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Article

Experimental Evaluation of Mechanical Compression Properties of Aluminum Alloy Lattice Trusses for Anti-Ice System Applications

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Abstract: Lattice structures have emerged as promising materials for aerospace structures applications due to their high strength-to-weight ratios, customizable properties, and efficient use of materials. These properties make them attractive for use in anti-ice systems, where lightweight and heat exchanging are essential. This paper presents an extensive experimental investigation into mechanical compression properties of lattice trusses fabricated from AlSi10Mg powder alloy, a material commonly used in casted aerospace parts. The truss structures were manufactured using additive manufacturing Selective Laser Melting technique and subjected to uniaxial compressive loading to assess their performance. The results demonstrate that AlSi10Mg lattice trusses exhibit remarkable compressive strength with strong correlations depending upon both topology and cells' parameters setup. The finding described highlights the potential of AlSi10Mg alloy as a promising material for custom truss fabrication, offering customizable cost-effective and lightweight solutions for aerospace market. The paper also emphasizes the role of additive manufacturing in producing complex structures with pointwise tailored mechanical properties.

Keywords: Additive Manufacturing (AM); Selective Laser Sintering (SLM); lattice structures; Design of Experiments (DOE)

1. Introduction

The aviation industry is under increasing pressure to reduce its carbon footprint due to the significant contribution to global greenhouse gas emissions [1–7]. With the projected growth in air travel, the demand for more fuel-efficient and environmentally friendly aircraft is becoming a priority for manufacturers. One way to achieve this goal is through the introduction of novel technologies such as Additive Manufacturing (AM) [8]. AM offers significant benefits to the aerospace industry, including reduced material waste, increased design flexibility, and faster production times [9]. The use of AM in aerospace products has the potential to significantly reduce the carbon footprint of aviation while also improving performance and reducing costs [10–15].

The exploration of lattice structures has been a significant topic in materials science and engineering, with comprehensive studies delving into various aspects of their mechanical properties [11,15–22]. Previous literature has extensively studied linear elasticity properties, such as stiffness in tension/compression and bending [23–25]. A key focus has also been on energy absorption and failure characteristics of lattice structures, addressing whether these exhibit quasi-brittle or elastoplastic behavior [23,25,26]. Moreover, the role of manufacturing inaccuracies on lattice structures, particularly geometric defects induced by Powder Bed Fusion (PBF), has also been meticulously investigated [27–29].

The hereafter discussed evaluation of mechanical compression of lattice trusses made with AlSi10Mg for an anti-ice system is just one part of a broader project involving several other studies and tests such as fatigue analysis and CFD simulations [30–33]. Furthermore, this project involves

cost evaluation studies, which aim to assess the economic feasibility of using lattice trusses made with AlSi10Mg for anti-icing systems in aircraft [16].

2. Materials and Methods

In the current study, an extensive experimental approach has been implemented, which considers different topological features and detailed statistical analysis for a bird-eye exploration of the structure-property relationships in lattice structures. The focus is on Aluminum Alloy AlSi10Mg, chosen for its printability in SLM machine as and for its suitability in mechanical properties for applications such as anti-icing systems in liner aircrafts. These systems encounter intense mechanical and thermal stresses at the leading edges during activation, due to running wet approach of the liquid water on the boundary layer [33].

For what concerns the specimens designed for compression tests, shape realized are parallelepipeds measuring $20 \times 20 \times 40 \text{ mm}^3$, in line with previous studies [34]. The dimensions of each specimen slightly varies based on cell sizes to ensure a finite number of cells on each side. The shape of the specimens, being twice as tall as the base's side, facilitates the examination of fracture behavior in various trabecular structures intended for the core of sandwich panels, minimizing the corner/edge effects.

A Full Factorial Designs (DoE) [35,36] has been employed, graphically reported in Figure 1 with three replicates for each specimen type. The two first specimens for each case have been directly tested while the third have been previously analyzed with non-destructive tests (CT Scan) to evaluate internal porosity, surface roughness and effective dimension.

Three parameters have been selected for each DOE are: Cell Type (shapes reported in Appendix A), Cell Size, and Relative Density. The Cell Type parameter affects the designed shape of the unit cell, three shapes have been selected, Bccz (a stretching structure) and two bending type: Octet truss and Rhombic Dodecahedron. The second parameter chosen has been the cell Size parameter which determines the length of the cell's sides (with all cells being cubic). Last Parameter chosen has been the Relative Density parameter which indicates the density of the elementary cells relative to the bulk of the reference cell. Numerical Data related to these parameters and the design approach are outlined in Table 1

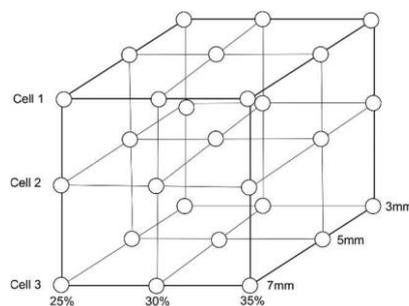


Figure 1. Doe Full Factorial Design.

Table 1. Truss Cells DOE Design.

Cell Type	Cell Size	Volume Fraction
Bccz	3	30
Bccz	5	25
Bccz	7	35
Rhombic	3	25
Rhombic	5	35
Rhombic	7	30
Octet	3	35
Octet	5	30
Octet	7	25

In the conducted research, lattice structures were fabricated utilizing SLM 500 Selective Laser Melting (SLM) system [37]. This additive manufacturing technique uses a high-power laser to sinter fine metal particles into a three-dimensional configuration. The process initiates with the laser selectively melting a layer of powdered material according to a predetermined 3D digital design. After each melting session, the powder bed is lowered, and a fresh layer of powder is spread. This procedure is iteratively executed, enabling the production of detailed and high-fidelity lattice structures.

The SLM 500 machine, permits the creation of metal components with notable mechanical attributes, intricate geometries, precise dimensions, and superior surface quality. Its incorporation of a quadruple laser system ensures rapid construction rates and efficient fabrication of large-scale components, suitable for aircraft's leading edges. For this study, the SLM 500's capabilities were leveraged to manufacture AlSi10Mg lattice trusses.

Compression testing was performed using a Zwick Roell testing machine equipped with a 50 kN load cell. The tests were executed at a constant speed of 1 mm/min, applying a pre-load of 1 kN. Figure 2 illustrates the machine's setup for the uniaxial compression tests.

Each specimen is described in the results section, identified by a label summarizing the cell's shape, size, and theoretical volume fraction. For example, the specimen labeled "Rhombic-5-25-2" features a rhombic dodecahedron geometry, a cell size of 5 mm, and a solid volume fraction of 25%. The final number in each specimen's label indicates the sequence of repetition, with this particular sample being the second in its series.

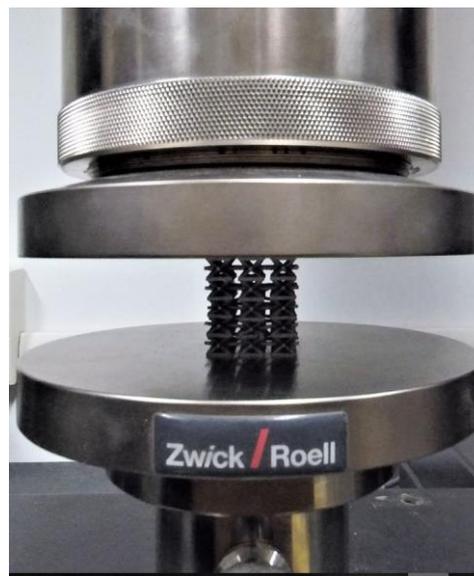


Figure 2. Setup of the Zwick Roell machine for uniaxial compression test on trabecular specimens.

In this study, the compression tests were analyzed using three distinct metrics derived from the stress-strain curves:

- Young's Modulus (E), calculated from the slope of the stress-strain curve within the linear elastic region;
- Maximum Stress (σ_{\max}), representing the highest point on the curve;
- Yield Stress ($\sigma_{0.2}$), defined as the stress value corresponding to a permanent plastic deformation of 0.2%.

The value of $\sigma_{0.2}$ is particularly significant as it marks the onset of the material's plastic deformation phase. This is determined by evaluating a parallel linear trend to the initial linear elastic portion of the specimen's behavior, intersecting at a deformation of 0.2%. Considering the variability in specimen dimensions, the main compressive property values are also normalized and presented in Appendix B.

3. Results

This section will present and discuss the experimental results obtained from the compression test. It will include both a quantitative report of the findings and a qualitative analysis of the rupture mechanisms observed. Detailed data on these outcomes is available in Appendix B while the Statistics related to DoE model are reported in Appendix D.

3.1. Bccz Specimens

Stress-strain behaviors for Bccz specimens are portrayed in Figure 3. Notably, specimen 3_35 achieved the highest maximum stress (σ_{\max}), showing significant plastic deformation with a strain at break reaching 15%. In contrast, Specimen 7_25, while registering the lowest σ_{\max} among the tested specimens, also exhibited minimal plastic deformation, not exceeding a 10% strain at break. Preliminary analysis indicates a predominant influence of cell size on the σ values, observing a decrement in mechanical performance from cell sizes 3 to 5 mm and subsequent stabilization moving from 5 to 7 mm cells. Additionally, a consistent upward trend is observed across all cell sizes in enhancing both Young's modulus and maximum stress correlating with the increase in density parameter. This relationships highlights the critical role of cell size and density in determining the material's mechanical characteristics. This aligns with the Gibson-Ashby model, which correlates an increase in relative density with enhanced mechanical properties.

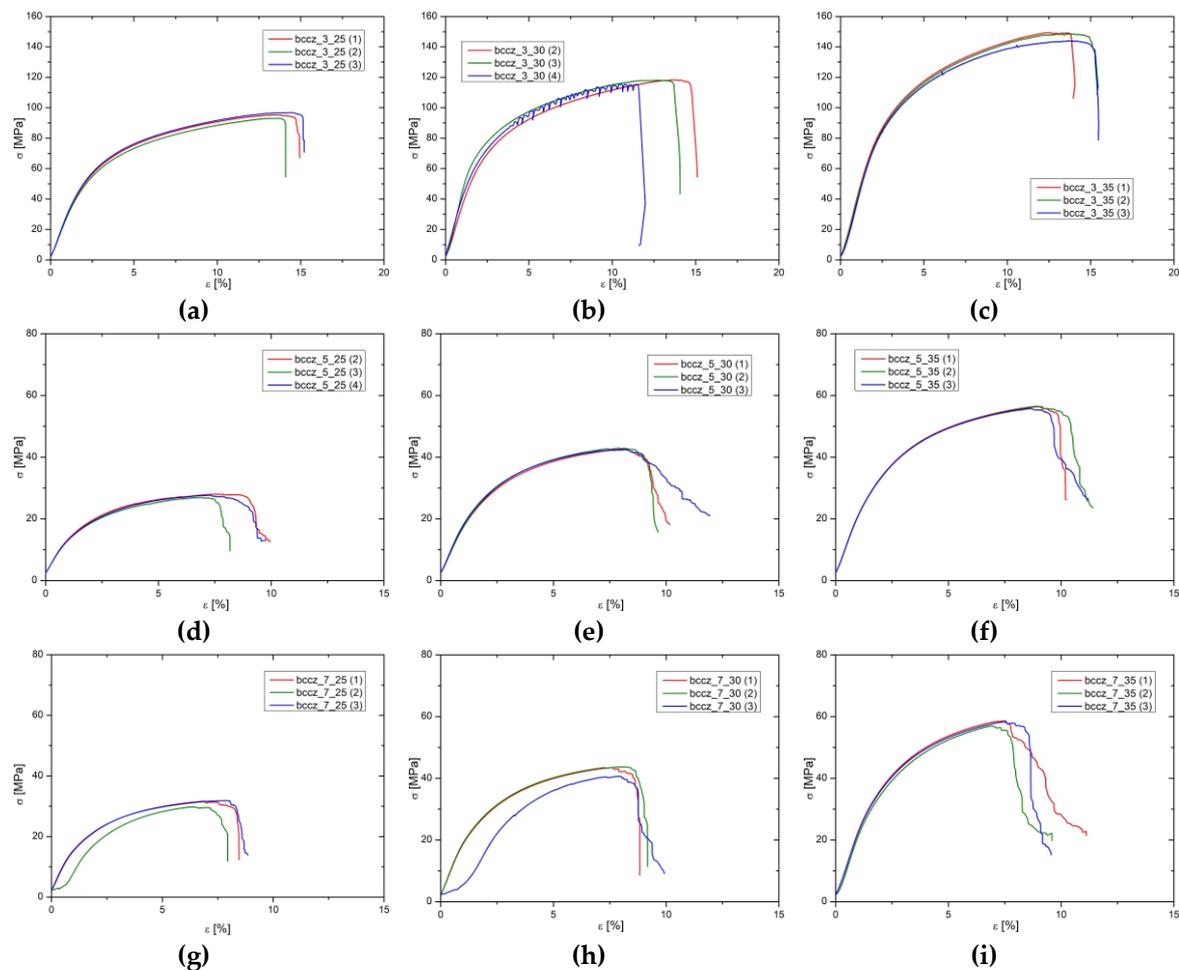


Figure 3. Stress-strain curves for Bccz specimens.

In mechanical failure of BCCZ trabecular structures, the initial collapse typically originates from buckling in the vertical struts, initiating on one side of the specimen approximately at mid-height. This leads to the subsequent failure of a trabecular plane, inclined at 45 degrees to the Z-axis, evident as a marked decrease in resistance on stress-strain curves. Post initial failure, the specimen exhibits

increased resistance, attributed to the stabilization of adjacent cell planes compressed together. The second failure phase, occurring at lower loads than the first, varies with cell size but appears independent of the solid volume fraction. This behavior is witnessed by Figure 4 while broken post-test specimens are reported in Figure 5.

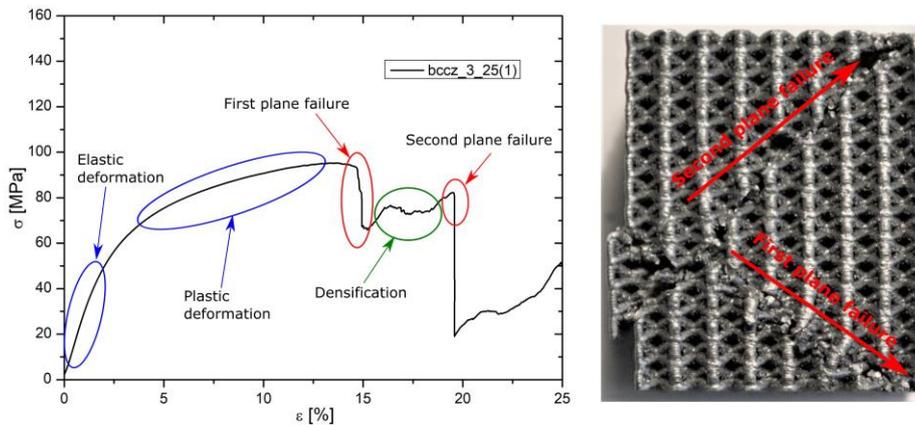


Figure 4. Phases of the compression test on Bccz-3-25-1 specimen.

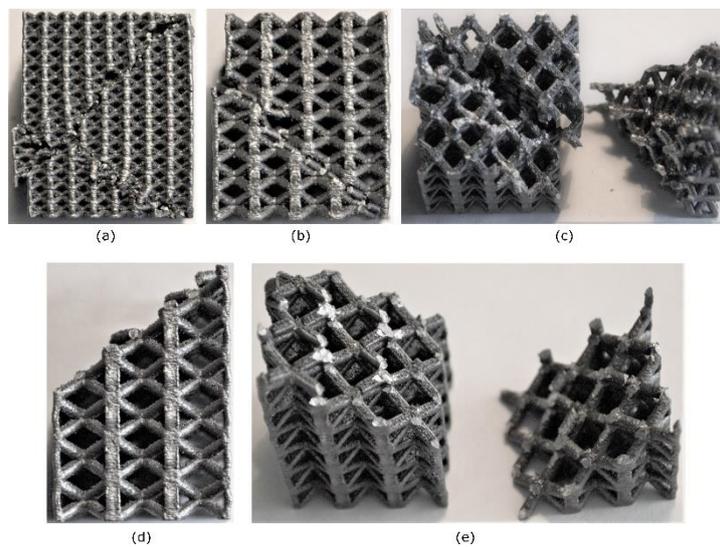


Figure 5. Tested specimens in different condition: a) Bccz-3-25-1 after the failure of the second plane, b) Bccz-5-30-2 after the failure of the first plane, c) Bccz-5-35-1 continuing the test after the failure of the first plane, d) Bccz-7-25-1 after the failure of the first plane and e) Bccz-7-30-2 after the failure of the first plane. Tested specimens: a) Bccz-3-30-3, b) Bccz-5-25-2 and c) Bccz-7-35-1.

3.2. Rhombic Dodecahedron Specimens

Figure 6 delineates the stress-strain relationships for specimens with Rhombic dodecahedron cell structures. These Rhombic cells, across varying sizes, demonstrate reduced deformation prior to reaching the point of failure compared to BCCZ, with failure occurring at approximately 7% strain, in contrast to the 13% observed before. Comparative analysis of these curves reveals that an increase in solid volume fraction is associated with linear improved mechanical performance; also a similar enhancement in mechanical properties is observed with the decreasing of cell size from 7 to 3 mm, albeit in a less linear fashion. In terms of trends for maximum stress (σ_{max}), yield stress (σ_{02}), and Young's modulus (E), the results are consistent with those observed in Bccz and Octet-truss specimens. However, the absolute values are lower for the Rhombic dodecahedron specimens, attributable to their distinct failure mechanism, which is bending-dominated rather than stretching-dominated (bccz).

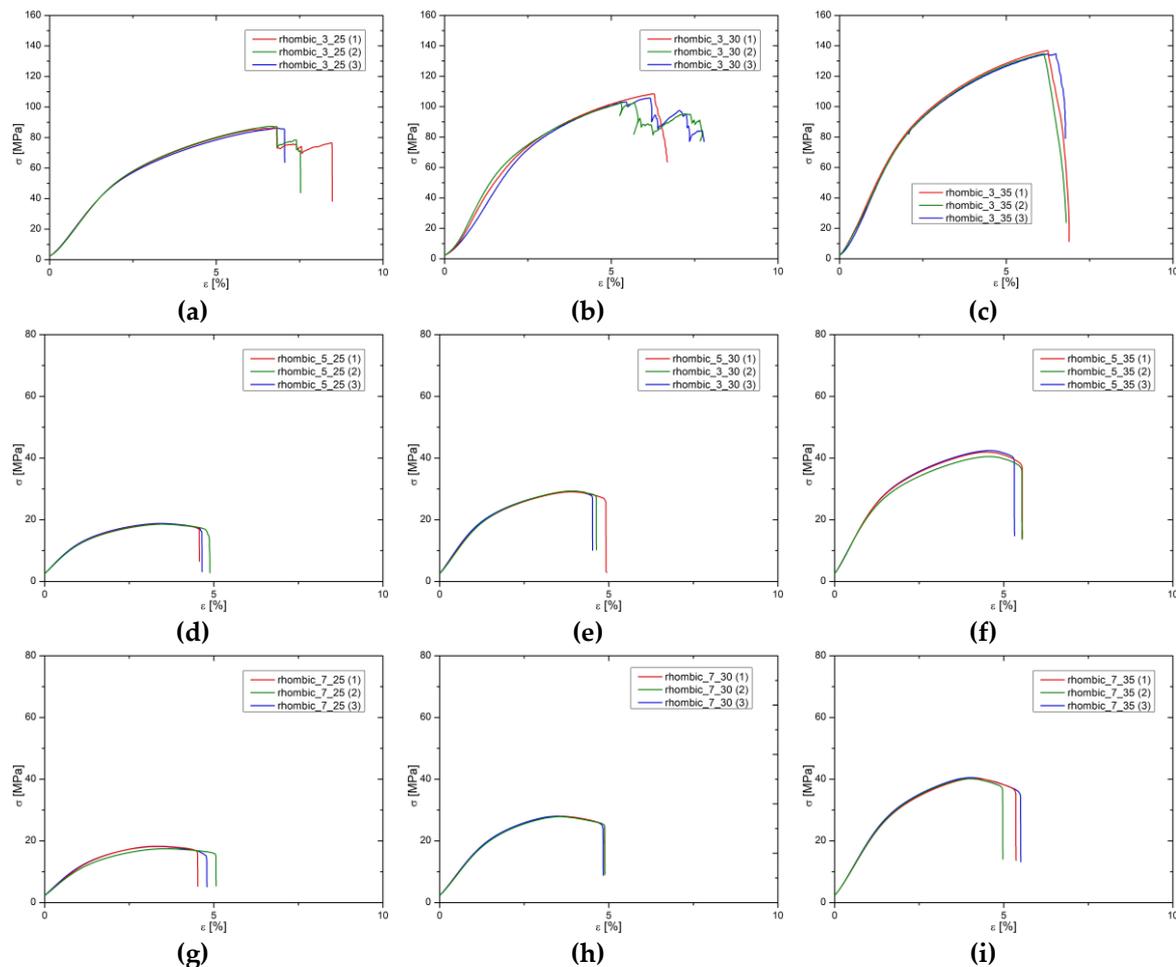


Figure 6. Stress-strain curves for Rhombic dodecahedron specimens.

The specimens featuring rhombic dodecahedron cell geometry exhibit a distinct failure mechanism compared to the Bccz specimens. In Rhombic specimens, failure consistently results from strut fractures at the nodes. This occurs as compression forces concentrate stresses where struts converge. The oblique struts gradually draw closer until they fracture at the node. Failure manifests through the collapse of a plane oriented at a 45-degree angle relative to the Z-axis. Notably, this collapse initiates from specimen's corner, contrasting with the Bccz specimens where failure typically originates from one side. Figure 7 illustrates this phenomenon, displaying several Rhombic specimens at different stages of breakage during the compression tests, highlighting the variation in cell size.

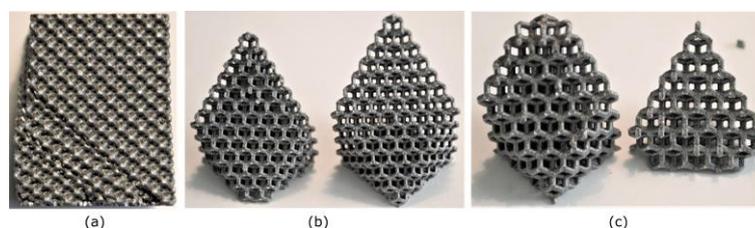


Figure 7. Tested specimens in different conditions: a) Rhom-3-25-1 after the failure of the second plane, b) Rhom-5-25-1 after the failure of the first plane and c) Rhom-7-30-2 after the failure of the first plane.

3.3. Octet-Truss Specimens

In the experimental evaluation of Octet truss structures, the 3mm cell size specimens distinctly exhibited superior mechanical properties, characterized by the highest stresses at rupture, yield

stress, and a Young's modulus, as shown in Figure 8. These graphs underscores also another time the positive impact of augmentation of relative density on the enhancement of mechanical properties in these structures. Comparatively, for what concerns cell size 5mm and 7mm cell sizes demonstrated lower mechanical stress-strain responses compared to 3mm cells. Their behavior of bigger cells resembles to be closer of a traditional truss structure, in contrast to the 3mm specimens which manifested characteristics akin to a porous solid. Notably, the strain levels for all specimens remained consistently below 10%, further distinguishing their mechanical response of bending trusses compared to stretching ones.

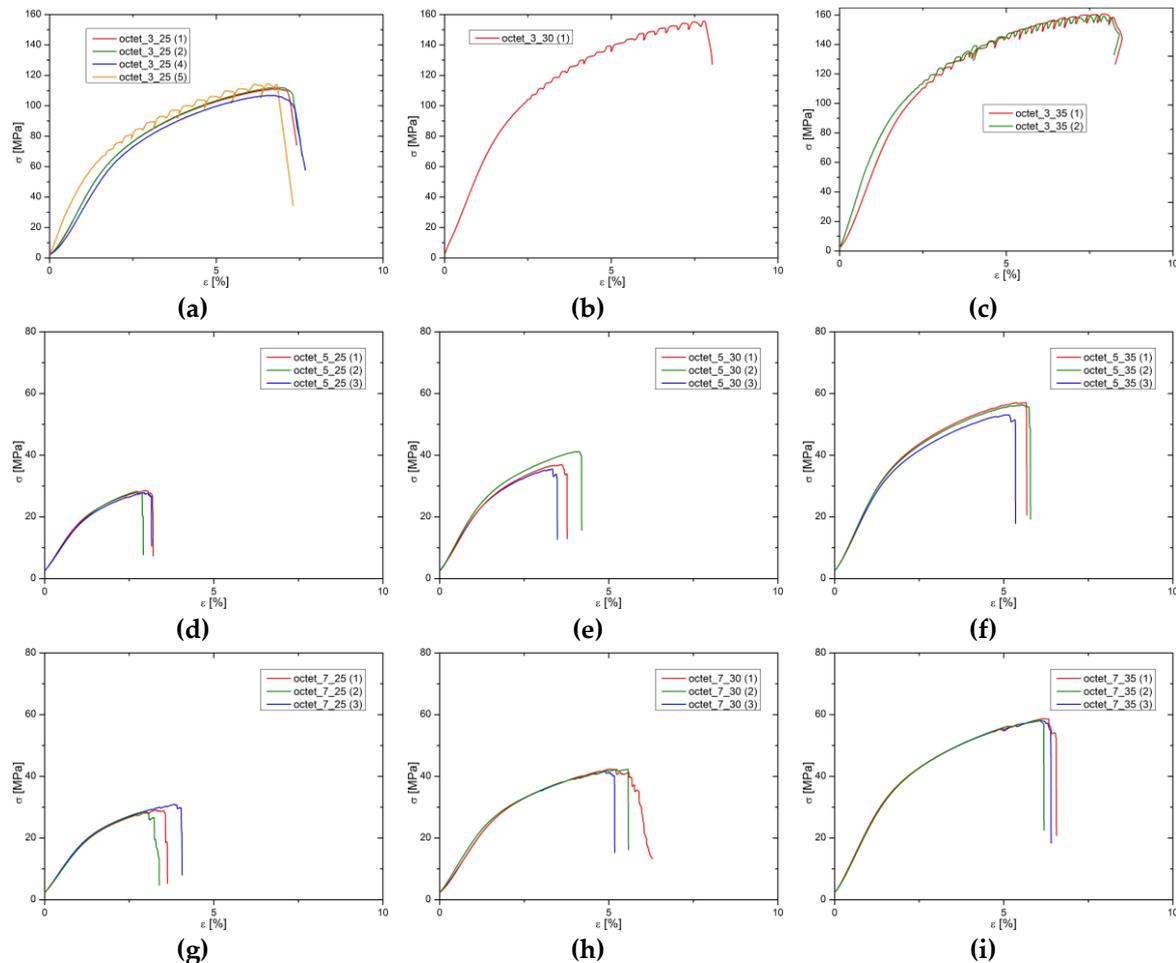


Figure 8. Stress-strain curves for Octet-truss specimens.

The mechanical response at the point of fracture for structures with Octet-truss cell geometry closely mirrors that of Rhombic dodecahedron cells, albeit under higher load conditions. Similar to Rhombic cells, failure in Octet-truss structures predominantly results from strut fractures at the nodes, as noted in reference [38]. This failure pattern arises from stress concentrations at the nodes under compression loads, leading to the gradual convergence and eventual breakage of oblique struts at these points. This observation aligns with the behavior noted in Rhombic cells, particularly in 3mm octet cells, which demonstrate a distinct response compared to their 5mm and 7mm counterparts. Post the initial plane failure, adjacent planes start to fail, resulting in the densification of the specimen. The collapse of these structures tends to be abrupt and noisy, occasionally leading to a violent separation of the parts.

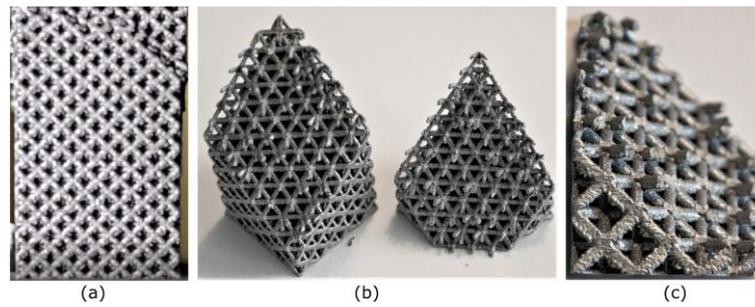


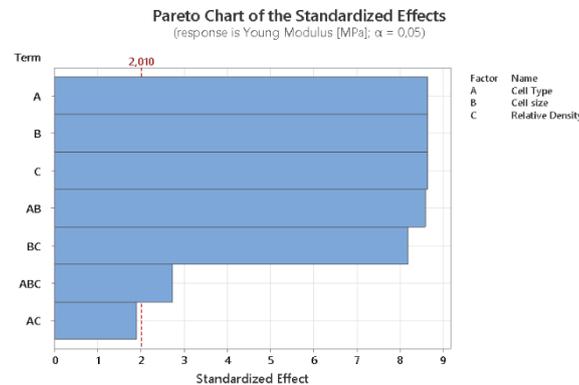
Figure 9. Tested specimens: a) Oct-3-35-1, b) Oct-5-30-2 and c) Oct-7-25-3.

4. Discussion

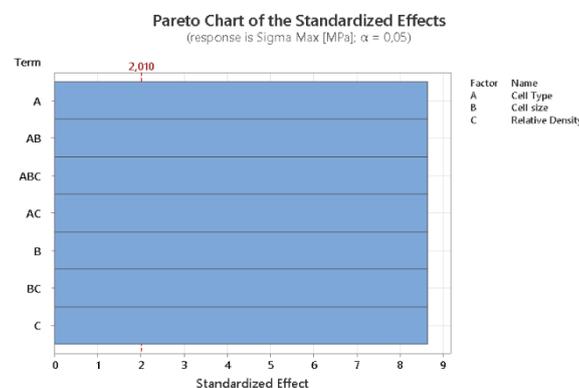
In the following section, an analysis and comparison of the data presented in the Results section will be conducted. This analysis aims to elucidate the impact of design parameters, thereby yielding novel insights into their influence.

4.1. DOE Analysis

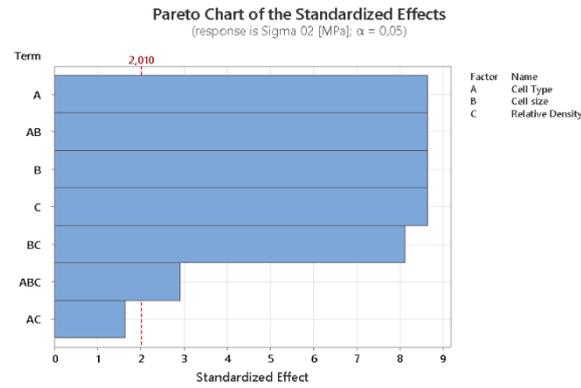
Observing the DOE Pareto analysis' results reported into Figure 10, it is possible to elaborate some inferences. Firstly, the Pareto chart, encompassing various responses such as Young's modulus, maximum stress, and yield stress, indicates that all the DOE design parameters chosen significantly influence these metrics. Cell size is the predominant parameter on both Young Modulus and Yield Stress while Cell Type influences mostly maximum stress. Relative density plays the less influential role in maximum stress while is the second most important parameter in Young Modulus and Yield stress.



(a)



(b)

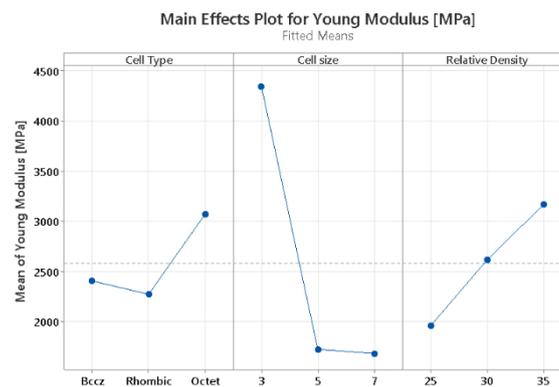


(c)

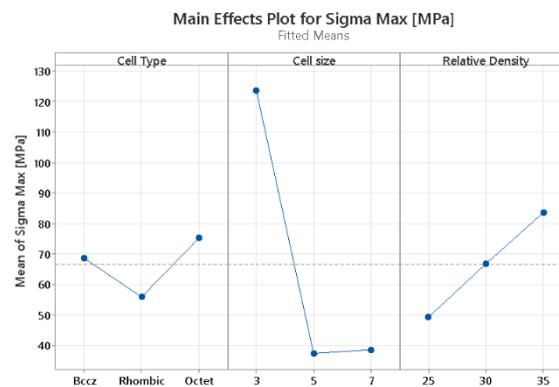
Figure 10. Pareto Chart Analysis.

Furthermore, an analysis of the main effects, evidenced in Figure 11 reveals that the octet lattice structure outperforms the other two cell types across all metrics selected, likewise due to its shorter beam length. In contrast, the rhombic lattice, primarily influenced by bending behavior with slender beams, exhibits inferior performance in maximum stress and Young's modulus compared to both BCCZ and octet structures. However, it surpasses the BCCZ structure in terms of yield stress.

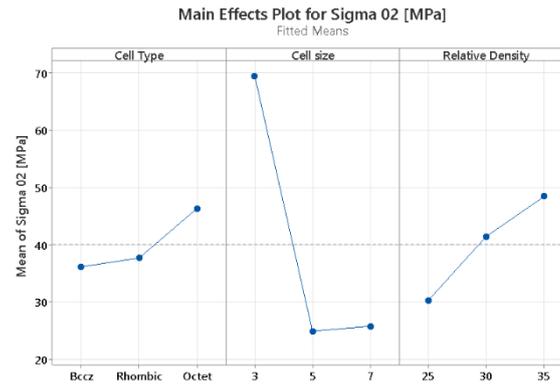
Additionally, an increase in relative density enhances almost linearly all the evaluated metrics.



(a)



(b)

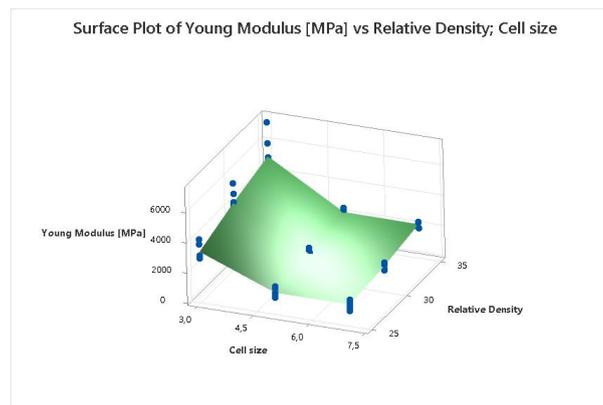


(c)

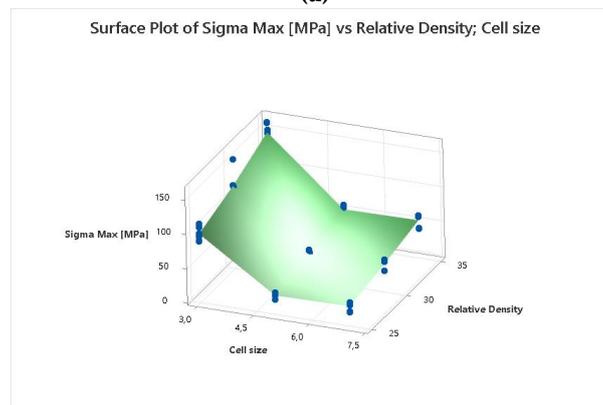
Figure 9. Main Effect Analysis.

Surface response behaviors, detailed in Figure 12, exhibit same non-linear characteristics, reported in the Main Effect

Analysis helping in the complexity to identify optimal configurations. The residual plots, reported in Appendix B, demonstrate a satisfactory normal distribution, indicating errors distributed randomly in order of tests and fits, which suggests minimal systematic errors across all metrics.



(a)



(b)

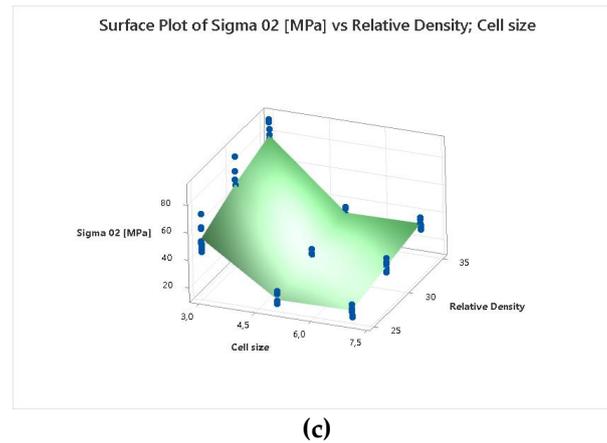


Figure 10. Surface Plot.

4.2. Comparison with Gibson-Ashby Model

The experimental data obtained for all the three cell geometries are compared with the Gibson-Ashby model for the foam with open cells. This model permits to calculate maximum strength and Young modulus of the trabecular specimens, using the density and the mechanical values of the foam and of the correspondent dense material. According to the Gibson-Ashby model, mechanical characteristics of the foam are reported as relative values with respect to the quantities of the equivalent bulk solid. The relative density, the compressive modulus and the compressive strength are calculated respectively using the equations:

$$\rho = \rho_f / \rho_s \quad (1)$$

$$\frac{E_f}{E_s} = C_1 \left(\frac{\rho_f}{\rho_s} \right)^2 \quad (2)$$

$$\frac{\sigma_f}{\sigma_s} = C_2 \left(\frac{\rho_f}{\rho_s} \right)^{1.5} \quad (3)$$

where ρ is the relative density, ρ_f is the density of the foam, ρ_s of the solid, E_f is the Young modulus of the foam, E_s is the Young modulus of the solid, σ_f is the compressive strength of the foam, σ_s is the compressive strength of the solid, C_1 and C_2 are specific constants that include all the geometric features of proportionality.

The mechanical values and properties of the fully dense AlSi10Mg alloy components were taken from EOS Datasheet [39]: the value of compressive σ_{\max} is equal to 392 MPa, E is 72.4 GPa and ρ is 2.67 kg/dm³.

Figure 12 provides a comparative analysis between experimental outcomes and predictions made by the Gibson-Ashby (GA) model for Bccz, Octet, and Rhombic specimens. In Figure 12a, the maximum stress (σ_{\max}) results are depicted, where the GA model's predictions are represented by lines, and the experimental data by points. It is observed that for specimens with lower density, the GA model aligns well with the experimental findings. However, as the density increases, the model's predictive accuracy diminishes. This deviation could be attributed to design and manufacturing factors. From a design perspective, the GA model is primarily tailored for low-density foam materials, leading to challenges in accurately predicting the behavior of high-density trabecular structures. On the manufacturing front, the discrepancy is particularly notable in specimens with 3 mm cell size. Computed Tomography (CT) analysis of these specimens revealed an effective dimensional increase, likely caused by the coalescence of surrounding powder during the melting phase. Figure 12b focuses on the results regarding Young's modulus (E), with the GA model represented by lines and experimental data by points. The correlation between the model and experimental data is less pronounced for Young's modulus. The correspondence is mainly observed

for low-density specimens but diverges rapidly, highlighting the distinct fracture behaviors of trabecular structures compared to foam. This disparity is especially apparent in Bccz cells, which, due to the failure of vertical struts, exhibit significant deformation before breaking and an elastic modulus lower than predicted by the model. For Octet and Rhombic cells, while this trend is observed at high densities, it is less pronounced. Their fracture mechanisms are more akin to those of foam, characterized by a brittle collapse of the cells.

Figure 13 focuses on the results regarding Young's modulus (E), with the GA model represented by lines and experimental data by points. The correlation between the model and experimental data is less pronounced. The correspondence is mainly observed for low-density specimens but diverges rapidly, highlighting the distinct fracture behaviors of trabecular structures compared to classical foam. This disparity is especially apparent in Bccz cells, which, due to the failure of vertical struts, exhibit significant deformation before breaking and an elastic modulus lower than predicted by the model. For Octet and Rhombic cells, while this trend is observed at high densities, it is less pronounced. Their fracture mechanisms are more akin to those of foam, characterized by a brittle collapse of the cells. Given these observations, it becomes evident that there is a need for a new, enhanced model that can more accurately predict the mechanical behavior of these complex structures, especially at higher densities. This scenario opens new research possibilities, inviting further investigation and model development to better understand and predict the behavior of trabecular structures under various conditions.

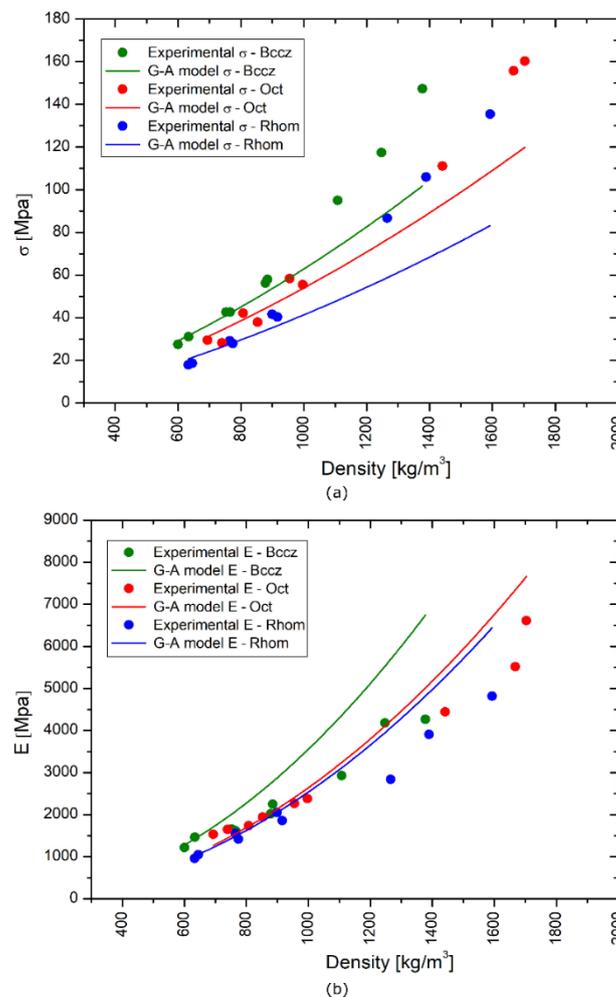


Figure 13. Comparison between experimental data for Bccz, Octet and Rhombic cells in AlSi10Mg with the values obtained from Gibson-Ashby model for the same cells in terms of a) σ_{\max} and b) E .

5. Conclusions

This research effectively demonstrates the viability of AlSi10Mg lattice trusses under compression, particularly when fabricated via additive manufacturing methods. Experimental analysis has shown these trusses possess considerable compressive strength, making them suitable for scenarios demanding mechanical robustness and long-term durability. The adaptability of lattice structures in terms of customization also contributes to enhanced functionality and safety in aircraft design.

The discussion of results highlights the significant impact of design parameters on the mechanical characteristics of lattice trusses. Notable variations in truss structures were observed, categorizing them into two primary groups based on bending and stretching behaviors. Additionally, it was discovered that relative density is a key factor that linearly amplifies mechanical properties. Disparities in mechanical performance and failure modes between denser cell structures (like 3mm cells) and larger cells (5 and 7mm) were also found. Furthermore, when comparing findings with the Gibson-Ashby model, a strong correlation for maximum stress (σ_{max}) in lower-density structures was noted. However, the correlation for Young's modulus was less pronounced, suggesting distinct fracture behaviors in trabecular structures compared to stochastic foams.

This study aligns with prior research in the field and introduces new insights, particularly through the application of a Design of Experiments (DOE) strategy. It is posited that incorporating lattice structures in aircraft design, coupled with advancements in additive manufacturing, can significantly reduce fuel consumption and operating costs, leading to more sustainable and economical transportation solutions. Future research should focus on delving deeper into the observed variability of mechanical properties and refining manufacturing techniques to optimize the efficacy of lattice trusses in applications like anti-icing systems and others where high strength-to-weight ratios are critical.

Author Contributions: Conceptualization, C.G.F and S.V.; methodology, C.G.F and S.V.; software, S.V.; validation, C.G.F and S.V.; formal analysis, C.G.F.; investigation, C.G.F and S.V.; resources, C.G.F.; data curation, C.G.F and S.V.; writing—original draft preparation, C.G.F and S.V.; writing—review and editing, C.G.F and S.V.; visualization, C.G.F.; supervision, P.M.; project administration, P.M.; funding acquisition, P.M. All authors have read and agreed to the published version of the manuscript.

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Conflicts of Interest: The authors declare no conflicts of interest.

Appendix A. Elementary Unit Cells Drawing

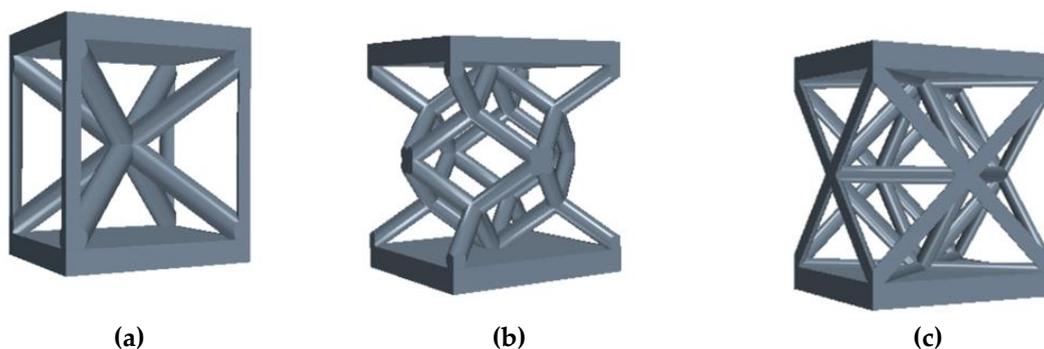


Figure 11. Elementary Unit Cells Drawing: (a) Bccz model; (b) Rhombic model; (c) Octet model.

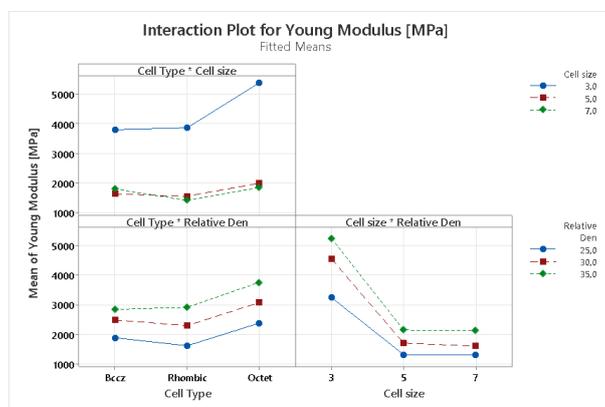
Appendix B. Experimental Collected Results

Table 3. Experimental results of compression tests.

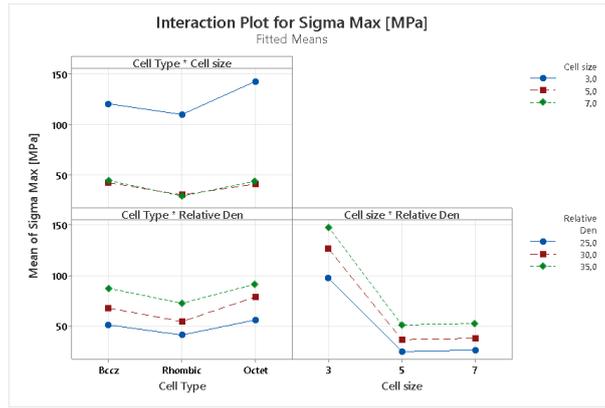
Cell Type	Cell size	Relative Density	Sigma Max [MPa]	Sigma 02 [MPa]	Young Modulus [MPa]
Bccz	3	25	95	48	2954
Bccz	3	30	116	59	3927
Bccz	3	35	149	77	4315
Bccz	5	25	25	15	1322
Bccz	5	30	42	25	1579
Bccz	5	35	57	31	2011
Bccz	7	25	32	18	1491
Bccz	7	30	44	24	1822
Bccz	7	35	59	32	2273
Rhombic	3	25	86	50	2847
Rhombic	3	30	108	72	4009
Rhombic	3	35	137	76	4840
Rhombic	5	25	19	15	1046
Rhombic	5	30	29	21	1522
Rhombic	5	35	42	30	2059
Rhombic	7	25	18	15	955
Rhombic	7	30	28	22	1403
Rhombic	7	35	40	30	1845
Octet	3	25	111	61	4070
Octet	3	30	156	89	5520
Octet	3	35	161	91	5902
Octet	5	25	28	24	1610
Octet	5	30	37	27	1906
Octet	5	35	57	36	2426
Octet	7	25	29	23	1502
Octet	7	30	42	29	1732
Octet	7	35	59	36	2289
Bccz	3	25	93	44	2837
Bccz	3	30	118	55	3804
Bccz	3	35	149	70	4264
Bccz	5	25	28	17	1224
Bccz	5	30	43	26	1565
Bccz	5	35	56	30	2020
Bccz	7	25	30	20	1282
Bccz	7	30	44	25	1826
Bccz	7	35	57	28	2251
Rhombic	3	25	87	51	2823
Rhombic	3	30	104	68	4254
Rhombic	3	35	135	80	4641
Rhombic	5	25	19	15	1001
Rhombic	5	30	29	22	1526
Rhombic	5	35	40	29	2006
Rhombic	7	25	17	14	908
Rhombic	7	30	28	22	1405
Rhombic	7	35	40	31	1840
Octet	3	25	112	62	4057
Octet	3	30	*	*	*

Octet	3	35	160	84	7321
Octet	5	25	28	23	1599
Octet	5	30	41	30	2011
Octet	5	35	56	36	2396
Octet	7	25	28	23	1507
Octet	7	30	42	28	1787
Octet	7	35	58	36	2240
Bccz	3	25	97	47	3000
Bccz	3	30	118	61	4821
Bccz	3	35	144	71	4224
Bccz	5	25	27	16	1233
Bccz	5	30	43	25	1707
Bccz	5	35	56	30	2018
Bccz	7	25	32	18	1626
Bccz	7	30	41	32	1337
Bccz	7	35	58	33	2219
Rhombic	3	25	86	52	2853
Rhombic	3	30	106	78	3464
Rhombic	3	35	135	89	4973
Rhombic	5	25	19	15	1096
Rhombic	5	30	29	22	1610
Rhombic	5	35	42	30	2090
Rhombic	7	25	18	15	1008
Rhombic	7	30	28	23	1440
Rhombic	7	35	41	31	1899
Octet	3	25	107	72	3747
Octet	3	30	*	*	*
Octet	3	35	*	*	*
Octet	5	25	29	22	1741
Octet	5	30	36	27	1922
Octet	5	35	53	35	2321
Octet	7	25	31	23	1595
Octet	7	30	42	29	1691
Octet	7	35	58	37	2260

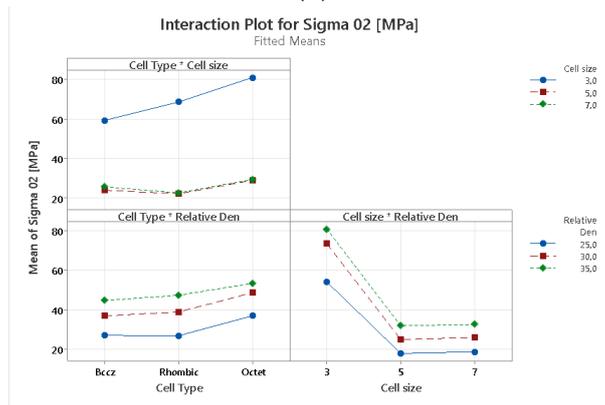
Appendix C. Interaction Plot and Residual Plot



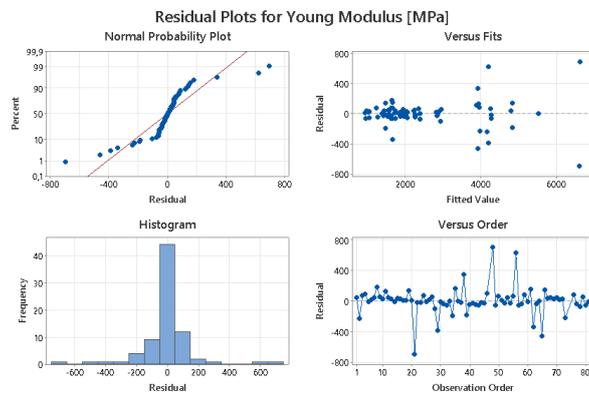
(a)



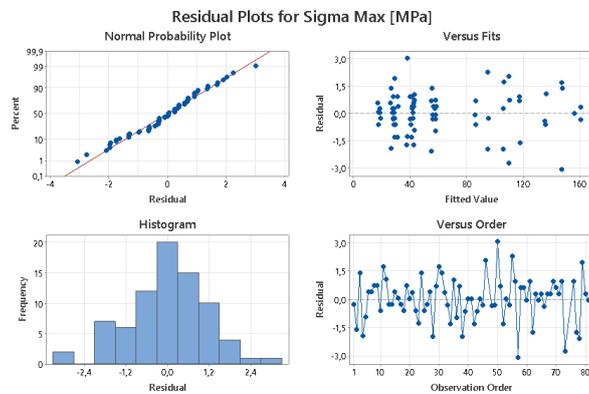
(b)



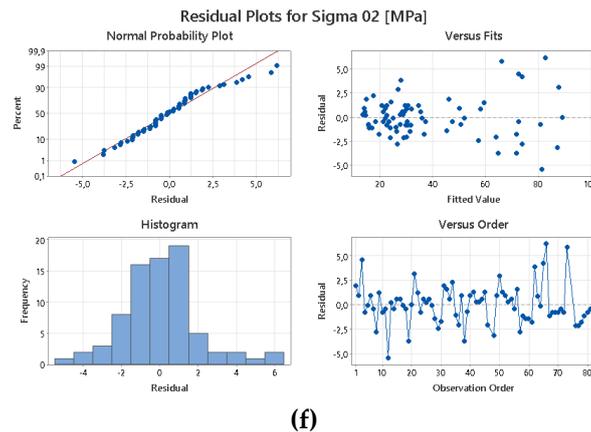
(c)



(d)



(e)



Appendix D. Model Summary Doe

Response	R-sq	R-sq(adj)
Young Modulus [E]	98,24%	97,24%
Sigma O2	99,12%	98,62%
Sigma Max	99,93%	99,88%

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