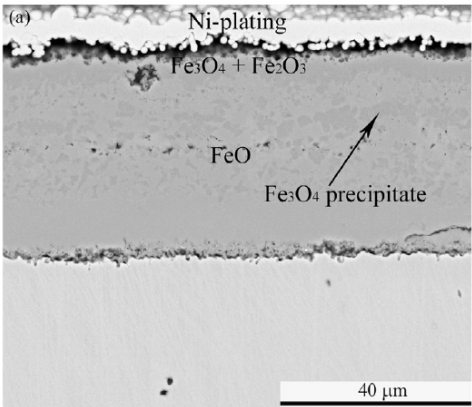


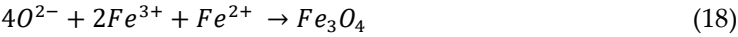
Figure 8. Cross-sectional SEM image of oxide scale formed on iron after 10 h of steam oxidation at 650 °C, where the outermost white layer is nickel-plated [42].

Rahmel & Tobolski et al. in 1965 [51] proposed the existence of pores at the Fe/FeO interface during the oxidation process, forming oxide bridges from the metal to the scale, which allows further oxidation of the metal without substantial inhibition. A mixture of H₂/H₂O is formed in these pores and through an oxidation/reduction mechanism O₂ is transported to the Fe surface.

An oxide of the Fe_nO_m type will normally contain a variety of defects. These defects are responsible for the transport of material through the oxide layer and therefore play a critical role in the oxidation process [6,55].

Yuan et al. [42] indicate in their research that during the initial period (before 5 hours), the existence of a single layer of columnar grains may allow relatively fast transport of OH⁻ ions across grain boundaries. Therefore, it is valid to assume that the hydroxyl ions interact with the surface Hematite and Magnetite, forming FeO.

However, some aspects remain to be investigated, such as that proposed by Stehle et al. [56], who indicate that at typical reaction temperatures (T> 327 °C), the mobility of oxygen and metal ions is expected to be very high, so diffusion limitations are not considered significant, except for the oxides that exceed the thickness of several microns, where the potential of iron III must be considered (eq.18) [42,56,57].



Another aspect that should be considered is what was stated by Saunders et al. [18], who indicate that the oxide growth, including the adsorption, dissociation, and diffusion of the reagents, are altered in the presence of water vapor, for which reason it will imply considering thermodynamics, the development of microstructures and the processes of transport. Table 6 shows the standard reduction potentials that have been considered in the electrochemical analyzes of the oxidation of iron and iron oxides in steam.

Table 6. Standard Reduction Potential.

Half reaction	Standard Reduction Potential, E° (V)	Ref.
H ₂ → 2H ⁺ + 2e ⁻	E° = 0,000 (V)	[58]
Fe → Fe ²⁺ + 2e ⁻	E° = - 0,447 (V)	[58]
Reaction		
3Fe + 4H ₂ O → Fe ₃ O ₄ + 4H ₂	-	[45]
2FeO + H ₂ O → Fe ₂ O ₃ + H ₂	E° = 0,356 (V)	[45]
2Fe ₃ O ₄ + H ₂ O → 3Fe ₂ O ₃ + H ₂	E° = 0,210 (V)	[30]
2Fe ₃ O ₄ + 4H ₂ O → 6FeOOH + H ₂	E° = 0,230 (V)	[30]

From these reactions, electrochemical reaction kinetics can be understood due to the importance it has for the application of metal oxide redox reactions in energy conversion systems such as chemical loop systems and H₂ storage [59].

In an activation regime, the speed of the electrochemical process is due to the transfer of electrons as the only variable that controls the speed of the global process [31], so the corrosion reaction can be expressed by the ionization of a metal. But the possibility that this reaction occurs spontaneously under real conditions forces us to study the energy changes associated with the reaction.

However, some H⁺ ions migrate into the metal-forming H₂, therefore, the presence of H⁺ ions can promote stress corrosion cracking, through the process of hydrogen embrittlement [60].

FeO nucleation and growth are enhanced by increasing oxygen pressure [55,61]. Kogan et al. [62] studied the dissociation of water at temperatures of 2000, 2200, 2500, and 2800 K, and a pressure of 0.05 (bar), and determined that 25% (at 2500K) and 55% (at 2800 K) of the Water vapor dissociates at constant pressure, increasing at high temperatures.

Kodama et al. [63] indicates that the separation of water by a thermochemical process at a pressure of 0.01 bar and 2000K is barely perceptible, and by increasing the temperature to 2500K, the yield of H₂ exceeds 15% at the same pressure. Young et al. [17] proposed that in pure steam or mixtures of water vapor and inert gas, the equilibrium value of p_{O2} (oxygen pressure) is determined by the degree of dissociation of H₂O, and that, in the case of pure steam, the dissociation of one mole of water produces x moles of H₂ and x/2 moles of O₂, and x is calculated from the equilibrium expression shown in Equation 19.

$$K_1^2 = \frac{x^3}{2(1-x)^2 \left(1 + \frac{x}{2}\right)} P_T \quad (19)$$

Where P_T is the total pressure and since K₁ is small (x << 1), therefore the before expression approaches:

$$x = \left(\frac{2K_1^2}{P_T} \right)^{1/3} \quad (20)$$

$$p_{O_2} = x P_T \quad (21)$$

Ehlers et al. [64] propose that Fe(OH)₂ formed within oxide scales at low oxygen pressure (P_{O2}), migrates to the outer surface, where due to higher pressure hematite (Fe₂O₃) is formed. Khanna et al. [52] propose that iron in the presence of oxygen when oxidized forms a mixed scale of three oxides (FeO, Fe₂O₃, Fe₃O₄), whose composition varies with the temperature and with the partial pressure of O₂ according to the iron phase diagram (Figure 9).

Ketteler et al. [65] in their research, consider a Fe-H₂O system, to determine the stability of iron oxide as a function of the partial pressure of water, indicating as the phase limit for water condensation the range 125 K (1x10⁻¹¹ mbar) up to 373K (1 bar). The calculated phase diagram is presented in Figure 10, as a function of oxygen pressure and temperature, according to the equilibrium constant of dissociation of water (2H₂O → 2H₂ + O₂), according to equation 22. The researchers conclude that water in its gaseous state acts as an oxidizing compound and that the stability of iron oxides is determined by the partial pressure of the decomposition of water into hydrogen and oxygen.

$$k_1 = \frac{p_{O_2} p_{H_2}^2}{p_{H_2O}^2} \quad (22)$$

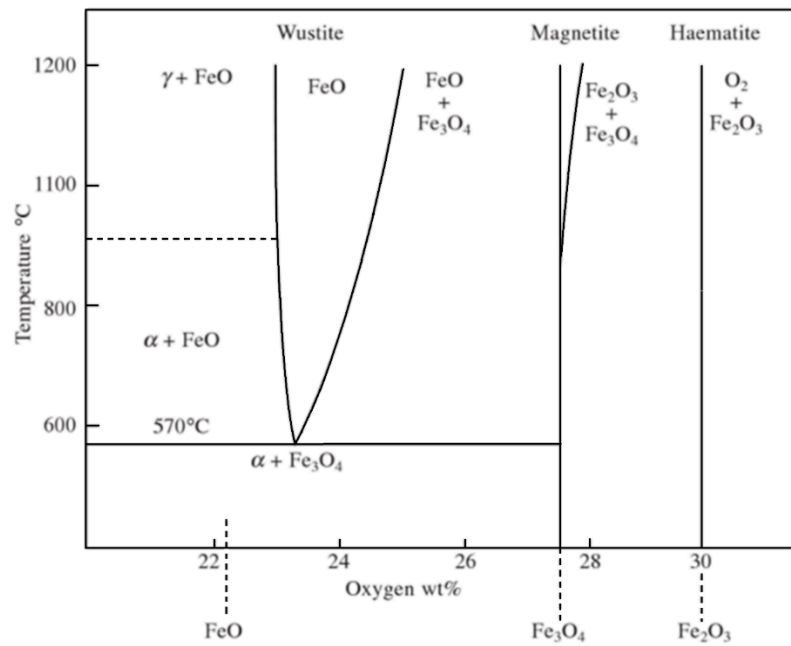


Figure 9. Phase diagram, depicting the stability of various iron oxides as a function of oxygen pressure and temperature [52,53].

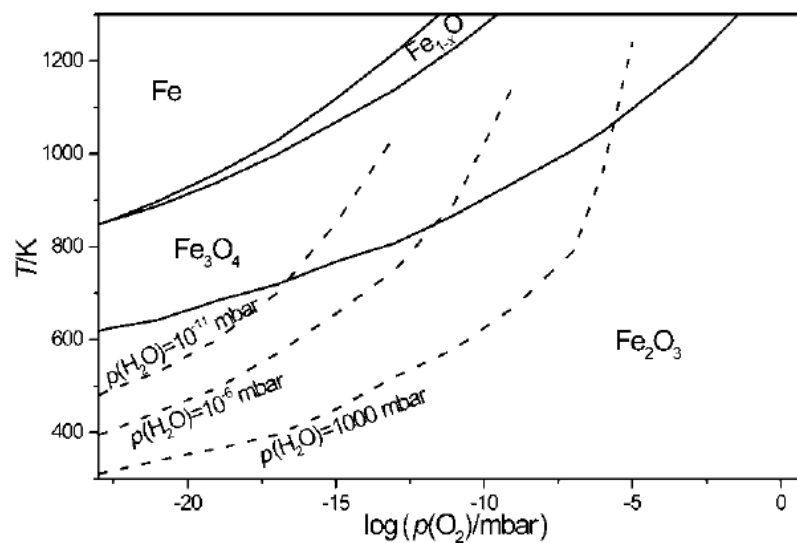


Figure 10. The phase diagram in a water atmosphere represents the stability of iron oxides [65].

5. Effect of pH on Oxidation and Temperature

The pH value is normally based on the equilibrium reaction of the dissociation of water ($\text{H}_2\text{O} \rightarrow \text{H}^+ + \text{OH}^-$), this reaction has an endothermic character, and its equilibrium shifts to the right side with increasing temperature.

Research indicates that at the temperature of 300°C (25 MPa), the concentrations of both H^+ (stable in acidic solutions) and OH^- (stable in basic solutions) [60] are approximately three orders of magnitude above those values of water at ambient temperature. Therefore, water has been considered both acidic and alkaline [66].

To produce green hydrogen from a thermochemical process (thermolysis), it is necessary to reach temperatures in the range of 800 – 1400 °C in the reactor [67], this temperature range has been reached in a solar thermal concentrator system of down beam.

Iron as a material for obtaining hydrogen has been considered for the thermochemical solar process because the conversion of Fe_3O_4 to FeO improves significantly with higher thermal reduction temperatures [42]. However, its application is complicated because in a thermochemical cycle (thermal reduction stage) the fusion of Fe_3O_4 occurs at temperatures above 2227°C (Figure 11) [7,68] and FeO melts at temperatures as low as 1370 [11] or 1400°C [16]. These factors complicate the design and operation of thermochemical reactors (STWS) based on $\text{Fe}_3\text{O}_4/\text{FeO}$ [11]. But the thermal oxidation stage (hydrolysis) is carried out at temperatures between $200 - 1000^\circ\text{C}$ [4,7,10,28,56]. According to the scientific literature, the temperature affects the process because the oxidation of iron by steam is thermodynamically favorable in the range of $127\text{-}527^\circ\text{C}$ [11,24] or in the temperature ranges of $650\text{-}750^\circ\text{C}$. [42], $700\text{-}850^\circ\text{C}$ [26], but reaction temperatures in the range of 717 to 1127°C [57] have also been investigated.

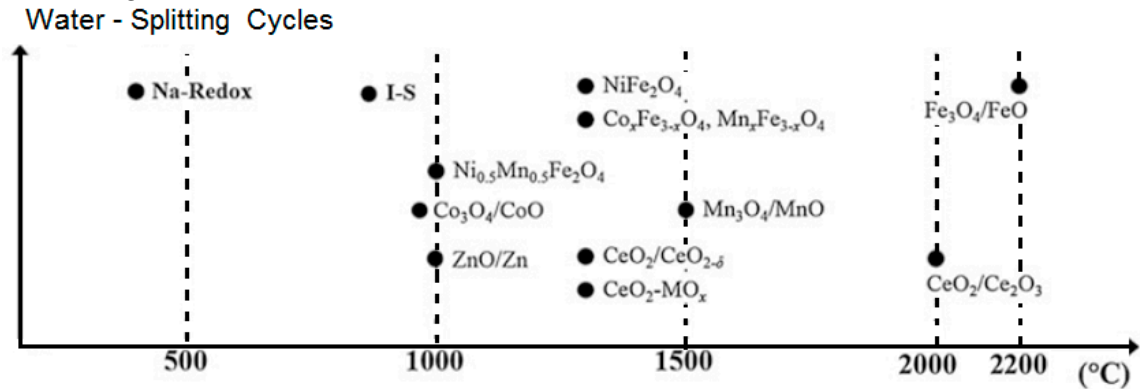


Figure 11. Operating temperature of typical thermochemical water splitting cycles in the reduction stage (adapted from Hiroki Miyaoka, 2016) [68].

6. Material Used in Fluidized Bed

It is important to highlight that the selection of iron oxide is because it is an environmentally friendly base material, with a relatively low cost and that it is generally used in the redox reaction due to its high oxygen-by-weight ratio in mass with sufficient reaction time [44].

From the reviewed scientific literature, it was determined that a temperature above 1300°C can be obtained when using a thermochemical solar reactor and some metallic material (cerium oxide or zinc oxide) in a fluidized bed [67,69]. The thermal powers obtained in reactors that use water-steam and different types of material in a fluidized bed are presented in Table 7. The iron oxide materials investigated for the separation of water in the oxidation stage at the laboratory level are presented in Table 8.

Gokon et al. [29], observed that the hydrogen production rate after steam injection reached a maximum value of $12.3\text{ Ncm}^3/\text{min}$ at 25 min, after which the hydrogen production rate decreased rapidly. This shows that the production of hydrogen through a fluidized bed is feasible.

Below is presented in Table 9, a list of thermochemical solar reactors, which have used a fluidized bed material and different gases for fluidization, considering the geometry of the reactor and the material used for its construction.

Table 7. Temperatures reached in a Thermochemical Solar Reactor with Fluidized Bed.

Thermal Power	Temperatures reached or required	Thermal Fluid	Fluidized Bed Material	Particle Size (μm)	Ref.
1 kW _{th}	1300 °C	Water/Steam	Coal – coke	140	[63,70–72]
				(200-300)	
				(300-500)	
				(500-710)	

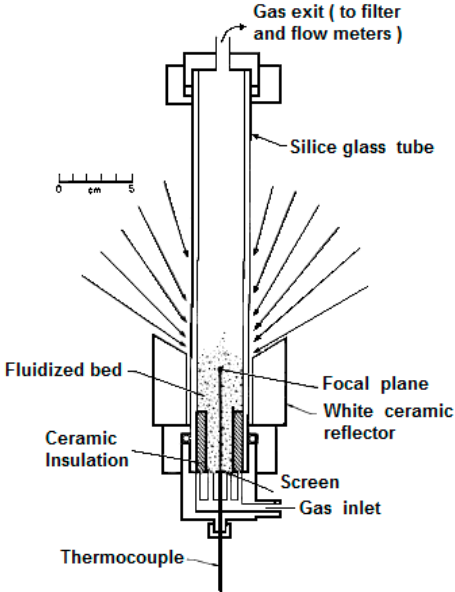
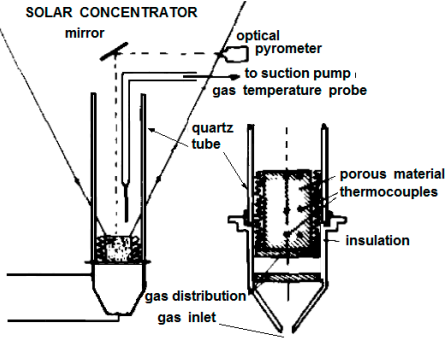
10 a 20 kW _{th}	2200 °C	Water/Steam	ZnO reagent powder	1 - 5	[73]
3 MW _{th}	1500°C.	Water/Steam	Cerium Oxide Material	< 300 (100 – 300)	[28,67,74,75]
110 kW _{th}	1400 °C	Gases	Non-stoichiometric cerium oxide particles	10 - 210	[69,76]
100 kW _{th}	960 – 1100 °C	Air	Quartz sand particles	100 - 500	[77]
250 kW _{th}	800 – 1000°C	Air	A mixture of coal-coke and quartz sand	100-300 (coal) 100 – 700 (sand)	[78]
30 kW _{th}	560 °C.	Air	A mixture of sand and basalt		[79]
450 kW _{th} 140,63 W _{th} (2kW _e) 231,32 W _{th} (4 kW _e)	770 °C.	Air	Isotropic materials		[80,81]
	250 °C	Air	Sand, ceramic casting media (carbo Accucast ID50) y SiC		[82]

Table 8. Iron Oxide Materials.

Iron Oxides Material		Sample/Particle Size (µm)	Temp. (°C) (time)	Partial Vapor Pressure	Thermal Fluid (flow)	Ref.
Magnetite	Fe ₃ O ₄	30 -50	575	-	-	[16,63]
		100 -125	673			
The partial substitution of iron in Fe ₃ O ₄ by Ni, Co, and Zr, to form mixed metal oxides.	(Fe _{1-x}) M _x) ₃ O ₄	Sample in a quartz tube	1000	-	-	[16,28,63]
	NiFe ₂ O ₄ NiFe ₂ O ₄ /m- ZrO ₂					
Magnetite supported on zirconium stabilized with cubic yttria	Fe ₃ O ₄ /c-YSZ	Ceramic foam	1100 (80 min)	75% of the steam pressure at 90°C, 1 bar	H ₂ O/N ₂ (10 Ncm ³ /min)	[76]
Commercial nickel ferrite supported on zirconium oxide	NiFe ₂ O ₄ / ZrO ₂	Sample on Pt cup	1000 (60 min)	steam pressure at 80°C, 1 bar	H ₂ O/N ₂ (4 mL/min)	[83]
Unsupported Commercial Nickel Ferrite	NiFe ₂ O ₄	212 -710	1,6 – 1,7 kW _{th} (10 - 92 min)	51%(0,51 atm) of the steam pressure at	H ₂ O/N ₂ (0,24 Ndm ³ /min)	[29]

					82°C, 1 atm		
Monoclinic magnetite supported on zirconium substrates	Fe ₃ O ₄ /m-ZrO ₂	Polished Fe Bar without oxidation	397 – 602 (180 min)	-	Nitrogen (100 cc/min) Argon (200cc/min) Liquid water (12,5 µL/min)	[56]	
	Fe ₂ O ₃ /ZrO ₂						
	Fe ₂ O ₃ /CeO ₂						
	Fe ₂ O ₃ /YSZ	150 -300	550	-			
iron oxide with various support materials, ZrO ₂ , CeO ₂ , yttria-stabilized zirconia (YSZ), and gadolinia-doped ceria (GDC)	Fe ₂ O ₃ /YSZ				H ₂ O/Ar (prop. 5:95) (300 mL/min)	[59]	
	Fe ₂ O ₄ /GDC						

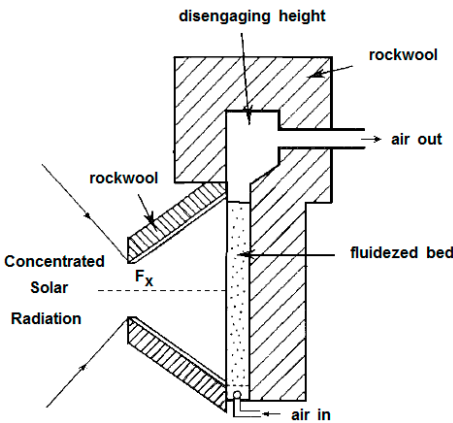
Table 9. Technologies of thermochemical solar reactors with fluidized bed.

Reactor Geometry	Bed /gas	Material	Reactor	Year/ Ref.
Diameter=5cm Height= 32cm	Charcoal /CO ₂	Silice glass tube		1983 [84]
	ZrO ₂ , SiC /aire,N ₂ ,CO ₂	clear quartz tube		1983 [85]

0.90 m high,
0.78 m wide,
with a radius
of curvature of
0.78 m

alumina
particles/air

refractory
stainless
steel (AISI
310)

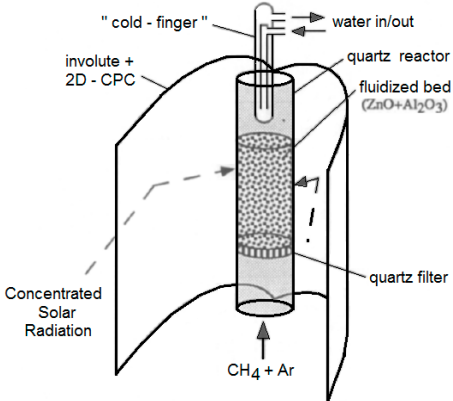


1988
[86]

Diameter=
2cm

ZnO+Al₂O₃/
CH₄+Ar

quartz tube

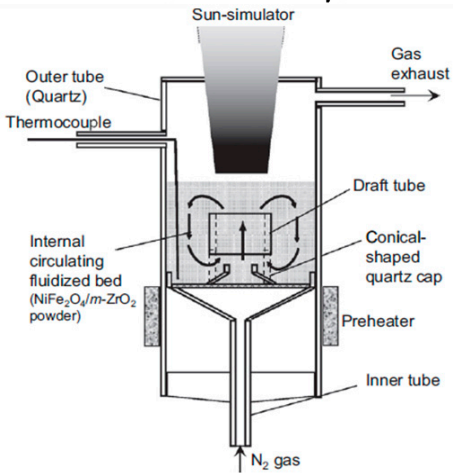


1995
[87]

Diameter= 45
mm, thickness
2.5 mm

NiFe₂O₄ –
ZrO₂ /N₂

quartz tube

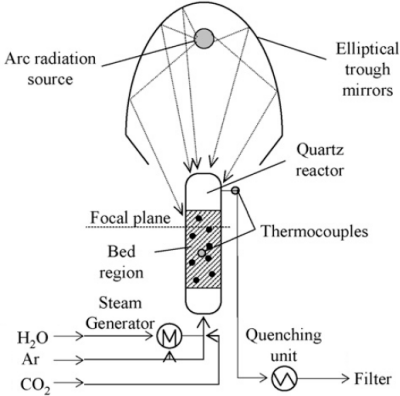


2008
[88]

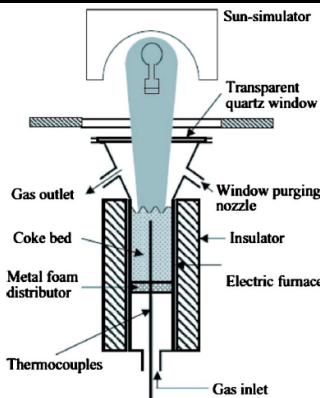
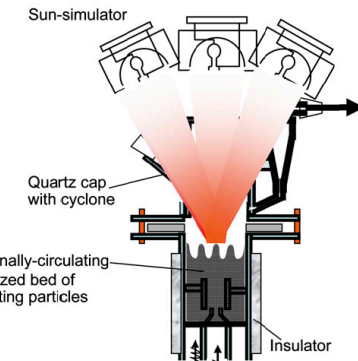
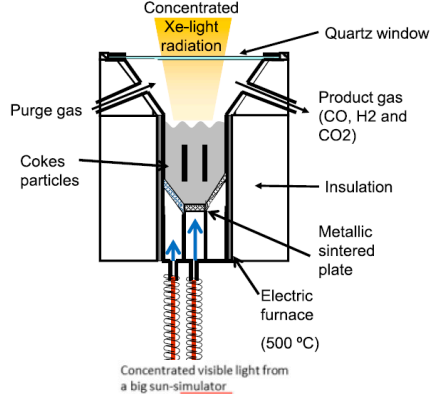
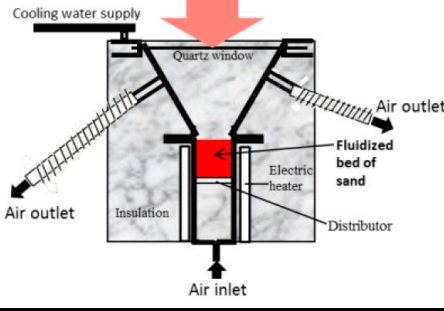
25mm in
diameter,
thickness
1.5mm, height
25cm.

CaO or
CaCO₃
/H₂O,Ar,CO₂

quartz tube



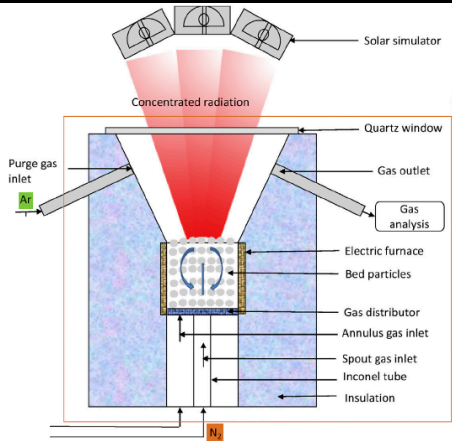
2009
[89]

420mm in length, 62.3mm in inner diameter, and 7mm in thickness	Coke/CO ₂	Stainless steel with a quartz window		2010 [70]
The inner diameter was 45 mm, and the thickness was 2.5 mm.	NiFe ₂ O ₄ /vapor	Stainless steel tube with a quartz cap		2011 [29]
length 420 mm, inner diameter 62.3 mm, thickness 7 mm	Coke/Ar-vapor CeO ₂ /N ₂	Stainless steel tube (SUS310S) with a quartz window		2015 [71,74]
42 cm diameter at the top	quartz sand particles/air	Inconel and stainless steel, with a quartz window		2016 [90]

0.2 m long and
0.3 m internal
diameter at the
top and 0.007
m thick

quartz
particles /N₂

Stainless
steel tube
with quartz
window (5
mm thick)

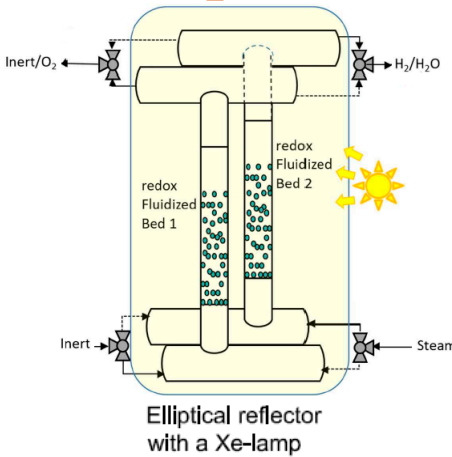


2018
[91]

35.6 cm long
and 2.54 cm
outer diameter

Iron
Aluminate
(FeAl₂O₄) / N₂

Silicon
Carbide
Cylindrical
Tubes (SiC)

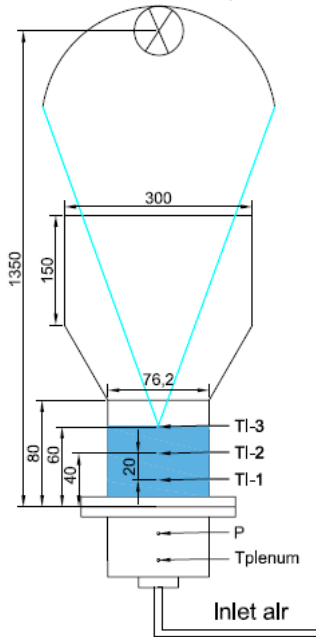


2019
[92]

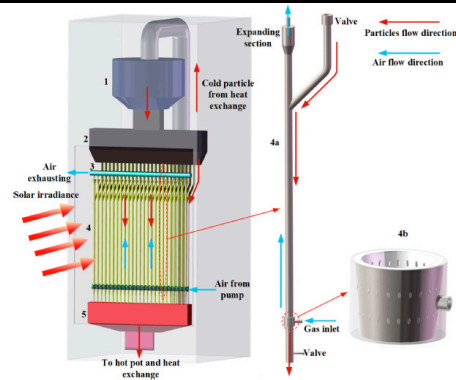
inside
diameter of
7.62 cm and a
height of 8 cm

sand, carbo
Accucast
ID50 y SiC
/air

Stainless
Steel tube
(304)



2020
[82]



Direct solar thermal water splitting using Fe electrodes at high temperatures is a promising advancement in the field of renewable energy and hydrogen production. By leveraging solar power

and affordable materials, it offers a sustainable and scalable solution to the challenges of clean hydrogen production. As this technology matures and overcomes its current challenges, it has the potential to play a vital role in the transition to a more sustainable and greener energy future, reducing our reliance on fossil fuels and lowering carbon emissions across various industries.

According to data collected from the review of scientific literature, iron will precipitate in the form of ferrous oxide, hematite and magnetite during the oxidation of iron in steam at high temperature, generating hydrogen. Therefore, it is valid to assume that iron, iron oxide or other metal oxides present in slags, such as copper slag, will also precipitate rust and hydrogen. The theoretical thermodynamic and electrochemical aspects of the reaction of metal oxides with water vapor are fundamental for the design of the solar reactor, as well as for an adequate selection of the catalyst metal.

Future recommendations will be aimed at determining whether mineral waste (copper slag) containing metal oxides (iron oxides) are feasible to be used as thermocatalytic materials for the production of hydrogen. For this reason, a morphological and thermochemical characterization of the copper slag produced in Chile will be carried out to determine its subsequent use in a water splitting using a thermochemical solar reactor.

Future research holds significant promise for advancing this technology and addressing various challenges. Here are some key areas of research that can help further develop this innovative approach: (i) Material Science and Corrosion Resistance, (ii) Efficiency Enhancement, (iii) Thermochemical Cycle Optimization, (iv) Integrated Energy Storage, (v) Economic Viability, (vi) Long-term Durability and Reliability, (vii) Scalability and Modular Design, (viii) Hydrogen Purity and Quality and (ix) Market Integration and Policy Support incentives to encourage the adoption of solar thermal water-steam splitting to make this technology more efficient, cost-effective, and environmentally friendly. By addressing these research areas, we can potentially overcome the current challenges and pave the way for its widespread adoption as a clean and sustainable method for hydrogen production.

Author Contributions: Conceptualization, M.F. and F.M.G.M.; methodology, M.F. and F.M.G.M.; validation, A.S., E.F., D.P. A.Sa. and N.T.; investigation, M.F., N.T. D.P. and F.M.G.M.; writing—original draft preparation, M.F. and F.M.G.M.; writing—review and editing, M.F. A.Sa E.F., D.P. A.S., N.T. and F.M.G.M.; visualization M.F., A.S., and F.M.G.M.; supervision, F.M.G.M. All authors have read and agreed to the published version of the manuscript.

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