Inorganic and Hybrid Perovskite Based Laser Devices: A Review

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Abstract: Inorganic and organic – inorganic (hybrid) perovskite semiconductor materials have attracted the worldwide scientific attention and research effort as the new wonder material in optoelectronics. Their excellent physical and electronic properties have been exploited to boost the solar cells efficiency beyond 23% and captivate their potential as competitors to the dominant silicon solar cells technology. However, the fundamental principles in Physics dictate that an excellent material for photovoltaic applications must be also an excellent light emitter. This has been realized for the case of perovskite based light emitting diodes (LEDs) but much less for the respective laser devices. Here, the strides have been made since 2014 are presented for the first time. The solution processability, low temperature crystallization, formation of nearly defect free, nanostructures, the long range ambipolar transport, the direct energy band gap, the high spectral emission tunability over the entire visible spectrum and the almost 100% external luminescence efficiency show perovskite semiconductors’ potential to transform the nanophotonics sector. The operational principles, the various adopted material and laser configurations along the future challenges are reviewed and presented in this paper.

Keywords: inorganic perovskites; hybrid perovskites; stimulated emission; laser devices

1. Introduction

Low temperature solution processed materials that exhibit (a) high optical and tunable absorption; (b) high optical gain; (c) high crystal quality; and (d) the ability to be integrated into various optical resonator configurations, are features very attractive for a primary gain medium on chip laser devices and applications. The perovskite materials, even though perovskites, have been discovered in 1839, have attracted the last ten years the scientific community’s attention and research efforts, thanks to these materials’ outstanding physical and electronic properties in combination of the ease of their processing. Perovskites are the new wonder semiconductor materials in solar cells. Perovskite based solar cells (PSCs) have skyrocketed their efficiency beyond 23% within a record period of time [1-6]. In 1994, researchers at IBM T. J. Watson Research Center (Yorktown Heights, N.Y.) developed light-emitting devices based on luminescent organic-inorganic halide perovskites, and in 1996 Boeing North America (Seal Beach, Calif.) created nonlinear optical crystals based on cesium-germanium halide salts. Many of the intriguing electronic and optical properties of halide perovskites such as high optical absorption coefficient, direct energy band gap, large oscillator...
strength, long carrier lifetime and high quantum efficiency are particular attractive for light emitting devices. What is more, the key parameter that works as a figure of merit for high performance photon emitter, the photoluminescence quantum yield (PLQY – the ratio of the emitted photons to those absorbed) of metal halide perovskites has significantly enhanced, reaching average values of 40% and 90% in bulk thin films [7,8] and nanoparticles (NPs) [9,10], respectively. Moreover, the functionality through controlling the compositional constitution of the perovskite semiconductor’s energy band gap, allows the coverage of the full spectrum using these materials as active layers in emitting devices. The solution processability of the hybrid or inorganic perovskite materials permits the fabrication of devices in large scale under lower cost, compared to the respective values the convectional fabrication methods, such as chemical vapor deposition resulted. Lasers are the youngest member of the family of perovskite optoelectronic devices. Perovskite lasers have entered the stage hand in hand with the vision to create low-cost, solution processed laser diodes. To highlight their potential as a gain medium within a laser system, it must be mentioned their lower amplified stimulated emission (ASE) carrier density (how easy a material can acquire net gain) threshold by one order of magnitude compared to those of (a) highly crystalline and high temperature grown ZnSe and CdS; and (b) this one of the solution processed organic semiconductors [11]. This figure of merit provides a strong impetus for optoelectronic applications using perovskite lasers. The low threshold values are attributed to the lower number of non-radiative pathways (directly linked to the existence of surface and core defects) compared to the other solution processed materials e.g. organic semiconductors. Another attractive feature the perovskite materials exhibit, is their functionality to tune their laser emission wavelengths over the entire visible range (390 – 790 nm) either (a) by modulating the halide constitution (controllable stoichiometry) [12]; (b) by varying the lead iodide concentration precursor during the perovskite fabrication process [13]; and (c) by varying or combining the cation element (e.g. using methylammonium (MA) or formamidinium (FA) or using both of them) [14]. Lasing in perovskites can be secured using various cavity/optical feedback architectures: (a) Distributed Feedback provided by gratings [15]; (b) Fabry – Perot resonators formed between the facets of the perovskite thin films [16]; (c) Scattering (in random lasing) provided by various grain boundaries [17]; (d) Whispering Galley Mode (WGM) cavities, exploiting total internal reflection [18]; and (e) polaritonic based laser devices [19]. Perovskite lasers have been demonstrated using a broad range of materials’ nanostructures such as microplatelets, nanowires, photonic crystal corrugations nanodots, nanocubes, microdiscs etc. All the important results are summarized onto tables that contain for each perovskite material all the important figures of merit: Lasing Threshold, Spectral Linewidth, Tunability, Q-factor, Gain coefficient. These tables will help the reader to observe the progress of the field, compare and view the characteristics of all the demonstrated perovskite laser devices.

Primary evidences of the lasing onset [20] include (a) a kink in the received optical output as a function of the optical pumping intensity; (b) a spectral shrinkage of the output signal; (c) a faster time resolved photoluminescent (TRPL) decaying due to the stimulated emission. The hybrid organic – inorganic perovskites combine the solution processability lend by their organic compound and at the same time the high crystallinity due to their inorganic part. However, the existence of the organic part is the main source of these materials’ instability under their exposure into the environmental hazards. This challenge is addressed with the employment of the inorganic perovskites that demonstrate superior stability under lasing operational conditions compared to their hybrid perovskites counterparts. The inorganic and hybrid perovskite are crystallized in temperatures compatible with flexible substrates (below 200 °C), characteristic that makes them compatible with the emerging printing electronics Industry.

In this review paper, which is the first attempt to concentrate all the published results in this topic, we cover the strides have made in the field and provide an insight in the major research effort has been launched since 2014 regarding the application of inorganic and hybrid perovskite materials, as a gain media in laser devices. The advantages the perovskite low dimensional (e.g. nanoplatelets, nanocubes, quantum dots and nanowires) configuration offer, like the formation of natural resonant cavities, are precious for the advancements of nanophotonics. The much less research work has been done in the field of perovskite-based laser devices (compared with the publication explosion in the
field of solar cells) was a surprise to us since physics implies that an excellent material for photovoltaic applications must also be an excellent light emitter (Figure 1). At the same time, this is an open window for more research to be implemented in this promising field. This paper attempts to disseminate and highlight perovskite semiconductors’ high potential as laser gain media through the challenges that field sets. The paper consists from three sections: (a) Hybrid Perovskite based laser systems; (b) Inorganic Perovskite based laser systems; and (c) Challenges and Proposed solutions to advance this very dynamic and with a high impact research topic.

Figure 1. Bar-plot showing the increasing number of published articles (red) and total citations (blue) containing the expressions “Halide Perovskite” and “Laser or Lasing” either in the title or in the abstract during the period 2013 and 2019. Source: Scopus bibliographic database (January 2019).

2. Literature Review Section

2.1. Hybrid Inorganic – Organic Perovskite Laser Devices

F. Deschler et al. [21] were the first ones that demonstrated the emission of stimulated light from a vertical cavity laser configuration that enclosed between its two mirrors a solution processed methylammonium lead mixed halide film (Figure 2a). The recorded photoluminescent quantum efficiency (PLQE) of the optically pumped films under high pumping densities (> 100 mW/cm²) was surprisingly high (~ 70%) showing the existence of very few nonradiative recombination pathways. Lasing operation at room temperature, for optical pumping energy above 0.2 μJ/pulse (using ns laser systems at 532 nm), was observed; it was reconfirmed through an order of magnitude increase in the slope of the output curve as a function of the pumping power.

Microdisc lasers (MDLs) (Figure 2a) are convenient laser cavity configurations since are characterized by (a) small mode volume; (b) high Q-cavity, that allow them to be easily embedded into nanophotonic circuits. Their operation is based on the whispering gallery mode (WGM) type resonator (Figure 2b-d), where the optical feedback is secured through the total internal reflection (TIR) along the perimeter of the disc. Q. Liao et al. [22] applied the principles of MDLs using as a gain medium hybrid inorganic – organic perovskites (MAPbI\(_3\)). The authors showed, a single longitudinal mode operation and tuning of the lasing emission wavelength achieved through the partial replacement of Br with Cl atoms; managing a tuning range from 525 nm to 557 nm (with decreasing x from one to zero in CH\(_3\)NH\(_3\)PbCl\(_x\)Br\(_{3-x}\) crystals). However, the PLQE was getting lower as the x increased (e.g. PLQE ~ 1% at x = 1 whereas PLQE~22% at x = 0). This was attributed to the increased number of traps as the x was approaching the unity value.
The demonstration of lasing from single crystal halide perovskite nanowires, provided nanolaser systems with strong photon confinement, excellent waveguide properties, high PLQE (~100%) and high Q-cavity factors (~3600). The demonstrated low threshold (~220 nJ/cm²) (Figure 3a), was a real breakthrough in the field of semiconductor NWs since it was the main obstacle for their wide use application and the introduction of electrically driven lasers. H. Zhu et al. [23] work showed that single crystal perovskite NWs (~20 μm length and with hundreds of nm in width) are ideal laser sources for the nanophotonic related Industry. A MAPbI₃ excellent crystal quality allowed the NWs two end facets to form a Fabry–Perot (FP) type of resonator, necessary for the building of the stimulated light at room temperature. The remarkable laser operation performance of the MAPbI₃ NWs was also confirmed from TRPL measurements; the lifetime of the spontaneous emission decreased above threshold from 150 ns to 5.5 ns as a result of stimulated emission (Figure 3b). Single and multimode operation was observed with the distance between the adjacent modes to decreased with the NWs length. The lasing output from such structures was observed to be linear polarized (degree of polarization (DOP) ~ 90%) with high polarization purity; the electric field oscillated in the plane (TM) of the NWs.

A step towards the realization of practical hybrid inorganic – organic halide perovskite semiconductor lasers was done by S. Chen et al. [24]. The authors of this work embedded a solution processed thin film (~130 nm the thickness) of polycrystalline CH₃NH₃PbI₃ between the facets of a two-dimensional photonic crystal in order to realize a photonic crystal (PhC) based microlaser. High degree of single mode operation, temporal and spatial coherence were observed during the lasing operation of the proposed set up. The excellent crystal quality of the solution processed utilized perovskite films secured excellent optical and electrical properties essential for pursuing low threshold laser devices. The practicability of the exhibited system was also confirmed from the pumping source; the authors employed ps laser pulses (lower cost devices than the fs ones) to create the necessary population inversion (optical gain) for lasing. Beyond the threshold (~68 μJ/cm²) the lifetime of the spontaneous emission was shortened by a 50-fold, from ns to ps range as a clear indication of a gain switch to the lasing regime. The degree of polarization (DOP) was close to 70% for pumping above the lasing threshold point. Tuning the output laser wavelength (for more than 30 nm) under fixed composition, was also achieved by controlling the PhC’s pitch periodicity. The fact that the laser operational conditions induce thermal stress of the perovskite active layers, two to three orders of magnitude higher compared to corresponding values in solar cells, convert laser-based devices an excellent platform for photostability measurements.

The fabrication of nanolasers based on the nonlinear properties of the hybrid perovskite semiconductors was realised by Z. Gu et al. [25]. Two photon optically pumped laser devices are attractive since can be placed within complex environments e.g. biological tissues and have a great impact on nonlinear optics; imaging, optical limiters and nanolasers due to the high resolution they
can provide. Z. Gu et al. work was the 1st one that demonstrated the two-photon absorption and lasing emission from microwire (10-30 μm in length) hybrid organic – inorganic perovskite semiconductors (MAPbBr3).

Auger recombination mechanism is one of the detrimental factors to achieve low threshold lasing values. This recombination mechanism is extremely strong under high optical pumping rates. Despite the fact of the work has been done in the field of perovskite lasers, the strong optical excitation power regime has not studied in depth. C. Zhang et al. [26] were the first ones that investigated this pumping regime. Moreover, they suggested that the incorporation of a thin layer of graphene next to the hybrid perovskite nanorods operated as an electron quencher; as a result, the holes and electrons were diffused in different regions. This assisted to avoid secondary absorption mechanisms that elevated the lasing threshold. The inclusion of a graphene layer on top of the deposition substrate the CH3NH3PbBr3 micro rods were deposited, resulted the reduction of lasing threshold (reduction by 20%) and the increase by four times of the output intensity under high pumping powers. This lasing operation improvement was attributed to the lowering of the Auger recombination rate due to the electron attractive nature of the graphene layer.

Realization of highly periodic perovskite laser arrays with variable cavity sizes, for large area optical gain applications, was a challenge regarding the demonstration of coherent multi-laser source for optical communication purposes. X. Liu et al. [27] managed to show direct lithographic patterning of perovskite based optical cavities onto traditional silicon substrates for optoelectronic applications, a task that prior to this work was requested complicated lithographic techniques. The authors of this work deposited onto h-BN/SiO2 substrates, MAPbI3 platelets of various lengths. The insulating h-BN (operated in this work as a buffer layer) was periodically patterned, deposited onto SiO2 and continuously the perovskite semiconductor based optical cavities nucleated onto this h-BN/SiO2 compound. The MAPbI3: platelet - arrays were optically pumped using fs laser pulses at 400 nm. The threshold level was measured at 11 μJ/cm2 (emission at 780 nm). TRPL measurements also indicated a fast recombination beyond threshold due to the lasing operation (where the lifetime of spontaneous emission reduced to ~ 30 ps). Single laser mode and Free Spectral Range (FSR) tunability were attainable through the modulation of the platelets’ size. Moreover, the authors realised a dependence of the achieved lasing thresholds as a function of the cavity length; as the platelet size increased the threshold decreased. This was attributed to the higher Q factors (better coupling of the spontaneous emission into the stimulated emission) achieved for the longer platelets. From the other hand and again due to FSR increase, single longitudinal mode operation was achieved for a 2 μm cavity. To overcome the high threshold values the short cavities suffered, the authors designed an asymmetrical shaped cavity. In this case the lasing threshold for single mode operation was lowered from 37 μJ/cm2 to 12 μJ/cm2 (asymmetrical shaped cavity).

One of the challenges the hybrid organic – inorganic perovskite semiconductors face is that are susceptible to environmental hazards; this is mainly due to the organic part of this composite materials. In laser-based devices this issue is mainly addressed with the employment of inorganic perovskite materials, mainly Cs based. P. Brenner et al. [28] motivated by the need of low cost, stable and tunable laser sources, proposed something different and innovative: the utilization of MAPbI3 distributed feedback (DFB) systems as stable and tunable laser source. The authors studied the lasing performance of a perovskite DFB lasers, fabricated in ambient atmosphere, directly on top of low-cost polymer gratings. Stable laser operation (of encapsulated devices) for 15 hrs under optical pumping (using ns DPSSL at 532 nm, 270 kW/cm2) without any obvious change of the emission wavelength (at 786.5 nm) was observed. This stability was considerably superior to other tunable active materials, such as organic semiconductors. While the laser threshold value was measured at 120 kW/cm2. Tunability of the laser wavelength from 781 to 794 nm was also demonstrated through modification of the grating’s period.

Random lasing scheme is very attractive for its low cost, simple processing schemes and the freedom to be realised without the use of any artificial designed resonant cavity. Random lasing light source with credentials to be infiltrated into biological tissues and to serve in pinpoint detection, has been rarely explored yet. Random lasing exploits the scattering effects in order to provide the necessary optical feedback and thus the optical gain. The emission of the induced coherent radiation
occurs in all the directions and thus this is another concrete difference with the formally defined directional laser operation. In perovskite polycrystalline materials the scattering is offered by the various interfaces formed between the different grains formed during the crystallization of the material. Z.F. Shi et al. [29] managed to demonstrate random lasing using CH$_3$NH$_3$PbI$_3$ (450 nm in thickness) films, as the gain medium, at room temperature. The gain medium was optically pumped using ns laser pulses at 355 nm. The laser threshold was measured at 102 μJ/cm$^2$. Coherent light emission was detected at RT and at 775 nm with a FWHM of 0.4 nm; the random lasing operation was confirmed by (a) the random spacing of the observed emission longitudinal modes; (b) the random change of the emission peaks with the optical pumping intensity; and (c) the multi-directionality of the output emission.

The improvement of the thermal and photo stability of the methylammonium hybrid organic inorganic perovskite materials is a challenge. Y. Fu et al. [30] proposed the use of formamidinium or of an alloy of methylammonium and formamidinium based hybrid halide perovskites (FA/MA composites assisted to fill in the gap of lasing wavelength previously unavailable with MA-based perovskites). Their proposal of solution processed formamidinium lead halide materials that demonstrated superior thermal stability compared to the methylammonium lead halide perovskites (due to better hydrogen bonding between the organic cation and the iodide anions) exhibited laser operational stability for more than 10$^4$ pumping 150 fs laser pulses at 402 nm and at room temperature (Figure 3d). Single mode laser operation at 800 nm was observed, with the threshold value at 6.2 μJ/cm$^2$. Spectral tunability of the output laser wavelength was also demonstrated through cation and anion substitution modifications (Figure 3c).

Another type of hybrid halide perovskite NWs was proposed by H. Yu et al. [31]. The fabricated single crystal hybrid perovskite nanowires were employed as an active layer in plasmonic based laser devices. The latter laser architecture is very attractive, and in comparison to Fabry – Perot (FP) nanowire based devices, can generate light below diffraction limit to subwavelength limit. The construction consisted from CH$_3$NH$_3$PbX$_3$ NWs deposited onto a silver film with a 10 nm thick magnesium fluoride (MgF2) spacer layer. These systems are of low cost, facile to produce and of high optical gain and are competitors & alternatives to corresponding high cost systems fabricating using VI and III-V inorganic semiconductor NWs. The laser mode of the CH$_3$NH$_3$PbX$_3$ NWs was confined in an ultra-small area, perpendicular to the nanowire axis. The NWs end facets operated as mirrors to form a FP optical resonator, necessary for lasing. The plasmonic CH$_3$NH$_3$PbX$_3$ NW laser output, compared to the photonic respective NW device’s (deposited directly on a quartz surface) output, was characterised by a lot of distinctive characteristics: (a) free of diffraction limit outputs (this is not the case for the photonic devices); and (b) linear polarization across the nanowire axis (while the photonic devices characterised by perpendicular to the optical axis polarization/ thin (less than 300 nm) nanowires were requested in order to have distinctive plasmonic devices). The CH$_3$NH$_3$PbX$_3$ NW laser devices were optically pumped employing 120 fs laser pulses at 400 nm with a repetition rate of 1 kHz. Depending on the size (length) of the CH$_3$NH$_3$PbX$_3$ NWs, the threshold varied from 13.5 to 56.3 μJ/cm$^2$ at room temperature whereas the wavelength was tuned from 767.76 to 796.75 nm. The threshold pump power was increased as the temperature increased. Lasing operation was observed up to temperatures of 43.6 °C.

An important figure of merit of any type of laser device is its laser threshold; it should be as low as possible in order to maximize the conversion of the pumping energy to output power. T.S. Kao et al. [32] discovered that the relative concentration of the PbI$_2$ played a major role in the determination of the excitonic and lasing performance of the MAPbI$_3$ materials. More particular the increase of the concentration of the PbI$_2$ up to 30 wt% resulted the lowering of the lasing threshold, mainly due to (a) increased volume of the gain media; (b) to the enhancement of the generated scattering effects (that provide the necessary optical feedback to observe stimulated emission) between the grain boundaries; and (c) the higher exciton binding energy (as thermal PL measurements showed). On the other hand, the increase of the PbI$_2$ concentration beyond an optimum concentration, due to the incomplete induced crystallization of the perovskite material, resulted in higher lasing threshold values. The random lasing, with threshold value at 230 μJ/cm$^2$, observed at 745 nm. The lasing
emission wavelength could be tuned from 745 to 800 nm by modulating the PbI₂ concentration between 30 and 40 wt.%. Trihalide perovskites possess very attractive physical properties that combined with their solution processability, convert them as very promising materials in nano-laser devices. V. Sarritzu et al. [33] exhibited CW and quasi CW laser operation using trihalide perovskite-based devices. All the systems were optically pumped using fs laser pulses (130 fs) for the case of CW and ns laser pulses (4 ns or 300 ns) for the quasi CW laser operation, respectively. The authors' main findings included (a) the dependence of the device’s lasing threshold with the square of lattice’s temperature; (b) the thermal equilibrium between the electron – hole plasma with the perovskite lattice when the system was pumped with fs laser pulses; and (c) the existence of a maximum lattice temperature above which there is no light emission due to high threshold values (losses mainly due to non-radiative recombination) when the system pumped using ns laser pulses. The lattice warming was attributed to (a) Auger non-radiative recombination; and (b) to quantum defect of the excited carriers (ration between the radiative recombination to the total number of excited carriers). This work highlighted the role and the importance of the deposition substrate not only on the crystallization of the perovskite but also as a heat dissipator.

The primary methods to tune a hybrid or an inorganic perovskite laser output are (a) through halide stoichiometry control (Figure 2e); (b) through the organic cation substance; and (c) through the concentration of lead iodide precursor. All these methods applied during the fabrication of the semiconductor. N. Zhang et al. [34] proposed something innovative since the proposed technique is a post-synthetic. The authors proposed the exposure of CH₃NH₃PbBr₃ microplates with chlorine in inductively coupled plasma, that enabled them to tune the emission wavelength by 50 nm. More importantly, the lasing threshold and the crystallization phase did not change after the application of the proposed process compared to the pristine sample. The increase of the process time from 0 to 60 s resulted in a blue shift of the absorption and the emission wavelengths. A linear relationship between these two experimental factors, with a slope of -0.75 nm/s and total tunability of 50 nm was experimentally extracted. The achieved wavelength tuning was attributed to the modifications of the mixture of halide concentration during the interaction of the hybrid perovskite with the chlorine ions. The lasing of the pristine hybrid perovskite was observed at 555 nm; the lasing threshold was measured at 3.4 μJ/cm² with FWHM beyond lasing threshold of 0.33 nm (Q-factor of 1667). The lasing wavelength could be tuned with the processing time; tuning from 555 nm (at 0 s) to 506 nm (at 60 s) was observed.

There is a strong demand in ultra-small laser sources due to the rapid progress of the optical integrated circuits. Up to now the nano-laser research was based on the optimization of individual devices. However high impact applications demand an increase of the throughput of nanolaser configurations. This requirement steams the development of nano-laser arrays. The challenges for this technology are numerous. Among them, are: (a) emission in uniform wavelengths; (b) high integration density; and (c) relatively high powers. S. Sun et al. [35] studied a way, to build using a simple method, a nanolaser perovskite (CH₃NH₃PbBr₃) based array that emitted high spectral purity output and high output powers. The lead halide perovskite semiconductor material was selected due to its high optical gain and high crystalline quality. The optical oscillator in this work, was consisted from the lead halide perovskite nanoribbon and the gold grating onto which the perovskite semiconductor has been placed. The refractive indices differences between the grating’s gold teeth and the air slot area around it, provided satisfactory waveguide properties to trap the light in the air slot areas and thus laser action to be observed. The nanolaser arrays secured low threshold due to the excellent waveguiding provided within the air-gaps between the gold made grating teeth. The system was pumped using 100 fs laser pulses at 400 nm. The lasing threshold was measured at 4.2 μJ/cm² at 553 nm with FWHM of around 0.7 nm beyond threshold point. The output was linear polarized (TM – the electric field perpendicular to the plane). The authors highlight that an essential parameter to secure all the desired properties a laser array should have (see above), was the length of the air gap between the grating teeth. There is an optimum value (350 – 400 nm) that secures the smallest FWHM; an indication of good trapping. The optimum air gap width secured an integration of arrays density of around 1250 – 1428/mm.
The fabrication of semiconductor nanowires (SNWs) laser arrays is very challenging due to difficulties to directly fabricate these constructions onto silicon substrates. The introduction of cost effective and massive fabrication processes of SNW laser arrays is of high priority for nanophotonic industry. P. Liu et al. [36] proposed a fabrication technique that can be directly applied onto any substrate. The authors managed to design and fabricate a template onto PDMS (polydimethylsiloxane) over which solution processed MAPbX₃ nanowire arrays were grown. The spatial confinement the proposed template offered, permitted the fully control of the nanowire dimensions (width, length and height) and interwire distance. The nanowires operation principles were governed by the Fabry - Perot cavity principles. These optically pumped (at 400 nm using 150 fs laser pulses) characterized by long lifetime, corresponded to the emission of 4 × 10⁷ fs laser pulses at 543.1 nm under ambient conditions. The onset of the lasing was accompanied by a substantially increase in the output and a narrowing of the emission spectral width (from 10 nm to 0.9 nm). The threshold value was measured at 18.3 μJ/cm² for PNWs of length of 30 μm. TRPL measurements also demonstrated the effect of the stimulated emission; below threshold the PL signal ceased within 3.6 ns whereas above threshold the PL signal ceased much faster and within 30 ps. As expected and due to higher provided optical gain, the threshold of this architecture was lowering as the cavity length got longer. From the other hand, the longer the length of the PNWs the longer its demonstrated operational stability.

S. Shunemann et al. [37] proposed for the first time an all-solution CH₃NH₃PbBr₃ and CH₃NH₃PbI₃ distributed feedback lasers based on a 3D photonic crystal nanostructure. The exhibited constructions demonstrated very low lasing threshold (1.6 mJ/cm²) and long term operational stability. The proposed fabrication technique is of low cost without the use of expensive and elaborate lithography to be needed. The DFB structure ensured the refractive index contrast, between 3D arrays of spheres of air and the perovskite template surrounding these spheres to provide the necessary optical feedback in order to reach lasing.

H. Cha et al. [38] was motivated by the request to reduce the cost of single mode laser devices based on single – crystalline III-V compound semiconductors. They proposed the use of solution processed halide perovskite alloy system, CH(NH₃)₂Pb(IₓBr₉₋ₓ). The emission wavelength extracted from the thin film of CH(NH₃)₂Pb(IₓBr₉₋ₓ) (200 nm thick, coated onto fused quartz substrates) films, could be controlled through the anion composition ratio x (could be tuned from 550 – 820 nm). The alloyed films exhibited a single mode lobed emission at 630 nm (optically pumped by a frequency doubled 532 nm Nd:YAG 400 ps laser, with repetition rate of 1 kHz). The most important acquired information through the PL measurements, was that the alloy was a completely different light emitting semiconductor material than the individual pristine constituents (FAPbBr₃ and FAPbI₃); its emission wavelength was between the emission wavelengths of the pristine light emitting semiconductors. The fused quartz substrates onto which the thin film of CH(NH₃)₂Pb(IₓBr₉₋ₓ) was spin coated (with SEM revealing grain sizes between 100 – 200 nm), were engraved with a grating in order to secure optical feedback. This optical feedback not only reduced the measured lasing threshold (3.5 μJ/cm²) but also determined the single mode lasing operation. The selection of the second order diffraction beam permitted the authors to have an additional output beam across the vertical direction of the films’ plane. The composition of the employed perovskite did not only affect the emissive wavelength but also determined the temperature range over which single mode operation appeared. For x = 0 and one, single mode operation demonstrated up to room temperature and up to 200K for the alloys (x = 0.6). It has been notified that as the temperature elevated the threshold value followed the same trend. The lasing output from the perovskite alloys was strongly linear polarized, with polarization plane to be parallel to the grating’s lines.

F. Chen et al. [39] demonstrated a solution processed CH₃NH₃PbBr₃ perovskite microsheet and microwire Fabry Perot nano-laser devices operating at room temperature. The Fabry Perot nature of the laser cavity allowed the demonstration of multi-mode and single mode operation by just modifying the optical resonator’s length. The demonstrated systems were optically pumped using 150 fs laser pulses at 325 nm at a repetition rate of 1 kHz. The lasing action was achieved for pump powers beyond 0.251 mW and lasing appeared in the sides of the microsheet. The lasing was accompanied by a sharp increase of the output intensity (transition from the amplified spontaneous
emission to stimulated emission) and a shrinkage of the FWHM of the output light: from 25 nm down to 0.41 nm. The emission wavelength was at 554 nm and the calculated quality cavity factor, Q – factor was estimated close to 1352. To lower the threshold level as much as possible and at the same time to secure single longitudinal mode operation, the optimum dimensions of the nanosheet should be calculated (in collaboration of course with the gain bandwidth of CH₃NH₃PbBr₃); the optimum width was estimated to 2.35 μm. Single mode operation was also achieved from CH₃NH₃PbBr₃ nanowires with width of 2.43 μm and length of 380 μm. The respective values of Q factor, lasing threshold level and output emission was 1924, at 0.357 mW and at 558 nm respectively.

The demonstration of continuous wave (CW) lasing operation by solution processed hybrid perovskite semiconductors is of high priority objective. CW lasing operation is considered the milestone just before an electrically pumped diode laser system will be realised. Recent attempts in the case of hybrid perovskites (MAPbI₃) for CW laser operation has been achieved at 160K. At these temperatures the lifetime of the CW lasing is very short and more particular of the order of few tens of nanoseconds. Y. Jia et al. [40] demonstrated MAPbI₃ thin films (of thickness of 120 nm) as gain media in a DFB (~120-nm-thick MAPbI₃ film onto a Λ = 385 nm period grating etched into a thin alumina layer on a high-thermal-conductivity sapphire substrate). CW optically pumped laser device operating at temperatures lower than 160 K, was demonstrated for more than an hour. The threshold level of this optically pumped (at 445 nm from a 920 ns long pulses) systems was determined at 17 kW/cm². The longer lifetime of the CW lasing was attributed to the higher values of population inversions can be achieved at low temperatures. Moreover, the long lasing in this MAPbI₃ systems was realised at temperatures where the perovskite crystal has an orthorhombic phase and was transformed to tetragonal when was pumped by high intensity laser pulses. The tetragonal phase inclusions trapped more efficient the carriers and thus achieving higher population inversions and longer-lived CW lasing. When the optical pumping was turned off, the tetragonal phase changed back to orthorhombic and the lasing ceased (at higher temperatures than 160K).

The integration of efficient light sources for the next generation silicon nitride (Si₃N₄) photonic platforms is quite challenging. Technology experts would like to overcome the obstacle of the use of external light sources in silicon nitride optoelectronic platforms; todays’ technology limitations lead to (a) higher coupling losses; and (b) higher required energy per transferred bit. These obstacles are possible to be tackled with the introduction of perovskites nanolasers. These optical devices are (a) possible to be embedded with Si₃N₄ waveguide wafers; and (b) do not require complex epitaxial bonding processes (as semiconductors III-V, request) that generate high costs. These credentials are attributed to their solution processability and their low temperature crystallization, that enable them to be easily integrated to existing photonic circuits. The 1st demonstration of an on chip nano-laser device was demonstrated in 2017 by P. J. Cegielski et al. [41]. The authors exhibited the integration of solution processed perovskite laser onto a silicon nitride photonic chip. The authors managed to embed methylammonium lead tri-iodide (MAPbI₃) in a pre-patterned micro-resonator and coupled the laser light into an intergraded photonic waveguide. The integrated laser light was optically pumped, with a threshold value of 19.6 μJ/cm² (optically pumped using 120 fs laser pulses at 645 nm). The authors selected to adopt, for their optical resonator, a racetrack type of cavity. The latter requires much lower lithographic resolution and enables the coupling of laser light into a silicon – nitride waveguide through the emitted evanescent field waves.

Three photon processes in perovskite microlasers were first studied in the work implemented by Y. Gao et al. [42]. The solution processed synthesized hybrid perovskite nanostructures dominated from microplates and microrods structures. The boundaries and the surfaces of the perovskite nanostructures were very smooth and uniform; property that assisted the optical feedback requested for the lasing operation. The nanostructures were optically pumped using 100 fs laser pulses at 1240 nm. The employed wavelength triggered the three-photon absorption by the perovskite nanostructures. Beyond a lasing threshold point (130 μJ/cm² for the microrods), a non-linear enhancement of the PL signal was observed. Lasing was accompanied by strongly linear TE polarised beam at 547 nm.

Miniaturisation of laser devices is a key concept to lower energy consumption, for faster, smaller and of higher density photon-based sources. The most prominent way to reach this objective, is to
adopt the plasmonic based device architecture. The latter can result in devices that break the diffraction limit because they exploit the plasmonic effect: the storage of energy in electron oscillations at the metal-dielectric interface. Another approach related to the plasmonic configuration, is the generation of oscillating waves in the space between metal nanoparticles and the dielectric (plasmonic waveguide-based spacers) surrounding the nanoparticle. This strategy is more energy advantageous since it is less susceptible to Ohmic losses. The stability, the tunability of the plasmonic waveguide-based spacers is primarily defined by the corresponding characteristics of the employed semiconductor. C. Huang et al. [43] applied hybrid organic–inorganic perovskite as a semiconductor material in a plasmonic waveguide-based spacer (MAPbX$_3$ /SiO$_2$/Au film). The total internal reflection provided in the interface between the perovskite and the air allowed the generation of whispering gallery modes (WGM) in the plane of the perovskite. The high reflectance in the interface between the perovskite & the air, the small oscillating volume and the high optical gain of the perovskite, resulted into a hybrid nanostructure perovskite-SiO$_2$/Au nanoscale coherent light source. This miniaturized laser system was pumped using a frequency doubled Ti : Sapphire laser (@400 nm, 100 fs pulse duration and 1000 Hz repetition rate). Below threshold the collected output signal from the hybrid spacer, had a spectral width of 40 nm centred at 770 nm; where the pump power exceeded 59.2 μJ/cm$^2$, the emission intensity ‘jumped’ to higher values and the emission spectrum was dominated by the sharp peaks. It was found that when the spacer operated at the plasmonic mode (for perovskite thicknesses below 50 nm), the threshold increased with the thickness of the device whereas when the nanolaser operated under the photonic laser mode the threshold kept much lower at larger thicknesses. The employed perovskite materials, in comparison with previous employed inorganic semiconductors CdS and GaN, can be tuned through halide elements stoichiometry; the detected output coherent light emission wavelengths could be tuned from 550 (CH$_3$NH$_3$PbBr$_3$) to 770 (CH$_3$NH$_3$PbI$_3$) nm. The option to construct a plasmonic nanolaser with an arbitrary cavity shape, as this was determined by the shape of the bottom Au pattern, was explored by the authors. This resulted in much lower laser threshold pump intensities as low as 30 μJ/cm$^2$ and control of the resonance frequencies as a function of the shape and the size of the bottom Au patterns. The authors also demonstrated that the resonances within hybrid plasmonic nanolaser can be precisely controlled by the shape and size of bottom Au patterns instead of the top semiconductors. As a result, by patterning the substrate into Au disks and Au strips, have experimentally realized the circular hybrid plasmonic nanolaser and the uniform plasmonic nanolaser array.

Random Lasing (RL) has only been demonstrated in halide perovskites only via linear one pulse (1P) and nonlinear two laser pulses absorption processes. Contrary the frequency up conversion of three pulses absorption and resultant RL is still lacking, despite of its great importance in high – order nonlinear optical applications. Moreover, the 1P pumped cases employ ultra violet (UV) light that is destructive for the samples under investigation. Additionally, due to the high absorption coefficient, the halide perovskites exhibited in this spectral region limit UV laser pulse penetration depth which is a disadvantage especially in biomedical imaging and sensing applications. Contrary, the non-linear up conversion of two or more photons demonstrate a lot of merits, to name few of them: (a) large penetration depth; (b) high spatial resolution; (c) small photo damage and photo bleaching. Moreover, the multi-photon pumped process has fascinating characteristics such as repression of unwanted absorption losses. In the study by G.E. Weng et al. [44], high quality CH$_3$NH$_3$PbBr$_3$ perovskite thin films (thickness of ~4.3 μm and grain sizes ranging from several hundred nanometres to a few microns), synthesized through a solution based one spin coating method, demonstrating a frequency up converted RL via direct nonlinear 3P absorption process. Apart of the aforementioned advantages compared to the 1P absorption induced lasing, the 3P frequency up conversion films demonstrated lower threshold compare to the systems that have been pumped with 1P. The absorption of the tested films was set on at 550 nm (indicating the films’ energy band gap @ 2.27 eV). The films excitation using femtosecond (fs) pulses at 1300 nm resulted in an incoherent RL emission signal at 545 nm, a clear indication of the non-linear 3P up conversion absorption. The necessary optical feedback, lasing requires, was resulted by random scattering effects provided by the polycrystalline grain boundaries. The onset of the observed incoherent RL was accompanied apart of a steep increase in the emission intensity at 550 nm with a simultaneous decrease of the spectral full
half width maximum (FWHM) of the detected optical signal. The pump threshold was measured approximately equal to 27 mJ/cm². At the same time an observed wavelength red shift (of maximum 2.8 nm) in the lasing output, as a result of many body interactions, was observed. The measured pulse width of the 3P pumped films, was measured using Kerr – based TRPL and found equal to 3.1 ps, which was the shortest observed in bromide perovskite-based lasers without any post processing. The production of ps laser pulses is one of the attractions of the perovskite-based laser devices. The ultrafast rise process of the RL pulse was benefited from the high material’s optical gain, while the shortest decay time was attributed to the short lifetime of the photons within the perovskite gain medium.

One of the key challenges to introduce electrical pumped perovskite-based lasers is to lower the lasing threshold point as much as possible. Optical feedback using diffraction gratings or distributed feedback (DFB) resonators is a very successful strategy that have already proved its success in inorganic semiconductor laser systems. N. Pourdavoud et al. [45] have applied the DFB approach on methylammonium lead bromide (MAPbBr₃) thin films-based laser systems. More particular, a linear photonic grating was directly imprinted onto the MAPbBr₃ active layer films using the thermal nano-printing technique (@100 °C). The thermal nano-printing technique also overcome internal limitations the hybrid perovskite demonstrates e.g. low intrinsic stability and their susceptibility to a few commonly used solvents, that other strategies have experienced in order to realise high Q – cavities such as wet-chemical lithography. As a result of the nano-printing linear Bragg grating (300 nm the periodicity of the grading, with a depth of 100 nm) onto the MAPbBr₃ film, a high Q resonator resulted. The thermally printed grating also resulted to the smothering of the MAPbBr₃ film surface to roughness below 0.6 nm (from the 46 nm the as cast spin coating MAPbBr₃ films exhibited). The smooth surface afforded a significant decrease of the threshold of the stimulated emission; In optically pumped (at 532 nm using a 0.3 ns pulsed laser source) DFB laser structures, thresholds as low as 3.4 μJ/cm² have been reached. The stimulated emission output could be tuned in the spectral region between 543.3 to 557.4 nm. The appearance of the lasing was accompanied with the shrinking of the FWHM of the output to 0.14 nm (compared to the 2.5 nm Amplified Spontaneous Emission has shown). The excellent material quality as a result also of the nano-printing technique was confirmed through SEM images that showed perovskite grain sizes of the order of 10 μm.

J. Wang et al. [46] provided a deep study of the modulation of laser threshold and emitting laser wavelength as a function of the temperature. Such a study was absent until the publication of this work. More particular, in this paper a MAPbI₃ semiconductor is embedded in a vertical Fabry – Perot (FP) microcavity (consisting from a top metallic mirror and a bottom dielectric distributed Bragg reflector (DBR) with a Q value of 312). This microcavity (PM) system was pumped by a continuous wave (CW) at 633 nm and the lasing threshold measured at 12.9 μJ/cm². Beyond laser threshold, a FWHM of 0.76 nm at 782 nm was measured (@ 25K). Due to suppress of the free exciton irradianve recombination as the temperature decreased (from 75K to 25K), the achieved lasing threshold was lowered (as a result of the weaker exciton – phonon interaction and defect scattering), a spectral red shift (as a result of the temperature dependent orthorhombic phase) was observed (750 – 760 nm) and the spectral coherence was increased. It was registered that beyond 55K the lasing operation ceased, and the stimulated emission was replaced by the spontaneous emission radiation (broader spectral emissions of lower intensity).

A. Salfadar et al. [47] were motivated from the challenge to fabricate low cost, solution processed and without the need of carefully engineered optical cavities, laser devices. The authors managed to show random lasing (single and dual laser mode operation) from a thin film of solution processed perovskite (CH₃NH₃PbI₃), deposited onto a nonpatterned glass substrate at room temperature. The demonstrated lasing threshold was 10 μJ/cm² and the pumping source was a pulsed frequency doubled YAG laser (pulse length of 400 ps, at 532 nm). The light amplification was optimised by finding the optimum point between intense optical scattering and scattering length. The random lasing was observed when constructive interference of the multi-scattered light was secured. It was essential in the random lasing the number of scattering events and the gain to interplay correctly (not too much or too low scattering effects). By simply controlling the solution processing (spin coating turns, dip time in solvent) the quality and the thickness of the perovskite film was fully controlled
and determined; and moreover, the whole random lasing operation was optimised (especially important the optimum crystallization of the perovskite as this was determined by anti-solvent extraction time). The laser waveguide was consisted from air – CH$_3$NH$_3$PbI$_3$ – glass and its length was of the order of 50 nm. The threshold value was increased as the time of exposure of non-encapsulated perovskite films in air and light (e.g. the laser output deceased to 80% of its initial value after its illumination with 105 pulses).

The development of nanophotonic devices that will open new avenues for realizing on-chip optical components has been realised. Perovskite semiconductors due to their attractive physical characteristics and especially their direct bandgap and high optical gain, are promising materials for realizing such an optical component. 3D cavity shapes such as cubic corner pyramids due to the internal effective optical coupling/trapping offer, provided by the three reflectors, show the way towards the realization of low threshold laser devices. Y. Mi et al. [48] provided a theoretical work on MAPbBr$_3$ cubic corner pyramids (the three facets operate as very efficient reflectors), chemical vapor deposited onto mica substrates, laser-based elements. Lasing from the perovskite pyramids was observed from 80 to 200 K within the threshold range from 92 μJ/cm$^2$ to 2.2 mJ/cm$^2$, after the pyramids have pumped by femtosecond laser pulses at 400 nm. To be mentioned that in comparison to the inorganic semiconductors, a blue shift of the output wavelength was observed as the temperature raised. This tendency was attributed to the thermal expansion of the crystal lattice and thus the increase of the energy bandgap. As in all other cases, the lasing action was accompanied with a substantial spectral reduction of the output: the FWHM decreased from 9.6 nm to 0.56 nm. No lasing has been observed for temperatures higher than 200 K. By capping an Ag layer beneath the mica substrate, the authors managed to reduce the threshold from 92 μJ/cm$^2$ to 26 μJ/cm$^2$ (@ 80 K) and room temperature lasing operation to be achieved (with threshold at 75 μJ/cm$^2$). The Ag substrate enhanced the optical feedback from below towards backwards to the MAPbBr$_3$ nano-pyramids, and thus room temperature lasing was simulated.

**Figure 3.** (a) Nonlinear increase of the output intensity beyond the laser threshold point; A spectral linewidth shrinking above threshold is also observed as a clear indication of the onset of the laser operation. (b) The transition from the spontaneous emission to stimulated emission is depicted on the reduction of the PL signal above threshold as the Time Resolved Photoluminescent measurements indicate. Reproduced with permission from H. Zu et al., Nature Materials; published by Nature Publishing Group, 2015. (c) Tunability of the emission laser light achieved through the halide constitution (controllable stoichiometry); (d) Operational Lasing Stability at room temperature as a function of pump laser shots. Reproduced with permission from Y. Fu et al., Nano Letters; published by American Chemical Society, 2016.
Table 1. Organic - Inorganic Perovskite Based Laser Device Characteristics; NA denotes for the Non-Available Information

<table>
<thead>
<tr>
<th>Material</th>
<th>Nanostructure Type /Cavity Architecture</th>
<th>Laser Threshold (μJ/cm²)</th>
<th>Emission Linewidth (nm)</th>
<th>Emission Wavelength (nm)</th>
<th>Q-factor</th>
<th>Gain Coefficient (cm⁻¹)</th>
<th>Tunability (nm)</th>
<th>Pumping Source</th>
<th>Reference (Year)</th>
</tr>
</thead>
<tbody>
<tr>
<td>MAPbI₃₋ₓClₓ</td>
<td>Thin Film /Vertical Cavity</td>
<td>0.2 μJ/pulse</td>
<td>N/A</td>
<td>775</td>
<td>NA</td>
<td>NA</td>
<td>0.4 ns @ 532 nm</td>
<td>F. Deschler et al. (2014)</td>
<td></td>
</tr>
<tr>
<td>MAPbI₃Brₙ</td>
<td>MDLs/WGM</td>
<td>3.6</td>
<td>1.1</td>
<td>557.5</td>
<td>430</td>
<td>NA</td>
<td>525 – 557</td>
<td>Q. Liao et al. (2015)</td>
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</tr>
<tr>
<td>MAPbI₃</td>
<td>NWs/Fabry Perot Cavity</td>
<td>220 nJ/cm²</td>
<td>0.22</td>
<td>787</td>
<td>3600</td>
<td>NA</td>
<td>400 - 800</td>
<td>H. Zhu et al. (2015)</td>
<td></td>
</tr>
<tr>
<td>MAPbI₃</td>
<td>Photonic Crystal/Fabry Perot Cavity</td>
<td>68.5</td>
<td>0.24</td>
<td>788</td>
<td>NA</td>
<td>NA</td>
<td>768 - 794</td>
<td>S. Chen et al. (2016)</td>
<td></td>
</tr>
<tr>
<td>MAPbBr₃</td>
<td>MWs/ Fabry Perot Cavity</td>
<td>674</td>
<td>0.8</td>
<td>546</td>
<td>650</td>
<td>NA</td>
<td>NA</td>
<td>100 fs @ 800 nm</td>
<td>Z. Gu et al. (2016)</td>
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<tr>
<td>MAPbBr₃</td>
<td>Micro-rods/graphene/F abry Perot Cavity</td>
<td>48</td>
<td>N/A</td>
<td>551.8</td>
<td>N/A</td>
<td>NA</td>
<td>NA</td>
<td>C. Zhang et al. (2016)</td>
<td></td>
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<tr>
<td>MAPbI₃</td>
<td>Platelets /WGM</td>
<td>11</td>
<td>N/A</td>
<td>780</td>
<td>1200</td>
<td>NA</td>
<td>NA</td>
<td>X. Liu et al. (2016)</td>
<td></td>
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<tr>
<td>MAPbI₃</td>
<td>Thin Films/DFB systems</td>
<td>120 kW/cm²</td>
<td>0.2</td>
<td>786.5</td>
<td>NA</td>
<td>NA</td>
<td>781 - 794</td>
<td>P. Brenner et al. (2016)</td>
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<tr>
<td>MAPbI₃</td>
<td>Thin Film/Random lasing</td>
<td>102</td>
<td>0.4</td>
<td>775</td>
<td>NA</td>
<td>NA</td>
<td>NA</td>
<td>Z.F. Shi et al. (2016)</td>
<td></td>
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<tr>
<td>(FA, MA)PbI₃ and (FA,MA)Pb(I, Br)</td>
<td>NWs/Fabry Perot Cavities</td>
<td>6.2</td>
<td>0.24</td>
<td>560 (FAPbBr₃)</td>
<td>2300</td>
<td>NA</td>
<td>500 - 790</td>
<td>Y. Fu et al. (2016)</td>
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<tr>
<td>CH₃NH₃X₃</td>
<td>Plasmonic NWs/Fabry Perot Cavities</td>
<td>13.5</td>
<td>5</td>
<td>785</td>
<td>151</td>
<td>NA</td>
<td>767 - 796</td>
<td>H. Yu et al. (2016)</td>
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<tr>
<td>MAPbI₃</td>
<td>Thin Films/Random lasing</td>
<td>230</td>
<td>3</td>
<td>745</td>
<td>NA</td>
<td>NA</td>
<td>120 fs @ 400 nm</td>
<td>T.S. Kao et al. (2016)</td>
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</tr>
<tr>
<td>MAPbI₃ &amp; MAPbBr₃</td>
<td>Thin Films / Fabry Perot Cavities</td>
<td>NA</td>
<td>NA</td>
<td>NA</td>
<td>NA</td>
<td>NA</td>
<td>130fs (CW) 4ns (quasi CW)</td>
<td>V. Sarritzu et al. (2016)</td>
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<tr>
<td>MAPbBr₃</td>
<td>Microplates / WGM</td>
<td>3.4</td>
<td>0.33</td>
<td>555</td>
<td>1667</td>
<td>NA</td>
<td>555 - 506</td>
<td>N. Zhang et al. (2016)</td>
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<tr>
<td>Material</td>
<td>Device Type</td>
<td>Band Gap (eV)</td>
<td>Thickness (nm)</td>
<td>Power (mW)</td>
<td>Duration (fs)</td>
<td>Wavelength (nm)</td>
<td>Ref.</td>
<td></td>
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<tr>
<td>MAPbBr₃</td>
<td>Nanoribbons / Fabry Perot Cavities</td>
<td>4.2</td>
<td>0.7</td>
<td>553</td>
<td>NA</td>
<td>NA</td>
<td>100 fs @ 400 nm</td>
<td>S. Sun et al. (2017)</td>
<td></td>
</tr>
<tr>
<td>MAPbX₃</td>
<td>NW arrays / Fabry Perot Cavities</td>
<td>18.3</td>
<td>0.9</td>
<td>543.1</td>
<td>500</td>
<td>NA</td>
<td>150 fs @ 400 nm</td>
<td>P. Liu et al. (2017)</td>
<td></td>
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<tr>
<td>CH₃NH₂PbBr₃</td>
<td>Photonic Crystal / DFB</td>
<td>1.6</td>
<td>0.15</td>
<td>545</td>
<td>NA</td>
<td>NA</td>
<td>0.5 ns @ 532 nm</td>
<td>S. Shuneman et al. (2017)</td>
<td></td>
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<tr>
<td>CH₃NH₂Pb (IₓBr_{1-x})₃</td>
<td>Thin Films / DFB</td>
<td>3.5</td>
<td>NA</td>
<td>630</td>
<td>NA</td>
<td>NA</td>
<td>400 ps @ 532 nm</td>
<td>H. Cha et al. (2017)</td>
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<tr>
<td>CH₃NH₂PbBr₃</td>
<td>Microsheet / Fabry Perot Cavities</td>
<td>0.251 mW</td>
<td>0.41</td>
<td>554</td>
<td>1352</td>
<td>NA</td>
<td>150 fs @ 325 nm</td>
<td>F. Chen et al. (2017)</td>
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<tr>
<td>CH₃NH₂PbI₃</td>
<td>Thin Films / DFB</td>
<td>17 kW/cm²</td>
<td>NA</td>
<td>785</td>
<td>NA</td>
<td>NA</td>
<td>920 ps @ 445 nm</td>
<td>Y. Jia et al. (2017)</td>
<td></td>
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<tr>
<td>CH₃NH₂PbI₃</td>
<td>Thin Film / Ring Cavity Laser</td>
<td>19.6</td>
<td>NA</td>
<td>791.5</td>
<td>650</td>
<td>NA</td>
<td>Possible</td>
<td>P.J. Cegielski et al. (2017)</td>
<td></td>
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<tr>
<td>CH₃NH₂PbBr₃</td>
<td>Microplates and Microrods / WGM</td>
<td>130</td>
<td>0.3</td>
<td>547</td>
<td>1800</td>
<td>NA</td>
<td>100 fs @ 1240 nm</td>
<td>Y.Gao et al. (2017)</td>
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<tr>
<td>CH₃NH₂PbX₃</td>
<td>Microplates/WGM</td>
<td>30</td>
<td>NA</td>
<td>770</td>
<td>3200</td>
<td>NA</td>
<td>100 fs @ 400 nm</td>
<td>C. Huang et al. (2018)</td>
<td></td>
</tr>
<tr>
<td>CH₃NH₂PbBr₃</td>
<td>Thin Film / Random Lasing</td>
<td>27 mJ/cm²</td>
<td>NA</td>
<td>545</td>
<td>NA</td>
<td>NA</td>
<td>35 fs @ 1300 nm</td>
<td>G.E. Wang et al. (2018)</td>
<td></td>
</tr>
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</table>
### 2.2. Inorganic Perovskite Based Laser Devices

Quantum dots (QDs) lasers are size tunable, low threshold laser devices with a great potential to be employed as a single photon sources (optical qubits) with applications in quantum computing. The optical properties of the organic–inorganic halide perovskites (CH$_3$NH$_3$PbX$_3$, X= I, Br, Cl) have shown their potential as efficient optical gain media. Y. Wang et al. [49] combined the advantages of the QD configuration, the photophysical properties of perovskites (high PLQE) and the superior stability against environmental hazards (moisture and oxygen) of the inorganic semiconductors, in order to demonstrate for first time, stimulated emission at room temperature from optically pumped cesium lead halide perovskite (CsPbX$_3$) QDs (~9 nm in diameter). The lasing threshold was measured at 22 μJ/cm$^2$ (with PLQE of 85%) which was at that period, one order of magnitude lower than the corresponding values of the CdS QDs. The lasing not only exhibited at room temperature, but it was also sustainable under atmospheric environment exposure; even after three weeks exposure to the air, the QDs exhibited the same lasing threshold value. Moreover the CsPbX$_3$ QDs film demonstrated excellent photostability, under the operational conditions; the CsPbX$_3$ QDs output was almost at the same lasing intensity level (~90% of the initial output), under optical pumping of 57 μJ/cm$^2$, for about 4.5 hrs. The tunability of the lasing wavelength achieved through the tuning of (a) the size of the QDs (e.g. the 5.5 nm QDs emitted light at 502 nm); and (b) the halide constitution of the perovskite (scan across the whole visible spectrum).

The work of S. Yakunin et al. [50] shed light to the photophysics of inorganic metal halides (CsPbX$_3$) in the form of uniform, size tunable nanocrystals. Low threshold and tunable lasing were demonstrated mainly due to the excellent crystal quality & high PLQE (~70 - 90%) and the compositional tuning of the halide components respectively. The authors demonstrated lasing operation under (a) WGM cavity configuration; (b) FP cavity; and (c) free of optical oscillator configurations (random lasing).

The hybrid perovskite NWs beyond their advantages demonstrate material instabilities: they are sensitive to environmental hazards such as moisture and oxygen. From the other hand, the inorganic Cs based perovskites offered an alternative that show superior stability under environmental operational conditions. Y. Fu et al. [51] employed CsPbX$_3$ (X= I, Br or Cl) and MAPb(BrCl)$_3$ alloyed composites NWs as an active layer to demonstrate stable lasing operation under ambient conditions. The authors tested various inorganic perovskite films under continuous optically pumped conditions; the inorganic perovskite semiconductor media retained their lasing output constant for more than $7.2 \times 10^9$ pumping laser pulses (eight hours of continuous operation). This stability

<table>
<thead>
<tr>
<th>CH$_3$NH$_3$PbX$_3$</th>
<th>Thin Film / DFB</th>
<th>3.4</th>
<th>0.14</th>
<th>555</th>
<th>NA</th>
<th>NA</th>
<th>0.3 ns @ 532 nm</th>
<th>N. Pourdavoud et al. (2018)</th>
</tr>
</thead>
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<tr>
<td>CH$_3$NH$_3$PbI$_3$</td>
<td>Thin Film / VCSEL</td>
<td>12.9</td>
<td>0.76</td>
<td>782</td>
<td>312</td>
<td>NA</td>
<td>750 - 760 fs @ 633 nm</td>
<td>J. Wang et al. (2018)</td>
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<tr>
<td>CH$_3$NH$_3$PbI$_3$</td>
<td>Thin Film / Random Lasing</td>
<td>10</td>
<td>45</td>
<td>NA</td>
<td>NA</td>
<td>70.1</td>
<td>NA</td>
<td>400 ps @ 532 nm</td>
</tr>
<tr>
<td>CH$_3$NH$_3$PbBr$_3$</td>
<td>Cube Corner Pyramids / Fabry Perot Cavities</td>
<td>75</td>
<td>0.56</td>
<td>530</td>
<td>NA</td>
<td>NA</td>
<td>400 nm</td>
<td>Y. Mi et al. (2018)</td>
</tr>
</tbody>
</table>
performance was substantially more robust than their organic–inorganic counterparts. The emission spectrum of MAPb(BrCl)₃ was tuned, as in other similar cases, through the entire visible spectrum, by controlling the stoichiometry of the alloyed perovskite material.

X. Tang et al. [52] combined the advantages of the inorganic perovskite semiconductors and the facile operational principles of the random lasing; they managed to demonstrate a very low cost and of low threshold (0.97 mJ/cm²) (Figure 4b) optically pumped miniature laser source (QDs of diameter of 9 nm). Random lasing operation, under optical excitation using a 100 fs laser source at 800 nm, was observed. The emission wavelength depended on the halide element selected; lasing peaks at 427 (CsPbCl₃), 527 and 539 nm were observed (Figure 4c).

The superior operational stability (60 minutes under operational conditions at ambient atmosphere), the tunability achieved through the halide stoichiometry and the excellent optical confinement of the 1D nature NW configurations, using single crystal inorganic perovskite semiconductors were demonstrated from K. Park et al. [53]. The low lasing (@ 300 nm) threshold approximately at 3 μJ/cm² was secured by the excellent crystallinity of the spin coated CsPbX₃ NWs (~ 10 μm long and 300 nm in diameter) and the high oscillatory strength of the 1D character of NWs. The suburb properties of the 1D optical confinement in combination of the excellent crystallinity were reflected on the high Q-value of 1300 of the optical resonators formed between the NW end facets. The experimental measurement of the non-single valued FSR of the laser oscillating modes suggested that the traveling waves within a NW were governed by the exciton-photon dispersion curve (polariton dispersion curve) where the experienced group refractive indexes are 6 -7 fold increase; as a result the oscillating modes appeared closer (smaller FSR) than the classical model predicted.

Y. Wang et al. [54] demonstrated for the 1st time, and beyond random and whispering gallery mode lasing, a vertical cavity emission surface laser (VCESL) – (an inorganic perovskite (CsPbX₃) nanocrystal). This work was exceptional since for the first time a single device demonstrated (a) low threshold; (b) operational stability; (c) single mode operation; and (d) output directionality (one of the most important advantages of a laser device). The laser emission demonstrated very high directionality, with the far field divergence angle to be ~ 3.6°. The short cavity length, secured the single longitudinal mode laser operation at 504 nm (linewidth of ~ 0.6 nm). The demonstrated VCESL was optically pumped using fs laser pulses at 400 nm. The exhibited system was very stable, with its output to retain the 80% of its peak power for more than one hour. Beyond its operational characteristics, these devices presented high flexibility, since this particular VCESL can be versatilely engineered by independent adjustments of the optical resonator and the composition of the perovskite semiconductor crystals (mainly due to the solution processability of the active layer). A step towards the demonstration of a CW perovskite laser device was the demonstration from the authors of quasi steady state lasing. This was achieved using 5 ns and at 400 nm laser pulses. The lasing threshold was at 900 μJ/cm² but as was said this was a big step towards CW optical and electrical pumped perovskite semiconductor lasers. These are important results, since indicate the roadmap for the development of low cost laser devices for the future.

The continuous evolvement and the upcoming commercialization of the integrated onto chip photonics circuits request the fabrication of low cost nano-laser sources. The work from X. He et al. [55] proposed a one step, room temperature, facile and of low-cost fabrication process of well-spaced aligned micro-disk (MDLs) Cs lead halide-based laser arrays. The proposed technique permitted the researchers of this work to control the size and the spacing of the fabricated onto quartz CsPbX₃ MDLs. Key element of the proposed technique was the employment of a PDMS cylindrical hole template that confined the fabrication of the CsPbX₃ in well predefined positions and sizes. The CsPbX₃ MDLs were optically pumped by a 400 nm, 150 fs laser pulses at 1 kHz. The lasing threshold was similar for all the constructed MDLs, measured around 10.3 μJ/cm² at 425 nm, an attractive feature since permitted the simultaneously operation of all the photonic elements onto the quartz substrate. The stochiometric tunability was also demonstrated and allowed, as in other cases the spectral tuning of MDLs output (from 416 nm to 530 nm). The fabrication of different wavelength output MDLs onto the same substrate was also exhibited; a configuration especially important for multi-coloured laser displays, laser lighting and sensing.
The employment of 3D structures, such as micro cubes, can offer lasing operation in three dimensions and therefore the laser action is less sensitive to the excitation direction. Another attractive feature of the perovskite material (independent of their structural set up, e.g. 1D or 2D or 3D) is the presence of nonlinear properties that allow the multi-photon pumping. The multi-photon pumping results in lower material damage and the resulted up conversion generates wavelength with (a) deeper penetration depth; and (b) higher spatial resolution. However, and regardless their superb optical properties, the hybrid perovskite semiconductors are very susceptible in hydrolysis and this hinders their commercialization. This stability task is addressed with the use of inorganic perovskite materials such as Caesium lead halide perovskite, CsPbX₃. Z. Hu et al. [56] exploited the advantages of the micro-tubes 3D character and the ability to multi-photon optical pumping them. The micro-cubes side dimensions (505 nm) & facets, operated as Fabry – Perot cavities that resulted optical cavities of very high Q values (~ 1100) and lasing operation of low threshold at room temperature. Moreover, for two photon optical pumping at 800 nm (using a Ti:sapphire, fs laser system) the threshold was approximately found at 439 μJ/cm². Above this pumping value, the presence of lasing operation was accompanied with a substantial decrease of the spectral width of the output (from 22 nm down to 0.46 nm) and an abrupt increase on the output. The authors through TRPL spectroscopy measurement showed that above threshold the carriers’ recombination (excitons recombination) was much faster than below threshold pumping operation due to the lasing effect (1.13 ns from 4 ns). The stability of the inorganic perovskite microcubes was tested under storage and illumination conditions. All the tests occurred in room temperature and under humidity conditions (~ 35-40%). The storage stability tests showed that after several months storage the micro-cubes retained their structure (XRD measurement used for this purpose). The photo stability tests acquired PL spectra for several months without showing any significant degradation. A constant ASE signal was measured for over ten hours indicating a significant operating lifetime under ambient conditions.

CsPbX₃ based thin films are simpler to be fabricated than nanowires or nanocrystals, however, contain a lot of pinholes that seriously degrade their lasing performance. This is the reason that despite of their fabrication simplicity, these systems demonstrate much lower quantum efficiency than the corresponding nanocrystals. C. Li et al. [57] realised that by mixing ZnO nanoparticles (smaller diameter than 35 nm) into the CsPbX₃ precursor’s solution addressed the aforementioned drawback; the resulted thin film (CsPbBr₃:ZnO, thickness of 210 nm) synthesized using the one step method, exhibited no pinholes or voids and as a result its luminescence performance improved compared to this one of the reference device. Criteria of the improved photoluminescence performance was the lower ASE (from 0.292 mJ/cm² to 0.207 mJ/cm² for the one photon pumping, and from 0.679 to 0.569 mJ/cm² for the two photon pumping) and lower lasing threshold for one (400 nm) and two photon (800 nm) fs pumping (50 fs, 1 kHz repetition rate, at 800 nm); This was attributed to mainly to morphological reasons: (a) Better crystallization (more intense and sharper XRD peaks) that is caused due to the improved nucleation the ZnO nanoparticles induced; (b) Smaller grain crystals (SEM measurements showed decrease of the grain size from 0.43 μm to 0.24 μm) that lead to shorter exciton diffusion lengths and thus less probability for exciton dissociation; (c) less surface roughness (AFM measurements showed a smoother films, the roughness decreased from 36.36 nm to 26.51 nm) that allow the more efficient pumping photons to be trapped within the material; and (d) to the shorter optical loops the signal is doing between the perovskite crystals. The PL emission spectra of the pristine and the CsPbBr₃:ZnO thin films is identical (@ 523.5 nm) but with the PLQY of the ZnO NPs devices to be higher: from 12% to 21%. The ASE signal from CsPbBr₃:ZnO thin films is approximately 4.25-fold higher than that of the pristine thin films.

The majority of the work have been done until today regarding the hybrid organic – inorganic perovskite lasers is characterised from an output emission in the NIR part of the spectrum. There is a lack of demonstrators in visible of low cost and stable perovskite lasers. Low cost, stable under ambient conditions, solution processed laser devices in visible are very important for applications including displays, high-density data storage and readout, and underwater communications. This gap will be filled by the inorganic perovskite materials. The latter materials compared to the hybrid ones, demonstrate some superior properties such as (a) higher current densities; (b) reduce heating effects; (c) superior thermal and environmental stabilities. J. Gong et al. [58] exhibited the unique...
merits of thin film (80 nm), solution processed inorganic perovskite semiconductors (CsPbBr₃ – PEO) as a gain medium in DFB laser configuration. Polyethylene oxide (PEO) was added into the precursor solutions to improve the film quality, resulting in an enhanced PLQY values. The CsPbBr₃ – PEO systems demonstrated single mode operation, high gain coefficient of 161.1 cm⁻¹ that with the combination of the provided optical feedback from the constructed nano-printed DFB resonator (with the Bragg grating period to be 360 nm), resulted in low lasing threshold values of 33 μJ/cm², of linear polarised (TE – optical polarization parallel in the grooves of the grating) emissions at 654 nm, with a linewidth of 4.9 nm. The samples were pumped by a 355 nm 90 pico-second laser.

Perovskite semiconductor nanowire lasers have already been demonstrated as a promising approach with well-defined one-dimensional geometry towards on chip integration laser sources. X. Wang et al. [59] work was motivated by the challenge to understand deeper the operational principles of these configurations: Understanding light matter interactions for both fundamental and practical reasons in designing low power consumption nanoscale light sources. The CsPbX₃ nanowires are vapour deposited on sapphire deposition layer whereas their average length was of the order of 10 – 20 μm and their diameter of 200 – 500 nm. The Rabi splitting (2g) provided a measure of the photon – exciton coupling; it can be expressed as g~√f/V where f is the oscillator strength and V is the mode volume. Since nanowire is a 1D construction, and due to strong confinement, the V takes small values and thus the Rabi splitting is expected to be strong. Moreover, due to high excitonic binding energy the oscillatory strength f is high, and this allows the stronger exciton – photon coupling within CsPbX₃ nanowires. Authors findings showed that it is these polaritons that were the origin of the demonstrating lasing under CW or pulsed optical pumping. Lasing emission under pulsed optical pumping (using fs laser pulses at 400 nm for the CsPbCl₃ and 470 nm for the CsPbBr₃ and CsPbI₃) was observed only by the two ends of the nanowires as a result of the strong coupling of the spontaneous emission with the FP resonator modes of the nanowire cavity. The stimulated emission appears for threshold pumping values of ~ 4 μJ/cm² in the case of CsPbBr₃. The FWHM of the dominating lasing mode at 535 nm, is of the order of ~ 0.20 nm. The achieved CsPbBr₃ nanowire lasers have a very high degree (98%) of linear polarization. Moreover, the emitted laser light could be tuned from 425 nm to 722 nm by varying the nanowires chemical stoichiometry.

Z. Liu et al. [60] reported the 1st laser device that was smaller in all dimensions than the emitted wavelength. Their research was motivated by (a) the requirement to develop optical components with the three of their dimensions in the nano scale, for applications on chip photonic information processing systems. This is a challenge since the miniaturization of the cavity length and the extremely high requirement of optical gain to overcome optical losses are difficult to realize; and (b) to address the temperature instability of the CsPbBr₃ semiconductor laser medium. The authors reported the building of a CsPbBr₃ perovskite nanocuboid laser, with sides’ length of an average of 400 nm, that operated under single longitudinal mode (emitted from a single nanocuboid perovskite source) operation at room temperature. The single mode operation was secured by the short FP distance between the facets of the nanocuboid that made the Free Spectral Range (FSR) longer than the gain bandwidth of the active medium. The flat and smooth end facets of the nanocuboids and the high gain of the material (502 cm⁻¹) assisted to achieve low threshold lasing operation. The reported threshold lasing values, were 40.2 and 374 μJ/cm² for one (pumped by a fs laser source at 400 nm, emitting laser light at 539 nm with a linewidth of 0.26 nm and a Q factor of 2075) and two photon (pumped by an fs at 800 nm laser source, emission at 539 nm with a linewidth of 0.29 nm and Q equal to 1859) excitation. The demonstrated quality factor of the nanocuboid laser systems was reported as high as 2075. Moreover, the authors exhibited a quiet stable laser device with temperature insensitive (22 ps – pulse width) lasing operation for temperatures between 180 and 380 K. The operational stability is one important issue for the practical exploitation of these nanolaser systems. The long-term stability under ambient conditions (23 °C, 40% relative humidity) was demonstrated. The presented system showed an excellent photo- and thermal stability without demonstrating any mode hopping after 160 min of uninterrupted excitation. The thermal stability of these systems was evaluated for the temperature window between 77 to 300 K by examining their PL signal after two photon excitation. It was found the impressive thermal stability of these systems is attributed to the strong exciton binding energy of the CsPbBr₃ nanocuboids. This exhibited stability, between 180 to
380 K, makes them a very promising material for light emitting devices that during their operation their temperature arises and any shift in the emitting wavelength is not desired at all.

An innovative synthesis of a graded composed CsPbBr$_{3-x}$I$_x$ NWs, that allowed, due to the increased Br/I ratio from the center to the ends, the dual wavelength laser emission was demonstrated by L. Huang et al. [61]. A tunable wavelength emission was exhibited with a blue shifted emission from the center towards the NWs edges (as a result of the wider bandgap semiconductors at the end while narrower are formed at the centre) (Figure 4d). The synthesis of such construction was challenging because the inorganic perovskite semiconductors due to their soft lattice and rapid anion exchange reaction favours homogeneous alloys to be formed rather inhomogeneous one. The authors exploited (a) the desynchronized deposition of Caesium lead halides; and (b) the temperature-controlled anion exchange reaction to manage the construction of the graded composited CsPbBr$_{3-x}$ NWs. The varied content along the NW was confirmed from XRD spectra. The CsPbBr$_{3-x}$NWs were pumped using fs laser pulses (150 fs, 1 kHz) at 400 nm. Dual color lasing, with a wavelength separation of 35 nm (from 521 nm to 556 nm) was observed as the micro-photoluminescence ($\mu$PL) measurements demonstrated. Both wavelengths appeared into the detected spectrum when the pumping fluence was above 51 $\mu$J/cm$^2$. The authors discovered that the emission wavelengths originated from the center and the ends of the NWs were dependent on (a) the dimensions (length and width of the NWs; following the tendency for further blue shifted as these dimensions are increased); and (b) the fabrication time which determined the ratio of Br/I; as a result of this parameter the nanowire started from deposition of rich in iodide, CsPbBr$_{3-x}$ (emission above 580 nm) and gradually take place by a Br rich sample (emission below 580 nm).

T. J. S. Evans et al. [62] were the first ones that demonstrated CW lasing from CsPbBr$_3$ NWs at 77 K. This is a step towards the realization of electrically pumped perovskite semiconductor lasers. NWs due to their geometry, emit a low laser mode volume that encourages a lot the light matter interaction between the generated excitons and the pumping photons (plasmons). In this regime there is no need of population inversion in order stimulated light to be emitted. This tackles the obstacles of high temperature instabilities the perovskite suffers under intense optical pumping to create population inversion. To be better studied, the perovskite nanostructures were optically pumped at room temperature using fs laser pulses at 454 nm. One of the most striking observations in this work was the increase of the FSR as the temperature of the perovskite samples was elevated from 77 K to 294 K. Such a behaviour contradicted the function of the NW as a photonic based FP cavity. The authors suggested that the CW lasing observed at very low temperatures was ascribed to generated polaritons revealed the strong matter interaction in this material. This work introduced the low power inorganic perovskite polaritonic based lasers.
Figure 4. (a) Single Mode Laser Operation at blue and red colour from CsPb(Br/Cl)\textsubscript{x} and CsPb(I/Br)\textsubscript{3} IPNCs under pump intensity of 38.2 and 30.5 µJ cm\textsuperscript{-2}, respectively. Reproduced with permission from Y. Wang et al., Advanced Functional Materials; published by Wiley-VCH, 2017. (b) The transition from the spontaneous emission to stimulated emission is accompanied by a simultaneous non-linear increase of the Intensity above threshold and at the same time a shrieking of the emission spectral linewidth; in this figure the lasing behaviour of CsPbBr\textsubscript{3} NC film. Reproduced with permission from X. Tang et al., Nano Energy; published by Elsevier, 2016. (c) Tunability of the emission PL light achieved through different composition. Reproduced with permission from X. Tang et al., Nano Energy; published by Elsevier, 2016. (d) Dual Wavelength Lasing through structure of composition-graded CsPbBr\textsubscript{3-x} NW and the graded bandgap. Reproduced with permission from L. Huang et al., Advanced Materials; published with Willey-VCH, 2018.

Table 2. Inorganic Perovskite Based Laser Device Characteristics; NA denotes for the Non-Available Information

<table>
<thead>
<tr>
<th>Material</th>
<th>Nanostructure Type / Cavity Architecture</th>
<th>Laser Threshold (\mu\text{J/cm}^2)</th>
<th>Emission Linewidth (nm)</th>
<th>Emission Wavelength (nm)</th>
<th>Q-factor</th>
<th>Gain Coefficient (cm^3)</th>
<th>Tunability (nm)</th>
<th>Pumping Source</th>
<th>Reference (Year)</th>
</tr>
</thead>
<tbody>
<tr>
<td>CsPbX\textsubscript{3}</td>
<td>QDs/WGM</td>
<td>22</td>
<td>NA</td>
<td>524.5</td>
<td>NA</td>
<td>98</td>
<td>Possible</td>
<td>100 fs @ 400 nm</td>
<td>Y. Wang et al. (2015)</td>
</tr>
<tr>
<td>CsPbX\textsubscript{3}</td>
<td>Nanocrystals / Fabry Perot Cavities and WGM cavities</td>
<td>5</td>
<td>0.15</td>
<td>530</td>
<td>10\textsuperscript{9}</td>
<td>450</td>
<td>440 – 700</td>
<td>100 fs @ 400 nm</td>
<td>S. Yakunin et al. (2015)</td>
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<tr>
<td>CsPbX\textsubscript{3} &amp; MAPb(BrCl)\textsubscript{3}</td>
<td>NWs / Fabry Perot Cavities</td>
<td>6.2</td>
<td>0.26</td>
<td>538</td>
<td>2069</td>
<td>NA</td>
<td>420 - 710</td>
<td>150 fs @ 402 nm</td>
<td>Y. Fu et al. (2016)</td>
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<tr>
<td>CsPbBr\textsubscript{3}</td>
<td>QDs / random lasing</td>
<td>0.97</td>
<td>6</td>
<td>539</td>
<td>90</td>
<td>NA</td>
<td>400 - 700</td>
<td>100 fs @ 400 nm</td>
<td>X. Tang et al. (2016)</td>
</tr>
<tr>
<td>CsPbX\textsubscript{3}</td>
<td>NWs / Fabry Perot Cavities</td>
<td>3</td>
<td>0.3</td>
<td>530 (of CsPbBr\textsubscript{3})</td>
<td>1300</td>
<td>NA</td>
<td>NA</td>
<td>100 fs @ 355 nm</td>
<td>K. Park et al. (2016)</td>
</tr>
<tr>
<td>CsPbX\textsubscript{3}</td>
<td>Thin Films / VCESL</td>
<td>9</td>
<td>0.6</td>
<td>504</td>
<td>NA</td>
<td>NA</td>
<td>400 – 700</td>
<td>100 fs @ 400 nm</td>
<td>Y. Wang et al. (2017)</td>
</tr>
<tr>
<td>CsPbX\textsubscript{3}</td>
<td>Micro-disc Arrays / Fabry Perot Cavities</td>
<td>10.3</td>
<td>NA</td>
<td>425</td>
<td>530</td>
<td>NA</td>
<td>416 – 530</td>
<td>150 fs @ 400nm</td>
<td>X. He et al. (2017)</td>
</tr>
<tr>
<td>CsPbX\textsubscript{3}</td>
<td>Micro-cubes / Fabry Perot Cavities</td>
<td>439</td>
<td>0.46</td>
<td>533</td>
<td>1100</td>
<td>NA</td>
<td>420 - 630</td>
<td>35 fs @ 800 nm</td>
<td>Z. Hu et al. (2017)</td>
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<tr>
<td>CsPbX\textsubscript{3} with ZnO NPs</td>
<td>Thin Films / Fabry Perot Cavities</td>
<td>569</td>
<td>NA</td>
<td>530</td>
<td>NA</td>
<td>NA</td>
<td>NA</td>
<td>50 fs @ 800 nm</td>
<td>C. Li et al. (2017)</td>
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<tr>
<td>CsPbBr\textsubscript{3} – PEO</td>
<td>Thin Films / DFB</td>
<td>33</td>
<td>4.9</td>
<td>654</td>
<td>NA</td>
<td>161.1</td>
<td>NA</td>
<td>90 ps @ 355 nm</td>
<td>J. Gong et al. (2017)</td>
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<tr>
<td>CsPbX\textsubscript{3}</td>
<td>NWs / Fabry Perot Cavities</td>
<td>4</td>
<td>0.2</td>
<td>535</td>
<td>2256</td>
<td>NA</td>
<td>425 - 722</td>
<td>100 fs @ 400 nm</td>
<td>X. Wang et al. (2018)</td>
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<tr>
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<td>Nanocuboids / Fabry Perot Cavities</td>
<td>40.2</td>
<td>0.29</td>
<td>539</td>
<td>2075</td>
<td>502</td>
<td>NA</td>
<td>100 fs @ 400 nm</td>
<td>Z. Liu et al. (2018)</td>
</tr>
<tr>
<td>CsPbBr\textsubscript{3}</td>
<td>NWs / Cavities</td>
<td>51</td>
<td>0.3</td>
<td>534</td>
<td>NA</td>
<td>NA</td>
<td>521 - 566</td>
<td>150 fs @ 400 nm</td>
<td>L. Huang et al. (2018)</td>
</tr>
<tr>
<td>CsPbBr\textsubscript{3}</td>
<td>NWs / Cavities</td>
<td>6 kW/cm\textsuperscript{2}</td>
<td>NA</td>
<td>532</td>
<td>2300</td>
<td>NA</td>
<td>1W, CW laser diode at 450 nm</td>
<td>T.J.S. Evans et al. (2018)</td>
<td></td>
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</table>
3. Conclusions

The area of perovskite based nanolasers is advancing since 2014 adopting various gain media and laser cavity configurations. It is obvious that the optical confinement the low dimensional perovskite material configuration secure, in combination of the superb intrinsic properties (e.g. solution processability, high absorption and emission coefficients in visible, excellent and defect free crystallinity, smooth end crystal facets, high carrier injection ratio, long bipolar diffusion lengths, low excitonic binding energy) properties and the advancements in device engineering, have replicate their potential into solar cells into laser devices. Moreover, the formation of natural optical cavities exploiting the smooth facets of the perovskite grains is another attraction this semiconductor materials offer towards the realization of simpler, of low cost and solution processed nanolaser systems. These properties have opened all the pathways for the application of these materials in low threshold, cavity free, high spectral purity and tunable stimulated emission devices.

The current optically pumped laser systems are the beginning of the trip. The aforementioned perovskite semiconductor properties and laser device performance indicate that the main challenge and ultimate target today to realize electrically pumped perovskite lasers will be a reality soon. However, other milestones should be addressed in order to evolve perovskite-based laser devices beyond research labs and garner market stakeholders. First of all, the identification of lead (Pb) free, efficient light emitters will be a breakthrough (for environmental purposes) allowing the easier commercialization of these devices. The exploitation of the high quantum yield, the spectral tunability & purity, the solution processability of the perovskite materials, will be fully exploited only if the operational stability under ambient conditions, especially in the case of organic – inorganic perovskite semiconductors, will be improved and the respective devices’ operational lifetime extended. For enhanced stability maybe 2D perovskites semiconductors are the solution. In general, 2D perovskite are more stable than 3D, but the photovoltaic performance of the 2D perovskite is still quite low, which is a challenge that should overcome for the laser devices as well. The main challenge apart of the environmental hazards (that can be addressed using high quality encapsulation coatings) the perovskite-based lasers face are the high temperatures the intense optical pumping and the laser emission generate. These high temperatures have a negative impact on the photostability of these materials. Graphene based materials can be applied in perovskite-based lasers, within the active layer or as a thin interlayer, and due to their high thermal conductivity to operate as heat sinksers and extend the lifetime of these devices. Moreover, the more efficient temperature handling will allow the optical pumping of these devices with ns laser sources reducing the operational costs and at the same time doing a step towards the continuous wave laser and electrical pumping. However more work should be invested into the understanding of the photo-degradation mechanisms of these materials in order to treat them more effectively. A step towards the realization of more stable laser gain media could be the demonstration of triple cation perovskite laser device that exhibit better stability as materials. Moreover, these materials provide a control of the achieved crystal quality as a function of the lead deficiency as well. The coupling of these materials with grating structures and the demonstration of single mode laser operation is expected to open the field of continuously tunable perovskite laser devices and the latter devices application in high demanded laser based spectroscopic applications. It is highly expected the research in perovskite-based laser systems will be intensified in the near future showing the potential of these materials beyond solar cells.

Author Contributions: Conceptualization, K.P.; Methodology, K.P.; Formal Analysis, K.P.; Investigation, K.P.; Data Curation, M.M.S. and K.P.; Writing-Original Draft Preparation, G.K. and K.P.; Writing-Review & Editing, M.M.S., T.M., A.P., G.K. and K.P.; Supervision, E.K. and K.P.; Project Administration, K.P.; Funding Acquisition, M.M.S and K.P.

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Conflicts of Interest: The authors declare no conflict of interest.

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