

**Supporting Information for**

**A facile synthesis of (PIM-polyimide)-(6FDA-durene-polyimide) copolymer as novel polymer membranes for CO<sub>2</sub> separation**

**Iqubal Hossain,<sup>1,2,3</sup> Abu Zafar Al Munsur<sup>1,2</sup> and Tae-Hyun Kim<sup>1,2,\*</sup>**

<sup>1</sup>*Organic Material Synthesis Laboratory, Department of Chemistry, Incheon National University, Incheon, 22012, Korea.*

<sup>2</sup>*Research Institute of Basic Sciences, Incheon National University, Incheon, 22012, Korea.*

<sup>3</sup>*Department of Chemistry, Ghent University, Sint-Pietersnieuwstraat 33,9000 Gent, Belgium*

**\*Corresponding author. E-mail address: tkim@inu.ac.kr (T.-H. Kim)**

## S-1. Materials

4,4'-(Hexafluoroisopropylidene) di-phthalic anhydride (6FDA), 2,3,5,6-tetramethyl benzene-1,4-diamine (durene), and 4,5-dichloro phthalonitrile (98%), were purchased from Tokyo Chemical Industry (TCI) Co. Ltd. (Tokyo, Japan) and were used as obtained. 5,5',6,6'-Tetrahydroxy-3,3,3',3'-tetramethyl-1,10-spirobisindane (97%) was obtained from Alfa Aesar. Acetic anhydride, toluene, and triethylamine were obtained from Sigma Aldrich. Methanol, ethanol, dimethylformamide, dimethylacetamide, potassium carbonate, and potassium hydroxide were purchased from DaeJung Chemicals & Metals Co. Ltd. in South Korea. 6FDA, durene, and  $K_2CO_3$  were dried under a vacuum at 60 °C for 24 h prior to use. Anhydride monomer of PIM (**An**) was synthesized following the literature method [00]. All chemicals, unless otherwise noted, were obtained from commercial sources and were used as received.

## S-2. Characterization and measurements

The **<sup>1</sup>H NMR** spectra were obtained on an Agilent 400-MR (400 MHz) instrument using *d*<sub>6</sub>-DMSO or CDCl<sub>3</sub> as a reference or an internal deuterium lock.

The attenuated total reflection Fourier transform infrared (**ATR-FTIR**) spectra were recorded using a Bruker Vertex 80v Hyperion 2000 ATR-FTIR spectrometer.

Molar masses were determined by gel permeation chromatography (GPC) using two PL Gel 30 cm × 5  $\mu$ m mixed C columns at 30 °C running in DMF and calibrated against polystyrene ( $M_n = 600 \times 10^6$  gmol<sup>-1</sup>) standards using a Knauer refractive index detector.

The thermal stability of the membranes was analyzed by thermogravimetric analysis (TGA) measurements conducted on a Shimadzu TGA-2950 instrument at a heating rate of  $10\text{ }^{\circ}\text{C min}^{-1}$  under a nitrogen flow.

The tensile properties were measured on a Shimadzu EZ-TEST E2-L instrument benchtop tensile tester using a crosshead speed of  $5\text{ mm}\cdot\text{min}^{-1}$  at  $25\text{ }^{\circ}\text{C}$  under 50% relative humidity. The engineering stress was calculated from the initial cross-sectional area of the sample and Young's modulus (E) was determined from the initial slope of the stress-strain curve. The membrane samples were cut into rectangular shapes  $40\text{ mm} \times 10\text{ mm}$  (total) and  $20\text{ mm} \times 10\text{ mm}$  (test area) in size.

The densities of the membranes ( $\text{g}\cdot\text{cm}^{-3}$ ) were determined experimentally using a top-loading electronic Mettler Toledo balance (XP205, Mettler-Toledo, Switzerland) coupled with a density kit based on the Archimedes principle. The samples were weighed in air and in a known-density liquid, high-purity heptane. The measurements were performed at room temperature using the buoyancy method, and the density was calculated as follows,

$$\rho_{\text{polymer}} = \frac{W_o}{W_o - W_l} \rho_{\text{liquid}}$$

where  $W_o$  and  $W_l$  are the membrane weights in air and in heptane respectively. Heptane sorption of the membranes was not considered due to their extremely low absorption properties.

The **X-ray** diffraction patterns of the membranes were measured using a Rigaku DMAX-2200H diffractometer operated at a scanning rate of  $4^{\circ}\text{ min}^{-1}$  in a  $2\theta$  range from  $5^{\circ}$  to  $30^{\circ}$  with Cu K $\alpha 1$  X-ray radiation ( $\lambda = 0.1540598$ ). The  $d$ -spacings were calculated using Bragg's law ( $d = \lambda/2\sin\theta$ ).

Tapping-mode **AFM** was conducted using a Bruker MultiMode instrument. A silicone cantilever with an end radius of <10 nm and a force constant of 40 Nm<sup>-1</sup> (NCHR, nanosensors, f=300 kHz) was used to image the samples at an ambient temperature.

### S-3. Gas permeation procedure

Permeation measurements of pure gas were taken using a high-vacuum time-lag measurement unit based on a constant-volume/variable-pressure method. All of the experiments were performed at a feed pressure of 2 bar (except for the pressure effect experiments which were carried out in the range of 100 mbar to 2 bar feed pressure) and a feed temperature of 30 °C. Before taking these measurements, both the feed and the permeate sides were thoroughly evacuated to below 10<sup>-5</sup> Torr (1.33×10<sup>-8</sup> bar) until the readout showed zero values for the removal of any residual gases. The downstream volume was calibrated using a Kapton membrane and was found to be 50 cm<sup>3</sup>. The upstream and downstream pressures were measured using a Baraton transducer (MKS; Model No. 626B02TBE) with a full scale of 10,000 and 2 Torr (13.3 and 2.7×10<sup>-3</sup> bar), respectively. The pressure on the permeate side was recorded as a function of time using a pressure transducer and passed to a desktop computer through a shielded data cable. The permeability coefficient was determined from the linear slope of the downstream pressure versus a time plot (dp/dt) according to the following equation,

$$P = \frac{273}{76} \times \frac{Vl}{ATp_0} \times \frac{dp}{dt} \quad (2)$$

where,  $P$  is the permeability expressed in Barrer (1 Barrer = 10<sup>-10</sup> [cm<sup>3</sup> (STP) cm.cm<sup>-2</sup>.s<sup>-1</sup>.cm<sup>-1</sup>.Hg<sup>-1</sup>]),  $V$  (cm<sup>3</sup>) is the downstream volume,  $l$  (cm) is the membrane

thickness,  $A$  ( $\text{cm}^2$ ) is the effective area of the membrane,  $T$  (K) is the measurement temperature,  $p_o$  (Torr) is the pressure of the feed gas in the upstream chamber, and  $dp/dt$  is the rate of the pressure change under a steady state. For each gas, the permeation tests were repeated more than three times, and the standard deviation from the mean values of the permeabilities was within ca.  $\pm 3\%$ . Sample-to-sample reproducibility was high and within  $\pm 3\%$ . The effective membrane areas were 15.9  $\text{cm}^2$ . The ideal permselectivity,  $\alpha_{A/B}$ , of the membrane for a pair of gases (A and B) is defined as the ratio of the individual gas permeability coefficients:

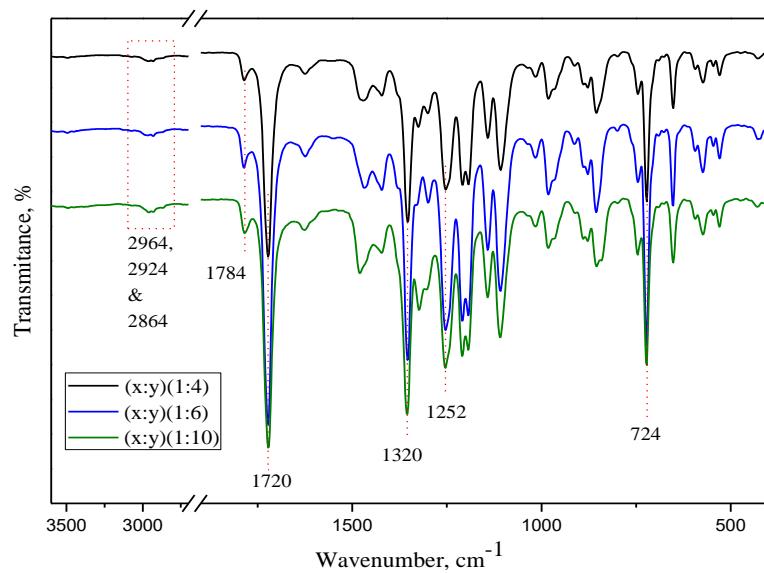
$$\alpha_{A/B} = \frac{P_A}{P_B} \quad (3)$$

The diffusivity and solubility were obtained from the time-lag ( $\theta$ ) value according to the equations

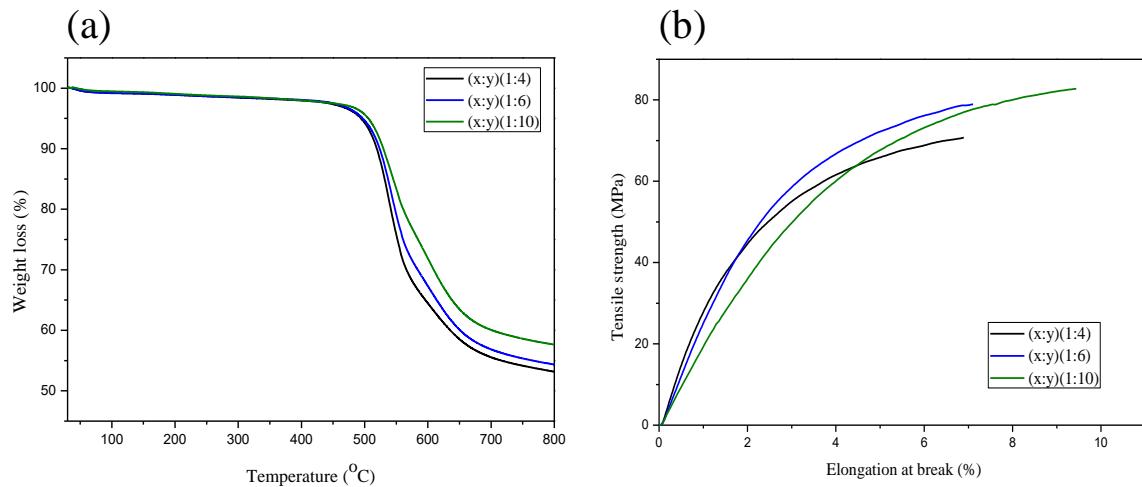
$$D = \frac{l^2}{6\theta} \quad (4)$$

$$S = \frac{P}{D} \quad (5)$$

where,  $D$  ( $\text{cm}^2 \text{ s}^{-1}$ ) is the diffusivity coefficient,  $l$  is the membrane thickness (cm) and  $\theta$  is the time lag (s), as obtained from the intercept of the linear steady-state part of the downstream pressure versus a time plot. The solubility,  $S$ , was calculated from Eqn. (5) with the permeability and diffusivity obtained from Eqns. (3) and (4).



**Figure S1.** ATR-FTIR spectra of the  $[(\text{PIM-PI})_x-(6\text{FDA-durene-PI})_y]$  copolymers with two different compositions ( $x:y = 1:4, 1:6$  and  $1:10$ )



**Figure S2.** TGA graph (a) and S-S curve (b) of the copolymer  $[(\text{PIM-PI})_x-(6\text{FDA-durene-PI})_y]$  membranes