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## Article

# Effects of Drilling Parameters and Mud Types on Wear Factors and Mechanisms of SM2535 Casings

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**Abstract:** This work aims to explore the impact of side loads, drill-pipe tool-joint (DP-TJ) speed (RPM), and mud type on the austenitic stainless steel SM2535-110 casing wear characteristics. Actual field drill pipe tool joints, casings, and drilling muds are used in this study. The results of the study show that under both types of lubrication, the wear volume increased with radial load and DP-TJ speed. SM2535-110 casing specimens tested under oil-based mud (OBM) lubrication had higher casing wear volumes than those obtained under water-based mud (WBM) lubrication. This unexpected behavior is mainly due to the increase in the surface hardness of the casing specimens tested under WBM. The results also show that the specific wear factor (K) values of specimens tested under WBM are in general 2 to 4 times higher than those obtained under OBM. While K values under WBM increase with both the side load and RPM, those under OBM show a sharp decrease with RPM. This behavior under OBM is due to this lubricant's higher viscosity and the change of lubrication regime from thin film to thick film lubrication at higher RPM. Scanning electron microscopy (SEM) and the digital microscopic imaging (DMI) of SM235-110 casing specimens show that an aggressive combination of adhesive, abrasive, and plastic deformation was observed under WBM, while the dominant wear mechanism under OBM is abrasive wear.

**Keywords:** friction; casing wear; wear factor; wear mechanisms; SM2535 steel; oil based mud; water based mud

## 1. Introduction

Casing wear is the process of a casing's wall gradually losing thickness because of the drill string's and casing's relative motion [1]. Conditions like the downhole forces, the total period the drill string has been in contact with the casing, the drill string speed, and the materials employed all affect how much casing wear occurs. This process is intricate and incorporates unpredictable abrasive, adhesive, and corrosive wear processes [2]. The main drilling characteristics that significantly impact casing wear at a given casing grade include rotary speed, rate of penetration (ROP), mud density, mud type, and bottom hole assembly [3].

Different grades of steel have been used for casing and the selection of the material is influenced by the operating conditions and drilling environment. From well testing through completion, high-pressure high-temperature wells present several obstacles to material design.

The amount of casing wear of several steel grades has been investigated in several studies, but its accurate estimate has proven to be quite difficult. This is because casing wear is a complicated process that depends on several parameters, including mud type, percentage of abrasives in the mud, tool joint hard-facing, tool joint diameter, drill string speed, contact pressure, and more [3–8].

Several experimental and numerical studies were performed to develop relationships between drilling parameters and casing wear factors and wear mechanisms. Zhang et al. [7] performed an experimental investigation of several casing steel grades to understand the effect of different drilling parameters on their wear behavior. They showed that the amount of wear increased with both side load and RPM. Moreover, they found that casing weight loss is inversely proportional to the casing steel grade. Huimei and Yishan [3] conducted a study to determine the effects of drilling factors on the casing wear and concluded that drill pipe spinning speed and drill pipe penetration rate are the most important factors in ultra-deep wells. Doering et al. [9] demonstrated that the new small-scale casing wear tester can produce wear mechanisms equivalent to both historical data produced by the DEA-42 test and from commercial down-hole service. Yu et al. [10] developed an equation that relates the maximum wear depth to the residual casing strength after accumulative wear. Chen et al. [11] have examined the impact of drill string length, speed of rotation, direction of travel, and density of the drilling fluid on friction and they established a mathematical model. Yu et al. [12] offered a theoretical and technological framework for the evaluation of casing integrity and casing configuration in directional wells subjected to in situ stress. Osman, et al. [13] in their research focused on the design, manufacture, control, and measurement of critical factors that contribute to casing wear, as well as the evaluation of wear volume. Osman et al. [14] carried out an experimental casing wear test with real drill pipe joints (DP-TJ) and P110 steel casing under various operating conditions and test environments in order to comprehend and evaluate casing wear during drilling. They found that the amount of wear increases with the increase of the side loading and DP-TJ spinning speed and both wear volume and wear factor obtained under WBM are about double those obtained under OBM. Similar results were reported by Osman et al. [15] for L-80 casing wear factors under similar drilling parameters. The results on P110 and L-80 revealed that the casing wear factor was lowered as rotating speeds increased for OBM. They attributed this reduction to the possible transition of the lubrication regime from thin to thick film lubrication with the increase in speed [14,15].

Materials with high strength are always needed for the fabrication of tubes, tubing, hangers, and tubular joints in case the well depths are more than 1 Km [16]. These types of oil wells can generate fluids with elevated chloride amount and high partial pressures of CO<sub>2</sub>, and H<sub>2</sub>S besides the elevated temperatures and pressures. Some reservoirs also contain harmful species like mercury and elemental Sulphur. Low-alloy steels that have high strength are not suitable for these applications, except for casing materials. Owing to its low overall corrosion rates in elevated CO<sub>2</sub> and H<sub>2</sub>S environments, high temperatures, and other environmental stresses, corrosion-resistant alloys (CRAs) are employed in these applications. Nevertheless, depending on the metallurgical and environmental circumstances, they may experience various types of environmentally assisted cracking [17,18].

The SM2535-110 casing material, which was introduced in the middle of the 1980s as an innovative development, has grown to become the standard for tubing and liner applications in harsh environments. Alloy SM2535-110 is a cold-hardened nickel-based alloy designed for corrosion resistance in situations with moderate chloride concentration and severely sour (H<sub>2</sub>S) conditions that need high strength up to 177°C (350°F). When used as downhole tube components, packers, and other subsurface equipment in sour wells with high-pressure and temperature (HPHT) conditions and Acid Gas Injection (AGI) wells, SM2535-110 exhibits greater resistance to the effects of H<sub>2</sub>S compared to stainless steels [19]. However, SM2535-110, being austenitic stainless steel (SS) is expected to have insufficient tribological properties, mainly because of its low carbon content. Moore [20] has found that the square root of the carbon content in steel is a linear relation of wear resistance.

All the above-mentioned studies on casing wear considered mainly martensitic steels such as P110, L80, and N80. Investigations of austenitic stainless-steel casing such as SM2535-110 are non-existent in the open literature. In this paper, a testing facility, developed in an earlier study [13] is modified and utilized to study the effects of three important drilling parameters (side loads, drill string RPM, and drilling fluid type) on casing wear depth, wear volume, and specific wear factors of SM2535-110 steel. The effects of these parameters on casing hardness, temperature and coefficient of friction were also investigated.

## 2. Experimental Procedure and Characterization

### 2.1. Experimental Procedure

A conventional lathe machine has been modified as shown in Figure 1 to carry out casing wear tests on SM2535-110. Various systems were designed and used to measure, control, and record test parameters such as applied side load, friction force, average casing temperature, and wear depth. The applied side load, which was determined by a dynamometer type 9139AA, was maintained, and controlled by a microprocessor and a stepper motor connected to a reduction gearbox. More details of the established casing wear testing setup can be found in our previous studies [13–15]. A 25 mm wide sixty-degree circular section of the casing was machined from a real casing pipe and used as a casing sample.

A waterproof DS18B20-type digital temperature sensor was introduced through a hole in the casing sample at around 5 mm from the interface between the casing and DP-TJ. Testing was performed under OBM and WBM real drilling fluids. Three altered side loadings of a magnitude of 1 KN, 1.2 KN, and 1.4 KN were chosen to result in pressures that are within the reported field conditions of 0.3 to 0.5 MPa. The rotational speeds of 115, 154, and 207 RPM are close to the common drill string speed range of 100 to 120 rpm [5,20]. The characteristics of the muds which were provided by a local oil company were reported in earlier studies [14,15]. Furthermore, a Rheometer 702e anton parr was used to measure OBM and WBM viscosity at 25°C. OBM had an average viscosity of 478 mPa.s at ambient temperature, while WBM had a viscosity of roughly 182 mPa.s.

The crescent's worn volume per unit width (mm<sup>3</sup>/mm) shown in Figure 2, may be estimated by equation (1) if the maximum depth of the wear (w) in mm, casing inner radius (R) in mm, and drill pipe outer radius (r) in mm are known.

$$\text{Wear volume} = 12 \left( \beta r^2 + 2\sqrt{Q(Q-R)(Q-S)(Q-r)} - R^2 \right) \quad (1)$$

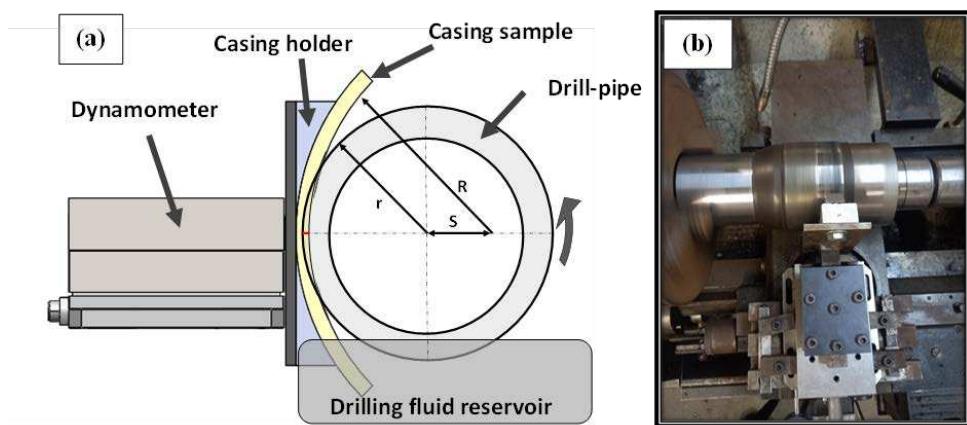
With:

$$Q = \frac{R+r+S}{2}, \quad S = R - (r - w),$$

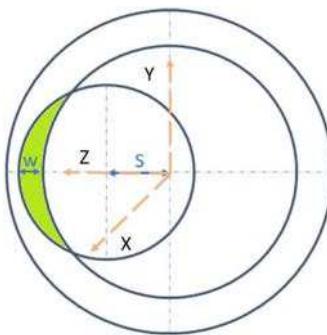
$$\cos \alpha = \operatorname{arctg} \left( \frac{R^2+S^2-r^2}{2RS} \right),$$

$$\beta = \operatorname{arctg} \left( \frac{R \sin \alpha}{R \cos \alpha - S} \right)$$

The maximum wear depth used in equation (1) was monitored and recorded during each of the tests, with a digital indicator that has a resolution and accuracy of  $\pm 0.004$  mm. The maximum wear depth values were then used to calculate the average wear volume.



**Figure 1.** The casing wear test setup where (a) is the 2D drawing of the setup and (b) is a Photo of the setup.



**Figure 2.** Crescent Shape Wear Groove.

The chemical compositions of the SM2535-110 casing steel, the drill pipe and DP-TJ utilized in this are illustrated in Table 1 and the mechanical properties along with dimensions of the casings are given in Table 2. An XD105 TM Pipe drill body grade that has a 100 mm (4 in) outer diameter and XT TM 39 tool-joint connection type with a 124 mm (4.875 in) outer diameter were used [22].

**Table 1.** Elemental compositions of SM2535 casing, drill pipe and DP-TJ [22].

Elements	SM2535-110	Drill Pipe	Counterface (DP-TJ)
Fe	35.5%	96.2%	92.4%
C	0.02%	0.4%	0.9 %
Cr	25.7%	1.4%	3.8%
Ni	32%	0%	0%
Mo	4.1%	0%	0%
Mn	0.7%	0%	0%

**Table 2.** Mechanical properties of SM2535-110 steel casing.

Casing property	unit	values
Ultimate tensile Strength value	(MPa)	≥792
Yield Strength value	(MPa)	758-965
Casing hardness Value	(HRC)	≤33
Outer DP Diameter	(mm) [in]	(245) [9 – 5/8]
Casing Thickness	(mm) [in]	(110) [0.44]

## 2.2. Characterization

The hardness of the as-received casing, drill pipe and DP-TJ was measured using the INNOVA-model 783-D Rockwell hardness machine, with a conical diamond indenter and applied load of 150 Kgf.

The chemical constituents of the tested samples were determined by Plasma optical emission spectrometry (ICP-OES PlasmaQuant® PQ 9000).

To investigate the wear characteristics of the tested specimens, Digital microscopic images (DMI) and scanning electron microscope (SEM) micrographs were captured using high-performance SEM type JSM-6610LV low vacuum with Oxford INCA, Energy Dispersive System (EDS), and Microscope Image Capture System (MICS) that has a magnification of 5 to 300000 times and a resolution of 3.00 nm.

### 3. Results and Discussions

#### 3.1. Elemental Composition and Hardness Evaluation

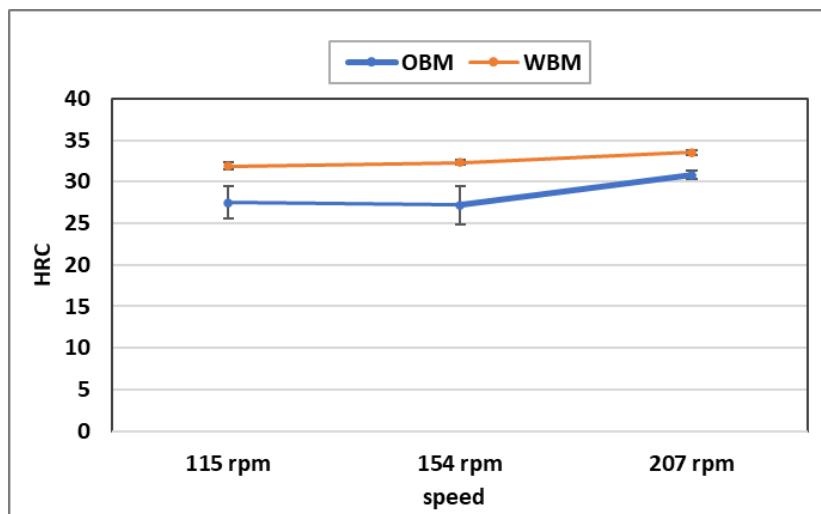
The chemical constituent of the SM2535-110 casing was determined at four distinct locations and the results showed that the tested casing specimens have Carbon, Chromium, Nickel, and Molybdenum contents of 0.02%, 25.7%, 32.0%, and 4.1%, respectively. These findings demonstrate strong concordance with the chemical composition of SM2535-110 casing steel as specified by API SPEC 5CT Standard [22].

The average values of ten hardness measurements of the received casing samples, drill pipe, and DP-TJ are illustrated in Table 3. The SM2535-110 casing sample's measured hardness values agree with those reported in API SPEC standard ( $\leq 33$  HRC).

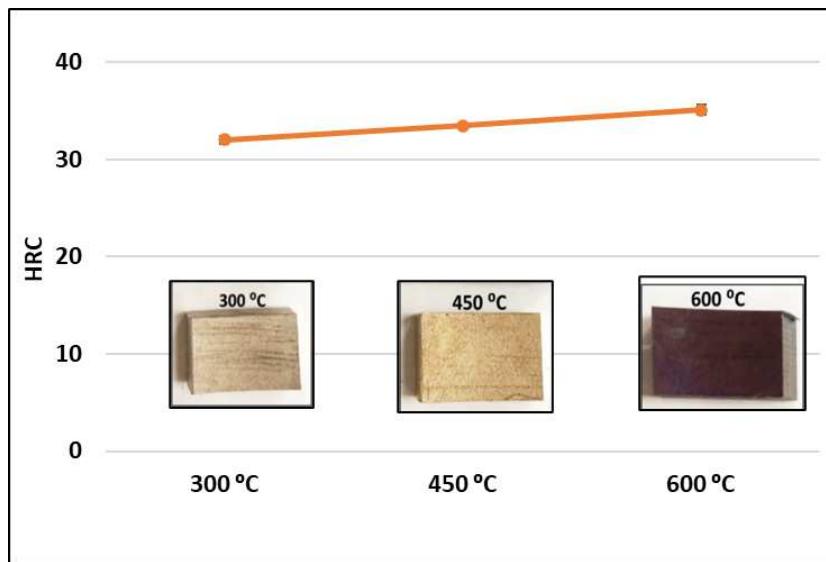
**Table 3.** Average hardness of the three materials and drill pipe.

	SM2535-110	Drill Pipe	Counterface (DP-TJ)
Hardness	$31.73 \pm 1.9$	$26.18 \pm 1.5$	$57.94 \pm 0.8$

The hardness values of SM2535-110 casing specimens tested under WBM and OBM at 1 KN and at 115, 154 and 207 RPM were also measured to investigate the effect of DP spinning speed on this property and the results are demonstrated in Figure 3. It is observed that as the rotational speed increases, the hardness increases, especially under WBM lubrication. This behavior can be directly related to the development of a hard-worked layer because of surface deformation. Moreover, the heating-cooling process that happens at the contact surface during the test may have improved the surface hardness. To evaluate the average contact surface temperature during the test, three specimens of SM2535-110 casing were sectioned and then heated inside a furnace to 300°C, 450°C, and 600°C for 30 min then rapidly cooled in water. The hardness of the sectioned specimens was then measured to study the heat treatment effect on the hardness of the SM2535-110 specimens and the results are shown in Figure 4. It is observed that the higher the heat treatment temperature, the higher the hardness. These results can be used to justify the hardness increase of SM2535-110 casing specimens tested under WBM lubrication which could indicate that the interface temperature during the casing wear test reached a temperature of about 300°C.



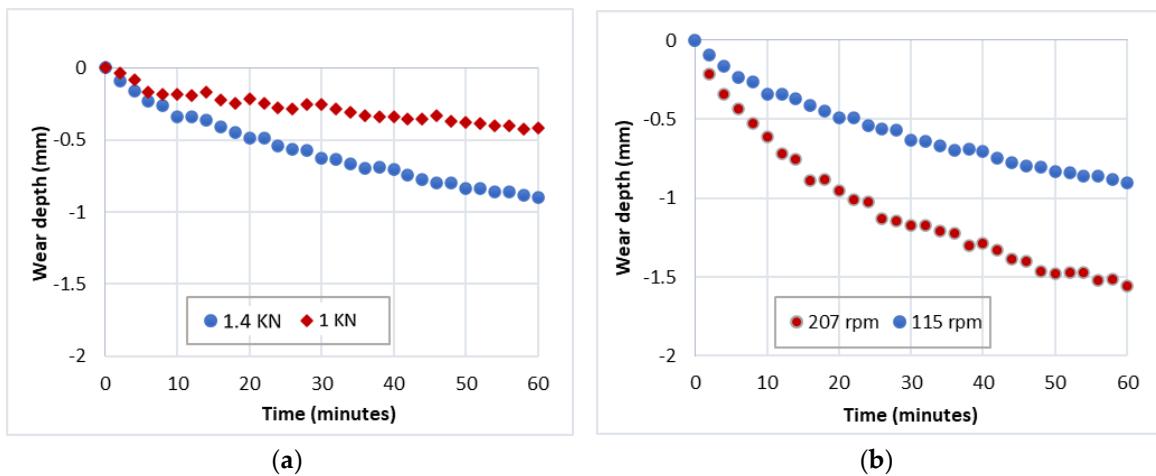
**Figure 3.** The hardness values of SM2535-110 casing tested at 1 KN and three speeds under both WBM and OBM lubrication.



**Figure 4.** The hardness of SM2535 casing after heating the specimens for 30 minutes each at various temperatures followed by rapid cooling in water.

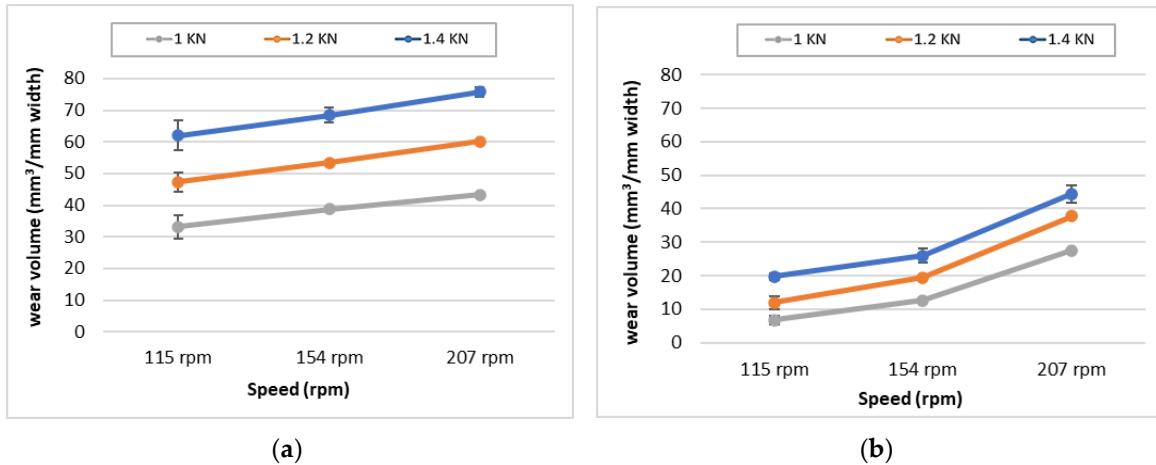
### 3.2. Casing wear depth and volume

The maximum wear depth was periodically obtained every two minutes with the digital indicator and the values were recorded. Figure 5(a) displays the maximum wear depth of the SM2535-110 measured under WBM at a fixed speed and two side loads (1 K and 1.4 KN). On the other hand, Figure 5 (b) displays the maximum wear depth measured under WBM at a fixed load and two rotating speeds. As can be observed, the maximum wear becomes an almost linear function of time after a wear-in period of around 10 minutes. The overall maximum wear depth after 60 minutes increases by more than 100% when the load passes from 1 KN to 1.4 KN at 115 rpm. However, after changing the speed from 115 rpm to 207 rpm under 1.4 KN applied side load, the total maximum wear depth has only increased by 50%.



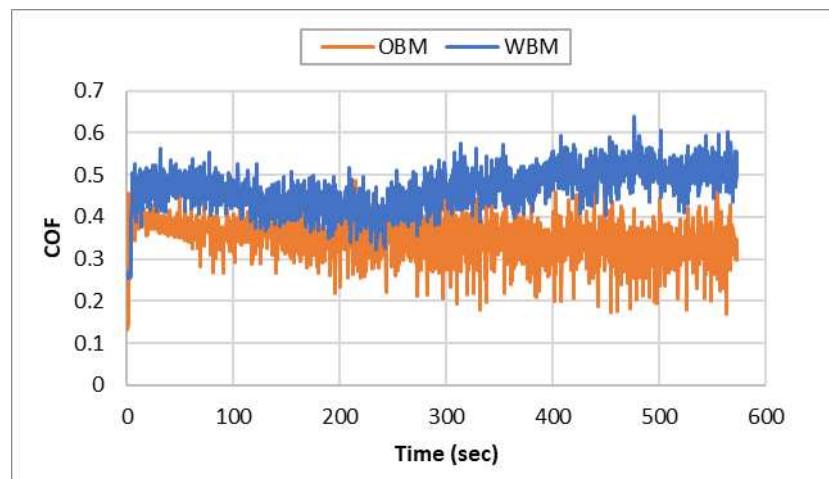
**Figure 5.** The maximum wear depth of the SM2535-110 casing wear measured under WBM at (a) 115 rpm and two loads (b) 1.4 KN and two speeds.

The volume loss of SM2535-110 casing specimens evaluated, using equation 1, at various loads and speeds, for OBM and WBM lubrication, is illustrated in Figure 6. It is observed that irrespective of the type of mud type, the wear volume increases with both the contact load and RPM. It is important to note that the high coefficient of friction of the DP-TJ/SM2535-110 casing specimens under both OBM and WBM caused the SM2535-110 casing sample to exhibit high wear volume.

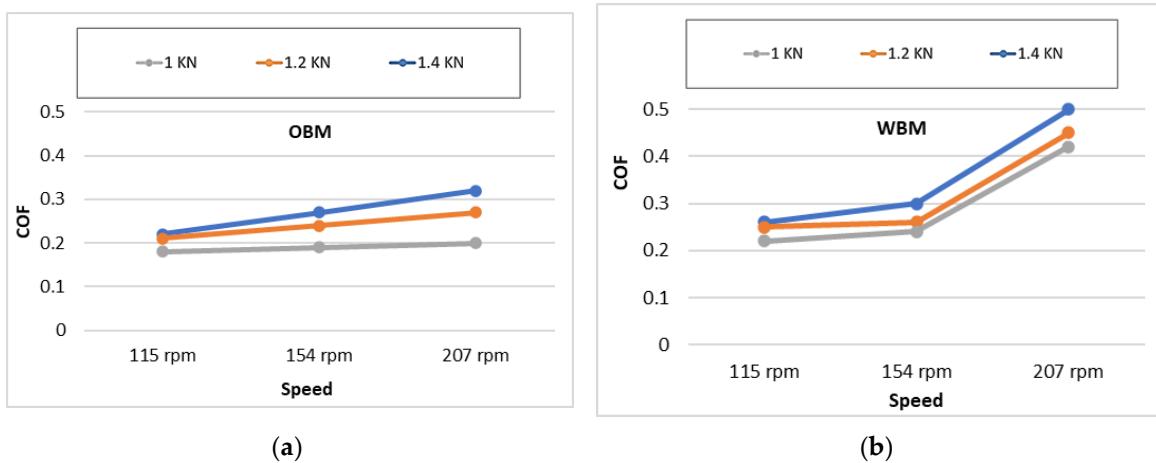


**Figure 6.** The wear volume of SM2535-110 casing specimens tested at different contact loads and three RPM values under a) OBM and b) WBM.

To understand how drilling mud type affects the COF at different operating conditions, friction-wear tests were conducted for all SM2535-110 casing specimens at (1.4 KN, 1.2 KN, 1 KN) and (207 rpm, 154 rpm, 115 rpm) under WBM and OBM. An example of the average values of the COF for the samples tested at 207 rpm and 1.4 KN under WBM and OBM is revealed in Figure 7. It is observed that the COF under WBM (0.42) is higher than COF under OBM (0.33). The average COF of SM2535-110 casings specimens tested at the different contact loads and DP speeds under WBM and OBM are illustrated in Figure 8. The general trend observed under both WBM and OBM is that the COF increases with both the side load and the rotational speed, N. The increase of COF with N can be explained by the increase of the Sommerfeld number with the rotational speed in the thick film hydrodynamic lubrication region since the test configuration is like that of the partial journal-bearing. Based on the McKee brother's analysis,  $\mu N/P$  must be greater than  $1.7 \times 10^{-6}$  (reyn rps/psi) to keep a thick film hydrodynamic lubrication;  $\mu$  this is the lubricant viscosity. In this case,  $\mu N/P$  equals 0.069 (reyn rps/psi) at 1.4 KN and 115 rpm proving that hydrodynamic lubrication is present [23].

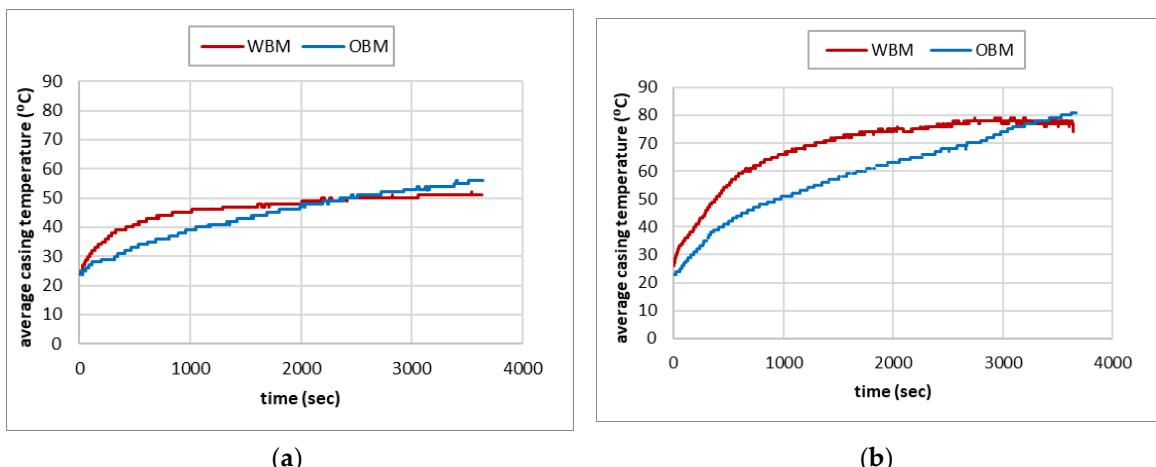


**Figure 7.** COF of the DP-TJ/SM2535-110 casing specimens tested at 154 rpm and 1.4 KN under WBM and OBM.

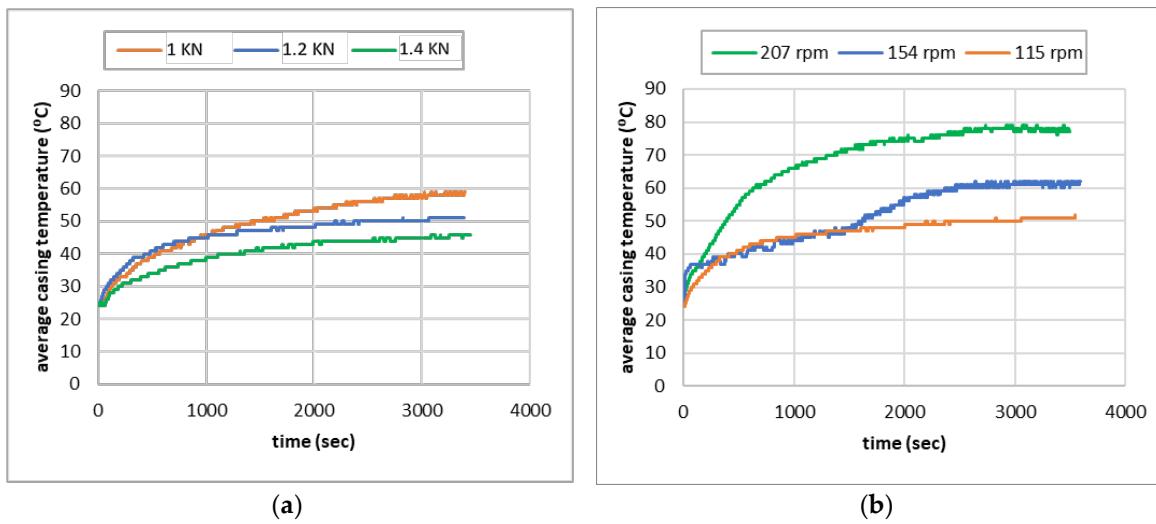


**Figure 8.** COF of SM2535-110 casing specimens at three applied side loadings and spinning speeds under a) OBM and b) WBM.

Furthermore, the average casing temperature profiles recorded during the test performed at 1.2 KN and 115 rpm and 207 rpm under OBM and WBM are presented in Figure 8 (a) and (b), respectively. It is obvious that under WBM, the average temperature of SM2535-110 casing started to rise at a higher rate compared to casing specimens tested under OBM during the first 10 minutes, then kept on increasing at a lower rate until it reached 78°C and almost remained constant. However, under OBM a completely different behaviour was observed since the temperature started to increase almost linearly and it reached an average temperature of 56°C at 115 rpm and 81°C at 207 rpm and kept on increasing. This could be directly related to the difference in the wear mechanism under OBM compared to WBM as it will be discussed in the casing wear mechanism section. Figure 10(a) displays the typical casing temperature profile for SM2535-110 casing specimens tested in a WBM environment at 115 rpm and different contact loads. The average SM2535-110 casing temperature is shown to steadily rise at various rates and the maximum temperature measured after 60 minutes decreases as the side load decreases. On the other hand, the average SM2535-110 casing temperature increases as the DP-TJ speed increases from 115 rpm to 207 rpm at the same radial load of 1.2 KN, as shown in Figure 10(b).



**Figure 9.** SM2535-110 casing specimens' average temperature measured under OBM and WBM during the wear test at 1.2 KN and rotational speed of (a) 115 rpm and (b) 207 rpm.



**Figure 10.** Average temperature of SM2535-110 casing specimens tested under WBM at a) at 115 rpm constant speed; b) at 1 KN constant load.

### 3.3. Specific wear factor of SM2535-110 casing

The casing specific wear factor (K) in  $\text{mm}^3/\text{Nm}$  was estimated, using the following equation.

$$K = V/PL \quad (2)$$

With:

V: volume loss ( $\text{mm}^3$ )

P: Radial load (N)

L: Total crossed distance (mm),

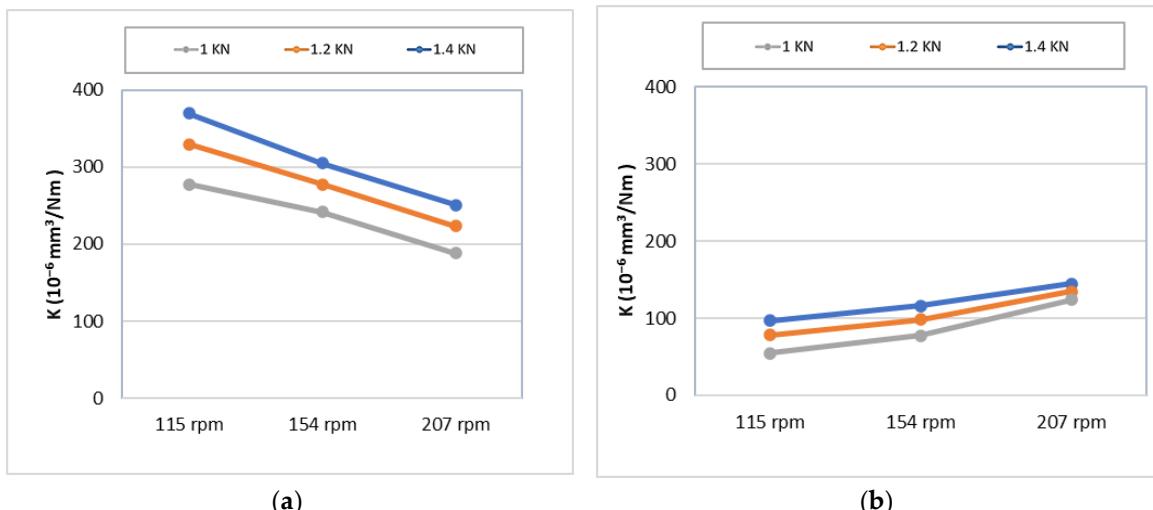
The estimated specific wear factor (K) values for SM2535-110 casing specimens tested under OBM and WBM and different rotational velocities and radial loads, are illustrated in Table 4.

**Table 4.** Specific casing wear factor (K) of SM2535-110 specimens.

Type of mud	Sample number	Speed (rpm)	Side Load (N)	K ( $10^{-6} \text{ mm}^3/\text{Nm}$ )
OBM	S1	115	1 K	277.13
	S2	115	1.2 K	329.53
	S3	115	1.4 K	369.52
	S4	154	1 K	241.66
	S5	154	1.2 K	277.42
	S6	154	1.4 K	304.82
	S7	207	1 K	188.38
	S8	207	1.2 K	223.17
	S9	207	1.4 K	251.06
WBM	S10	115	1 K	54.88

S11	115	1.2 K	78.48
S12	115	1.4 K	106.27
S13	154	1 K	77.33
S14	154	1.2 K	98.12
S15	154	1.4 K	115.81
S16	207	1 K	124.05
S17	207	1.2 K	134.85
S18	207	1.4 K	144.4

The specific wear factors obtained for SM2535-110 casing specimens, tested under the above-mentioned conditions are shown in Figure 11. Under OBM casing specific wear factor varied from  $140.9 \times 10^{-6} \text{ mm}^3/\text{Nm}$  at 115 rpm and 1 KN side load to  $295.26 \times 10^{-6} \text{ mm}^3/\text{Nm}$  at 207 rpm and 1.4 KN. On the other hand, under WBM the SM2535-110 casing-specific wear factor varied from  $60.12 \times 10^{-6} \text{ mm}^3/\text{Nm}$  at 115 rpm and 1 KN to  $142.43 \times 10^{-6} \text{ mm}^3/\text{Nm}$  at 207 rpm and 1 KN. These K values are 10 to 50 times higher than those obtained for martensitic steel casing materials P110 and L80 reported in earlier studies [15,16]. This is mostly because of the minimal carbon content of SM2535-110 compared to P110 and L80. As reported earlier, the wear resistance of steel is a linear function of the square root of the carbon weight per cent [21]. Furthermore, austenitic stainless steels are known to have insufficient tribological properties.

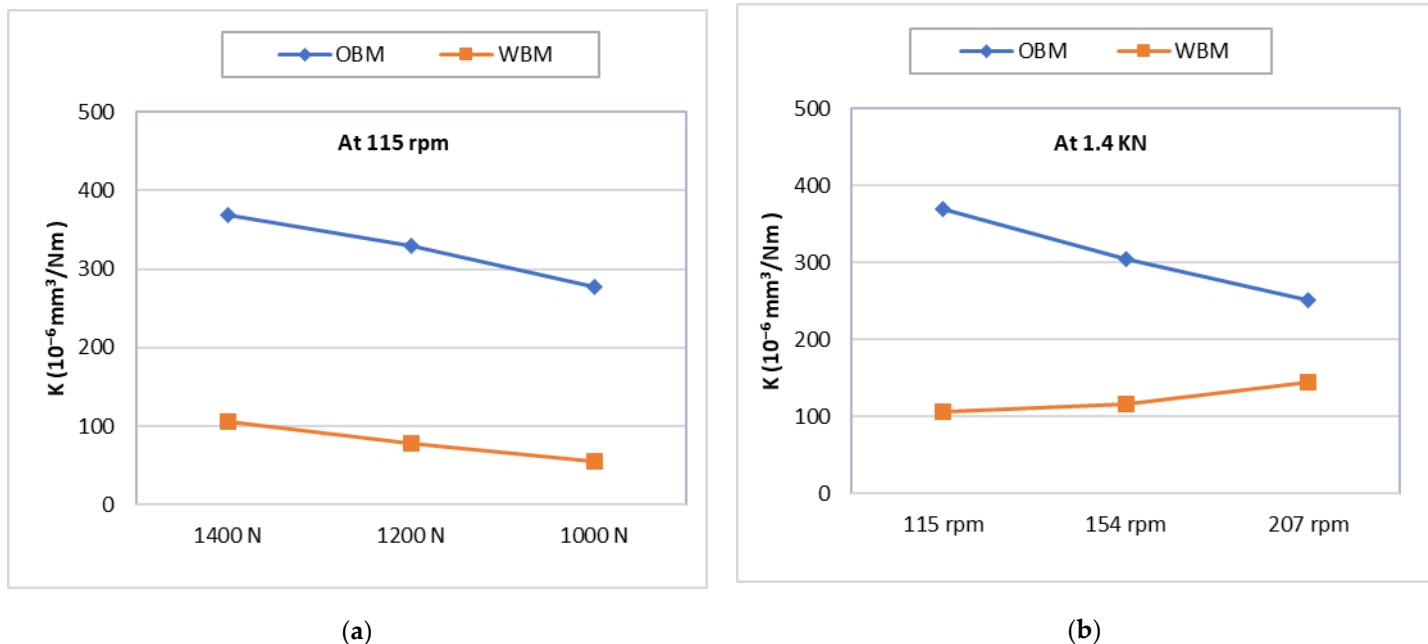


**Figure 11.** Specific wear factor of SM2535-110 casing specimens examined at different loads and speeds under a) OBM and b) WBM.

### 3.4. Drilling Mud Type Effect on Wear Factor

Unexpectedly, the value of the specific wear factor found for SM2535-110 specimens tested at 115 rpm and side loads of 1 KN, 1.2 KN and 1.4 KN, under OBM is almost twice as high as that obtained for WBM at the same speed as illustrated in Figure 12(a). This is mainly due to the increase of the surface hardness of the casing specimens tested under WBM which was subjected to rapid heating and cooling process during the test and the formation of a thin film protective layer since the WBM contains  $\text{Cl}^-$  [7]. Similarly, the K values SM2535-110 casing specimens obtained at 1.4 KN for

the three values of DP-TJ speed, under OBM are higher than those obtained at the same contact load under WBM as shown in Figure 12(b).

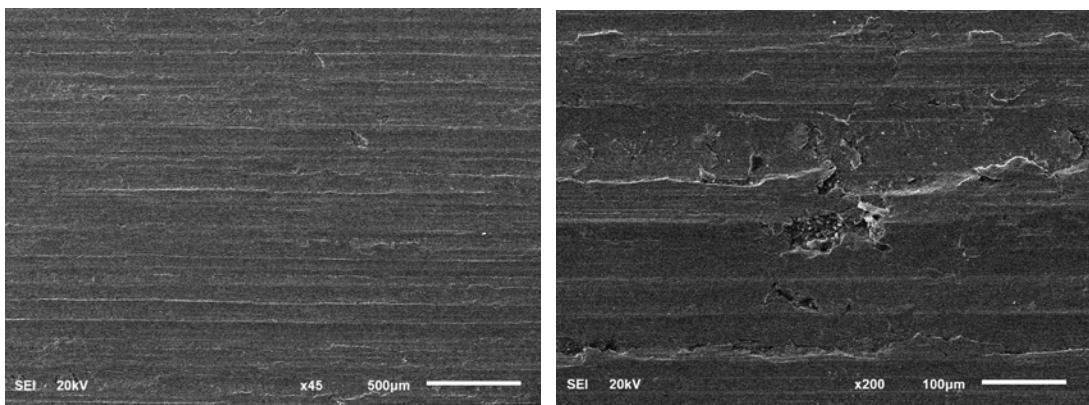


**Figure 12.** Specific wear factor of SM2535-110 casing tested under WBM and OBM at: a) 115 rpm; b) 1.4 KN.

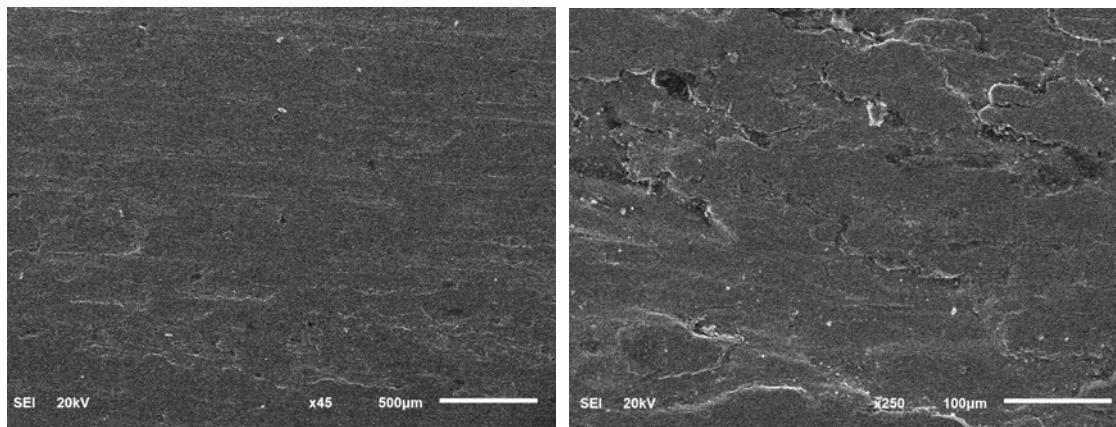
### 3.5. Wear mechanisms of SM2535-110 casing

SEM micrographs of the worn surfaces of SM2535-110 casing specimens, tested under OBM and WBM at 207 rpm and 1.4 KN are presented in Figure 13 and Figure 14, respectively. These micrographs reveal that abrasion is the leading wear mechanism under OBM, while under WBM a combination of adhesion and plastic deformation was observed. It is worth mentioning that under WBM lubrication at different spots, the material was detached from the surface and reattached again as proven by EDS analysis carried out at the worn area, illustrated in Figure 15. It can be that the newly reattached and squeezed flakes have the same composition as the original material. On the other hand, some solid particles originating from the WBM were observed on the surface which is believed to be the cause of the three-body abrasive wear.

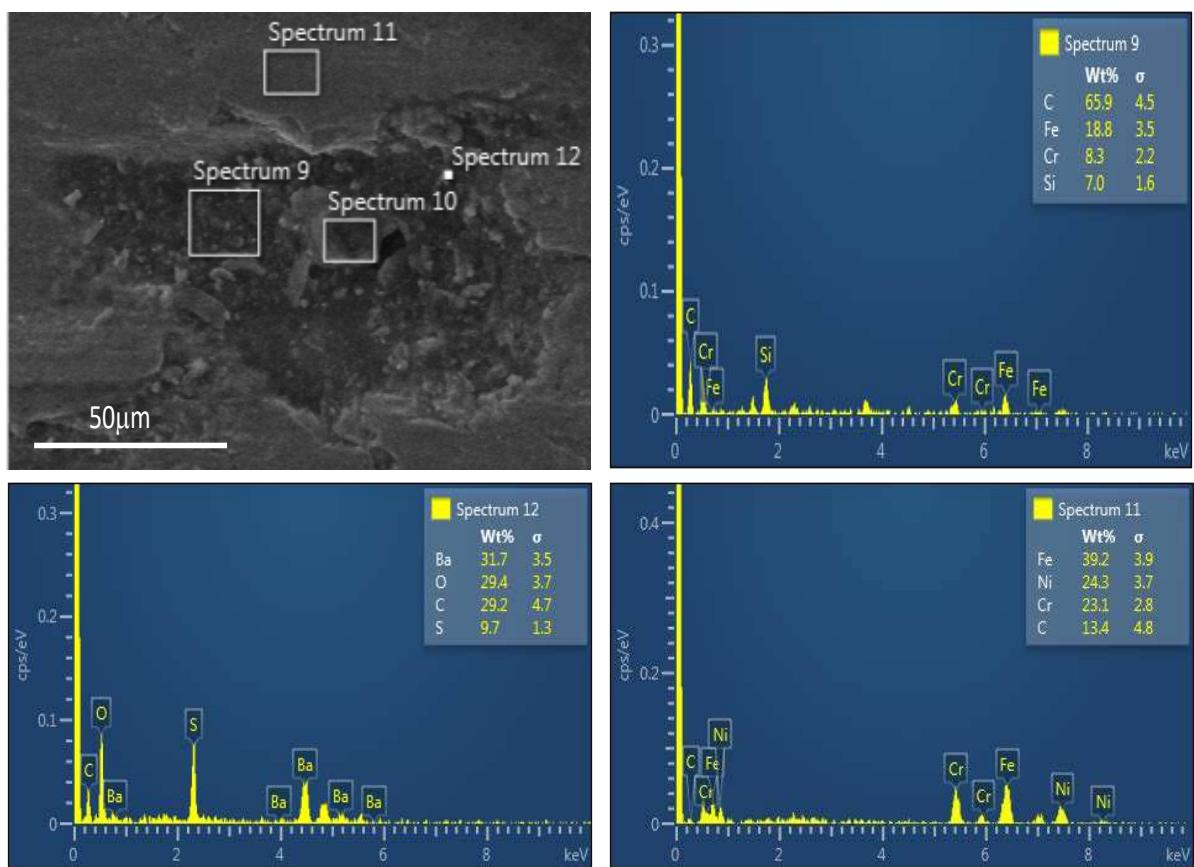
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**Figure 13.** SEM micrographs of SM2535-110 casing specimen tested at 207 rpm and 1.4 KN under OBM.

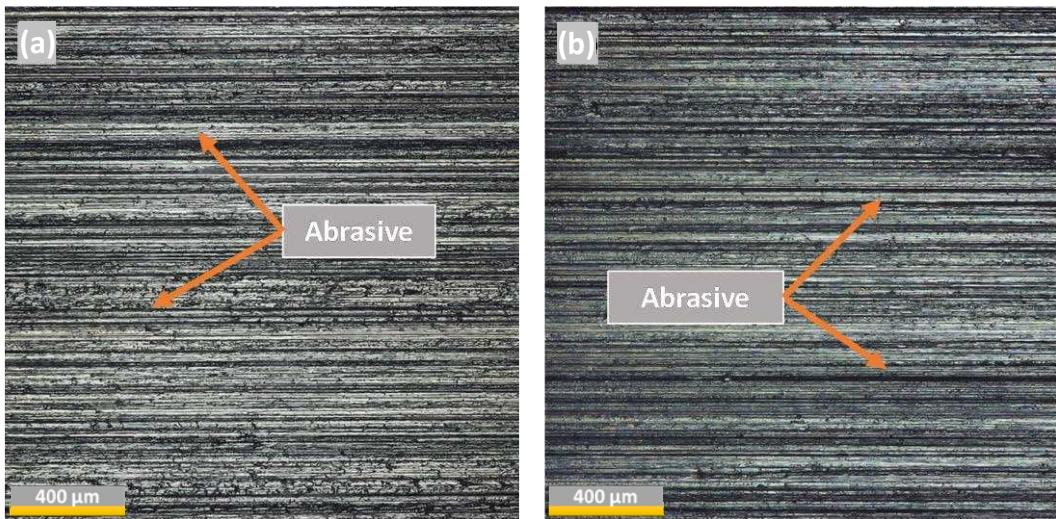


**Figure 14.** SEM micrographs of SM2535-110 casing sample subjected to 1.4 KN at 207 rpm and under WBM.

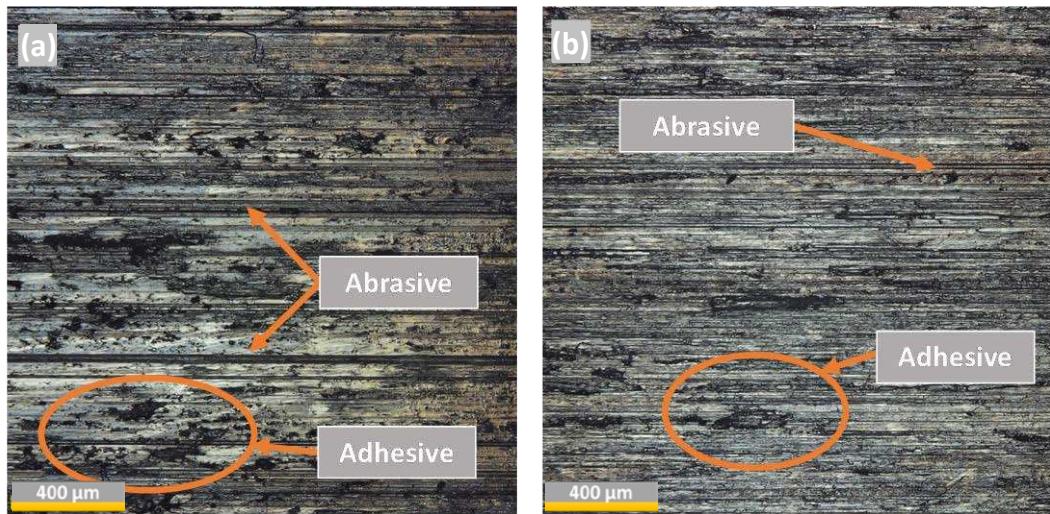


**Figure 15.** EDS analysis of worn SM2535-110 casing specimen subjected to a side load of 1.4 KN at 207 rpm and under WBM lubrication.

To explore the effect of the side loading on the SM2535-110 casing wear mechanism, DMI images were captured at the worn area for selected specimens. The DMI of the worn casing samples tested under OBM at 154 rpm and two contact loads of 1 KN and 1.4 KN are illustrated in Figure 16. It can be seen that under OBM, abrasive wear is dominant with the severity of the wear increasing with the side load. The DMI of the specimens tested under WBM at 154 rpm and two side loads of Figure 17 reveal that both adhesion and abrasion were noticed at low and high side loads. Yet, the adhesive wear becomes more dominant as the side load increases from 1 KN to 1.4 KN and the severity of the wear increases as well.

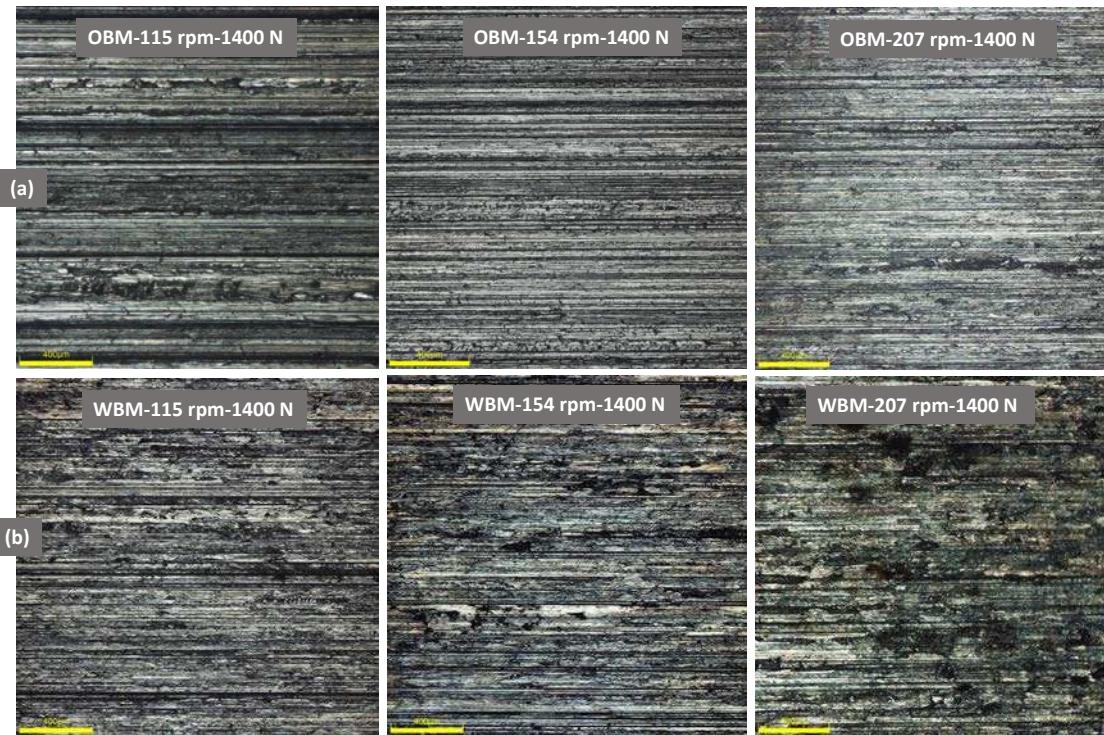


**Figure 16.** DMI of SM2535-110 casing specimens tested at 154 rpm under OBM conditions at a) 1.4 KN and b) 1 KN.



**Figure 17.** DMI of SM2535-110 casing specimens tested at 154 rpm under WBM conditions at a) 1.4 KN side load, and b) at 1 KN side load.

Figure 18 illustrates the side load effects on the wear mechanisms of SM2535 tested at 1.4 KN under OBM and WBM. It is noticed that at 1.4 KN and a lower speed of 115 rpm under OBM, the primary wear mechanism is abrasive wear, and the abrasive wear becomes dominant as the DP-TJ speed increases to 207 rpm. As opposed to that, a hard-worked reactive layer is observed under WBM at lower and higher speeds and both adhesion and abrasion are observed. It is also noticed that as the DP-TJ speed increases, the wear becomes more aggressive along with the deformed reactive layer as shown in Figure 18(b). This hard-worked reactive layer justified the lower wear volume and specific wear factor obtained for SM2535-110 specimens tested under WBM compared to OBM since it improved the wear resistance [9].



**Figure 18.** DMI of SM2535-110 casing specimens tested at 1.4 KN and different speeds under a) OBM and b) WBM.

#### 4. Conclusions

Casing wear experiments were carried out on austenitic stainless SM2535-110, under OBM and WBM under three rotational speeds and three different side loads. The experimental results revealed that:

1. Under both OBM and WBM, the specific wear factor increases with increasing side load due to the increase in the real area of contact.
2. The casing wear factor decreases with increasing rotational speed for specimens tested under OBM while it increases for specimens tested under WBM. The first is mainly due to the change in the lubrication regime from boundary to hydrodynamic lubrication at higher speeds, while the second is due to the lower viscosity of the WBM and the higher COF.
3. The increase in surface hardness of specimens tested under WBM resulted in a lower specific wear factor compared to that obtained under OBM.
4. The primary wear mechanism under OBM was abrasion, whereas the primary wear mechanism with WBM lubrication was adhesion and plastic deformation, especially at high side loads.

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