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Transparent and water-resistant composites prepared from acrylic resins ABPE-10 and acetylated NFC as FOLED substrate

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15 Abstract: The ANFC/ABPE-10 composite film was prepared by ABPE-10 impregnating into ANFC 16 films under negative pressure. And the composite film had meted high performance FOLED 17 substrate requirement even when ANFC dosage was high for approximately 70%, which was 18 consistent with for low cost efficiency, recyclability, and environmentally friendly. The enhanced 19 properties of ANFC films were mainly because of the nature of ABPE-10 itself and the IPN 20 structure formed between ABPE-10 and ANFC film. The transparency of composite films with 21 different ANFC dosage was significantly increased from 67% to 88% by UV-Vis analysis. The 22 composite film inherited the properties of AFNC, obtaining low CTE characteristics and ductile 23 compact structure. The contact angles of ANFC films increased by 102% from 49.2° to 102.9° after 24 dipping ABPE-10. Additionally, the composite films had outstanding mechanical properties such 25 as tensile strength 173.72 MPa, Young's modulus 4.06 GPa, and elongation at break 5.81%.

Keywords: Acetylated nanofibrillated cellulose; Acrylic resins ABPE-10; Composite films; Flexible
 organic light-emitting device substrate; Interpenetrating polymer network

28

29 1. Introduction

30 Organic light-emitting diode (OLED) provide unique features such as wide viewing angle, high 31 efficiency, low power consumption, high response speed, and low cost, which make to develop 32 highly portable and flexible OLED possible [1,2]. Flexible OLEDs are typically made of the anode, 33 cathode, and organic material films, including electron injection layer, organic emitting layer, hole 34 injection layer between anode and cathode, as well as at least a transparent electrode in order to get 35 the shiny surface [3,4]. Flexible organic light-emitting diode (FOLED) has broad application 36 prospects in information, energy, healthcare, defense, and other fields because of its flexible, high 37 efficiency, and low fabrication cost [5]. Meanwhile, FOLED display will have a profound impact on 38 the application of wearable and portable devices, and will be widely used with the continuous 39 development of personal intelligent terminals [6]. With the growing sophistication of organic 40 light-emitting materials and device technologies, FOLED is regarded as the most promising 41 technology for future displays [5,7].

42 FOLED is fabricated based on flexible substrate, and is operated in flexible substrate, so the 43 FOLED substrate is a very important part of flexible display device. It needs to provide mechanical 44 support for the equipment, at the same time to facilitate processing photonic and electronic to 45 achieve the specific performance of FOLED equipment, thus the quality of the substrate will 46 ultimately determine the performance of device, life expectancy and production methods. FOLED 47 substrate should have the advantages of smooth, flat, good dimensional stability, high flexibility, 48 good thermal stability, high transparency, and barrier properties [7]. What's more, substrate 49 materials would have an impact on the following processes: electrode deposition, barrier coating, 50 patterning exposure, and thin-film transistor fabrication. Therefore, how to select the proper 51 substrate materials is very critical. In the current FOLED substrates, some synthetic soft polymer 52 matrix are expected to replace the glass substrate, while the large coefficient of thermal expansion 53 (CTE) limit their use. Cellulose is the most abundant renewable raw materials on earth. The 54 nanofibrillated cellulose (NFC) that is prepared from natural fibers via mechanical grinding is 55 composed of aggregate of linear microfibrils with higher aspect ratio. It has the tendency of 56 self-linking through hydrogen bonding interaction between fibers, and the three-dimensional rigid 57 network can be formed within matrix [8]. At the same time, NFC has the advantages of good 58 flexibility, non-friable, low CTE $(8 \times 10^6 / K)$, light weight (density 1.5 g/cm³), large surface area 59 (>50 m²/g), high mechanical properties, and biodegradable, compared with conventional glass 60 materials. What's more, the thin film prepared from NFC is transparent and has high strength [9], 61 which has great application potential in high technology fields such as conductive NFC film [10] and 62 electronic substrate [11]. When the addition amount of NFC is low (\leq 5%), NFC composites as 63 FOLED substrates still has superior thermal stability, mechanical property, barrier property, and 64 recyclability [12]. In addition, the sheet-by-sheet processing of FOLED substrate can be replaced by 65 large-scale roll-to-roll processing [13]. Therefore, the cost-efficient and environmentally friendly 66 FOLED substrate produced by NFC is attracting intensive research and commercial interesting.

67 Our previous studies also found that acetylated nanofibrillated cellulose (ANFC) films as 68 FOLED substrates had many advantages such as smooth surface, high flexibility, good thermal 69 properties, and mechanical properties. However, the light transmittance of NFC films was about 70 70% and had not reached the requirement of FOLED substrates 80% [7], and its water resistance also 71 needed to be enhanced. Acrylic resins ABPE-10 is a kind of colorless, transparent, light-cure, and 72 water-insoluble resin. Compared with the traditional heat cure, light cure technology has the 73 characteristics of fast curing speed, high production efficiency, good physical mechanical properties, 74 and curing at room temperature [14]. Also, the prices of light cure equipment and energy 75 consumption are low, so acrylic resins ABPE-10 is known as an environmentally friendly green 76 material. Okahisa et al. [11] impregnated NFC film in acrylic resins and tetrahydrocyclopentadiene 77 dimethacrylate, and obtained the composite material with high transparency. Nogi and Yano et al. 78 [15,16] prepared flexible substrate by compositing bacterial-cellulose nanofibers films and acrylic 79 resins, also the nanocomposites had excellent performance in optical properties, dimensional 80 stability, and thermal performance. However, these studies rarely describe the combination 81 relationship between ANFC film and acrylic resins ABPE-10, as well as rarely report how to improve 82 the transparency of NFC films produced by mechanically ground NFC for preparing the highly 83 transparent FOLED substrates. Furthermore, the combination of ANFC film and ABPE-10 to 84 improve the water resistance of ANFC membranes as FOLED substrates has also been rarely 85 studied.

86 The ABPE-10 contains carboxylic acid ester and phenyl structure (in Fig 1), belonging to the 87 amphiphilic substance. The carboxylic acid ester structure of ABPE-10 can form hydrogen bonding 88 with the hydroxyl groups (-OH) in the NFC, enhancing the binding between NFC film and ABPE-10 89 , and improving the evenness of ABPE-10 on the surface of NFC film. On the one hand, the phenyl 90 structure of ABPE-10 after drying can also improve the oxidation resistance and water resistance of 91 FOLED substrate. Additionally, acrylic resins ABPE-10 contains double functional groups 92 (-CH=CH₂) structure, which can accelerate UV-light polymerization speed. In order to further 93 improve the light performance, smoothness, and water resistance of FOLED substrate based on

94 ANFC, we produced the composites of ANFC/ABPE-10 with high ANFC content by simulating the 95 papermaking process, which would provide a low cost-efficient and environmentally friendly 96 method to prepare FOLED substrate. The ANFC film and ABPE-10 interaction mechanism in 97 composite materials are shown in Fig 1. The acrylic resin ABPE-10 with photoinitiator is dipped and 98 swelled into the ANFC film by negative pressure, and the ABPE-10 polymerized in situ under 99 UV-light crosslinks immediately with ANFC film to form a composite film. The three-dimensional 100 network structure is composed of ANFC network and ABPE network, which are partially interlaced 101 on a molecular scale by hydrogen bonding but not covalently bonded to each other. Thus, the 102 structure of ANFC/ABPE-10 composite film should be interpenetrating polymer network (IPN).

103 In this study, the ANFC film was impregnated in acrylic resins ABPE-10 under the condition of 104 negative pressure, then was cured by UV-light. Herein, transparent and bendable ANFC/ABPE 105 composite film with high ANFC content (approximately 70%) was prepared as FOLED substrate. 106 The combination relationship between ANFC film and ABPE-10 was illustrated, also the 107 transmittance and water resistance of ANFC film as FOLED substrate were improved.





Fig 1. The schematic representation of IPN ANFC/ABPE-10 composite film

109 2. Materials and Methods

110 2.1. Materials

Bleached softwood kraft pulp (Pinus khasys), provided by Yun-jiang Forestry & Pulp Mill Co., ltd. (China, Yunnan), contains 96.90% cellulose, 3.50% hemicellulose, and less than 0.1% lignin. It was utilized as the raw material. 1-hydroxycyclohexyl phenyl ketone (analytically pure) as photo initiator, 2-hydroxyethyl acrylate (analytically pure) as reactive diluent, and 2,2-bis[4-(acryloxy polyethoxy) phenyl] propane (acrylic resins ABPE-10) were obtained from the company of Shin Nakamura Chemical Co., Ltd. (Tokyo, Japan).

117 2.2. Preparation of ANFC

The 3 wt% pulp was ground through a grinder (Super Masscolloider MKZA 10-15JIV; Masuko Sangyo Co., Ltd., Saitama, Japan) at 1,500 rpm for 30 min. Then, the pulp after grinding was diluted to 0.2 wt% pulp suspension and passed through a high-pressure homogenizer (GJJ-0.06/40; Keju fluid equipment manufacturing Co., Ltd., China). The conditions of homogenization were followed: passing 2 times at 0 bars, passing 3 times at 400 bars, and passing 3 times at 600 bars.

123 The NFC suspension was replaced repeatedly with acetone through vacuum filtration in order 124 to obtain the NFC acetone suspension. Similarly, the NFC toluene suspension was obtained. NFC 125 acetylation was conducted when 25 mL toluene, 20 mL acetic acid, and 0.1 mL perchloric acid were 126 putted into the NFC (83 wt%, 1.0 g bone dry) in sequence. Then, 3 mL acetic anhydride was added 127 to the NFC and was stirred continuously for 1 hour at room temperature. After acetylation, the

128 ANFC was washed thoroughly with ethanol and distilled water by centrifugal separation,129 respectively.

130 2.3. Preparation of ANFC films

131 The ANFC slurry (0.2, 0.3, 0.4, 0.5, 0.6 g bone dry) was diluted to 0.2 wt%, respectively. And the 132 suspension of diluted ANFC was stirred for 2 hours in order to ensure its dispersion, respectively. 133 Then, the dispersed ANFC was vacuum filtered with G2 sand core funnel (90 mm diameter), which 134 was padded with a layer of hydrophilic polyterafluoroethylene organic filter membrane (0.22 um 135 pore size, 90 mm diameter) in advance. The wet ANFC film was taken out together with the organic 136 filter membrane after filtering, covering another organic filter membrane on the other side of ANFC 137 film. Also, the filter papers were covered on the surface of organic filtering films before drying in 138 order to speed up the removal of moisture. The wet ANFC film was pressed from both sides with 139 glass in order to obtain flat film during the drying. Then the film was dried at room temperature for 140 12 h, and moved to a vacuum drying of 55° C for 24 h.

141 2.4. Preparation of IPN ANFC/ABPE composite films

142 The ANFC films were impregnated in acrylic resins ABPE-10 under a pressure of -0.09 MPa for 143 12 h. ANFC films were taken out after impregnating, and were used a small coating machine to 144 scrape the extra ABPE-10 on the surface. Then the films were solidified for 3 minutes under the UV 145 light of 1000 W, and the ANFC/ABPE composite films were obtained. At last, the films were 146 balanced more than 24 h under 25° C and 50% humidity.

147 2.5. Analysis

148 2.5.1. Transmission Electron Microscopy (TEM)

The morphology, dimension, and yield of NFC and ANFC were studied by TEM at 100 kV (TECNAI G2 F30, USA). The concentration of NFC and ANFC suspension was diluted to 0.01 wt%, and dispersed for 30 min with ultrasonic, respectively. A small amount of diluted dispersion was carefully dropped on a 200 mesh copper net. After drying at room temperature, sample was dyed with phosphotungstic acid stain for 20 min. The morphology distribution and particle size analysis of samples were carried out using the analyzing software of Nano Measurer 1.2.5.

155 2.5.2. Determination of acetylation degree

The degree of substitution (DS) of samples were measured by ¹H-NMR spectroscopy (Bruker AC III HD600, Germany) [17]. ¹H-NMR spectra were measured with an aspectrometer by tetramethylsilane as the internal standard at 256 scanning times and 500 MHz. The sample was dissolved in DMSO-d6, and the DS was calculated according to the formula of Goodlertt [18].

160 2.5.3. Optical properties

161 Using the single lens reflex (SLR) camera nikon d7100, the apparent transmittances of films 162 were photographed in a well-lit laboratory. A UV-Vis Spectrometer Lambda 950 (PerkinElmer, 163 USA) was used to measure the light transmittance of films in visible wavelength range from 380 nm 164 to 780 nm. The samples were cut into 10 mm×10 mm, and placed at 25 cm from the outlet of the 165 integral sphere.

166 2.5.4. Apparent morphology

167 The apparent morphology and tensile fracture-surfaces of the samples were sprayed with gold, 168 then were observed by the field emission scanning electron microscopy (FE-SEM, Hitachi 169 high-technologies corporation SU 8020, Japan).

170 *2.5.5. Contact angle*

171 The contact angles were measured using contact angle meter (DSA100, Germany). The 172 specimen dimensions were 40 mm in length and 10 mm in width. A droplet of distilled water was 173 deposited on the flat film. The contact angle was measured on 3 different points and the average 174 values were calculated.

175 2.5.6. *Thermal performance*

The CTE of films was measured by a thermomechanical analyzer (Q400, TA Instruments, U.S.). The measurement conditions were followed: specimens area 25 mm \times 3 mm, pull 0.03 N, temperature from 30°C to 150°C with the heating rate of 5°C/min. The test was conducted under nitrogen condition, and each sample was circulated three times. The CTE values were determined by the average value of the second run and the third run in order to eliminate the residual stress of the membrane material. CTE values were given as the average of three independent determinations for each sample.

183 2.5.7. Mechanical properties

The Young's modulus, tensile strength, and elongation at break of the samples were measured using a Shimadzu AG-X testing machine (Kyoto, Japan). The specimen dimensions were 25 mm in length and 3 mm in width. The measurement conditions were followed: load sensor 50 N, gauge length 20 mm, and stretching rate 1 mm/min. Three test samples were measured as well as the data reported were an average of tests.

- 189 3. Results and Discussion
- 190 3.1. Characteristic analysis of ANFC

191 TEM micrographs of NFC and ANFC were displayed in Fig 2, and both had a great aspect 192 ratio. The dimensions of NFC and ANFC were obtained by measuring at least 100 individuals from 193 the TEM images. From the Fig 2 (a), there was more than 80% diameter of individual NFC 194 estimated to be within the range of 5-30 nm. Approximately 80% diameter of ANFC had an average 195 between 5-20 nm in Fig 2 (b). The result that acetylation had a little effect on fiber dimension was 196 similar to the finding of Jonoobi et al. [19]. In our earlier study, ANFC film had the best 197 performance when the DS of ANFC was 0.24 shown in Fig 2 (b), and more detailed analysis were 198 described in our previous work [7].



205 3.2. The effects of different ANFC dosages on the optical properties of composite films

Transparency is the most crucial property for bottom emissive displays, and 80% total light transmission of FOLED substrate in the visible light range of 380-780 nm is required [20,21]. The composition and thickness of the composite films are described in the Table 1.

	NFC (%)	ANFC (%)	ABPE (%)	Thickness (um)
Neat ABPE	—	—	100	42
Neat NFC	100	_	_	42
Neat ANFC	—	100	—	42
ANFC/ABPE	—	53	47	42
ANFC/ABPE	—	59	41	42
ANFC/ABPE	—	68	32	45
ANFC/ABPE	—	74	26	47
ANFC/ABPE	_	77	23	48

209

Table 1. The composition and thickness of the films

The transparency of samples such as neat ANFC film, neat ABPE-10 film, and ANFC/ABPE-10 composite films were illustrated by photographing and UV-Vis Spectrometer analysis (in Fig 3). At the same time, since the ANFC/ABPE-10 composite films with different ANFC dosages had similar apperance images, only the ANFC/ABPE-10 composite film with ANFC dosage 68% was presented as a representative in here. From the Fig 3, although the bee on the red rose of background image could be displayed through these films, the transparency of (a) neat NFC film and (b) neat ANFC film was lower than that of (c) neat ABPE-10 film and (d) ANFC/ABPE-10 composite film, and the

transparency of (c) neat ABPE-10 film was slightly higher than that of (d) ANFC/ABPE-10 film.



Fig 3. The effects of different ANFC dosages on the transparency of the samples, and the apparent photographs of (a) neat NFC film, (b) neat ANFC film, (c) neat ABPE-10 film, and (d) ANFC/ABPE-10 composite film.

As shown in Fig 3, the transparency of neat NFC film was the lowest compared to others films, followed by the neat ANFC film, indicating that acetylation could slightly improve the transparency of films. The transparency of ANFC/ABPE-10 composite films with different ANFC dosage, compared with that of neat ANFC film, was significantly increased from 67% to 88% at a wavelength of 600 nm at about 45 um thickness, indicating that the combination of ABPE-10 and ANFC was very good for improving the transparency of ANFC film. The main reason was that the ABPE-10 filled the gap of ANFC film and decreased the surface roughness of film, suppressing the

228 scattering of photons on the surface of film and resulting in the low scattering index [16]. The 229 results were comparable with the findings of Okahisa et al. [11]. Mimicing the light scattering path 230 on different surfaces of films are shown in Fig 4. With the increase of ANFC dosage in the 231 composite film, the transparency of ANFC/ABPE-10 composite films was decreased to varying 232 degrees. It was because that the dense networks structure in films was formed through hydrogen 233 bonding between fibers, which made it difficult to impregnate more acrylic resins ABPE-10 (as 234 shown in Table 1.). The results led ABPE-10 not to well fill the surface porosity of ANFC film and 235 decreased its surface soomthness, and further causing different degrees of light scattering. At the 236 same time, the increase of ANFC dosage led to a slight increase in the thickness of films, eventually 237 also causing a fall in the transparency of composite films. But when the ANFC dosage in composite 238 film was less than 68%, the transparency of ANFC/ABPE-10 composite film was still attained to 239 80%, meeting the requirement of FOLED substrate. Compared with what Okahisa at al. studied, the 240 ANFC dosage was low around 35%-40% when the transparency of composite films were similar to 241 us [11]. Considering that the cost efficiency, recyclability, and environmentally friendly, the dosage 242 of ANFC should be increased as much as possible without prejudice to transparency requirements.



- Fig 4. The light scattering path of films: (a) neat ANFC film, (b) neat ABPE-10 film, (c) ANFC/ABPE-10
 composite film (68% ANFC)
- 245 3.3. The effects of different ANFC dosages on the apparent qualities of composite films

246 Some surface qualities including roughness, cracks, and cleanliness are important to guarantee 247 the integrity of subsequent barrier and conductive layers. The cracks in substrate may lead to the 248 form of pinholes on the thin films of electrode, creating dark spots in OLEDs. Also, the defects of 249 cracks would be serious when the displays were bent. In order to determine the apparent 250 morphology, stable performance, and the bonding degree of ANFC and ABPE-10, both the 251 apparent morphology and tensile cross-section of films were analyzed by FE-SEM. Since the 252 apparent morphology, tensile cross-section, and contact angle of ANFC/ABPE-10 composite films 253 with different ANFC dosages had similar results, only the ANFC/ABPE-10 composite film with 68% 254 ANFC dosage was present as a representative in here. The results are shown in Fig 5.









279 It could be seen from Fig 5 that the surface of (c) neat ABPE-10 film was most smooth and 280 cleanliness, followed by (d) ANFC/ABPE-10 composite film, under the same magnification of 1.0 K 281 times. Also, some small cracks were observed on the surface of (b) neat ANFC film. Compared with 282 (b) neat ANFC film, the surface smoothness of (d) ANFC/ABPE-10 composite film was greater, and 283 the surface crack had been largely filled, but the surface smoothness was slightly lower than the (c) 284 neat ABPE-10 film. On the basis of these results, we concluded that acrylic resins ABPE-10 could fill 285 the gap of ANFC film well and improved the smoothness of ANFC film. The results further 286 confirmed that acrylic resins ABPE-10 improved the transmittance of the composite film, which was 287 consistent with the observation from Fig 3.

288 Through the observation for tensile cross-section of (g) neat ABPE-10 film, it could be found 289 that the cross-section was smooth and flat. Also, the fracture direction of the cross-section was 290 consistent and emerged obvious rigidity structure. Under the external load, the ability of such 291 materials to resist being torn and bent was small, leading to rapid expansion of cracks according to 292 the cracks propagation direction. These features of the material presented a typical characteristic of 293 brittle fracture [22]. Thus, neat ABPE-10 film was not suitable for preparing FOLED substrate. 294 Comparing with the image of (e) neat NFC film, the (f) neat ANFC film could be more clearly found 295 that the fracture surface was separated into flakes, scales, and layers, which proved that the fracture 296 was a result of ductile tearing and the ANFC film was tougher material. Fortunately, the composite 297 films prepared by brittle ABPE-10 and ductile NFC/ANFC presented obvious ductile tearing in the 298 Fig 5 (h) and (k), which was due to the fact that the fiber had some outstanding properties including 299 good expansivity, superior flexibility [23], abundance hydrogen bonds, and the three-dimensional 300 network structure of the NFC itself, delaying the breakage. The results would be beneficial to obtain 301 excellent strength performance of FOLED substrate. At the same time, some holes appeared on the 302 cross-section of NFC/ABPE-10 composite film in Fig 5 (h), indicating that the binding of unmodified 303 NFC with ABPE-10 was slightly inferior, compared with the ANFC/ABPE-10 composite film in Fig. 304 5 (k). The compact structure, flakes state, and wiredrawing shape were more obvious and 305 outstanding in ANFC/ABPE-10 composite film than that of Fig 5 (h), which indicated that

acetylation improved the compatibility of NFC and ABPE-10. Furthermore, the structure of the composites in Fig 5 (k) showed the IPN structure and presented good interaction and compatibility with the matrix. This structure was often used in the preparation of nanocellulose-based IPN hydrogels with good moisture stability and mechanical performance [24]. Obviously, the ductile compact structure of this substrate would greatly reduce the formation of pinholes on the thin films of electrode, and strengthen the bending performance of displays.

312 At the same time, acrylic resin ABPE-10 also significantly improved the water resistance of 313 ANFC film. The contact angle of ANFC film increased by 102% from 49.2° to 102.9° after dipping 314 ABPE-10, as shown in Fig 5 (m) and (n). On one hand, the higher water resistance of 315 ANFC/ABPE-10 composite film was due to superior hydrophobic property of acrylic resin 316 ABPE-10. On the other hand, acrylic resin ABPE-10 covered the surface of ANFC film and filled the 317 gaps between nanofibers, which contributed to reduce the hydrophilicity of cellulose by decreasing 318 the exposure of surface hydroxyl. Moreover, the capillary effect of the fiber surface was severely 319 weakened as a result of the decrease of microtube of fiber itself and the interfiber pore. We even 320 ventured to guess that these results should also improve the gas barrier property of ANFC film in a 321 way.

322 3.4. The effects of different ANFC dosages on the thermal performance of composite films

Thermal stability of substrate is also an important issue that needs to be solved. Improved dimensional stability of composites is desired for FOLED substrate thanks to its varying temperature during application and process. A low CTE is also beneficial to make dimensionally stable designs for devices, and the CTE of FOLED substrate should be less than 20 ppm/K [11,20]. So, the CTE of samples need to be evaluated by thermomechanical analyzer (TMA) and be shown in Table 2.

329

Table 2. The effects of different ANFC dosages on the CTE of different films

ANFC dosage (g)	CTE (ppm·k-1)
Neat ABPE-10	128.40 ± 2.47
Neat NFC	15.05 ± 0.68
Neat ANFC	5.43 ± 0.35
53% ANFC	15.32 ± 1.10
59% ANFC	15.37 ± 0.75
68% ANFC	13.26 ± 0.58
74% ANFC	11.25 ± 0.48
77% ANFC	10.91 ± 0.50

330 As shown in Table 2, the CTE of neat ANFC film was about 5.43 ppm/K, and the thermal 331 stability was more excellent compared to that of neat NFC film. The CTE of neat ABPE-10 film was 332 extremely high for 128.40 ppm/K approximately, which would lead to unstable dimensions during 333 application. However, the CTE of the ANFC/ABPE-10 composite films had fallen sharply compared 334 to neat ABPE-10 film, ranging from 128 ppm/K to 11 ppm/K, which could be comparable to glass 335 [25]. Simultaneously, when the dosage of ANFC in ANFC/ABPE-10 composite film was increased 336 from 53% to 77%, the CTE of composite films were slightly decreased. These results indicated that 337 the dosage of ANFC with low CTE characteristics played an important role in the decrease of CTE 338 of composite films. This was because that the thermal stability of composites mainly depended on 339 the thermal properties of material and the enhancer [20]. Additionally, the three-dimensional 340 network structure of composite films was strengthened with the increase of ANFC weight 341 percentage, further causing the variation of composites' density [26], thus limiting the thermal 342 expansion of the composite films.

343 3.5. The effects of different ANFC dosages on the mechanical properties of composite films

The high mechanical properties of FOLED substrate are of great significance to keeping the robustness and flexibility of FOLED, and meeting the preparation requirement of roll-to-roll [27]. Fig 6 and Fig 7 show the mechanical properties and flexibility of different films, respectively.





354

Fig 6. The effects of different ANFC dosages on the mechanical properties of different films



Fig 7. Flexibility of different films

355 The ultimate strength of composite materials, especially nanocomposites, mainly depends on 356 the nature and volume fraction of the component, and the adhesiveness and compatibility of 357 polymer matrix and additives, known by the study of Dufresne [28]. As shown in Fig 6, compared 358 with the neat ABPE-10 film, the mechanical properties of ANFC/ABPE-10 composite films were 359 improved dramatically. And when ANFC dosage increased to 68%, the tensile strength, Young's 360 modulus, and elongation at break of the composite film increased by 3.94 times, 9.68 times and 1.47 361 times, respectively. Compared with the neat ANFC film, both tensile strength and elongation at 362 break of ANFC/ABPE-10 composite films were significantly improved. This was mainly due to the 363 fact that the formation of IPN structure between fiber and ABPE-10 strengthened the stability of 364 composite films. Importantly, those tiny pores and cracks on the surface of ANFC film were filled 365 by ABPE-10 when the film was impregnated in ABPE-10 under a certain pressure, enhancing the 366 compactness of film matrix, and thus improving its mechanical properties, which was similar to the 367 findings of Yoko Okahisa [11]. However, when the ANFC dosage was more than 68%, the 368 mechanical properties of composite films had fallen sharply. One of main reasons was that the 369 cracks and holes on the surface of ANFC film could not be well covered with the reduced 370 adhesiveness of ANFC and ABPE-10, with the decreasing of ABPE-10 dosage in the composite film

(shown in table 1), eventually resulting in the non-uniform force and deteriorating the mechanical properties. Additionally, both the neat ANFC film and ANFC/ABPE-10 composite film presented an outstanding flexibility in Fig 7. On the contrary, the flexibility of neat ABPE-10 film was bad, which brought about the breakage of film, which were consistent with the research of Nogi et al. [29]. Notably, this was also supported by the apparent morphology and tensile cross-section of films in Fig 5.

377 3.6. Performance comparison of different polymer FOLED substrates

378

Table 3. Performance comparison of different polymers FOLED substrates [16,30,31]

OLED	tensile strength	Young's	elongation at	transmittance	CTE
substrates	(MPA)	modulus (GPA)	break (%)	(%)	(ppm • K-1)
ANFC/ABPE-10	173.72	4.06	5.81	82.53	13.26
composite film					
BC/PU composite	(0.50	6.00	1.90	82.00	
film	69.50				-
BC/epoxy resin	325.00	20.00	0.02	84.00	6.00
composite film					
Polyethylene		2 00 2 70	7.00	82.00	20.00.100.00
terephthalate	-	2.00-2.70	7.90	83.00	20.00-100.00

379 Through the experiment results analysis above, we concluded that when ANFC dosage was 380 about 68% in composite film, the performance of FOLED substrate could be maximally improved, 381 the results were: transmittance 82.53 %, contact angle 102.9°, CTE 13.26 ppm/K, tensile strength 382 173.72 MPa, Young's modulus 4.06 GPa, elongation at break 5.81%, and good flexibility. From Table 383 3, the CTE of polyethylene terephthalate was very larger than other polymers and was not suitable 384 for substrate requirements. And the transmittance of all polymer FOLED substrates at the 385 wavelength of 600 nm was nearly the same. Meanwhile, the uniform deformation or stable 386 deformation of BC/epoxy resin composite film was poor thanks to its low elongation at break. 387 Thence, the mechanical properties of the ANFC/ABPE-10 composite films were exceedingly 388 excellent, compared to the BC/PU and BC/epoxy resin composite film in Table 3. In spite of the fact 389 that the CTE of ANFC/ABPE-10 composite films was higher than BC/epoxy resin composite film, it 390 was less than 20 ppm • K⁻¹ meeting the requirements of the substrate. In summary, considering that 391 the cost efficiency, recyclability, and environmentally friendly, the dosage of ANFC should be 392 increased as much as possible without affecting the properties of FOLED substrate. The 393 ANFC/ABPE-10 composite films prepared in this study combined the following advantages: 394 cheapness, superior thermal stability, great flexibility, good biodegradability, high transparency, 395 good smoothness, as well excellent mechanical performance, and could be used as high 396 performance FOLED substrate.

397 4. Conclusions

The ANFC/ABPE-10 composite film was prepared by ABPE-10 impregnating into ANFC films under negative pressure. And the performance of the thin composite film (45 um) could maximally be improved and had meted the FOLED substrate requirement when ANFC dosage was high for approximately 70%. The properties of ANFC films were enhanced mainly because of the nature of ABPE-10 itself and the IPN structure formed between ABPE-10 and ANFC film. In fact, the composite films displayed high transparency (up to 80%), low CTE (13.26 ppm/K), and good water resistance. Additionally, the composite films had outstanding mechanical properties such as tensile

- strength 173.72 MPa, Young's modulus 4.06 GPa, elongation at break 5.81%. Therefore, we suggest
 that ANFC/ABPE-10 composite film is potential candidate for FOLED substrate.
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