Fullerene Aggregation in Thin Films of Polymer Blends for Solar Cell Applications

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Abstract: We report on the effects of the film morphology on the fluorescence spectra for a thin film including a quinoxaline-based co-polymer (TQ1) and a fullerene derivative (PCBM). The ratio between the polymer and the fullerene derivative, as well as the processing solvent were varied. Beside the main emission peak at 700 nm in the fluorescence spectra of thin films of this phase-separated blend, a broad emission band is observed with a maximum at 520 - 550 nm. The intensity of this emission band decreases with an increasing degree of mixing in the film and becomes most prominent in thicker films, films with high PCBM content, and films that were spin-coated from solvents with lower PCBM solubility. We assign this emission band to aggregated PCBM.

Keywords: Fullerene aggregation; Organic solar cells; Fluorescence; Organic photovoltaics

1. Introduction

The interest for polymer based solar cells has grown rapidly during the last two decades. This technology shows several advantages, all related to its easy manufacturing by solution-based processes, compared to other solar energy harvesting technologies. Today single junction polymer solar cells of binary blends have reached a record efficiency of about 14 % [1-2] and tandem cells even higher, above 17% [3].

The photoactive layer consists of a mixture of an electron donating polymer and an acceptor molecule. One of the most studied class of acceptor molecules is fullerene derivatives, such as [6,6]-phenyl-C61-butyric acid methyl ester (PCBM) or [6,6]-phenyl-C71-butyric acid methyl ester (PC71BM), but more recently non-fullerene acceptor molecules or polymers gain in popularity. The donor and acceptor material are typically processed together from a chlorinated and/or aromatic solvent into a thin liquid film. The morphology of the donor/acceptor active layer is formed during deposition when the solvent evaporates and can later be altered through post-production treatments [4,5]. The morphology will affect the charge transfer from the donor to the acceptor, as well as the charge transport to the electrodes and is, consequently, a crucial factor determining the solar cell performance.

When a polymer/fullerene blend phase separates, polymer-rich and fullerene-rich domains are formed. The degree of phase separation will depend on several factors, e.g. the solubility of the two materials in the solvent [6,7], their mutual miscibility [8,9], and the rate of drying [10,11]. The interaction between the fullerene derivative and the conjugated polymer, as for instance expressed by the Flory-Huggins interaction parameters, plays a significant role in the phase separation [8]. By varying the solvent used for deposition, the solubility and the drying kinetics are affected simultaneously, and the change in morphology is complicated to predict [6,12-15]. Microscopy techniques, such as atomic force microscopy (AFM), are very suitable to study domain structures of 50 nm up to several micrometre in diameter. For systems where liquid-liquid phase separation prevails, coarser domain structures will be achieved on if the drying time is prolonged by e.g. slower spin-coating speeds [10,11] or solvent annealing. Blends of a quinoxaline-based donor polymer (TQ1)
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[16] and PC70BM (Figure 1) [6, 17, 18], has shown to be a material combination that yields solar cell efficiencies up to 7% [19].

In order to access information about the structures on the molecular scale, it is of importance to combine microscopy techniques with spectroscopy, e.g. fluorescence spectroscopy, which yields information on the molecular scale. Here we report on the aggregation of PC70BM, as studied by fluorescence spectroscopy. Fluorescence spectroscopy has successfully been used for characterisation of these [17,18] or other [20,21] blends earlier. The optical properties are correlated with the phase separated domain structure as imaged by AFM.

2. Materials and Methods

PC70BM (purity 99%) was purchased from Solenne, The Netherlands. TQ1 was prepared according to previously published procedures [16]. TQ1 had a number-average molecular weight of $M_n \sim 34,000$ (polydispersity index $\sim 2.7$) as measured with size exclusion chromatography with a polystyrene standard. Chlorobenzene (CB) and chloroform (CF) (analytical grade) were purchased from Merck, ortho-dichlorobenzene (oDCB) (analytical grade) was purchased from Sigma-Aldrich. All solvents were used as received.

Solubility measurements

Saturated solutions of PC70BM were prepared in the three solvents. After equilibration, the solutions were thoroughly centrifuged and an aliquot of known volume of the supernatant was withdrawn. The saturation concentrations were estimated by gravimetric analysis of the dry content after complete solvent evaporation.

Thin film preparation

Thin films were spin-coated from CF (12.5 mg/ml), CB (20 mg/ml), or oDCB (25 mg/ml) solutions. Concentrations in brackets are the total material concentration in solution in which the ratio between TQ1 and PC70BM was altered. Solutions were gently heated prior to spin-coating. For fluorescence spectroscopy and AFM, silicon wafers were used as substrates, cleaned using the standard RCA-method [22-24], without the final HF-etching step, so that a clean and hydrophilic surface is achieved.

The film thickness was approximately 90 nm (± 10 nm), as measured by AFM, if nothing else is stated.

Steady-state fluorescence measurements

Steady-state fluorescence spectra were collected on a SPEX FL3-11 TAU fluorimeter, purchased from Gammadata (Sweden). Measurements were performed on thin films in front-face mode at 22° angle relative to the incident excitation light, in order to prevent reflected excitation light. The excitation wavelength was 380 nm. All samples were kept in the dark prior to the measurements and the spectra were recorded at room temperature and in ambient air.

Atomic force microscopy (AFM)

Images were collected in tapping mode with a Nanoscope IIIA Multimode AFM (Bruker) using a TESPA-V2 n-doped silicon cantilever.

3. Results and Discussion

3.1. Solubility of PC70BM

The solubility of PC70BM in oDCB was determined by gravimetric analysis to be 66 mg/ml, compared to 56 mg/ml in CB and 34 mg/ml CF [18]. The vapour pressures of the solvents are 0.14 kPa, 1.5 kPa, and 26.3 kPa for oDCB, CB, and CF, respectively [25,26].

3.2. Fluorescence measurements

To study the relationship between morphology and optical properties, both the solvent for deposition and the ratio between TQ1 and PC70BM were altered. It has been shown earlier that the phase separated domains become larger when the fraction of PC70BM is increased in the blend [17] and that the domain size depends on the choice of solvent, for films with equal thickness [6]. Hansson et al. have shown, by STXM measurements, that the composition of the phase separated domains is rich in PC70BM and, by AFM imaging, that the domains are larger when CF is used as solvent than
when oDCB is the solvent [6]. The AFM images shown in Figure 2 show that the phase-separated, PC\textsubscript{\textcircled{BM}}-rich, domains become larger when a less good solvent for the fullerene derivative is used, in line with the earlier findings. This is more pronounced for the 1:3 blend than for the 1:1 blend in the series of decreasing PC\textsubscript{\textcircled{BM}} solubility, i.e. oDCB, CB, and CF. Emission spectra, Figure 3, obtained upon excitation at 380 nm, of the blend films show, apart from the main TQ1 emission peak at 700 nm [17,18], a very broad emission band with a maximum between 520 nm and 550. For pure TQ1 in solution, this emission band is absent when TQ1 is dissolved in CF, CB or oDCB [18], which is also found for pure TQ1 films (spectrum not shown). The only emission in the visible region from TQ1 dissolved in those solvents is found around 650-700 nm, thus slightly blue-shifted in comparison with the emission from TQ1 films [17,18]. For TQ1:PC\textsubscript{\textcircled{BM}} blend films, we have shown that when the PC\textsubscript{\textcircled{BM}}-rich domains grow, the intensity of the emission band between 500 and 550 nm increases [18].

A similar trend in the emission peak as in the previous studies is observed if the amount of PC\textsubscript{\textcircled{BM}} was increased (Figure 3a) or if a less good solvent for PC\textsubscript{\textcircled{BM}} (CF) is used (Figure 3b). Further, the maximum of the 500-550 nm emission band shifts to longer wavelengths, i.e. from approximately 520 nm to 550 nm, Figure 3. Cook et al. showed that films of pure PC\textsubscript{\textcircled{BM}} give rise to a broad emission band at approximately 500 nm, which is absent when the fullerene derivative is well distributed in a polystyrene matrix (1:7 fullerene:polystyrene) [27]. They assign this emission to the occurrence of aggregated PC\textsubscript{\textcircled{BM}}. In addition, Jamieson et al. observed this peak in films of blends with a high fullerene content in a semi-conducting polymer matrix relevant for solar cell applications [28]. The emission peak observed in the present PC\textsubscript{\textcircled{BM}}-based blends seems to be similar to the one reported for PC\textsubscript{\textcircled{BM}}. This series of results indicates that the 520-550 nm emission band in strongly phase-separated TQ1:PC\textsubscript{\textcircled{BM}} is likely to emanate from aggregated PC\textsubscript{\textcircled{BM}}, and that samples with larger domains include a larger portion of aggregated PC\textsubscript{\textcircled{BM}}. The nature of these aggregates is not clear from these results. Earlier literature suggest that nanocrystals of fullerenes can be formed upon spin-coating [29]. Probably, the shift from 520 nm to 550 nm in emission maximum is due to a gradual development of the aggregates from a less organised cluster of PC\textsubscript{\textcircled{BM}} molecules (in films coated from CB and in films with 1:1 ratio) to more compact aggregates in films with 1:3 ratio or in films prepared from CF. For the present results, we cannot draw conclusions about the location of the PC\textsubscript{\textcircled{BM}} aggregates, even though it is likely to assume that they are formed inside the PC\textsubscript{\textcircled{BM}}-rich domains.

To further underpin this interpretation, films of the 1:3 TQ1:PC\textsubscript{\textcircled{BM}} blend with different thicknesses (50-130 nm) were prepared [11] by changing the spin-coating speed, (Figure 4). Thicker films, resulting from slower spin-coating, demonstrated larger PC\textsubscript{\textcircled{BM}}-rich domains, as shown in the AFM images (Figure 4a). The corresponding emission spectra (Figure 4b) show that when the film thickness is increased from 90 nm to 130 nm, the intensity of the emission band increases and the emission maximum shifts from 520 nm to about 540 nm. Decreasing the film thickness from 90 nm to 50 nm does not have a significant effect on the emission band. This confirms that the domains have to reach a certain size before the aggregate emission becomes detectable.
3.3. Figures

Figure 1. Chemical structure of TQ1 and PC$_{60}$BM.
**Figure 2.** AFM micrographs (5×5 µm) of thin films of 1:1 TQ1:PCBM (a-c) and 1:3 TQ1:PCBM (d-f) spin-coated from CF (a,d), CB (b,e), and oDCB (c,f). Scale bar indicates 1 µm. Surface roughness (RMS) as well as the corresponding height scale is added to each image.
Figure 3. Fluorescence emission spectra of thin films measured with $\lambda_{ex} = 380$ nm of (a) 1:1 (black) and 1:3 (red) TQ1:PC$_{70}$BM spin-coated from CF (b) 1:3 TQ1:PC$_{70}$BM spin-coated from CF (red), CB (black), and oDCB (blue).
Figure 4. (a) AFM micrographs (5×5 µm) of thin films of 1:3 TQ1:PCBM spin-coated from CB with the indicated film thicknesses. A height scale is added to each image. (b) Corresponding fluorescence emission spectra for films with indicated thicknesses.
4. Conclusions

In this study we have investigated the relationship between the detailed emission spectra analysis of TQ1:PC_{70}BM blends and the morphology of these blends. Solvents with lower PC_{70}BM dissolution capacity as well as an increased PC_{70}BM concentration yield larger phase-separated domains. In these larger domains, PC_{70}BM aggregates are prone to be formed, resulting in a broad and rather weak emission between 520 and 550 nm upon excitation at 380 nm. For better PC_{70}BM solvents and at lower PC_{70}BM concentrations, the fullerene derivative stays well-mixed with the polymer and no emission between 500 and 550 nm is detected. Further investigation by, e.g. high resolution microscopy techniques is needed to determine the sizes and locations of these aggregates.

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