

## Article

# Porous TiO<sub>2</sub>/Carbon Dots Nanoflowers with Enhanced Surface Areas for Improving Photocatalytic Activity

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**Abstract:** Electron-hole recombination and narrow range utilization of sunlight limit the photocatalytic efficiency of TiO<sub>2</sub>. We synthesized a carbon dots (CDs) modified TiO<sub>2</sub> nanoparticles (NPs) with flower-like mesoporous structure, *i.e.*, porous TiO<sub>2</sub>/CDs nanoflowers. Among such hybrid particles, CDs worked as photosensitizers for the mesoporous TiO<sub>2</sub> and enabled the resultant TiO<sub>2</sub>/CDs nanoflowers with a wide-range light absorption. Rhodamine B (Rh-B) was employed as a model organic pollutant to investigate the photocatalytic activity of the TiO<sub>2</sub>/CDs nanoflowers. The results demonstrated that the decoration of CDs on both TiO<sub>2</sub> nanoflowers and P25 NPs enabled a significant improvement of the photocatalytic degradation efficiency compared with the pristine TiO<sub>2</sub>. TiO<sub>2</sub>/CDs nanoflowers with porous structure and larger surface areas than P25 showed a higher efficiency owing to prevent local aggregation of carbon materials. All the results revealed that the introduced CDs and the unique mesoporous structure, large surface areas and loads of pore channels of the prepared TiO<sub>2</sub> NPs played important roles in the enhancement of the photocatalytic efficiency of the TiO<sub>2</sub>/CDs hybrid nanoflowers. Such TiO<sub>2</sub>/CDs composite NPs also opens a door for photodegradation, photocatalytic water splitting and enhanced solar sunlight as light source.

**Keywords:** porous TiO<sub>2</sub>; hybrid TiO<sub>2</sub>/CDs; photocatalysts; photodegradation; large surface areas

## 1. Introduction

The pollutants, such as inorganic nitrogen oxides and organic dyes from gas or liquid phases, may cause severe environmental and health problems, hence complete degradation and elimination of them is of great importance.<sup>1-3</sup> Removal of pollutants via photocatalytic oxidation process using semiconductor materials and harnessing solar energy has been gaining increasing interest in the last decade, because it is a promising green technology.<sup>4-5</sup> Among them, titanium oxide (TiO<sub>2</sub>)-based materials are believed to be the most reliable materials for photocatalytic reactions due to its low toxicity, high chemical stability, availability, abundance and low cost.<sup>6-9</sup> However, the application of pure TiO<sub>2</sub> is limited by its relative low solar photo-conversion efficiencies, because its wide band gap (3.2 eV for anatase and 3.0 eV for rutile) requires photocatalytic activation only by ultraviolet irradiation (< 4% of the total solar spectrum),<sup>10</sup> and the high electron-hole recombination rate in TiO<sub>2</sub> particles results in low quantum efficiency of photocatalytic reactions.<sup>11</sup> Therefore, great efforts have been directed towards modifying TiO<sub>2</sub> to extend its photocatalytic activity into the visible region and suppress the recombination of photo-generated electro-hole pairs, and thereby enhance solar energy conversion and improve the photocatalytic efficiency. To achieve this target, various strategies have been adopted including doping with transition metals ions (*e.g.* V, Cr, Fe),<sup>12</sup> coupling with narrow band gap semiconductors,<sup>13-14</sup> depositing noble metals,<sup>15-16</sup> and incorporation of non-metal elements.<sup>3, 6, 17-19</sup> Doping TiO<sub>2</sub> with metal ions narrowed the bandgap due to the formation of

intermediate energy levels in the band gap.<sup>6</sup> However, the metal doped materials lack the necessary thermal stability, exhibit atom diffusion and a remarkably increased electron/hole recombination of defect sites, which results in a low photocatalytic efficiency.<sup>12</sup> Non-metal doping TiO<sub>2</sub> has since been proved to be far more successful, and numerous reports on TiO<sub>2</sub> doped with B, F, N, S, I and C have demonstrated a significant improvement of the visible-light photocatalytic efficiency.<sup>1, 11, 20-21</sup>

Among these, the decoration of TiO<sub>2</sub> with carbonaceous nanomaterials like carbon nanotube,<sup>22</sup> C60,<sup>23-24</sup> graphene,<sup>25-26</sup> *etc.* is being increasingly investigated. The resultant TiO<sub>2</sub>/carbon composites were demonstrated in the literature with an improved optical absorbance and better photocatalytic activities in the visible light range when compared with pristine TiO<sub>2</sub>.<sup>6, 27</sup> Carbon dots (CDs), 0-D nanodots, have recently emerged as a new member of the nano-carbon family, and have drawn great attention due to their unique properties.<sup>28-29</sup> Like traditional semiconductor quantum dots, CDs show excellent photoluminescence, great resistance to photo-bleaching and good chemical stability. In addition, they possess the advantages of well water dispersibility, non-toxicity, electron-accepting and transport properties, and facile production of low cost.<sup>30-31</sup> To date CDs have been applied to couple with TiO<sub>2</sub> to extend their light absorption to visible range for a better photocatalytic performance. Researcher believed that CDs can improve optical absorption, trap charge carrier, and hinder the recombination of the photo-generated electron-hole pairs.<sup>32-37</sup> In addition CDs with up-conversion property may even broaden their absorption to NIR light range and enable them with NIR photocatalytic activity.<sup>33</sup> However, up to now, most of the reported TiO<sub>2</sub>/CDs photocatalysts were mainly focused on employing TiO<sub>2</sub> particles with nanotube, nano-belts, or spherical morphology.<sup>32-35</sup> These TiO<sub>2</sub> nanocomposites are normally possess low surface areas which tends to cause uneven distribution and local aggregation of carbon materials, making them the recombination center of photo-generated carriers, reducing the utilization of light.<sup>11</sup>

To solve this problem, we introduce wide visible to NIR absorption range CDs into the mesoporous TiO<sub>2</sub> matrix with precise pore size of around 5 nm to simultaneously utilize the advantages of CDs and mesoporous structured materials with high surface areas. The porous structure and high surface areas may increase the effective contact areas between the pollutants and the active surfaces of photocatalytic particles, reducing local CDs aggregate, which can significantly improve catalytic efficiency.<sup>22</sup> Herein, TiO<sub>2</sub> with mesoporous structure and higher surface areas than the commercial P25 TiO<sub>2</sub> was synthesized by using hydrothermal method and calcination method. CDs were fabricated by simple hydrothermal carbonation of biomass, and then introduced them into the mesoporous TiO<sub>2</sub> particles by self-assembly process, thereby forming TiO<sub>2</sub>/CDs hybrid particles. The photocatalytic behaviour of as-synthesized TiO<sub>2</sub> and TiO<sub>2</sub>/CDs samples towards the degradation of rhodamine B (Rh-B) were studied and exhibit higher catalytic efficiency compared to the commercial P25 TiO<sub>2</sub>.

## 2. Materials and Methods

### 2.1. Preparation of TiO<sub>2</sub> mesoporous particles.

2 mL of titanium butoxide was dispersed in a solution of 20 mL NH<sub>3</sub>H<sub>2</sub>O (37%) dissolved with 0.28 g KCl. The suspension was stirred for 30 minutes, transferred to Teflon-sealed autoclave reactor and maintained at 180 °C for 4 h. The precipitates were then washed with ethanol and DI-water for a few times to remove free ions, and freeze-dried. Finally, the resulting dry powder was calcined by a conventional method, in which the calcinations was conducted at temperature at 550 °C for 4 h in air with a heating ramp of 10 °C/min.

### 2.2. Preparation of carbon dots.

0.7 g D-(+)-glucose was dissolved in 20 mL ethanol and placed in a Teflon-lined, stainless steel autoclave, which underwent treatment at 200 °C for 6 h. The dark brown and yellow solution obtained was centrifuged at 10000 rpm for 20 min to remove the

solution containing fluorescent CDs from the solid black precipitate. The solution of CDs was then filtered using standard syringe filters.

### 2.3. Preparation of TiO<sub>2</sub>/CDs photocatalysts.

The as-prepared mesoporous TiO<sub>2</sub> particles were coated with CDs by soaking them in the CD solution in ethanol for 24 h before removing and rinsing with clean ethanol to remove any poorly adhered CDs.

### 2.4. Characterization of the photocatalysts.

Several techniques were employed for the characterization of the samples. In order to determine the crystal phase composition of the as-prepared TiO<sub>2</sub> particles, X-ray diffraction measurements were carried out at room temperature using a Bruker D5000 with Cu K $\alpha$  radiation in the region  $2\theta = 15-70^\circ$ . The morphology and surface characteristics of the TiO<sub>2</sub> particles were investigated by using a Scanning Electron Microscopy (SEM, FEI Inspect-F). Samples were observed using an accelerating voltage of 20 kV, a spot size of 3.5, and a working distance of approximately 10 mm. The microcrystalline morphology and structure of the samples was analysed by Transmission Electron Microscopy (TEM) and High-Resolution (HR) TEM, using a JEOL JEM-2010 electron microscope operating at 200 kV. BET surface area determination was obtained by measuring N<sub>2</sub> adsorption-desorption isotherm using an Autosorb-IQ2-MP-C system (Quantachrome Instruments, USA). During the BET analysis, the samples were degassed at 150 °C for 24 h prior to nitrogen adsorption measurements. The specific surface area was calculated by the multipoint Brunauer-Emmett-Teller (BET) method, and the pore-size distribution was calculated based on Quenched Solid Density Function Theory (QSDFT) using the adsorption branch. Absorption spectra of the products (CDs, TiO<sub>2</sub>, and TiO<sub>2</sub>/CDs hybrid particles) were measured using PerkinElmer Lambda LS 35 UV/Vis spectrometer. The fluorescence spectra of CDs and TiO<sub>2</sub>/CDs were measured using the fluorescence spectrometer PerkinElmer LS 55, with a slit width of 10 nm both for excitation and emission. Fourier transform infrared (FTIR) spectra were recorded on FTIR spectrometer 100, Perkin-Elmer, collecting data scanning from 4000 to 400 cm<sup>-1</sup> at a spectral resolution of 4 cm<sup>-1</sup>.

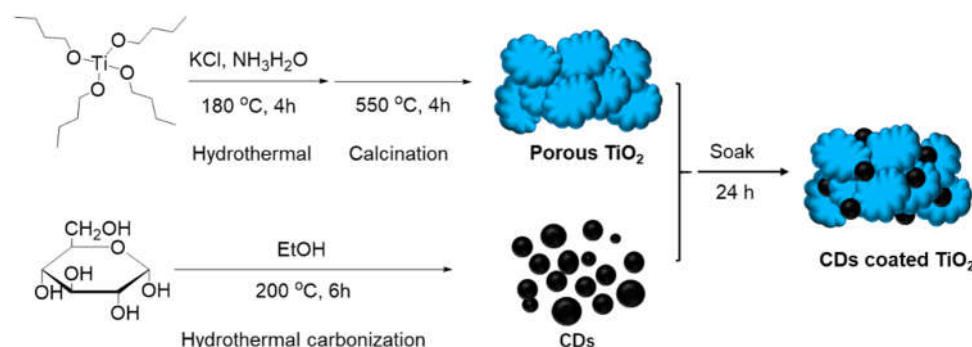
### 2.5. Photocatalytic degradation experiments.

The photocatalytic activity of the synthesized pure TiO<sub>2</sub> and TiO<sub>2</sub>/CDs was tested under a Xenon lamp ( $\lambda > 340$  nm,  $\sim 110$  mW cm<sup>-2</sup>). Rh-B solution with concentration at 20  $\mu$ g/mL was prepared in DI-water. Typically, 10 mg powder samples were suspended in 30 mL Rh-B solution. Prior to the light irradiation, the suspension was kept in the dark for 30 min under magnetic stirring to reach the adsorption-desorption equilibrium between Rh-B molecules and the photocatalysts. Then the above suspension kept in a quartz cuvette was exposed to the irradiation light ( $\lambda > 340$  nm). An ice bath was applied to ensure that the temperature change of the suspensions was less than 5 °C. With magnetic stirring, 3 mL of the dispersion was taken out at regular intervals and centrifuged at 10000 rpm for 5 min, and then the supernatant solution was collected and analysed using a UV-Vis absorbance spectroscopy. For comparison, the photocatalytic reactions were carried out with the catalyst of P25 and P25/CDs under the same procedure. The percentage of degradation was calculated as  $C_t/C_0$ , where  $C_t$  is the concentration of the remaining dye solution at each irradiated time interval, while  $C_0$  means the concentration of Rh-B solution after keeping it in the dark for 30 min in the presence of any photocatalyst.

## 3. Results and Discussion

As depicted in Scheme 1, the synthesis of porous TiO<sub>2</sub>/CDs photocatalyst was realized with a few steps. The TiO<sub>2</sub> with porous structure was fabricated through a combined hydrothermal reaction (180 °C) and calcination method (550 °C), using titanium butoxide, KCl and NH<sub>4</sub>OH as starting materials. The TiO<sub>2</sub> allows a controlled interface and nanocrystal growth under such high thermal treatment. CDs were produced by

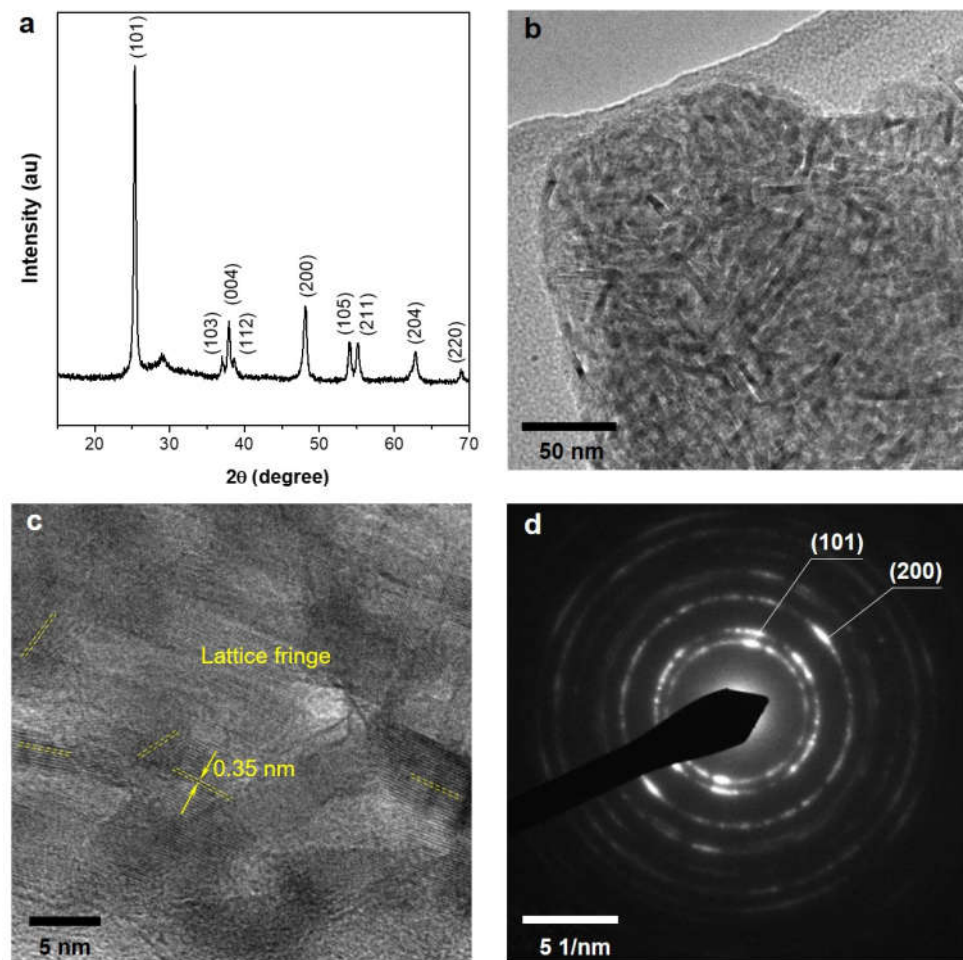
hydrothermal carbonisation of D-(+)-glucose at 200 °C and were simply introduced into porous TiO<sub>2</sub> by self-assembly method. The formed TiO<sub>2</sub>/CDs was then used for degradation study of organic dyes.



**Scheme 1.** Schematic diagram of hybrid TiO<sub>2</sub>/CDs preparation procedure.

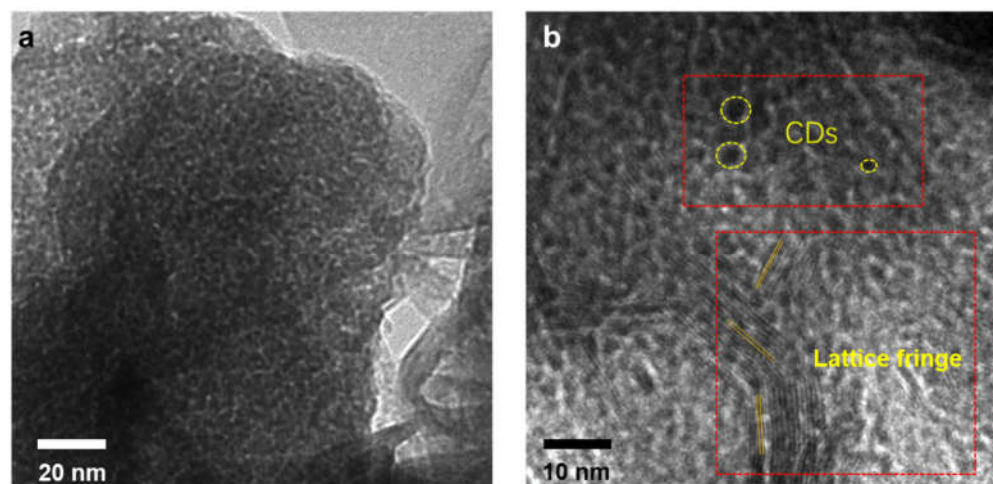
XRD measurement of the as-prepared TiO<sub>2</sub> particles was carried out to investigate their crystalline structure as shown in Figure 1a. XRD pattern exhibited strong diffraction peaks at 25.4°, 37.8°, 48°, 53.9°, 55.1° and 62.7°, which correspond to (101), (004), (200), (105), (211), and (204) faces of anatase TiO<sub>2</sub>.<sup>38</sup> Notably, there were no apparent peaks at the position of 27.58°, 41.38, 44.18, which are the characteristic peaks for rutile TiO<sub>2</sub>. It shall be mentioned that anatase and rutile are two type of TiO<sub>2</sub> polymorphs which are typically being used for photocatalytic application. Anatase structure is preferred due to its significantly higher photocatalytic activity than the rutile phase for several reasons: (1) anatase has a larger band-gap than rutile, which provides anatase with a higher redox potential; (2) anatase has a higher area density of surface hydroxyls, which slows the recombination rate of photogenerated electron-hole pairs.<sup>22, 28</sup> Even though the real causes of the better photocatalytic properties of anatase are not yet fully understood, it is speculated that charge-carrier mobility in anatase is 89 times faster than that in rutile, which is around 80 cm<sup>2</sup> V<sup>-1</sup> s<sup>-1</sup>.<sup>22, 39-40</sup> Hence, we expected that the as-synthesised TiO<sub>2</sub> particles that were mainly composed of anatase phase TiO<sub>2</sub> may possess a superior photocatalytic activity. The scanning electron microscopy (SEM) image showed porous and aggregated particles which were self-assembled by petal-shaped TiO<sub>2</sub> nanoparticles (Figure S1a), while the transmission electron microscopy (TEM) image in Figure 1b displayed a network of porous interconnected channels due to these constituent agglomerated petal-shaped TiO<sub>2</sub> nanoparticles, demonstrating a sponge-like and mesoporous architecture. High resolution (HR)-TEM image of the TiO<sub>2</sub> materials showed clear lattice fringes of the TiO<sub>2</sub> particles, indicating an interplanar spacing of 0.35 nm (Figure 1c), which matched well with the (101) plane of anatase TiO<sub>2</sub>. In addition, our HRTEM image indicated the crystallite of TiO<sub>2</sub> are in indeed nano-sized and the aggregated particles were with through pore structures formed by these nanocrystals, as can be verified from the contrasts in the HRTEM image (Figure 1c). The crystallographic structure of the TiO<sub>2</sub> material was further characterized by selected area electron diffraction (SAED), as displayed in Figure 1d. With a bright 101 ring, the diffraction rings can be indexed perfectly to the anatase phase of TiO<sub>2</sub>.<sup>41</sup> Both the HRTEM and SAED analyses demonstrated a perfectly crystallized TiO<sub>2</sub> materials, which was correspondent well with the XRD results.





**Figure 1.** (a) XRD spectra of the obtained porous  $\text{TiO}_2$  (b) transmission electron microscopy (TEM) image, (c) High Resolution Transmission Electron Microscope (HRTEM) image and (d) selected area electron diffraction (SAED) pattern of the porous  $\text{TiO}_2$ .

When CDs were introduced into porous  $\text{TiO}_2$  by self-assembly method, porous  $\text{TiO}_2/\text{CDs}$  nanoflowers structures were formed (Figure S1b). To confirm that the porous  $\text{TiO}_2$  are incorporated with CDs, Energy Dispersive X-ray (EDX) spectrometry were conducted. Peaks of C, O and Ti elements are shown in the EDX spectra of hybrid  $\text{TiO}_2/\text{CDs}$  nanoflowers (Figure. S2), in which C is from the assembled CDs, demonstrating that CDs is successfully incorporated into the porous  $\text{TiO}_2$ . Therefore, CDs coated porous  $\text{TiO}_2$  are successfully fabricated. (TEM) image also confirmed that a network mesoporous architecture of porous interconnected channels with carbon dots incorporated (Figure 2a). HRTEM image of the  $\text{TiO}_2/\text{CDs}$  showed a clear lattice spacing of 0.35 nm which corresponded to the (101) plane of anatase  $\text{TiO}_2$  (Figure 2b) and the interface of amorphous carbon dots and continuous lattice fringes  $\text{TiO}_2$  (Figure 2b).

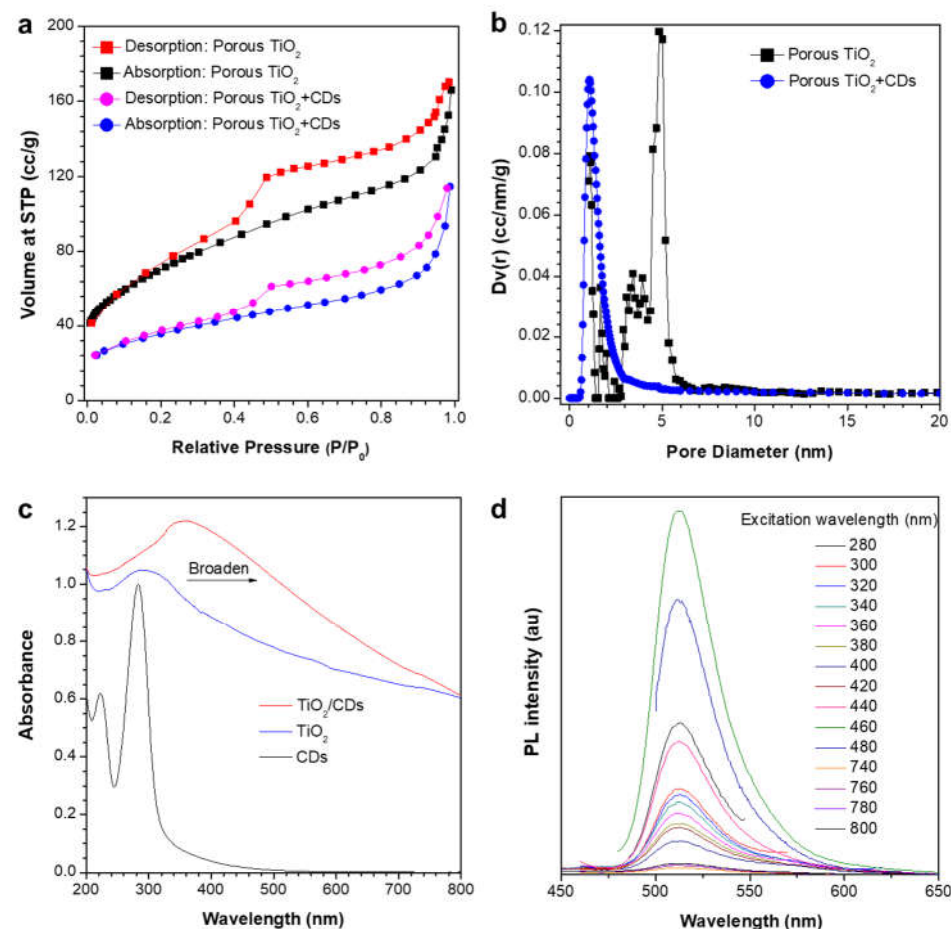


**Figure 2.** (a) TEM image and (b) HRTEM image of porous carbon dots/TiO<sub>2</sub>.

The surface area and the pore characteristics of the TiO<sub>2</sub> product were investigated by measuring the nitrogen isotherms (adsorption-desorption loop), as shown in Figure 3a. The hysteresis loop of the prepared TiO<sub>2</sub> demonstrated a typical type IV isotherm, revealing the characteristic of mesoporous materials. The specific surface area of porous TiO<sub>2</sub> was calculated to be 235.86 m<sup>2</sup>/g, which is greatly larger than the commercial TiO<sub>2</sub> (P25) (50 m<sup>2</sup>/g, data provided by manufacturer). The pore size distribution of the sample was determined by the Barret-Joyner-Halenda (BJH) equation<sup>42</sup>, which suggested a well-defined porous structure (Figure 3b). Pore size distribution shows that the majority of the pores are around 5 nm in diameter, consistent with our HRTEM results. The BET results also showed that the mesoporous channels remain open. Such open mesoporous architecture with connected pore system and large surface area plays an important role in catalyst design for the ability to improve molecular transport of reactants and products.<sup>27</sup> After integrated with CDs, the surface area decreased to 124.923 m<sup>2</sup>/g. The average pore size reduced to 1.07 nm, revealing the CDs were incorporated into the large pores of TiO<sub>2</sub>.

The UV-vis absorption spectra of the pure CDs, pure porous TiO<sub>2</sub> and TiO<sub>2</sub>/CDs were shown in Figure 3c-d. Pure CDs exhibited two strong absorption peaks in the UV region tailing into the visible range until  $\lambda = 550$  nm. The curves of porous TiO<sub>2</sub> and TiO<sub>2</sub>/CDs showed strong light absorption at the UV and visible wavelength of 200-800 nm. Notably, compared with pure porous TiO<sub>2</sub>, the TiO<sub>2</sub>/CDs showed a wider peak and higher absorption intensity, indicating a significant enhancement of light absorption. The peak position of porous TiO<sub>2</sub> located at  $\sim 290$  nm, while that of TiO<sub>2</sub>/CDs shifted to  $\sim 365$  nm. Similarly, P25 TiO<sub>2</sub> with CDs incorporation demonstrated a broader peak, higher absorption intensity and a longer wavelength of the peak position than pure P25 TiO<sub>2</sub>, as can be seen from Figure S3. For both our synthesized TiO<sub>2</sub> and the commercial P25 TiO<sub>2</sub>, their light absorbance was extended and enhanced with the import of CDs. The reason for the enhancement of the UV and visible light absorption could be attributed two parts: (1) the TiO<sub>2</sub>/CDs performed like "dyade" structure, where the TiO<sub>2</sub> and CDs may form a joint electronic system, which gave rise in synergistic properties;<sup>6</sup> (2) the CDs themselves which could absorb UV light and part of visible light.<sup>10</sup> The above results indicated that we demonstrated the design of photocatalysts to enable to harness the use of the full spectrum of sunlight. The porous TiO<sub>2</sub> with high surface area possessed better absorbance of the UV light and even the visible light than the commercial P25. Moreover, the sensitization of TiO<sub>2</sub> with CDs could significantly enhance their UV-vis light absorption and reduce the extinction in the IR region. The as-prepared pure CDs exhibited an excitation-dependent down-conversion photoluminescent (PL) behaviour (Figure S4). It is interesting that no position shift of PL emission peaks of TiO<sub>2</sub>/CDs upon excitation from 280 nm to 800 nm (Figure 3d). The peak is around 512 nm, which further proved the TiO<sub>2</sub>/CDs photocatalyst could absorb wide spectrum of sunlight. Moreover, the maximum emission for TiO<sub>2</sub>/CDs was found at 460

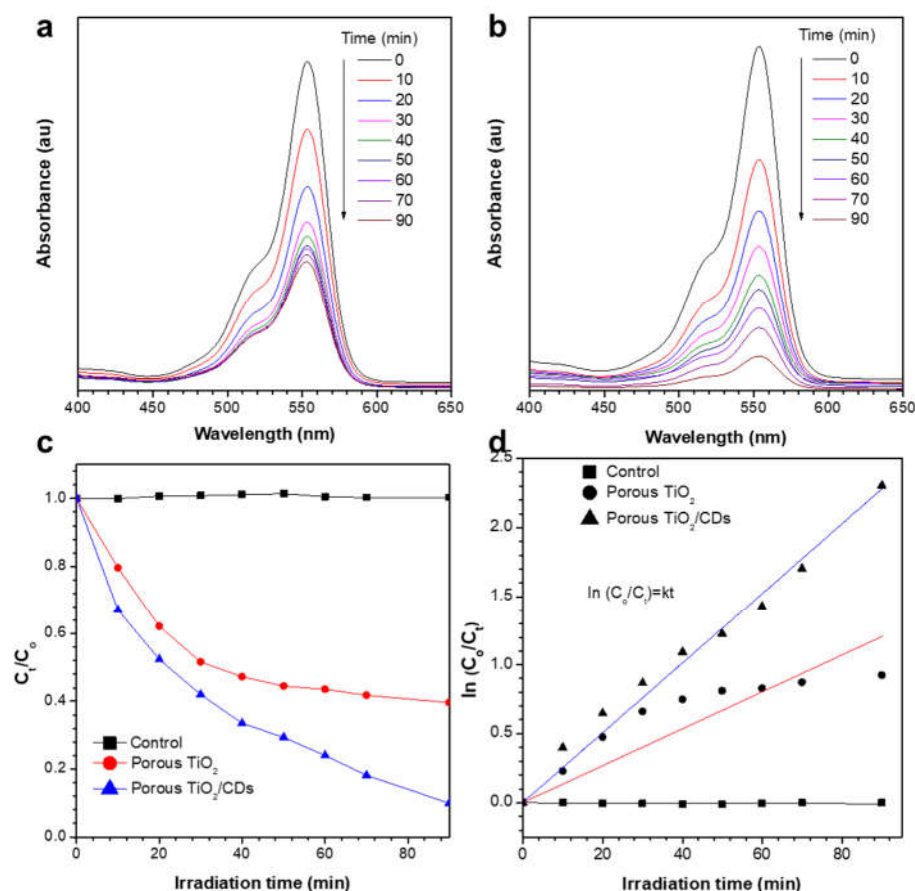
nm excitation, which further confirmed that the CDs attached on porous TiO<sub>2</sub> has a prefer absorbance at 460 nm. This also amplify the photocatalytic property of the composite.



**Figure 3.** (a) Nitrogen adsorption-desorption isotherm curves of porous TiO<sub>2</sub> and porous TiO<sub>2</sub>/CDs particles. (b) Pore size distribution curves of the porous TiO<sub>2</sub> and porous TiO<sub>2</sub>/CDs particles. (c) UV-Vis of CDs (from D-glucose with ethanol), porous TiO<sub>2</sub> and porous TiO<sub>2</sub>/CDs nanocomposite particles. (d) Photoluminescence spectrums of the porous composite TiO<sub>2</sub>/CDs (add up-conversion).

The photocatalytic activities of the prepared porous TiO<sub>2</sub> and porous TiO<sub>2</sub>/CDs samples were evaluated by the degradation reaction of organic dye Rh-B. The concentration of Rh-B was calculated as a function of the irradiation time by measuring the absorbance intensity changes at 554 nm with a UV-vis spectrophotometer. To eliminate the influence caused by adhering of Rh-B on the surface of composite NPs, samples were permitted to be fully saturated with Rh-B solution in dark environment under magnetic stirring until the Rh-B concentration of the supernatants remained constant. After full adsorption in the pores, the photocatalytic activities of the samples were evaluated by measuring the time-dependent photodecomposition of Rh-B aqueous solutions upon the irradiation, as shown in Figure 4a,b. The absorption intensity at  $\lambda = 553$  nm decreased significantly for both sample systems when prolonging the irradiation time. It was found that, TiO<sub>2</sub> incorporated with CDs worked more effective for Rh-B degradation. After 90 mins of irradiation, Figure 4c showed that around 40% of Rh-B was still maintained in the pure TiO<sub>2</sub> system, however, only around 10% of Rh-B was not degraded in the TiO<sub>2</sub>/CDs system. The degradation rates of Rh-B dye over TiO<sub>2</sub> without and with CDs incorporation. The corresponding degradation rates of Rh-B dye over different samples were displayed in Figure 4d, indicating a higher degradation rate of TiO<sub>2</sub>/CDs than that of TiO<sub>2</sub>. In addition, contrast experiments were carried out using commercial P25 as photocatalysts, and the same procedure for the reduction of Rh-B was employed (Figure S5). P25 TiO<sub>2</sub> with CDs

incorporation demonstrated a better degradation efficiency and a higher degradation rate. The results indicated that around 22% of Rh-B in P25/CDs sample while around 50% in P25 samples were left after 90 mins irradiation.

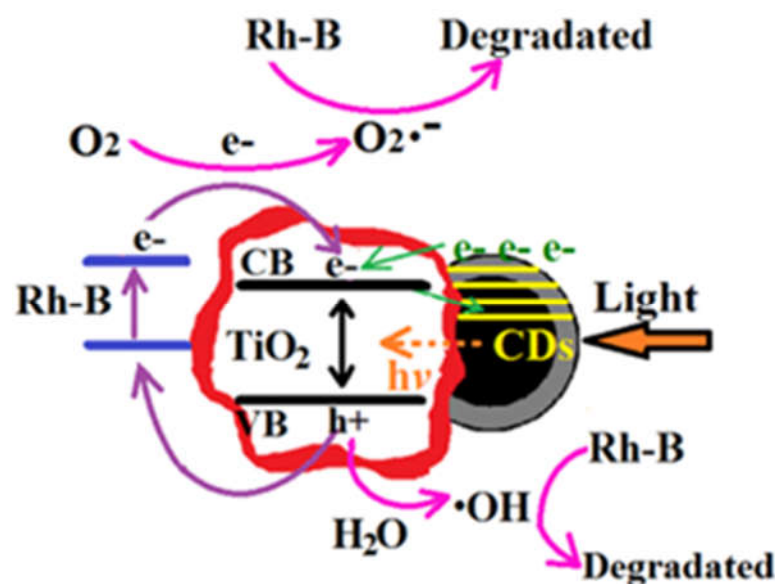


**Figure 4.** UV-Vis spectra of RhB-degradation with (a) porous TiO<sub>2</sub> and (b) porous TiO<sub>2</sub>/CDs under light irradiation for different time. (c) Photocatalytic degradation curves of porous TiO<sub>2</sub> and porous TiO<sub>2</sub>/CDs at different time under light irradiation. (d) Pseudo-first-order fitted degradation of RhB by porous TiO<sub>2</sub> and porous TiO<sub>2</sub>/CDs.

The above analysis revealed that both the synthesized TiO<sub>2</sub> and commercial P25 TiO<sub>2</sub> samples with CDs showed the higher photocatalytic activity of the corresponding bare TiO<sub>2</sub> materials upon the light irradiation, thus proving that indeed a special behaviour is to be expected, due to the intrinsic properties of CDs and the possible interaction between the CDs and TiO<sub>2</sub>. The possible mechanism could be explained as follows. Firstly, CDs incorporation could increase the amount of the total light absorption, which should be from both CDs and Ti species. As demonstrated in Figure 3c, the absorption edge of the TiO<sub>2</sub> exhibits red-shifted after bonding with CDs, indicating that the hybrid TiO<sub>2</sub>/CDs particles have an extended light absorption spectrum and can absorb more light during the photocatalytic process. Secondly, CDs could create an intragap localized states of C 2p situated around the valence band edge of TiO<sub>2</sub> (Figure 5),<sup>30</sup> and NIR light irradiated on CDs can be up-converted to visible light (Figure 3d), therefore, the Ti species could also be excited by the visible light and even the NIR light so as to produce electron-hole pairs and generate  $\cdot\text{O}_2^-$  and  $\cdot\text{OH}$ . Thirdly, CDs are able to serve as electron acceptors and donors,<sup>20</sup> wherein the photo-induced electrons can transfer from CDs to TiO<sub>2</sub> surfaces, and then the redundant electrons on TiO<sub>2</sub> can transfer back to the CDs, as illustrated in Figure 5. To be more precise, on the one hand, CDs can serve as electron reservoirs to trap the photo-induced electrons from TiO<sub>2</sub>, facilitating the efficient separation of electrons and holes and thereby improving their photocatalytic activity. On the other hand, CDs could



work as photosensitizers to  $\text{TiO}_2$ , sensitizing  $\text{TiO}_2$  through the possible formed  $\text{Ti-O-C}$  bond between the  $\text{TiO}_2$  and CDs, and inject the photo-generated electrons to the conduction band of  $\text{TiO}_2$ .<sup>43</sup> When the electron is transferred to surface-absorbed  $\text{O}_2$ , active species of  $\cdot\text{O}_2^-$  will form. Consequently, the integration of CDs and  $\text{TiO}_2$  could efficiently promote the transfer of photo-generated electrons, reduced the electron-hole recombination rate, and then greatly enhance the generation of active species on the surface.<sup>30</sup> Furthermore, the synthesized  $\text{TiO}_2/\text{CDs}$  demonstrated better photocatalytic activity than P25/CDs sample, demonstrating that the microstructure of particles plays an important role in their efficient photocatalytic activity. Specifically, mesopores in  $\text{TiO}_2$  particles influenced the photocatalytic effect to a great extent, which was mainly due to two aspects: (1) large surface area resulting from the mesoporous structure provided plenty of active sites and enabled a better adsorption of the Rh-B molecules; (2) mesoporous channels provided paths for dye molecules to diffuse easily and reach to the active sites.<sup>44</sup> The larger surface area of porous  $\text{TiO}_2$  was proved by the BET data (Figure 3a,b). All the above proposed causes contributed to the final excellent photocatalytic activity of the porous  $\text{TiO}_2/\text{CDs}$  hybrid materials.



**Figure 5.** Schematic description of the proposed photocatalysis mechanism.

#### 4. Conclusions

We hope that the present strategy for constructing CPL-active materials in the condensed matter states will open numerous opportunities for applications in photonic devices. The photo-degradation of Rh-B demonstrated an excellent photocatalytic activity of the  $\text{TiO}_2/\text{CDs}$  hybrid particles, which was much higher than the pristine porous  $\text{TiO}_2$ . With CDs incorporation, P25 also showed an increased photocatalytic activity than that without CDs. It's believed that new chemical and optical properties of the  $\text{TiO}_2/\text{CDs}$  nanohybrid were introduced by CDs, which synchronously made a great contribution to the enhanced photocatalytic activity. The photocatalytic differences between mesoporous  $\text{TiO}_2/\text{CDs}$  and P25/CDs indicated that the nanostructure of the  $\text{TiO}_2$  particles played a valuable role in their photocatalytic activities. Compared with the commercial P25, our synthesized porous  $\text{TiO}_2$  with large amount of pores and extremely high surface areas ( $235.86 \text{ m}^2/\text{g}$ ) displayed a better behaviour in the photo-degradation of Rh-B.

**Supplementary Materials:** Figure S1: SEM image of (a) porous  $\text{TiO}_2$  and (b) hybrid  $\text{TiO}_2/\text{CDs}$ ; Figure S2: (a) EDX spectra of hybrid  $\text{TiO}_2/\text{CDs}$  after CDs incorporation. Inset: SEM image of hybrid  $\text{TiO}_2/\text{CDs}$ . (b-d) EDS mapping profiles of the rectangular area in panel (a) for (b) carbon, (c) titanium and (d) oxygen. Figure S3: UV-Vis spectrums of pure P25 and P25 with CDs. Figure S4: Photoluminescence spectrum of CDs with different excitation (add up-conversion). Figure S5: UV-Vis spectra

of RhB-degradation with (a) P25 and (b) P25/CDs under UV light at different irradiation time. (c) Photocatalytic degradation curves of P25 and P25/CDs at different time under UV light irradiation. (d) Pseudo-first-order fitted degradation of RhB by P25 and P25/CDs.

**Author Contributions:** H. Gao conceived the idea and guided the whole study. F. Song and H. Gao carried out the experiments, including the preparation of TiO<sub>2</sub> mesoporous particles, carbon dots, TiO<sub>2</sub>/CDs photocatalysts. Hao Sun conducted the experiments of SEM, TEM and EDX. Hailong Ma carried out the photocatalytic degradation experiments. All authors were involved in result discussion and manuscript writing. F. Song and H. Gao drafted the manuscript with inputs and comments from all authors

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**Conflicts of Interest:** The authors declare no conflict of interest.

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