Reduction of Anisotropic Volume Expansion and the Optimization of Specific Charge Capacity in Lithiated Silicon Nanowires

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Abstract: This computational research study analyzes the increase of the specific charge capacity that comes with the reduction of the anisotropic volume expansion during lithium ion insertion within silicon nanowires. This research paper is a continuation from previous work that studied the expansion rate and volume increase. It has been determined that when the lithium ion concentration is decreased by regulating the amount of Li ion flux, the lithium ions to silicon atoms ratio, represented by x, decreases within the amorphous lithiated silicon (a-Li_xSi) material. This results in a decrease in the volumetric strain of the lithiated silicon nanowire as well as a reduction in Maxwell stress that was calculated and Young’s elastic module that was measured experimentally using nanoindentation. The conclusion as will be seen is that as there is a decrease in lithium ion concentration there is a corresponding decrease in anisotropic volume and a resulting increase in specific charge capacity. In fact the amplification of the electromagnetic field due to the electron flux that created detrimental effects for a fully lithiated silicon nanowire at x = 3.75 which resulted in over a 300% volume expansion becomes beneficial with the decrease in lithium ion flux as x approaches 0.75 which leads to a marginal volume increase of ~25 percent. This could lead to the use of crystalline silicon, c-Si, as an anode material that has been demonstrated in many previous research work to be ten times greater charge capacity than carbon base anode material for lithium ion batteries.

Introduction

The lithiated silicon nanowire has the potential of being a great advancement in anode material in lithium ion batteries (LIBs). As well documented in the current literature of research on LIBs, the specific charge capacity (scc) of lithiated silicon has been measured to be greater than 10 times that of carbon base anode batteries. Unfortunately the overwhelming volume expansion in excess of 300% of the silicon nanowire during lithiation has made this material ineffective for the future of lithium ion batteries due to the resulting fracture and failure of this material. This volume expansion appears to occur during full lithiation described by the lithium-silicon material Li_xSi where x = 3.75 defines the state where the lithiated silicon nanowire is at full lithiation. When this occurs the volume expands at an uncontrollable anisotropic rate where the <110> crystallography direction could possibly increase 12 times greater than that of the <111> orthogonal direction within the silicon nanowire. In this computation research work it will be demonstrated that this volume expansion could be avoided if the lithium ion flux rate is decrease where x < 3.75. The paper will draw heavily on the previous work done by Boone [1, 2] however information will come from other sources that will be indispensable in the calculation of volumetric strain and hence the volume expansion.

Analysis

For this research work, there will be a special notation that will be used throughout this study to indicate the orthogonal directions that are essential element in the presentation of this paper. As an example, the mathematical variables and functions that have directional characteristics will have subscripted notations that will indicate which orthogonal direction is being represented. For an example:

\[
A_{ij} = A_{(Orthogonal\ direction)} \quad i=j=1\ or\ 2\ or\ 3; \quad (1)
\]

\[
A_{11} = A_{(110)}
\]

\[
A_{22} = A_{(111)}
\]

\[
A_{33} = A_{(112)}
\]
The model of the computational research that will be presented is based on a single silicon diamond crystal lattice with lithium ions and electron flux diffusing in opposite directions through the stationary silicon atoms [1, 2]. There will be two main independent variables that will be utilized throughout this work. As previously discussed, the independent variable \( x \) in \( \text{Li}_x\text{Si} \) will be the lithium ion concentration for lithiated silicon where \( x \) is the ratio between lithium ion and silicon atoms. The second independent variable will be known as the average negative charge differential \( \bar{n}_c \), defined as the difference between the numbers of negative free electrons or electron flux and the number of positive lithium ions or lithium ion flux per unit volume that flows into our computational model that is being studied.

The volume expansion and geometric configuration of the silicon nanowire at full lithiation can best be demonstrated by deriving a set of equations from the Cassini oval geometry [2, 3]. In this study, the computational model simulated the nanowire volume increase slightly above 300\% upon the conclusion of lithium ion insertion. There was a volume change \( \Delta V_{ij} \) in each of the three orthogonal directions of \(<110>, <111> \) and \(<112>\). However, this study will focus exclusively on \( \Delta V_{11} \) since it was calculated that approximately 96\% of the volume increase was in that \(<110>\) direction.

\[
\Delta V_{(110)} = \Delta V_{11} = V_{\text{max}} \frac{\Delta \bar{n}_1^2}{\Delta \bar{n}_{\text{max}}^2}
\]  

(2)

with \( V_{\text{max}} \) being the total maximum volume increase and is defined as \( V_{\text{max}} = \Delta V_{11} + \Delta V_{22} + \Delta V_{33} \) (Figure 1). In terms of the maximum increase in volume, \( V_{\text{max}} \) can be thought of as being approximately equivalent to \( \Delta V_{11} \) at \( \bar{n}_c=6 \) because there is a negligible volume expansion in the \(<111>\) and \(<112>\) directions. The three volumetric strains are defined \( \varepsilon_{11}, \varepsilon_{22}, \varepsilon_{33} \) in each of the orthogonal directions. The definition of \( \Delta \bar{n}_{11} \) is the decrease in the length of the transition state vector \( \bar{n}_{ij} \) that can be further explained in [2].

\[
V_{\text{max}} = \varepsilon_{11} + \varepsilon_{22} + \varepsilon_{33}
\]  

(3)

\[
V_{\text{max}} \approx \Delta V_{11} = \varepsilon_{11} \quad \varepsilon_{22} \approx \varepsilon_{33} \approx 0 \quad \text{at} \quad \bar{n}_c=6
\]  

(4)

Since there is only one primary direction that will be analyzed, namely \(<110>\) direction, therefore Young’s Modulus \( \mathcal{Y}_{\text{Li}_x\text{Si}} \) will be used in the calculation of volumetric strain \( \varepsilon_{11} \).

\[
\varepsilon_{11} = \frac{E_{11}}{\mathcal{Y}_{\text{Li}_x\text{Si}}}
\]  

(5)

where \( E_{11} \) is the Maxwell stress equation defined in [1]. The determination of Young’s Modulus \( \mathcal{Y}_{\text{Li}_x\text{Si}} \) came from experimental data in the research paper Wang et.al [4] where a nanoindentation apparatus was used to measure \( \mathcal{Y}_{\text{Li}_x\text{Si}} \) at various concentration \( x \) values under dry and wet conditions. The values used in this research paper under the lithiated dry conditions are as follows:

<table>
<thead>
<tr>
<th>LiSi concentration</th>
<th>( x=0.75 )</th>
<th>( x=1.00 )</th>
<th>( x=1.50 )</th>
<th>( x=2.25 )</th>
<th>( x=3.00 )</th>
<th>( x=3.75 )</th>
</tr>
</thead>
<tbody>
<tr>
<td>Young’s Modulus (Pa) ( \mathcal{Y}_{\text{Li}_x\text{Si}} )</td>
<td>( 5\times10^9 )</td>
<td>( 6.3\times10^9 )</td>
<td>( 8\times10^9 )</td>
<td>( 9\times10^9 )</td>
<td>( 10\times10^9 )</td>
<td>( 12\times10^9 )</td>
</tr>
</tbody>
</table>
**Figure 1.** Most of the volume increase is in the <110> direction as displayed in the red rectangle area. Reprinted from Mathematical and Computational Applications - MDPI, October 2017. Copyright by the author under the terms and conditions of the Creative Commons Attribution (CC BY 4.0).

**Figure 2.** The maximum volume increase at $\bar{n}_c=6$ for each corresponding concentration $x$ which is equal to the ratio of the number of lithium ion to silicon atoms ($Li/Si$) for Li$_x$Si composite material. As the concentration $x$ decreases there is a corresponding decrease in the amount of volume change.
The display in figure 2 are the volume changes at different lithium ion concentration levels defined by $x$. As $x$ decreases so does the maximum volume at $\bar{n}_c=6$. The definition of the current within the lithiated silicon nanowire is defined as

$$I_{\text{current}} = \frac{\bar{n}_c e^2 \bar{E}_{11}}{a m_{Li} \omega_{Li}} \quad \text{where} \quad \bar{N}_c = \bar{n}_c \alpha^3 \quad (6)$$

The current is defined by the silicon lattice constant $a$, the electron charge constant $e$, the electric field $\bar{E}_{11}$ in the $<110>$ direction, lithium ion mass $m_{Li}$, average negative charge differential $\bar{N}_c$, and the angular frequency $\omega_{Li}$ of the lithium ion.

![Current (I)](image)

**Figure 3.** The current within the lithiated silicon nanowire for $\bar{n}_c < 6$ is approximately constant at any $x$ value. However at $\bar{n}_c \geq 6$ the electron flux experience an optical amplification through an increase of the electromagnetic field. The current increases nonlinearly as the concentration $x$ increases.

The specific charge capacity $q_{scc}$ is defined as

$$q_{scc} = (1 + \varepsilon_{11}) \frac{N_{\alpha}}{\bar{N}_{Li} m_{Li}} e \quad \text{where} \quad \bar{N}_c = \alpha \bar{N}_c \quad (7)$$

$$\alpha = \frac{\omega_{ij}}{\omega_{Li}} \quad \text{where} \quad \omega_{ij} = \frac{e \bar{E}_{11}}{\hbar (3n_c^2 \bar{n}_c)^{\frac{3}{2}}} \quad \omega_{Li} = \frac{\hbar (3n_c^2 \bar{n}_c)^{\frac{3}{2}}}{2m_{Li}} \quad (8)$$
where in addition to the parameters stated above for the current, $\bar{N}_{Li}$ is the number of lithium ions within the computational model. The parameter alpha $\alpha$ is the ratio between the electron’s angular frequencies $\omega_{ij}$ and the angular frequencies of the lithium ions. Most of the electrons are considered to be in the minimum conduction band of the quantum harmonic oscillator except for the electrons that are energetic enough to transition to a higher energy band. However, this transition is temporary that last a fraction of a second as the electron emits a photon and returns to its previous state within the minimum conduction band. In this study the alpha parameter has been calculated at an average of $\alpha = 10^3$ which is based on the range of $\alpha$ values between $\bar{n}_c=2$ to $\bar{n}_c=8$ and the average range of $x=0.75$ to $x=3.75$ of the LixSi lithiated silicon material (figure 4). This alpha of $10^3$ is when $n_c$ varies between 2 and 8 and when the average $x$ ratio of Li/Si is between 0.75 and 3.75. The average of alpha is also calculated for constant values of $n_c$ and LixSi in figure 4. As an example if the average negative charge differential is constant at $\bar{n}_c=5$ with a varying lithium ion diffusion rate between $x=0.75$ and $x=3.75$ than the corresponding average $\alpha$ is 1040. If the diffusion rate of LixSi is constant at $x=1.50$ and average negative charge differential varies between $\bar{n}_c=2$ and $\bar{n}_c=8$ than $\alpha=1240$.

Alpha $\alpha$ can be interpreted as the velocity of the electron flux being greater than the velocity of the diffusion rate of the lithium ions by the parameter $\alpha$. This can be seen by the definition of the phase velocity of electron $v_e = \omega_{ij} * k$ versus phase velocity of lithium diffusion rate $v_{Li} = \omega_{Li} * k$ where $k$ is the wave number. The $q_{sc}$ is stated below in figure 5 and 6 versus the concentration $x=Li/Si$ and volume change $\Delta V_{11}$ respectively.

![Alpha Chart](image)

<table>
<thead>
<tr>
<th>Average Li-xSi concentration</th>
<th>$x=0.75$</th>
<th>$x=1.50$</th>
<th>$x=2.25$</th>
<th>$x=3.00$</th>
<th>$x=3.75$</th>
</tr>
</thead>
<tbody>
<tr>
<td>Average Alpha $\bar{\alpha}$</td>
<td>1780</td>
<td>1120</td>
<td>929</td>
<td>707</td>
<td>609</td>
</tr>
</tbody>
</table>

**Figure 4.** In this computational study $\alpha=1000$ is used in equations 7 and 8 and is defined as the average of $\alpha$ between the range of $\bar{n}_c=2$ and $\bar{n}_c=8$ and the average lithiated silicon LixSi range between $x=0.75$ and $x=3.75$. The average value of alpha $\alpha$ is also given for the constant $\bar{n}_c=2$ through $n_c=8$ and for LixSi constant $x=0.75$ through $x=3.75$. 
Figure 5. The highest specific charge capacity $q_{ssc}$ is when $\bar{n}_c=6$ is at $x \geq 2.25$, however at $x < 2.25$ the $q_{ssc}$ steadily increases for all the $\bar{n}_c$ values. At $\bar{n}_c \leq 6$, the $q_{ssc}$ at optical amplification reaches the experimental limit of 4200 milliamp-hour per grams.

Figure 6. The specific charge capacity $q_{ssc}$ is dependent on the change of volume. As it can be seen for each $\bar{n}_c$ value, the lower the volume increase correspond to a low concentration $x$. When $x < 1.00$, the $q_{ssc}$ becomes exponentially large to the point of where the concentration is $x=0.75$. 

\[ \text{Specific Charge Capacity of Lithiated Silicon} \]

\[ \text{Specific Charge Capacity vs Volume Change} \]
Summary

As was demonstrated, the specific charge capacity $q_{sc}$ of the lithiated silicon nanowire can be increased to experimental limits and possibly beyond if the concentration $x$ is reduced during lithiation process. With the reduction of the concentration $x$ also comes a decrease in the volume expansion. The increase in volume should be less than 50% increase of the original silicon nanowire volume. If it is possible to design a lithium ion battery where the lithium ion flux concentration rate can be regulated below $x=2.25$ and at the same time work in concert with electron flux, crystalline silicon, c-Si, can be used as a great improvement in charge capacity for lithium ion batteries.

References


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