

1 *Type of the Paper (Review)*

2 **A Comparison between Semisolid Casting Methods** 3 **for Aluminium Alloys**

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9 **Abstract:** Semisolid casting of Aluminum alloys is growing. For magnesium alloys, Thixomoulding
10 became the dominant process around the world. For aluminium processing, the situation is
11 different as semisolid processing of aluminium is more technically challenging than for
12 magnesium. Today three processes are leading the process implementation, The GISS method, the
13 RheoMetal process and the SEED process. These processes have all strengths and weaknesses and
14 will fit a particular range of applications. The current paper aims at looking at the strengths and
15 weaknesses of the processes to identify product types and niche applications for each process
16 based on current applications and development directions taken for these processes

17 **Keywords:** Semisolid casting; RheoMetal process; GISS process; SEED process; production;
18 capability; surface treatment; heat treatment; tool-life; productivity

20 **1. Introduction**

21 Semisolid casting has developed strongly over the years with many different processes being
22 developed. In general, all these processes can be divided into two main routes for semisolid casting,
23 thixocasting and rheocasting [1]. The principal difference between these is that the Thixocasting
24 route involves the reheating a material into a semisolid state. Rheocasting, on the other hand, utilizes
25 a standard melt, cooled into a semisolid state. Thixocasting gained, through Thixomoulding a wide
26 commercial implementation for Mg alloys.[2] Thixocasting requires special pretreatments of the
27 billets to be reheated and struggles with material and process cost for aluminium. Rheocasting does
28 not have the same difficulties with process cost, not materials costs as standard alloys can be used
29 and are now expanding strongly for aluminium [3]. Driven by the needs to produce high quality,
30 lower costs parts for primarily the electronics and automotive industries, several different
31 rheocasting processes have been developed. Examples of these processes are the New Rheocasting
32 process (NRC) [4,5], Sub-Liquidus Casting process (SLC) [6,7], Semisolid Rheo Casting process (SSR)
33 [8–10], rheo-die casting [7], Gas-Induced Superheated-Slurry process (GISS) s[11], Rapid-S or the
34 RheoMetal process [8], Swirling Enthalpy Equilibration Device process (SEED) [12–18] and many
35 more. The leading processes with the strongest industrialization are the GISS, RheoMetal and SEED
36 processes. [3].

37 The processes GISS, RheoMetal and SEED, can be differentiated based on the solid fraction of
38 the solid fraction entering the shot sleeve/die cavity. GISS uses the lowest fraction and SEED uses the
39 highest. The quality of a slurry also depends on the morphology determine this. All these processes
40 are combined with HPDC processing. This also implies that all these processes will be very similar to
41 HPDC and can be combined with vacuum processing resulting in that all three processes are highly
42 capable.

43 In common for all three processes is that the primary phase has a decisive influence on the
44 mechanical properties of the casting. For Al-Si-Mg alloys, the mechanical properties are dominated
45 by the dissolved Mg content in the primary slurry particles. [19] This fact makes the fabrication of
46 the slurry particles very important for the component performance in many different aspects. The

47 intricate differences in the slurry making process between GISS, RheoMetal and SEED processes
 48 make them better for some applications and more difficult in others. Important to remember is that
 49 this is valid for both physical properties such as thermal conductivity [20] and mechanical properties
 50 [14], making the matching between component and process important.

51 The current paper aims to identify common traits, product types and niche applications for
 52 GISS, RheoMetal and SEED processes based on current applications and development directions
 53 taken for these processes.

54 2. Materials and Methods

55 The method used was a literature survey and interviews with the developers of the processes.
 56 The features that are common for all three processes were identified to clarify the significant generic
 57 benefits of semisolid casting. The individual characteristics were then used to identify the specific
 58 strengths of each process in order to give and indications of the best choice of process for a particular
 59 product group.

60 3. Results

61 The main results are found in tables 1-3. The leading products in production are collated in
 62 Table 3. to give an idea of what is commercialized. The main common characteristics are collated,
 63 table 1, to provide an idea of the generic benefits of semisolid casting based on the proven benefits of
 64 the processes discussed in this paper. The specific capabilities are collated in Table 2 to provide a
 65 foundation for the choice of processes based on component requirements.

66 3.1. GISS Technology

67 The Gas Induced Super-heated Slurry process or (GISS) process is the market leader with more
 68 than 100 licensed units used for parts used in the automotive, heavy truck, military, electronics and
 69 medical industries.

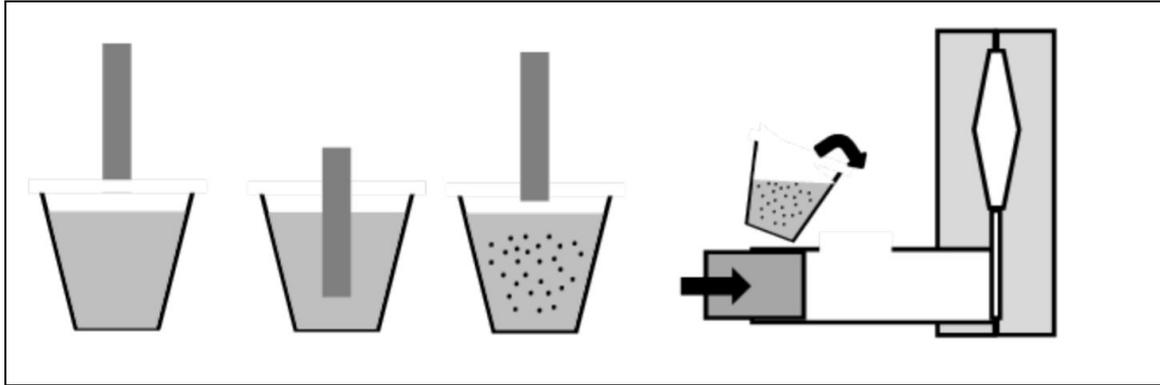
70 The GISS process is a process that offers a cost-effective, quick installation and is an entry-level
 71 process with a low threshold for implementation of semisolid casting. The process steps in the GISS
 72 process are as follows, Figure 1: [17]

- 73 1. The melt is ladled from the furnace with a 10 to 20 K superheat
- 74 2. A porous graphite body is immersed for 5-20 s with Nitrogen gas seeping through this porous
 75 body. The gas bubbling results in local cooling and nucleation of the primary solid-phase to
 76 initialize the slurry formation. It should here be noted that the gas seeping out through the
 77 graphite body hinders metal from sticking to its surface to facilitate a clean retraction of the rod
 78 from the melt.
- 79 3. As the graphite body was removed, the slurry precursor is directly poured into the shot sleeve
 80 where the slurry forms and is immediately cast.
- 81 4. The ladle is cleaned returns to the processing step 1)

82
 83 The primary characteristics of the process are that it runs on a relatively low fraction starting at
 84 some 5% solid fraction in the ladle and following cooling in the shot sleeve typically resulting in 25%
 85 to 30% solid fraction making is a low solid fraction process. The lower solid fraction and rapid
 86 cooling during filling together with a large number of nuclei generate conditions that do not give the
 87 time nor promotes the formation of dendrites. The main control her is the time for the gas bubbling
 88 that initiates the solid phase generation and thus acts as viscosity control and the gate speed through
 89 the second phase speed in the HPDC machine. The ratio gate speed/gas bubbling tome is essentially
 90 identical to the Reynolds number and thus directly related to the level of turbulence during the die
 91 filling process. It is the possibility to control this that allows for the manufacturing of parts with
 92 intricate shapes and thin-walled parts down to 0.5mm wall thickness.[21]

93 A wide range of materials is found in the applications ranging from conventional casting alloys
 94 such as A356 to wrought material such as AA6061 and AA7075. In terms of post-processing, both T6
 95 and anodizing are possible depending on alloy choice.

96



97 **Figure 1.** The GISS process (Courtesy GISSCO)

98 The GISS process set-up and process characteristics are such that it can be directly implemented
 99 into an existing conventional HPDC production. This provides a low implementation threshold and
 100 also good example with the possibility for direct comparison and identification for the benefits of
 101 semisolid casting to conventional HPDC casting. Noticeably, as claimed by all processes, is that the
 102 thermal load on the die is reduced. The first and direct benefit is an increased die-life, as thermal
 103 fatigue, and erosion dominate die life in HPDC casting. The introduction of the GISS process has
 104 resulted in a die-life extension up to a factor 4 compared to the corresponding HPDC process- This
 105 has resulted in that the die-life in the GISS process has exceeded 400k parts produced in a single die.
 106 The combination of reduced heat input and a reduced fill speed furthermore allows for a reduction
 107 of die lubrication/release agents with up to 40%. Die spraying reduction will directly reduce spray
 108 duration and will reduce cycle time. Additionally, as the slurry is semisolid the time for
 109 solidification in the die cavity reduces shortening the time to part ejection. Reduced duration for the
 110 part ejection and spray duration has reduced cycle time with as much as 20%. To achieve this, it is
 111 essential to manage slurry generations properly to ensure that there is no infringement on. From a
 112 practical standpoint this also involves planning robot actions so that there is no waiting time due to
 113 the slurry preparation.[21]

114 In terms of internal soundness, porosity level can be significantly reduced by the introduction
 115 of the GISS process together with the optimization of the lubrication spray cycle. The reduced
 116 porosity allows for the delivery of parts in F, T5 and T6 states. Reduced porosity also enables
 117 weldability as in HPDC the main hinder for welding is entrained gas in the die cavity during filling.
 118 It is well-known that porosity is the main reason for rejection in HPDC processing; the use of the
 119 GISS process has resulted in rejection reductions from 30% down to 5%.[21]

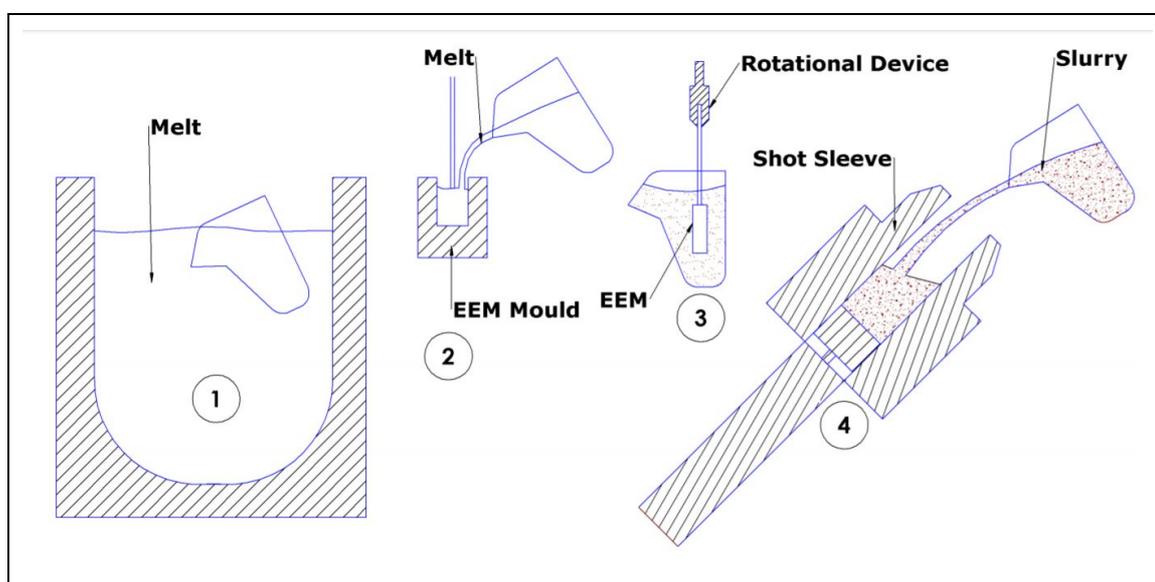
120 Intrinsic advantages originating from the thin-wall capability is that parts can be redesigned for
 121 semisolid casting weight reduction may be possible improving the sustainability of the produced
 122 components with increased resource efficiency and reduced energy content. The increased internal
 123 soundness allows for a process change from gravity die casting or low-pressure die casting to
 124 semisolid casting resulting in cycle time reduction from typically 4-8 minutes down to 1-2
 125 minutes.[21]

126 3.1. RheoMetal process

127 The RheoMetal process, with more than 30 machines delivered, have primarily found
 128 applications within heat sinks and LED fittings but is now finding new applications within the area
 129 of heavy trucks and automotive components. There are also application examples from marine
 130 equipment and sports equipment as well. The main characteristic is that a thick high solid fraction
 131 slurry is made in a very short time. The rapid slurry generation is enabled through the use of melting
 132 consumable or Enthalpy Exchange Material (EEM) to cool the melt into the semisolid region. The
 133 rapid slurry generation results in a highly non-equilibrium solidification in the slurry processing
 134 and a hard to predict solid fraction but once trimmed is very stable and repeatable. The

135 repeatability is due to that the balance is not temperature-controlled but rather mass-based. The
 136 process steps are as follows:

- 137 1. A preheated ladle is filled from the furnace with a melt with a superheat of typically 20°C, but
 138 depending on the melt delivery set up
- 139 2. Typically, 5-8% us teemed off from the ladle and cast around a steel rod into a cylindrical shape
 140 and placed in a carousel. This material is to be used as EEM.
- 141 3. A rod with an EEM previously cast (normally 6 cycles earlier as the carousel typically holds 6
 142 EEMs to allow degating and cooling), is immersed under rotation into the ladle with the
 143 remaining melt. The rotation provides the required shear to turn the solidified particles
 144 non-dendritic. The EEM is stirred until complete melting which typically takes 5-40 s. In newer
 145 systems, a secondary stirring is added for improved slurry homogeneity.
- 146 4. The slurry in the ladle is directly poured into the shot sleeve and injected into the mould cavity.
- 147 5. The ladle is cleaned and returned to preheating and returns to the processing step 1)
- 148 6. The steel rod is cleaned and returned to the casting station for a new EEM casting in step 2)



150 **Figure 2.** The RheoMetal process (Courtesy Comptech)

151 The RheoMetal process is classified as a high fraction semisolid casting process with its
 152 typically 30-45% solid phase in the slurry. The normal slurry making time is within 20 s. This short
 153 duration for the shearing will create a slurry with a solid phase is less globular compared to other
 154 processes. For high strength and more demanding applications, the introduction of a short
 155 secondary stirring has improved the slurry quality to generate very low levels of porosity. The
 156 reduced porosity has also allowed for pressure-tight castings to be produced, eliminating the need
 157 for impregnation.

158 Like all semisolid processes, thin-walled components are possible, and the RheoMetal process
 159 has shown that 40 mm high walls down to 0.35mm thicknesses can be produced industrially even
 160 for as complex products as a radio filter. Compared to the other processes, fatigue loaded
 161 thick-walled components are also being produced with wall thicknesses above 10 cm being in
 162 production. Again, this was achieved by a strong management of the slurry quality.

163 The material variety for the RheoMetal process is comprehensive, and Al-8Si, A356, A357, A319,
 164 Magsimal 59 have all been realized, as well as the wrought alloy 6082. Experimentally, alloys down
 165 to 0.45%Si have been cast successfully in full industrial-scale.

166 Critical understanding, for the RheoMetal process, are the consequences of the deviation from
 167 equilibrium. The first issue is that as the EEM is immersed into the melt, a freeze-on layer forms. This
 168 layer has a composition which is given by the composition of the solidus line in the phase diagram
 169 at the slurry forming temperature. This temperature is often just a just 2-5°C below the liquidus of the

170 melt. The level of solutes in the slurry particles is significantly lower than for other processes and has
 171 two direct consequences. Firstly, thermal conductivity is increased with up to 17% compared to the
 172 same material cast using HPDC or gravity die-casting. The downside is that yield strength is
 173 commonly slightly lower than for HPDC casting, but ductility is improved.

174 In terms of die-life, the same abundance of die-life data does not exist for the RheoMetal process
 175 as it does for the GISS process due to that most of the RheoMetals products have not been HPDC cast
 176 before and direct comparison is not possible. The main reason for an increase die life is the intrinsic
 177 heat in the slurry entering the die. The higher solid fraction RheoMetal process and the associated
 178 lower amount of intrinsic heat suggest die-life should be at least that seen for the GISS process.
 179 Die-life is also depending on part size and geometry as well as on the use of lubrication agents and the
 180 air blowing cycles. Being able to reduce this also reduce the die cooling action driving the generation
 181 of tensile stress in the surface, which also may affect the die life achievable.

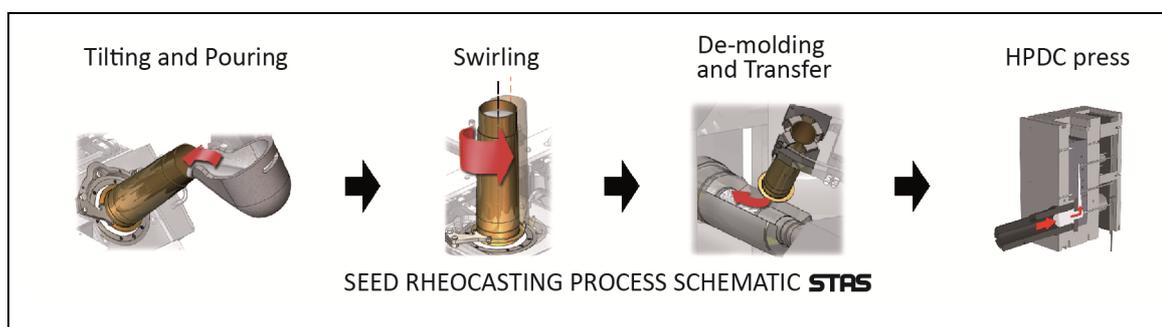
182 The use of sand cores in conjunction with the RheoMetal process has been successfully tested
 183 but not yet been fully utilized commercially.

184 3.1. SEED process

185 The SEED process has shown many applications, but the market penetration is not clear.
 186 High-performance heavy-duty components for the automotive, heavy truck and sports industries
 187 have been targeted for the development of the SEED process. This type of application requires
 188 high-quality slurry. To achieve the best possible quality, the duration for the slurry making process
 189 is significantly longer for the SEED process compared to the GISS and RheoMetal processes. The
 190 longer duration of the shearing phase gives SEED an advantage in the ability to generate a good
 191 quality slurry. The higher slurry quality also enables for even higher levels of solid fraction than the
 192 GISS and RheoMetal processes. The main process steps are

- 193 1. A clean slurry making container is filled from the furnace.
- 194 2. The slurry making container with the melt is placed on an oscillating table to create the swirling
 195 flow of the melt inside the container to provide the required shear in the melt to produce the
 196 globular microstructure
- 197 3. Older systems had a draining stage to allow the high fraction solid slurry to be removed from
 198 the container. In newer systems, this step is no longer needed making the processing time
 199 shorter
- 200 4. The slurry is poured from the container into the shot sleeve, and cast using high pressure die
 201 casting equipment
- 202 5. The ladle is cleaned and returned to preheating and returns to the processing step 1).

203



204 **Figure 2.** The SEED process (Courtesy STAS)

205 As part of the process design, there is a relation between the shot weight and the wall thickness
 206 of the container. The basic principle is that there should be a thermal equilibrium between the
 207 container and the slurry and to some extent the process is not purely temperature controlled but also
 208 mass controlled to provide a stable and repeatable process. The slurry making time range from 100 s
 209 up to 160 s, hence at least 3 slurries need to be under processing in order to not interfere with the
 210 process as common cycle times for an HPDC casting process is in the range between 30 s and 90 s.

211 The high solid fraction targets high-performance parts and not ultra-thin walls components.
 212 The viscosity of a 50% solid fraction melt makes thin-walled casting, and the minimum wall
 213 thickness for SEED is 0.75mm, thicker than what the GISS and RheoMetal process have been used to
 214 produce.

215 The material variety for the SEED process is similar to both RheoMetal and the GISS process but
 216 more focused towards alloys suitable for heavy-duty and high-performance application and as such
 217 not as comprehensive. SEED is, on the other hand, the only process that has targeted B206, which is
 218 one of the highest-performing alloys that can be cast. Unique to the SEED process is also that it has
 219 been used to cast Duralcan composite material and that it also has proven to improve fatigue
 220 performance with up to a 22% performance increase.

221 As for the RheoMetal process, existing data for die-life improvement is limited, but again as
 222 solid fraction is higher in the SEED process than for the other processes, and in theory, it should
 223 produce the highest die-life improvement provided that thermal fatigue is the limiting die life factor.

224 4. Discussion

225 3.1. Generic features

226 The semisolid casting processes have recently started to expand after many years struggling to
 227 find applications significantly. The reason for the current change is several but perhaps the most
 228 important change is due to that conventional HPDC processing has started to struggle to provide
 229 parts with the required features and properties. New applications for automotive components have
 230 large series, and permanent mould casting lacks sufficient productivity and cost-efficiency. It
 231 appears as the industry is at a pivoting point searching for a solution. The common traits of the
 232 processes hold a promise of a solution to many of these difficulties, table 1.

233 All semisolid casting processes have in common that heat is removed from the melt and a solid
 234 phase is precipitated. The first and most noticeable effect is that viscosity is increased in the presence
 235 of the solid phase. The Reynolds number, Re , Eq. (1) is a characteristic measure for the level of
 236 turbulence and an increased viscosity will result in a reduction of Re indicating a reduced level of
 237 turbulence.

$$238 \quad Re = \frac{v\rho D_H}{\mu}$$

Eq (1)

239 Here v is speed, ρ is density, D_H is the hydraulic diameter/characteristic length of the system,
 240 and μ is viscosity. In semisolid casting the increase viscosity is often also combined a reduction of
 241 the injection speed, adding to the reduction of Re and turbulence. This was also the first main focus
 242 of the research as it provides the core of the process control for the injection phase, solid fraction and
 243 injection speed and also the primary contributor to the yield improvements that can be seen, table 1.

244 The heat management is slightly different between the different processes. In the GISS process,
 245 this takes place in the ladle but perhaps foremost in the shot sleeve where the solid phase fraction
 246 increases significantly, In the RheoMetal process this occurs in the pouring ladle and the SEED
 247 process in the unique preparation crucible. Independent of this, the first consequence is that the
 248 intrinsic heat of the melt entering the mould cavity is significantly reduced and the thermal load of
 249 the mould materials is reduced. Die-life will be inherently improved. Die-life extension up to 4 times
 250 is possible and that a die-life of more than 400000 shots have been realized in production., table 1.

251 The fact that the amount of heat entering the die is reduced will also reduce the die thermal
 252 distortion. The part contraction will be similar as for the HPDC processing, but the reduced thermal
 253 distortion in the die also allows for a reduced release agent usage due to a lesser amount of relative
 254 motion between the mould and casting. This affects the die cavity atmosphere as the off-gassing from
 255 die-spray residues will be less than otherwise can cause significant rejection rates. [22] This reduces
 256 the available material for entrainment with or without a vacuum system during casting and reduces
 257 the requirements on venting. Similarly, reduced spraying also allows for a reduction of cycle-time. It
 258 should here be noted that the solid fraction does not strongly affect the cycle time as the
 259 solidification time in most cases is significantly shorter than the spraying cycle-time. The solid

260 fraction is important only for systems with a large biscuit dimension. The thermal management thus
 261 affects both the part rejection rate and cycle time indirectly, table 1.

262 The increased viscosity and the reduction of release agent usage together reduced tool
 263 distortion will create conditions for production with reduced rejections from porosity and fewer
 264 process interruptions as the flash formation and sticking tendencies are reduced as well. The most
 265 important part is also to realize that entrainment porosity will be located randomly and is hard to
 266 control. The pressure inside the entrained gas is also of the same order of magnitude as the die cavity
 267 pressure at the end of the intensification period. Removing these the only porosity that will remain is
 268 shrinkage porosity. Shrinkage porosity has the advantage that it can be managed through part
 269 geometry design and as such is manageable. It will also be reduced, as only 50 to 80% of the
 270 solidification takes place in the die cavity, reducing the feeding requirement. It should here be noted
 271 that due to the presence of the solid phase, feeding is more complicated and requires more research
 272 to be fully understood.

273 Porosity reduction will also have the most profound effect on part performance and
 274 post-processing capability. Reduction of the entrained gas will allow for heat treatment and welding
 275 as it is the entrained gas that causes the main issues for these processes with blistering and poor
 276 weldment quality.

277

Table 1. Collation of Common traits. [23–26]

Common element	Comment
Tool life	Reduced thermal load on dies improve die-life with GISS and SEED proving up to 4 times that of HPDC
Lubrication/release agent used	GISS has proven reductions of 40 %
Cycle time reduction	GISS has proven cycle time reductions of 20% compared to an HPDC cycle
Process yields	GISS shows from 30% to 5% and RheoMetal from scrap rates of 20% to well below 1%.
Productivity	Many applications mean a change from gravity die casting or low-pressure die casting to rheocasting with cycle time changes from 4-8 minutes to 1-2 minutes in Rheocasting as it is based on the HPDC cycle. Productivity is also increased due to yield increase
Weight reduction	The thin-walled capability allows for significant weight reduction (Radio filter down to 72% of HPDC cast version)
Weldability	Porosity reduction gives increased weldability
Heat treatment	All processes have proven that F, T5 and T6 conditions are possible

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279 The ability to fill combined with a high viscosity in a shear-thinning or even thixotropy allows
 280 for greater variation in section thickness than for conventional casting. This is one of the main issues
 281 for the electronics industry that is heavily depending on heatsinks with cooling fins. These fins can
 282 be made thinner with optimized distancing using semisolid casting and was also among the first
 283 industry sectors where this was used. Sustainability of the casting process is thus also significantly
 284 improved, from the process-yield increase, together with the possibility of part weight reduction
 285 that together drives resource efficiency. Besides, the reduction of release agent usage also results in
 286 reduced use of silane, siloxane and resins.

287

288 3.1. *Specific capabilities*

289 In terms of process capabilities, the different processes have different strengths making them
290 more suitable for different applications and easy to implement. The main difference between the
291 process resides in a specific manner the solid fraction in the slurry is generated with some very
292 intricate differences between the processes and the resulting material characteristics and process
293 capabilities.

294 The GISS process results in a slurry with a low fraction solid of the melt entering the mould
295 cavity (5-25%). The actual treatment results in approximately 5% solid phase that acts as seeds that
296 will aid nucleation during solidification in the shot sleeve and the die cavity. This results in
297 relatively well-rounded particles, but the low fraction will make the cooling rates and process
298 conditions similar to HPDC processing, compared to the other processes. This means that
299 segregation patterns and mechanical properties will be similar to the HPDC process materials except
300 for a significant reduction of porosity, Table 2.

301 The RheoMetal processing is generating a slurry rapidly with an extreme deviation from
302 equilibrium. In contrast to both the GISS and the SEED process, the RheoMetal process has an
303 element of Thixocasting included. The EEM that melts is not fully molten but equilibrated with the
304 other particles in the slurry. That means that the slurry consists of particles that originate from both
305 reheating and solidification, making the RheoMetal process a hybrid between Rheocasting and
306 Thixocasting. Another deviation is the dendritic freeze on layer formed on the EEM, that forms and
307 disintegrates during the slurry preparation. The primary deviation from equilibrium is
308 chemistry-based. Due to that, the majority of the solid phase is formed at very high temperature; the
309 primary slurry particles are lean in solutes. For an A356-type of alloy, the time to homogenize the Si
310 and Mg content in the slurry particles of a 70 μ m diameter is approximately 30s.[27] This is of the
311 same order of time as it takes to make the slurry. For the RheoMetal process, it is difficult to
312 accurately predict the amount of solid phase produced due to the kinetics of the alloy and the
313 processing system. Under production, the shot temperatures vary less than 1°C, even though furnace
314 temperature is much less controlled than this. The solid fraction varies from 30-40% typically, and
315 the flowability of the slurry limits the upper limit due to that the particles are not as round as for
316 GISS and SEED processes. It should here be noted that the latest developments of the RheoMetal
317 process also includes a secondary stirring step to improve slurry homogeneity and particle
318 roundness., Table 2

319 The materials produced using the SEED process will be closer to equilibrium as the duration of
320 the slurry making process is longer with the typical range of 100 s -160 s. The direct consequence of
321 this is the generation of well-rounded particles richer in solutes. The improved shape of the particles,
322 compared the RheoMetal process, allows the SEED process to run at higher solid fractions than the
323 other processes (35-50%).[24]

324 The deviation from equilibrium seen in firstly the RheoMetal process and to a lesser degree in
325 the GISS process with reduced amounts of dissolved solutes results in improved thermal
326 conductivity. In the RheoMetal process, an increase by as much as 17% compared to an HPDC cast
327 material with the same composition can be seen, table 2.

328 The reduced porosity from primarily entrainment porosity allows for heat treatment. Since T6
329 treatments are possible, a high productivity alternative permanent mould casting is found resolving
330 productivity issues for high-performance components. All three processes do well in the T6
331 condition in terms of mechanical response. In the as-cast and T5 condition, slight differences are
332 arising from the slurry making process differences. The RheoMetal process producing a primary
333 phase low in solutes gives a slightly lower strength but better ductility material compared to the
334 GISS and SEED processes. Similarly, the reduced amounts of solutes will also affect the T5 response,
335 where a slightly lower strength is to be expected in the RheoMetal process compared to GISS and
336 SEED processes. It should be noted that this is component dependent and will be depending on the
337 efficiency of the use of water quenching on ejection for parts that should be T5 heat treated table 2.

Table 2. Collation of Process capabilities. [23–26]

Capability measure	GISS	RheoMetal	SEED
Anodizing and surface treatment	Colour anodizing possible Anodizing of 7xxx, 6xxx alloys and Al-Mg alloy	Thick anodizing layers possible	Anodizing possible.
Fatigue resistance	-	Excellent with thick-walled component	- Excellent. Ex. Up to 22% increase in fatigue life (turbo impeller case)
Wall thickness	From >10cm down to less than 0.5mm, most common 1-3mm	From >10cm down to less than 0.35 mm, most common 2-3mm	Down to 0.75mm
Proven alloy capability	Casting alloys: A356, Al-Si7, A380, A383, Silafont 36, Magsimal 59, A390, Pure Aluminum	Casting alloys: Al-8Si, A356, A357, A319, Magsimal 59	Casting alloys: A356, A357, 319S, B206
	Wrought alloys: 6063, 6061, 5082, 7075	Wrought alloys: 6082	Wrought alloys: 6061
Strength	Normal strength Moderate T5 response Excellent T6 response	Softer as-cast condition, Excellent elongation Moderate T5 response Excellent T6 Response	Normal strength Excellent Elongation Excellent T6 response
Other notable achievements	Colour anodizing to thin-walled and thick-walled components Used in gravity die casting with a cycle time reduction of up to 20% Pressure tight castings without impregnation Improved thermal conductivity by up to 15% Flexibility to switch between rheocasting and HPDC-	Experimental casting tested successfully down to 0.45%Si Pressure tight castings without impregnation Use of sand cores Improved thermal conductivity by up to 17% Suitable for 20-40kg slurries Excellent slurry homogeneity due to secondary stirring Flexibility to switch between rheocasting and HPDC	High-Solid fraction up to 50% Process range from 2kg to 18kg slugs. Other alloys: Duralcan composite

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Fatigue performance has for long been the weak point of castings due to porosity, and only permanent mould casting has had some success in this area. Semisolid processing is changing this. GISS that operates at lower fractions solid compared to RheoMetal and SEED will be more subjected to the existence of porosity. RheoMetal and SEED operate at a high level of solid-phase, reducing entrainment porosity effectively and will thus have a better potential of achieving excellent fatigue resistance as porosity is the leading cause of fatigue crack initiation followed by oxides. That this is the case can be seen in applications with RheoMetal being used in heavy truck components with heavy fatigue loads and SEED with an impeller with 22% improvement in fatigue life, Table 2 and 3.

All processes are capable of casting wrought alloys for strength and also for the possibility to anodize for protection and appearance. GISS has developed colour anodized motorbike brake callipers in 7075. For the RheoMetal process, the choice of direction has been slightly different where work has focused both to cast with reduces Si amounts in the material and to increase the anodizing capability to understand better the relationship between the base material and the quality of the

353 anodized layer. Inoculation and strontium treatment together with a high fraction solid allows for
354 better anodizing outcome also for the Silicon alloyed materials, table 2.

355 GISS is furthermore the only process capable of being used with permanent mould casting due
356 to the lower solid fraction that may be generated, table 2.

357 3.3. Industrial applications

358 GISS has found the broadest range of application and also has the most comprehensive range of
359 alloys tested with a proven capability. This has its foundation with the commercial success of the
360 GISS process. The SEED process has chosen a different set of alloys with copper-rich aluminium
361 alloys designed for strength and performance. This has been a strategic decision to focus on the
362 automotive industry and high-value components, table 3.

363 Based on the current usage of the processes GISS has an excellent cover of the automotive
364 process with a slight dominance of passive components such as chain coves and oil pans, but also
365 pressure-tight application such as compressor housing. The RheoMetal process also achieves
366 pressure tightness. The SEED process, on the other hand, has more applications such as shock
367 towers, control arms and turbo impeller with components that are subjected to high dynamic loads.
368 This implies that for the automotive industry, the GISS and RheoMetal process have found more
369 applications relying on thin-walled capability with weight reduction as one feature as well as
370 reduced porosity for cost-effective impregnation free applications. The SEED process has targeted
371 more critical high strength and high-performance components. Herein lies also the alloy capability
372 and the use of the B206 alloy, table 2 and 3.

373 For heavy-duty truck components, GISS relies on thin-walled and shape replication capability
374 through gearbox castings. SEED and RheoMetal have found application in more fatigue loaded
375 components. The RheoMetal process, in particular, has found application in thick-walled
376 heavy-duty fatigue loaded components replacing both cast irons and forged aluminium
377 components, table 3.

378 In terms of marine components and military components, very few applications have been
379 realized, with simple shape component realized by GISS and a slightly more complex shaped part by
380 the RheoMetal process to relieve die sticking and production problems associated with the
381 Magsimal59 alloy and as such use the reduced intrinsic heat in the slurry, table 3.

382 Electronics components often have limited mechanical properties requirements, and the focus is
383 more on low-cost and thermal conductivity requirements. High thermal conductivity offer means
384 difficult to cast alloys, and thermal transfer means complicated shape driving the capability to cast
385 complex shapes using un-castable alloys where machining solutions are common. GISS having a
386 short run-in and introduction cycle have found a niche in the electronics industry with a reduction of
387 process yield improvements for the complex shape products. GISS is thus often used to improve
388 existing production issues. A slightly different approach and the benefit were found for the
389 RheoMetal process. The use of an EEM results in a higher slurry temperature and as a consequence
390 making the primary phase solute lean. This particular effect results in that the RheoMetal process
391 improves thermal conductivity, allowing an as-cast thermal conductivity that generally would
392 require a heat treatment with other casting processes. This attractive feature also has made a niche
393 for the RheoMetal process in electronics. The main application is, however, for new product projects.
394 The SEED process is also capable, but due to a lesser focus on highly complex shaped electronics
395 components and thin-walled capability, SEED has fewer applications in this area. The longer slurry
396 making sequence would also cause a lesser deviation from equilibrium ant, thus not reap the same
397 benefits of increase thermal conductivity as seen in the RheoMetal process, table 3.

398 In the medical component area, there are only parts produces by GISS as a lightweight
399 prosthetics focusing on internal soundness and weight without compromising part performance.

400 In the sports industry, bicycle components are the entry lever and motorbike components. All
401 processes are capable, but with the high-performance target of the SEED process, there are more
402 product examples for motorbike applications for SEED than for GISS and RheoMetal process. Here

403 the SEED process has seen applications in the most challenging high-performance application
404 through complex shape structural motorbike parts, table 3.

405 The RheoMetal process is the only process that has found application in machinery
406 manufacturing where the thick-walled capability is used in an application where a steel insert was
407 over-moulded, table 3.

408 **Table 3.** Collation of components in production.[23–26]

Application area	GISS	RheoMetal	SEED
Automotive	Auto gearbox		Brackets
	Brake system components	Compressor parts	Control arm
	Chain covers	Cooling units for power electronics	Engine bearing cap
	Engine block		Engine bracket
	Oil pan		Shock towers
	Steering wheels		Turbo impeller
Electronics	Handphone covers	Heat sinks	
	Hard disc drive housing	Radio filters 4G and 5G	Heat sinks
	Heat sinks		
	Radio filters 4G and 5G		
Heavy Duty Truck components		CAB mounts	Battery holder
	Truck gearbox	Muffler holders	Brake calliper
			Brackets
			Knuckle
			Skeleton joint
Machinery	-	Machine parts with steel inserts	-
Marine application	Sacrificial anode	Winch housing	-
Medical components	Prosthetics	-	-
Military components	Cast 7075 composite armour plate	-	-
Sports	Bicycle components		Motocross frame structural components (steering knuckle and others)
	Motorcycle parts	Bicycle components	Wheel knuckle (Quad)

409

410 5. Conclusions

411 The main conclusions drawn in this comparison can be made in the following areas

- 412
- 413 • Process differences
 - 414 • Process capabilities
 - 414 • Application areas

415 5.1. Process differences

416 The difference between the processes resides in the actual generation of the slurry particles as
417 this lays a foundation for the slurry characteristics. The first difference is that the GISS process
418 generates only 5% solid phase in the ladle and the rest is a fast, dynamic process in the shot sleeve
419 and die cavity while RheoMetal and SEED directly create more solid phase in the ladle. The
420 consequence is that GISS generates a material more similar to HPDC processed material. This
421 similarity also results in that the transition from HPDC to semisolid processing is relatively quick
422 and easy. The rich nucleation and rapid cooling support the creation of relatively well-rounded
423 particles.

424 RheoMetal and SEED generates a high fraction solid in the ladle/crucible, and the slurry
425 properties are changing less dynamically in the shot sleeve and are dominated by the conditions in
426 the ladle. The main difference is the time to process the slurry. RheoMetal typically takes 10-30
427 seconds to make the slurry while SEED takes 180s. The primary particles in the RheoMetal are
428 formed through kinetics, and the solid phase is far from equilibrium compared to the SEED process.
429 This makes the RheoMetal processed slurry particles lean in solutes that alters thermal conductivity
430 and heat treatment responses compared to what is seen in the SEED process. The longer processing
431 time for the SEED process allows the generation of more well-rounded particles, allowing SEED to
432 be operated at higher solid fractions than GISS and SEED.

433 *5.1. Process capabilities*

434 The process capability can be seen as 1) Shape capability, 2) Material properties capability 3)
435 Productivity capability

436 Compared to HPDC, all processes have improved shape capability and especially thin-walled
437 capability. In actual thin-walled capability, the RheoMetal process has reached furthest with 0.35mm
438 on a radio filter.

439 The materials property capability is a broad field and resides in both actual improvements in
440 the material properties as well as in alloy capability. GISS has the broadest range of proven
441 capability and is as such the most flexible in terms of choice. SEED has targeted high strength alloys
442 and is as such, the choice for strength if copper is acceptable as an alloying element.

443 The most detrimental influence on mechanical properties and part performance is porosity. All
444 processes deliver materials with improved soundness compared to HPDC Defects caused by gas
445 entrainment is reduced as the solid fraction of the slurry is increased. It should here be noted that
446 solid fraction is not the only factor dominating the entrainment defects, but other elements such as
447 the use of a vacuum during casting and the amount of release agent also have a strong influence on
448 the occurrence of entrainment defects. Shrinkage porosity is part geometry dependent, but the
449 presence of solid-phase will reduce the overall solidification shrinkage. It should also be noted that
450 just increasing the solid fraction may not always decrease porosity as solidification characteristic and
451 feeding resistance becomes important. Increase fraction will, however, always distribute porosity
452 and make pores smaller supporting increased fatigue resistance explaining the benefits seen in
453 RheoMetal and the SEED processes.

454 Productivity change compared to HPDC is challenging to measure, and the only bulk of data
455 existing originates from the GISS process where direct comparisons were made. Both cycle-time and
456 process yield improvements were realized. It is not clear if these can be realized to the same amounts
457 for the RheoMetal and SEED processes. This requires a reduction of the release agent usage.
458 Increasing the solid fraction means, however, that the heat received into the die is reduced and that
459 should reduce distortion and allow for a reduced release agent usage. In theory, this is possible but
460 convincing proof of actual achievement is missing. Die life data has a similar relationship to the
461 amount of solid fraction with an increasing fraction reducing the thermal load and as such, should
462 improve die-life. GISS has actual achievements recorded that can be compared. Again, in theory, this
463 is also possible for RheoMetal process and SEED process, but the actual convincing proof is still
464 missing.

465 *5.1. Application areas*

466 GISS has so far found the widest applications which likely is due to its ease of introduction. This
467 is also the likely reason for that there exist more direct comparisons between HPDC and GISS.
468 RheoMetal and SEED appear to be introduced to new projects where HPDC is unable to support the
469 requirements on the component.

470 GISS has been applied in almost all areas in terms of applications, but the dominant field is
471 within the electronics industry. This suggests that the main benefits of the GISS process reside in
472 productivity improvements and the capability to produce complex-shaped sound products
473 effectively. The immediate proven gains are found in rejection rates, cycle times, release agent usage

474 and foremost also in die-life improvements. The electronics industry also benefits from that the
 475 primary phase is somewhat lean in solutes and can provide improved thermal conductivity of the
 476 material compared to HPDC

477 The RheoMetal process has two main areas where it is applied, electronics and in the heavy
 478 truck component manufacturing. The electronics applications are found in China where it is the
 479 complex-shape capability together with the significantly improved thermal conductivity, due to the
 480 reduced amounts of dissolved solutes in the solid phase, that drives the application. The European
 481 heavy truck industry uses the reduced porosity to make heavy sectioned components competing
 482 with cast iron and forged aluminium component that are under fatigue load. These are in
 483 heat-treated conditions and often a T5 condition. This has taken some development effort in process
 484 control and timing. The T5 response in the RheoMetal process is more difficult compared to the
 485 other processes due to the reduced amount of solutes in the slurry phase and is entirely dominated
 486 by the Mg content in the slurry particles. [28]

487 The SEED process is capable of producing heatsinks just as GISS and RheoMetal, but due to the
 488 relatively high solid fraction used the extremely thin-wall capability has not been achieved. The
 489 longer processing time does not support the significant improvements in thermal conductivity seen
 490 in the GISS and RheoMetal process. The focus has been to draw benefit from the high solid fraction
 491 and to use this to cast high-performance alloys and to drive its implementation toward extremely
 492 demanding parts. These parts are often complex shape, relatively thin-walled and will often require
 493 T6 heat treatment. The achievable mechanical properties are higher than what is seen for GISS and
 494 RheoMetal using alloys such as B207.

495 **Funding:** This research was funded by the Knowledge Foundation, grant number 20170066.

496 **Acknowledgements:** The author is deeply indebted to Professor Jessada Wannasin with GISSCO, Dr Pascal
 497 Cote with STAS and to Dr Magnus Wessén, with the RheoMetal company as well as Per Jansson and Staffan
 498 Zetterström with Comptech and last but not least Mrs. Chen Qiurong with the RheoComp Technology
 499 company for providing information on applications and capabilities of their processes to make this work
 500 possible. This also includes providing the illustrations used in the current paper.

501 **Conflicts of Interest:** The authors declare no conflict of interest. All data were presented and evaluated without
 502 interference from the funding agency nor the process owners.

503

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