

1 Article

2 Synthesis of solution-stable PEDOT-coated 3 sulfonated polystyrene copolymer 4 PEDOT:P(SS-co-St) particles for all-organic 5 NIR-shielding films

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12 **Abstract:** We prepared poly(3,4-ethylenedioxythiophene) (PEDOT)-coated sulfonated polystyrene
13 copolymer particles as efficient heat-shielding agents, which showed strong near-infrared (NIR)
14 absorption, with high solid contents and good solution stability. The poly(styrene sulfonate
15 -co-styrene) (P(SS-co-St)) copolymers were successfully synthesized via radical solution
16 polymerization, and PEDOT-coated P(SS-co-St) (PEDOT:P(SS-co-St)) was synthesized via
17 Fe⁺-catalyzed oxidative polymerization. PEDOT:P(SS-co-St) was characterized by nuclear magnetic
18 resonance and Fourier transform infrared spectroscopies. The particle size and morphology of
19 PEDOT:P(SS-co-St) were examined using transmission electron microscopy, dynamic light
20 scattering, and zeta potential measurements. The maximum NIR-shielding efficiency of the film
21 was 92.0% with 40% transmittance. The high solution stability of PEDOT:P(SS-co-St) make it an
22 ideal candidate for heat-insulating materials that find application in semi-transparent
23 heat-insulator-coated windows.

24 **Keywords:** Conducting polymer; Poly(3,4-ethylenedioxythiophene); PEDOT; Core-shell particles;
25 Polystyrene; NIR shielding film

26

27 1. Introduction

28 Energy crisis is one of the most urgent problems threatening the society, and extensive effort is
29 being devoted to improve energy efficiency and the responsible use of energy resources. Therefore,
30 recent research has focused on the application of green technology in the fields of architecture and
31 transportation to address these concerns. Semi-transparent heat insulation coatings are an essential
32 next-generation green technology. Importantly, semi-transparent heat-insulator-coated windows
33 serve distinct roles under different conditions. Transparent windows that are commonly used in
34 buildings and transportation readily permit bi-directional heat flow. The energy distribution in solar
35 irradiation consists of 5% in the ultraviolet range (UV; 280-400 nm), 45% in the visible range (400-780
36 nm), and 50% in the near-infrared range (NIR; 780-2700 nm). Heat-shielding windows can not only
37 block most of the NIR light from passing into a structure and keep it cool during summer, but also
38 reduce heat loss during winter. NIR protection is required in many applications, as NIR light causes
39 thermal aging and can lead to adverse health effects in humans [1,2]. Thus, the development of
40 infrared-shielding materials has implications in energy use reduction and human health.

41 Heat-insulating materials usually block heat transfer by absorbing or reflecting infrared (IR)
42 light. As NIR-shielding materials, conducting polymers (CPs), including conjugated polythiophene
43 (PT), polyaniline (PANI) [3], and polypyrrole (PPy) [4], have attracted significant attention in recent
44 years. Compared with some transparent thermal insulation coatings based on metals (such as Ni,

45 Ag, and Au), semiconducting oxides (such as tin-doped indium oxide (ITO) and antimony-doped tin
46 oxide (ATO)) [5,6], and tungsten oxides (M_xWO_3) [7], CPs are more promising for commercialization
47 as NIR-shielding films due to their excellent processability, high flexibility, and low cost. CPs are
48 frequently used in optoelectronic devices, including light emitting diodes (LEDs), solar cells, and
49 touch panel displays, as they are relatively inexpensive, lightweight, and easily processable [8-10].
50 The unique properties of CPs implicate them in a nearly unlimited range of potential applications.
51 CPs have long been studied in electrical and electronic applications for their ease of manufacture
52 and excellent physical and electrical properties [11-13]. Among them, CPs, especially
53 poly(3,4-ethylenedioxythiophene) : poly(styrene sulfonate) (PEDOT : PSS) has attracted a great deal
54 of attention in various applications such as transparent electrodes, solar cells, light-emitting diodes,
55 and thermoelectric devices for reducing the processing cost or replacing expensive oxides.

56 In this paper, we report the synthesis of PEDOT:P(SS-co-St) (PEDOT =
57 poly(3,4-ethylenedioxythiophene)) with a 50:50 molar ratio of poly(styrene sulfonate) to styrene for
58 use as a metal-free NIR-shielding material. The hydrophobic styrene was first incorporated into the
59 sodium salt of 4-styrenesulfonic acid (NaSS), yielding the amphiphilic copolymer P(SS-co-St).
60 P(SS-co-St) can be easily polymerized via its vinyl functional groups, as well as processed as a
61 monomer with styrene via radical polymerization and emulsifier-free copolymerization [14,15]. The
62 amphiphilic copolymer P(SS-co-St) was readily synthesized via polymerization in aqueous solution.
63 Then, the obtained P(SS-co-St) copolymer was used as a polymeric template in the oxidative
64 polymerization of 3,4-ethylenedioxythiophene (EDOT), and the conductive dispersion product
65 PEDOT:P(SS-co-St) exhibited colloidal stability and a high solid content. Furthermore, the
66 heat-shielding properties of PEDOT:P(SS-co-St) composite films were evaluated. The maximum
67 NIR-shielding efficiency of the film was 92.0% with 40% transmittance.

68 2. Materials and Methods

69 2.1. Materials for P(SS-co-St) and PEDOT:P(SS-co-St) Synthesis

70 4-Styrenesulfonic acid sodium salt hydrate (NaSS), styrene (St; 99%) and 3,
71 4-ethylenedioxythiophene (EDOT; 97%) were purchased from Sigma-Aldrich and were used
72 without further purification. Sodium persulfate (NaPS) and iron(III) sulfate pentahydrate
73 ($Fe_2(SO_4)_3 \cdot 5H_2O$) were obtained from Sigma-Aldrich and used without further purification as
74 reaction initiators. Double-distilled deionized water was used in all experiments.

75 2.2. Preparation of P(SS-co-St) copolymers

76 In the first step, a series of P(SS-co-St) copolymers containing one styrene per polymer chain
77 were prepared via radical copolymerization in the presence of the water-soluble initiator NaPS. The
78 styrene content in P(SS-co-St) was controlled by adding 50 mol% styrene to the
79 poly(styrenesulfonate) (PSS) polymers. Polymerization was performed in a four-necked
80 double-jacketed glass reactor (150 mL) equipped with a mechanical stirrer, and was carried out at 80
81 °C for 12 h. In the reactor, NaSS was dissolved in 100 mL of ultra-pure water (UPW) and stirred at
82 250 rpm. The initiator solution was prepared by dissolving 0.065 g NaPS in 2 mL UPW, and was
83 added to the reaction vessel within 15 min. After the reaction period, the solution-phase copolymer
84 product was ion-exchanged by mixing it with 31 g of positive ion exchangers for 1 h at room
85 temperature to remove unreacted initiator, and was then filtered through a 30 μm mesh.

86 2.3. Synthesis of PEDOT:P(SS-co-St)

87 The reactor was purged with nitrogen gas (N_2 ; 99.999%) and mixed at 300 rpm. The initiators,
88 0.772 g NaPS dissolved in 2 mL UPW and 0.0053 g $Fe_2(SO_4)_3$ dissolved in 3 mL UPW, were added to
89 the reactants. After adding the EDOT monomer, the color of the solution gradually changed from
90 yellow transparent to dark blue. The reaction was allowed to proceed for 24 h at 15 °C, with stirring
91 at 300 rpm. After approximately 24 h of polymerization, PEDOT:P(SS-co-St) was purified via ion

92 exchange by mixing the solution with positive and negative ion exchangers for 2 h at room
93 temperature, followed by filtering it through a 30 μm mesh.
94

95 2.4. Characterization of *P(SS-co-St)* and *PEDOT:P(SS-co-St)*

96 The chemical structures of PSS and *P(SS-co-St)* were analyzed by ^1H nuclear magnetic
97 resonance (NMR) spectroscopy (Biospin Avance II, Bruker) at 400 MHz and by Fourier-transform
98 infrared spectroscopy (FT-IR) (Vortex 70, Bruker) over the wavenumber range of 2000–400 cm^{-1} and
99 with a resolution of 2 cm^{-1} . The samples used for FT-IR and NMR were dried at 60 $^\circ\text{C}$ in a vacuum
100 oven. NMR analysis was performed on solutions of PSS and *P(SS-co-St)* diluted in DMSO- d_6 . FT-IR
101 analysis was performed using PSS and *P(SS-co-St)* in powder form mixed with KBr powder and
102 palletized. The FT-IR absorption spectrum was recorded in transmittance mode.

103 The particle size and size distribution were measured using dynamic laser scattering (DLS;
104 Zetasizer Nano S, Malvern Instruments). The morphology of the particles was observed using
105 transmission electron microscopy (TEM; JSM 100CXII, UHR, JEOL). The NIR-shielding films were
106 prepared by first filtering *PEDOT:P(SS-co-St)* through a nylon syringe filter (5.0 μm) to remove
107 impurities. The purified *PEDOT:P(SS-co-St)* solution was then directly coated on A4-sized PET
108 substrates using the Mayer rod coating technique. First, the *PEDOT:P(SS-co-St)* solution was
109 dropped onto the upper surface of the substrate, and the Mayer rod was quickly rolled over the
110 *PEDOT:P(SS-co-St)* solution. Finally, the wet film of *PEDOT:P(SS-co-St)* spread over the surface of
111 the substrate was dried at 150 $^\circ\text{C}$ for 2 min in a convection oven. Transmittance at 550 nm and
112 absorption between 350 and 2000 nm were measured using a UV-Vis-NIR spectrophotometer
113 (V-770, JASCO Corporation). Infrared bulbs (250 W, Philips), black boxes, and thermometers were
114 used to obtain temperature variation curves for the heat-shielding films. Briefly, a box measuring
115 300 mm x 210 mm x 150 mm made of black acrylate was prepared, and a thermometer was placed in
116 the black box and used to measure variations in temperature. NIR-shielding film was used to cover a
117 glass window on the upper side of the box. The distance between the black box and the infrared
118 lamp was 500 mm.

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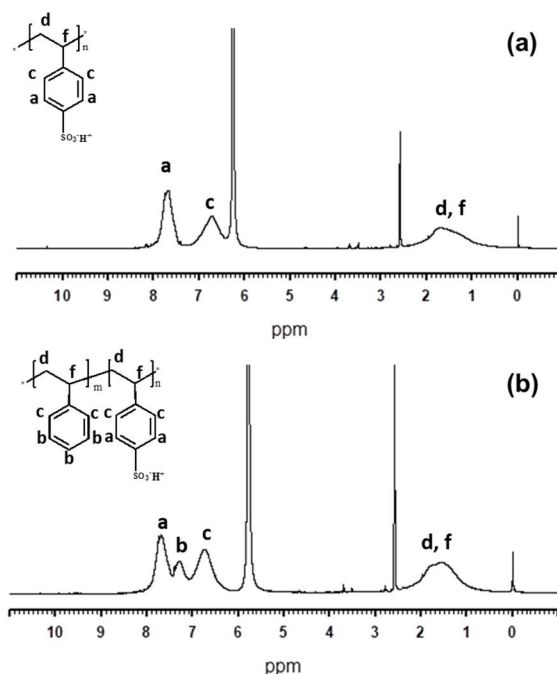
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132 **3. Results and Discussion**133 *3.1. Synthesis and characterization of P(SS-co-St) copolymers*

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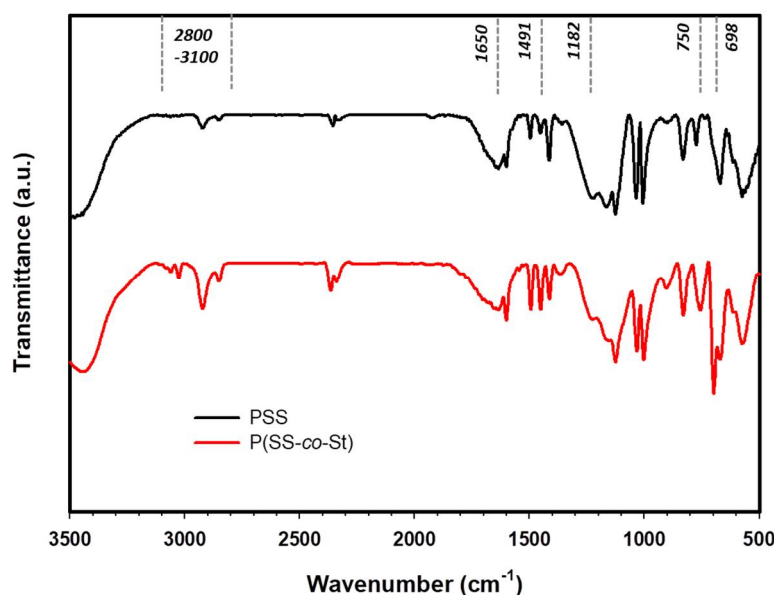


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Figure 1. ¹H NMR spectra of (a) PSS and (b) P(SS-co-St).

137 The hydrophobic styrene monomer influenced the water-solubility of PSS polymers in the
138 continuous phase. The solution of the synthesized polymer with added styrene was translucent, and
139 the originally hydrophilic polymer became amphiphilic upon addition of the styrene. The solubility
140 of the copolymers in water controlled the core-shell morphology, via the introduction of
141 hydrophobic styrene and hydrophilic NaSS monomers. Furthermore, the ¹H NMR spectra of the PSS
142 and P(SS-co-St) (Figure 1) were analyzed in order to determine the copolymerization mechanism.
143 The aromatic region of the spectra, along with the polymer structures (inset) and the peak
144 assignments, indicate that peak b (Figure 1) corresponds to the styrene in the copolymer [16,17]. The
145 effect of styrene incorporation into the PSS polymer backbone is thus clearly exhibited in the NMR
146 spectra. Analysis of the relative peak intensities of the NMR spectra of the polymers also provides
147 information on the reactivity of the monomers during copolymerization.



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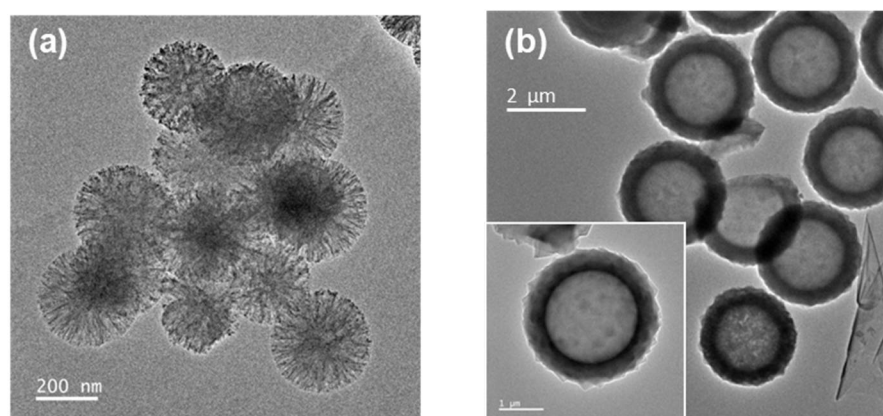
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Figure 2. FT-IR spectra of PSS and P(SS-co-St) polymers.

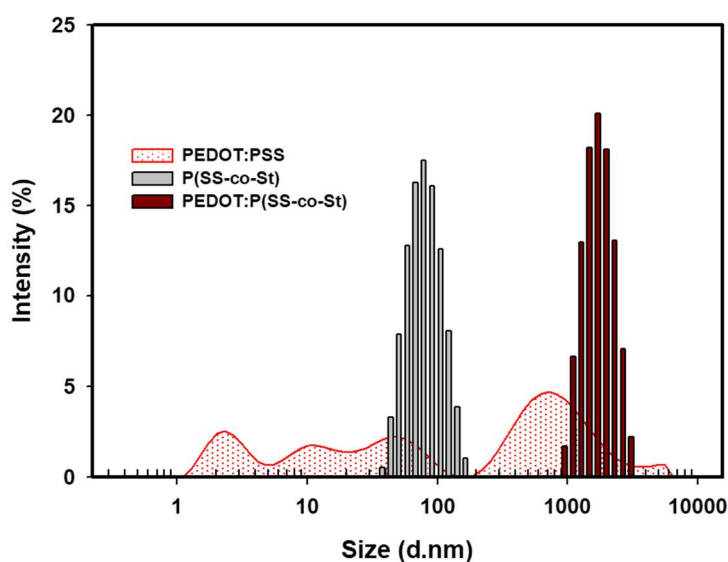
150 A comparison of the FT-IR spectra of the PSS and P(SS-co-St) polymers is shown in Figure 2.
151 The peaks in the 2800-3100 cm^{-1} region represent various vibration bands of polystyrene. The
152 characteristic vibration bands of polystyrene are present in the spectrum of the P(SS-co-St)
153 copolymer [16]. In addition, a previous study has reported that benzene ring stretching modes are
154 observed at 1491 cm^{-1} , out-of-plane ring deformation vibrations appear at 698 cm^{-1} , and that the peak
155 at 750 cm^{-1} is characteristic of the out-of-plane bending vibration of C-H groups in mono-substituted
156 benzene rings [18]. The peaks at 698 cm^{-1} , 750 cm^{-1} , and 1491 cm^{-1} increased in intensity as the styrene
157 was added to the polymerization. These observations provide evidence that the synthesized
158 P(SS-co-St) copolymer contained polystyrene. Both the spectra of PSS and P(SS-co-St) exhibited
159 vibration bands at 1650 cm^{-1} , which were assigned to water bound to the sulfonic groups of PSS. The
160 intensity of this peak was much lower in P(SS-co-St) compared to that in pristine PSS, implying a
161 relatively lower amount of absorbed water in the copolymer. The peak at 1182 cm^{-1} was assigned to
162 an asymmetric stretching vibration in $-\text{SO}_3$ resulting from the sulfonic acid hydrates ($-\text{SO}_3\cdot\text{H}_3\text{O}^+$)
163 [19]. However, the intensities of these two peaks 1650 cm^{-1} and 1182 cm^{-1} decreased when St was
164 added to the polymer. These aforementioned peaks were identifiable in the spectra of both PSS and
165 P(SS-co-St), meaning that the polymers contained $-\text{SO}_3$ groups with different ionization states.

166 The size and morphology of P(SS-co-St) and PEDOT:P(SS-co-St) particles were determined
167 from TEM images (Figures 3(a) and (b)). P(SS-co-St) consisted of spherical particles with size ranging
168 from 80 to 200 nm. Figure 3(b) shows a TEM image of PEDOT:P(SS-co-St) spheres; a uniform PEDOT
169 layer was observed on the surface of the P(SS-co-St) spheres. We obtained PEDOT:P(SS-co-St)
170 core-shell particles using sulfonated PS microspheres as templates in the reaction system. Compared
171 with the particle size shown in Figure 3(a) (200 nm), the size of the spherical PEDOT:P(SS-co-St)
172 particles increased to 2 μm , which is because the PEDOT polymerization occurred along the radial
173 direction from the surfaces of the P(SS-co-St) particles. This shows that the sulfonated PS
174 microspheres served as templates. As explained in a previous report [20], a PSS layer formed around
175 the PS spheres when PS template particles were first sulfonated with concentrated sulfuric acid. This
176 is because sulfonation increased the polarity of the surfaces of the PS spheres. Therefore, the surfaces
177 of the sulfonated PS microspheres exhibited negative charge when dispersed in water. After the
178 introduction of the EDOT monomer, the P(SS-co-St) spheres were surrounded by PEDOT⁺ through
179 electrostatic forces. PEDOT⁺ and the excess PSS then formed PEDOT:PSS around the PS spheres.

180 Therefore, PEDOT:P(SS-co-St) core-shell microspheres were obtained as a PEDOT:PSS layer covered
 181 the PS template.
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186 **Figure 3.** TEM images of (a) P(SS-co-St) and (b) PEDOT:P(SS-co-St); (c) DLS particle size distribution of
 187 PEDOT:PSS, P(SS-co-St), and PEDOT:P(SS-co-St).

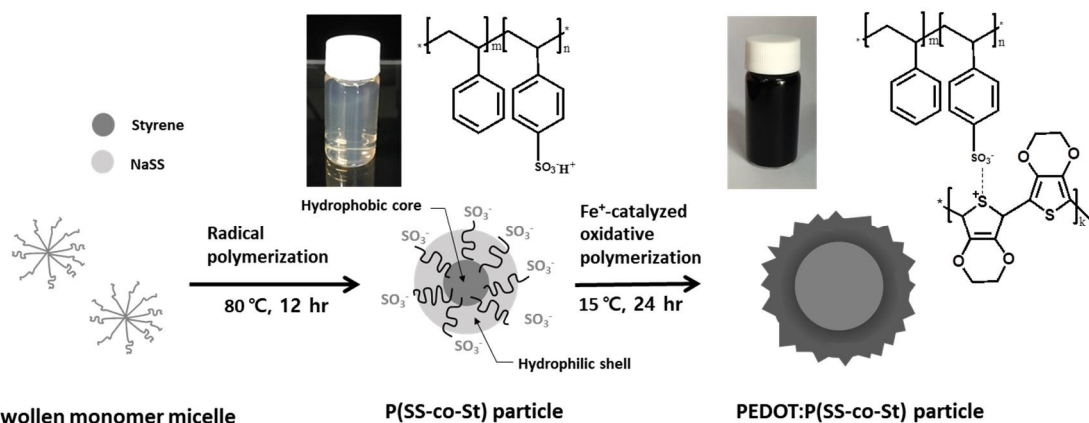
188 **Table 1.** Z-average particle diameter, polydispersity index (PDI), and zeta potential for P(SS-co-St),
 189 PEDOT:P(SS-co-St), and PEDOT:PSS

Material	Z-average d (nm)	PDI	Zeta potential (mV)
P(SS-co-St)	68.93	0.236	-74.9
PEDOT:P(SS-co-St)	1695	0.261	-70.4
PEDOT:PSS	56.84	0.726	-40.4

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191 Figure 3(c) shows the particle size and surface properties of the particles made of the various
 192 synthesized polymers in H₂O, as determined by DLS and zeta potential measurements [21]. The
 193 average diameter of P(SS-co-St) and PEDOT:P(SS-co-St) particles were 68.93 nm and 1695 nm,
 194 respectively. Moreover, the P(SS-co-St) and PEDOT:P(SS-co-St) particles exhibited zeta potentials of
 195 -74.9 mV and -70.4 mV, respectively. These results indicate that the surfaces of the P(SS-co-St) and
 196 PEDOT:P(SS-co-St) particles were rich in sulfonfyl groups [22]. On the other hand, PEDOT:PSS, with
 197 a solid content of 2%, exhibited an irregular particle size distribution and zeta potential of -40.4 mV,

198 indicating relative instability in solution. The result of DLS measurements of PEDOT:PSS indicated
 199 primary nanoparticles with traces of clusters. PEDOT:PSS with a solid content of 2%, which is higher
 200 than that of commercial PEDOT:PSS, can be predicted to exist in an unstable cohesive form in water
 201 due to its relatively low absolute Zeta potential. This indicates that PEDOT:PSS with a 2% solid
 202 content would exist in an unstable form one month after being synthesized. The first step in
 203 synthesis of PEDOT:P(SS-co-St) composites involved preparation of a stable emulsion from
 204 surface-sulfonated PS particles by self-assembly of PS particles at the water-monomer interface. The
 205 hydrophilicity of styrene and NaSS was adjusted by controlling the composition of the particles;
 206 therefore, $-\text{SO}_3\text{H}$ groups on the particle surfaces act as stabilizers for the template particles when
 207 dispersed in water [23-25]. The entire preparation process is illustrated in Scheme 1.
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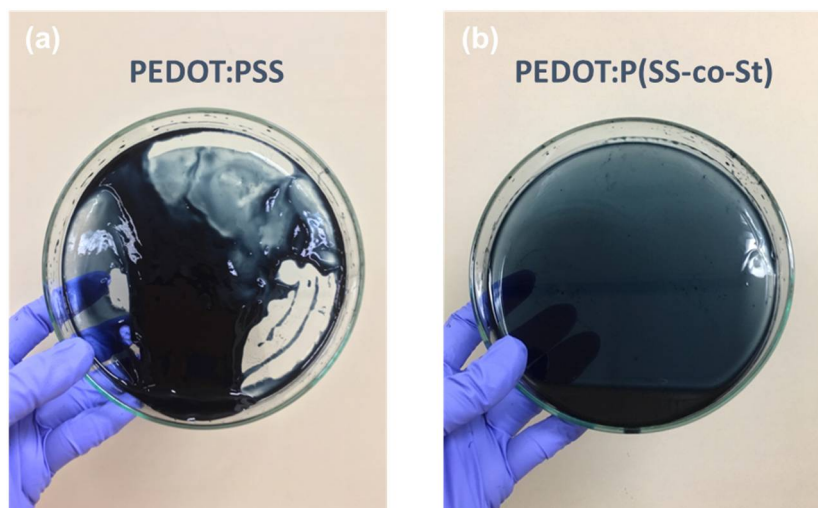
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Scheme 1. Schematic of the formation of P(SS-co-St) and PEDOT:P(SS-co-St).

211 The relatively higher solid content in the aqueous PEDOT:PSS in this study decreased its
 212 solution stability, resulting in high viscosity and gelation. For this reason, the solid content in
 213 commercial PEDOT:PSS (PEDOT to PSS ratio of approximately 1:2.5) is approximately 1.3%. The
 214 higher solid content requires an extremely well-controlled coating processes in order to guarantee
 215 high-quality film formation.

216 We synthesized CP dispersions under the same conditions with the PEDOT:PSS with a 2% solid
 217 content to compare the solution stabilities of PEDOT:PSS and PEDOT:P(SS-co-St). After the addition
 218 of 5% DMSO, the electrical conductivity of PEDOT:PSS was about 300 S/cm, whereas that of
 219 PEDOT:P(SS-co-St) was 250 S/cm, which is lower than that of the PEDOT:PSS. The
 220 high-solid-content PEDOT:PSS and PEDOT:P(SS-co-St) dispersions were stored for at least one
 221 month after synthesis, and the aged dispersions were then poured into Petri dishes to compare the
 222 stabilities of the dispersions; results are shown in Figure 4. The PEDOT:PSS dispersion partially
 223 gelled, while the PEDOT:P(SS-co-St) dispersion remained stable. The stability of
 224 PEDOT:P(SS-co-St) was thus remarkably enhanced due to its relatively high solid content (2%).
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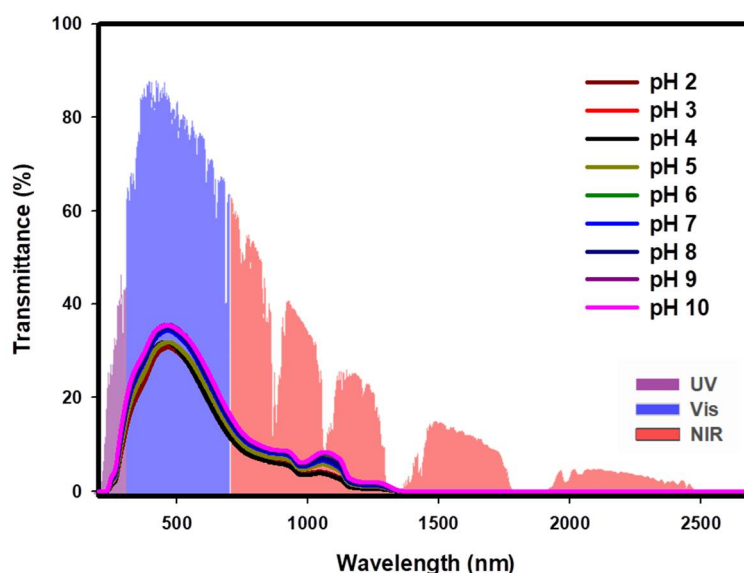
227 **Figure 4.** Solution-stability comparison of (a) pristine PEDOT:PSS and (b) PEDOT:P(SS-co-St).

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3.2. Heat-shielding efficiency of PEDOT:P(SS-co-St) NIR-shielding films

230 NIR-shielding films composed of organic materials are cost-effective and have simple
231 manufacturing processes. PEDOT has attracted much interest because of its high heat-shielding
232 efficiency, great thermal stability, and high transparency when used in thin films [26,27]. However,
233 like other conjugated polymers, PEDOT is insoluble in common organic solvents and water. In this
234 study, this was resolved by using PSS as a charge-balancing dopant during polymerization of EDOT,
235 and thus the resultant PEDOT:PSS dispersions were stable and had high conductivity [28].
236 However, PSS is a strong acid with a pH of less than 2, which adversely affects its lifetime and
237 performance in many applications. Neutralization of PEDOT:PSS through the use of a base like
238 NaOH is thus necessary to improve its lifetime and performance [27,29].

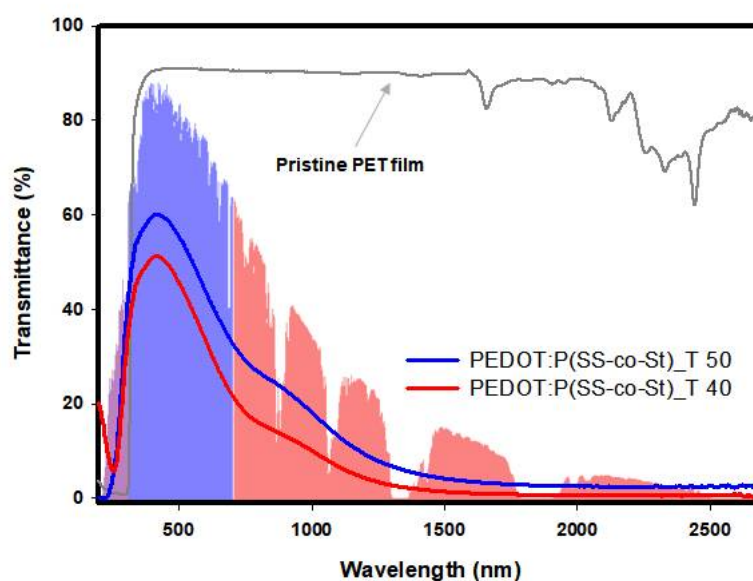


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240 **Figure 5.** Transmittance spectra of PEDOT:P(SS-co-St) solutions of different pH. The standard solar
241 irradiation spectrum is provided for comparison.

242 Yuta et al. suggested that the structure and electrical conductivity of PEDOT:PSS are dependent
243 on pH [30]. Therefore, the NIR-shielding capability of an aqueous PEDOT:P(SS-co-St) dispersion
244 depends on pH. The optical properties of the aqueous dispersions were examined using UV-vis-NIR

245 spectroscopy after adjusting the pH with 1 M NaOH (Figure 5). The NIR-shielding effect of
 246 PEDOT:P(SS-co-St) in solution was confirmed over a pH range from 2 to 10; there were no
 247 significant differences in the transmittance of the dispersions at different pH values. To determine
 248 the transmitted efficiency (TE) in the visible region (400–780 nm) and shielding efficiency (SE) in the
 249 NIR region (780–2700 nm), the standard solar spectrum is shown for comparison in Figures 5 and 6.
 250 As the concentration of NaOH in the aqueous solutions was increased, the PEDOT:P(SS-co-St)
 251 concentration was diluted. In addition, the transmittance in the NIR region slightly increased, to
 252 almost the same extent as NIR shielding efficiency. This confirmed that the NIR-blocking capacity
 253 remained almost constant with increasing pH. The neutral-pH PEDOT:P(SS-co-St) dispersion was
 254 thus considered suitable for use in practical applications.
 255



256

257 **Figure 6.** Transmittance spectra of PEDOT:P(SS-co-St) films with transmittance of 40% and 50% at
 258 550 nm. For comparison, the standard solar irradiation spectrum and the transmittance spectrum of
 259 the pristine PET film are provided.

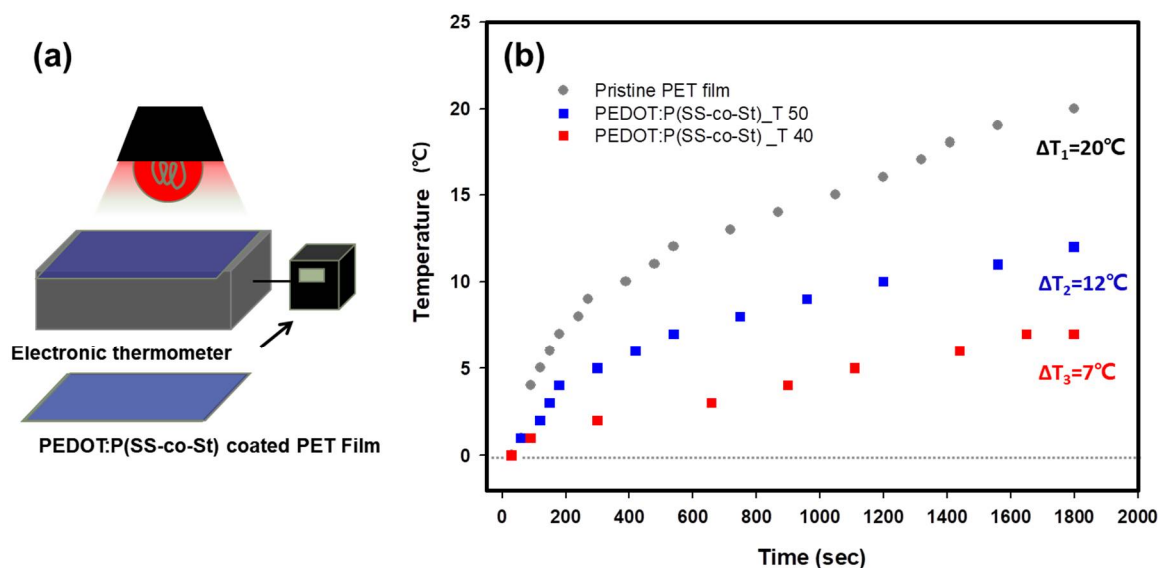
260 **Table 2.** Transmittance in the visible region and shielding efficiency in the NIR region of selected
 261 PEDOT:P(SS-co-St) films.

Sample	Transmittance (%) (at 550 nm)	NIR shielding efficiency (%) (780 – 1440 nm)
T 50	50.05	86.25
T 40	39.89	92.05

262

263 The NIR-shielding efficiency of the PEDOT:P(SS-co-St) films with different transmittance
 264 profiles were also examined by UV-vis-NIR measurements; the results are presented in Figure 6 and
 265 Table 2. We fabricated a heat-shielding film by varying the transmittance (40% and 50% at 550 nm)
 266 of the film, and confirmed the thermal efficiency of the films. The absorption at 789 nm is due to
 267 polaronic or bipolaronic absorption by the PEDOT shell [31,32]. As a result, the NIR-shielding effect
 268 of the PEDOT:P(SS-co-St) film depends on the film thickness. The thermal barrier effect improved as
 269 the transmittance at 550 nm was reduced from 50% to 40% and as the film thickness increased. In a
 270 previous study, we described the method for determining the shielding efficiency and presented the
 271 results [3]. Using this method, the NIR-shielding efficiency of the films were calculated to be 92.0%
 272 for the 40%-transmittance film and 86.2% for the 50%-transmittance film. Chen et al. previously
 273 reported that a Polypyrrole–Polyacrylic acid (PPy–PAA) film was able to block 92.1% of light in NIR

274 region, but transmitted 32.1% of visible light [4]. The PEDOT:P(SS-co-St) films exhibited good
 275 transparency in the visible region and strong absorption in the NIR region.



276

277 **Figure 7.** (a) Schematic illustration of the thermal insulation experiment, wherein a window in a
 278 sealed box was covered with a blank glass slide, commercial PET film, or PEDOT:P(SS-co-St)-coated
 279 PET Film. (b) Temperature changes of the air inside the box under the PET and
 280 PEDOT:P(SS-co-St)-coated PET films.

281 Based on the aforementioned optical properties of the PEDOT:P(SS-co-St) films, it is reasonable
 282 to suggest that these films show great potential for application as NIR-shielding films. Figure 7(a)
 283 illustrates the thermal insulation experiment which was conducted, wherein a window of a sealed
 284 box was covered with bare PET or PEDOT:P(SS-co-St)-coated PET films and placed under an IR
 285 lamp. Figure 7(b) shows the temperature variations inside the box after irradiation for 2000 s, and
 286 indicates the overall variation between the initial and final temperatures. When the box was covered
 287 with a glass slide and a bare PET film was placed over the glass, the temperature increased
 288 significantly as irradiation time progressed ($\Delta T_1 = 20^\circ\text{C}$). The bare PET film thus did not provide any
 289 NIR shielding. This configuration was thus taken as a reference for comparison with the synthesized
 290 films. In contrast, the heating rate in the box covered with the glass slide and a PEDOT:P(SS-co-St)
 291 film was much lower than in the reference configuration. Further, the heating rate inside the box
 292 decreased distinctly as the film thickness was increased. After irradiation for 2000 s, the overall
 293 temperature variation in the box covered with the PEDOT:P(SS-co-St) film with 50% transmittance
 294 (T 50 in Figure 7(b); $\Delta T_2 = 12^\circ\text{C}$), was lower than that for the reference configuration. Further, the
 295 temperature change in the box covered by the film with lower transmittance (40%; T 40) was even
 296 lower ($\Delta T_3 = 7^\circ\text{C}$). All PEDOT:P(SS-co-St) films were thus shown to provide good thermal
 297 insulation.

298 4. Conclusions

299 PEDOT:P(SS-co-St) dispersions were synthesized using a two-step process involving
 300 emulsifier-free emulsion copolymerization and oxidative polymerization. The PEDOT:P(SS-co-St)
 301 dispersions exhibited excellent multifunctional performance with high solid contents, stability, and
 302 NIR insulation capacity. The FT-IR and NMR results for PSS and P(SS-co-St) confirmed their
 303 successful polymerization. The size and morphology of the particles were determined via DLS
 304 measurements, zeta potential measurements, and inspection of TEM images. It was more difficult to
 305 stabilize the solution with higher solid content, which is an important factor in the coating process of

306 CP films. A comparison between the synthesized copolymers and existing CPs showed that the
307 limits of the existing CPs were exceeded by the polymers synthesized in this experiment. The
308 UV-vis-NIR measurements revealed the specific NIR-blocking capacity of the solutions as functions
309 of pH and film thickness. A PEDOT:P(SS-co-St)-coated PET film was used as the window on a sealed
310 black box and was illuminated with IR light. The interior air temperature increased to a smaller
311 extent (ΔT_2 and $\Delta T_3 = 12\text{ }^\circ\text{C}$ and $7\text{ }^\circ\text{C}$, respectively) than in the case wherein the window was covered
312 with a bare PET film ($\Delta T_1 = 20\text{ }^\circ\text{C}$). As a result of these properties, solution-stable PEDOT:P(SS-co-St)
313 copolymers have great potential for use as semi-transparent heat insulation coatings.

314 **Author Contributions:** Conceptualization: Wonseok Cho; Data curation: Soeun Im; Formal analysis: Soeun Im;
315 Investigation: Chanil Park, Jooyoung Kim, and Minseok Jeong; Supervision: Jung Hyun Kim; Visualization:
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323 References

- 324 1. Vierkötter, A.; Krutmann, J. Environmental influences on skin aging and ethnic-specific
325 manifestations. *Dermato-endocrinology* **2012**, *4*, 227-231.
- 326 2. Krutmann, J.; Morita, A.; Chung, J.H. Sun exposure: What molecular photodermatology tells us about
327 its good and bad sides. *Journal of Investigative Dermatology* **2012**, *132*, 976-984.
- 328 3. Park, C.; Im, S.; Cho, W.; Kim, Y.; Kim, J.H. Facile synthesis of p (edot/ani): Pss with enhanced heat
329 shielding efficiency via two-stage shot growth. *RSC Advances* **2018**, *8*, 12992-12998.
- 330 4. Chen, X.; Yu, N.; Zhang, L.; Liu, Z.; Wang, Z.; Chen, Z. Synthesis of polypyrrole nanoparticles for
331 constructing full-polymer uv/nir-shielding film. *RSC Advances* **2015**, *5*, 96888-96895.
- 332 5. Matsui, H.; Hasebe, T.; Hasuike, N.; Tabata, H. Plasmonic heat shielding in the infrared range using
333 oxide semiconductor nanoparticles based on sn-doped in₂o₃: Effect of size and interparticle gap. *ACS*
334 *Applied Nano Materials* **2018**, *1*, 1853-1862.
- 335 6. Miao, D.; Jiang, S.; Shang, S.; Chen, Z. Effect of heat treatment on infrared reflection property of
336 al-doped zno films. *Solar energy materials and solar cells* **2014**, *127*, 163-168.
- 337 7. Liu, T.; Liu, B.; Wang, J.; Yang, L.; Ma, X.; Li, H.; Zhang, Y.; Yin, S.; Sato, T.; Sekino, T. Smart window
338 coating based on f-tio 2-k x wo 3 nanocomposites with heat shielding, ultraviolet isolating, hydrophilic
339 and photocatalytic performance. *Scientific reports* **2016**, *6*, 27373.
- 340 8. Im, S.; Kim, W.; Cho, W.; Shin, D.; Chun, D.H.; Rhee, R.; Kim, J.K.; Yi, Y.; Park, J.H.; Kim, J.H.
341 Improved stability of interfacial energy-level alignment in inverted planar perovskite solar cells. *ACS*
342 *applied materials & interfaces* **2018**.
- 343 9. Cho, W.; Hong, J.K.; Lee, J.J.; Kim, S.; Kim, S.; Im, S.; Yoo, D.; Kim, J.H. Synthesis of conductive and
344 transparent pedot: P (ss-co-pegma) with excellent water-, weather-, and chemical-stabilities for organic
345 solar cells. *RSC Advances* **2016**, *6*, 63296-63303.
- 346 10. Kim, W.; Kim, S.; Chai, S.U.; Jung, M.S.; Nam, J.K.; Kim, J.-H.; Park, J.H. Thermodynamically
347 self-organized hole transport layers for high-efficiency inverted-planar perovskite solar cells. *Nanoscale*
348 **2017**, *9*, 12677-12683.

- 349 11. Kim, S.; Kim, S.Y.; Kim, J.; Kim, J.H. Highly reliable agnw/pedot: Pss hybrid films: Efficient methods
350 for enhancing transparency and lowering resistance and haziness. *Journal of Materials Chemistry C* **2014**,
351 2, 5636-5643.
- 352 12. Cho, W.; Im, S.; Kim, S.; Kim, S.; Kim, J.H. Synthesis and characterization of pedot: P (ss-co-vtms) with
353 hydrophobic properties and excellent thermal stability. *Polymers* **2016**, 8, 189.
- 354 13. Kim, Y.; Cho, W.; Kim, Y.; Cho, H.; Kim, J.H. Electrical characteristics of heterogeneous polymer layers
355 in pedot: Pss films. *Journal of Materials Chemistry C* **2018**, 6, 8906-8913.
- 356 14. Kim, J.; Chainey, M.I.; El-Aasser, M.; Vanderhoff, J. Emulsifier-free emulsion copolymerization of
357 styrene and sodium styrene sulfonate. *Journal of Polymer Science Part A: Polymer Chemistry* **1992**, 30,
358 171-183.
- 359 15. Sen, A.K.; Roy, S.; Juvekar, V.A. Effect of structure on solution and interfacial properties of sodium
360 polystyrene sulfonate (napss). *Polymer international* **2007**, 56, 167-174.
- 361 16. Arunbabu, D.; Sanga, Z.; Seenimeera, K.M.; Jana, T. Emulsion copolymerization of styrene and sodium
362 styrene sulfonate: Kinetics, monomer reactivity ratios and copolymer properties. *Polymer International*
363 **2009**, 58, 88-96.
- 364 17. Lee, D.K.; Park, J.T.; Roh, D.K.; Min, B.R.; Kim, J.H. Synthesis of crosslinked polystyrene-b-poly
365 (hydroxyethyl methacrylate)-b-poly (styrene sulfonic acid) triblock copolymer for electrolyte
366 membranes. *Macromolecular research* **2009**, 17, 325-331.
- 367 18. Kim, S.H.; Kim, J.H.; Choi, H.J.; Park, J. Pickering emulsion polymerized poly (3,
368 4-ethylenedioxythiophene): Poly (styrenesulfonate)/polystyrene composite particles and their electric
369 stimuli-response. *RSC Advances* **2015**, 5, 72387-72393.
- 370 19. Brijmohan, S.B.; Swier, S.; Weiss, R.; Shaw, M.T. Synthesis and characterization of cross-linked
371 sulfonated polystyrene nanoparticles. *Industrial & engineering chemistry research* **2005**, 44, 8039-8045.
- 372 20. Zhu, C.; Chen, Z.; Ni, C.; Yu, J.; Huang, B.; Shan, M. Degradation of sodium polystyrene sulfonate and
373 the radical initiated polymerization of styrene under ultrasonic irradiation. *Polymer-Plastics Technology*
374 *and Engineering* **2011**, 50, 1262-1265.
- 375 21. Ryu, H.W.; Kim, Y.S.; Kim, J.H.; Cheong, I.W. Direct synthetic route for water-dispersible
376 polythiophene nanoparticles via surfactant-free oxidative polymerization. *Polymer* **2014**, 55, 806-812.
- 377 22. Horikawa, M.; Fujiki, T.; Shirotsaki, T.; Ryu, N.; Sakurai, H.; Nagaoka, S.; Ihara, H. The development of
378 a highly conductive pedot system by doping with partially crystalline sulfated cellulose and its electric
379 conductivity. *Journal of Materials Chemistry C* **2015**, 3, 8881-8887.
- 380 23. Yao, Q.; Wilkie, C.A. Thermal degradation of blends of polystyrene and poly (sodium
381 4-styrenesulfonate) and the copolymer, poly (styrene-co-sodium 4-styrenesulfonate). *Polymer*
382 *degradation and stability* **1999**, 66, 379-384.
- 383 24. Fan, X.; Niu, L.; Xia, Z. Preparation of raspberry-like silica microcapsules via sulfonated polystyrene
384 template and aniline medium assembly method. *Colloid and Polymer Science* **2014**, 292, 3251-3259.
- 385 25. Lee, S.M.; Lee, S.J.; Kim, J.H.; Cheong, I.W. Synthesis of polystyrene/polythiophene core/shell
386 nanoparticles by dual initiation. *Polymer* **2011**, 52, 4227-4234.
- 387 26. Chang-Jian, C.-W.; Cho, E.-C.; Yen, S.-C.; Ho, B.-C.; Lee, K.-C.; Huang, J.-H.; Hsiao, Y.-S. Facile
388 preparation of wo3/pedot: Pss composite for inkjet printed electrochromic window and its
389 performance for heat shielding. *Dyes and Pigments* **2018**, 148, 465-473.
- 390 27. Yoon, H.J.; Jeong, S.Y.; Lee, S.; jic Shin, G.; Choi, K.H. Near infrared shielding properties of pedot: Pss
391 with different additives.

- 392 28. Kim, J.; Jung, J.; Lee, D.; Joo, J. Enhancement of electrical conductivity of poly (3,
393 4-ethylenedioxythiophene)/poly (4-styrenesulfonate) by a change of solvents. *Synthetic Metals* **2002**,
394 126, 311-316.
- 395 29. Cho, A.; Kim, S.; Kim, S.; Cho, W.; Park, C.; Kim, F.S.; Kim, J.H. Influence of imidazole-based acidity
396 control of pedot: Pss on its electrical properties and environmental stability. *Journal of Polymer Science*
397 *Part B: Polymer Physics* **2016**, 54, 1530-1536.
- 398 30. Mochizuki, Y.; Horii, T.; Okuzaki, H. Effect of ph on structure and conductivity of pedot/pss.
399 *Transactions of the Materials Research Society of Japan* **2012**, 37, 307-310.
- 400 31. Han, M.G.; Foulger, S.H. Crystalline colloidal arrays composed of poly (3, 4-ethylenedioxythiophene)-
401 coated polystyrene particles with a stop band in the visible regime. *Advanced Materials* **2004**, 16,
402 231-234.
- 403 32. Han, M.G.; Foulger, S.H. Preparation of poly (3, 4-ethylenedioxythiophene)(pedot) coated silica core-
404 shell particles and pedot hollow particles. *Chemical Communications* **2004**, 2154-2155.
- 405
- 406