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Influence of Concentration Fluctuations on Relaxation Processes in Spin Glasses

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- Abstract: Using the unique combination of atomically resolved atom probe tomography (APT)
- ² and volume averaged neutron (resonance) spin echo (NRSE and NSE) experiments, the influence
- ³ of nano-scaled clusters on the spin relaxation in spin glasses was studied. For this purpose, the
- ⁴ phase transition from the paramagnetic phase to the spin glass phase in a Fe-Cr spin glass with
- a composition of $Fe_{17.8}Cr_{82.2}$ was studied in detail by means of NRSE. The microstructure was
- 6 characterised by APT measurements, which show local concentration fluctuations of Fe and Cr on a
- ⁷ length scale of 2 to 5 nm, which lead i) to the coexistence of ferro- and anti-ferromagnetic clusters
- and ii) a change of the magnetic properties of the whole sample, even in the spin glass phase, where
- spins are supposed to be randomly frozen. We show that a generalized spin glass relaxation function,
- which was successfully used to describe the phase transition in diluted spin glasses, can also be used
- ¹¹ for fitting the spin dynamics in spin glasses with significant concentration fluctuations.
- ¹² Keywords: spin glasses; disordered systems; magnetism; neutron spin echo

13 1. Introduction

Spin glasses show frustrated magnetic interactions due to stochastically oriented spins and, 14 thus, possess no magnetic long-range order. Typically, phase transitions in spin glasses occur at 15 temperatures below 60 K. In spin glasses the disorder in the system at high temperatures, which is 16 typically paramagnetic, reappears at low temperatures in a frozen state. Due to the frustration of the 17 magnetic states in the spin glass phase, slow decay processes on large time scales may be observable 18 [1]. Inelastic and quasi-elastic neutron scattering capture an important part of these slow relaxation 19 processes [2,3]. On these data, models are based to describe the relaxation process in spin glasses. 20 Pickup et al. suggested a generalized spin glass relaxation process [4] by connecting the probabilistic 21 Weron model [5] and a model, which describes highly disordered systems on the basis of Lévy-stable 22 distributions, firstly introduced by Tsallis [6]. Pickup used this model to describe successfully the 23 relaxation processes in the diluted spin glass systems $Au_{1-x}Fe_x$ and $Cu_{1-x}Mn_x$ [4]. In this way it was 24 shown that processes in spin glasses are similar to those in other complex disordered systems. In a 25 few spin glass systems, the so called cluster spin glasses, clusters of magnetic order coexist with the 26 magnetic amorphous spin glass phase [7]. However, only little is known about the relaxation processes 27 in those cluster spin glasses. 28

- ²⁹ In this contribution we study the relation between microstructure leading to magnetically ordered
- ³⁰ clusters and the averaged spin relaxation in a Fe-Cr sample, which shows a temperature driven phase
- transition to a cluster spin glass (SG) phase. For this purpose we combine two high resolution methods,
- namely the atomically resolved atom probe tomography (APT) [8–10] and the high resolution neutron

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(resonance) spin echo spectroscopy (N(R)SE). Additionally, magnetization measurements complete the
 characterization of the specimen.

APT measurements show that nanometer sized Cr rich regions coexist next to Fe rich regions.

³⁶ These may result in magnetically ordered clusters even in the spin glass phase. The shown results

- ³⁷ support the assumption, that the Weron model introduced by Pickup et al. [4] for diluted spin glasses
- ³⁸ like $Au_{1-x}Fe_x$ and $Cu_{1-x}Mn_x$ still works for spin glasses with inhomogeneities in the microstructure,
- ³⁹ if some of the parameters are restricted, however, the physical interpretation of the results may differ.
- The model will be explained in detail below. It is shown that even small scale inhomogenities in the microstructure may change the overall relaxation processes of spin glasses significantly. The relaxation
- ⁴² process is then governed by different contributions of hierarchically ordered cluster interactions and

⁴³ parallel spin interactions compared to diluted spin glasses.

44 2. Experimental details

For this study a Fe-Cr alloy with a nominal composition of $c_{\text{Fe}} = 14.5$ and $c_{\text{Cr}} = 85.5$ at.% was used. 45 The sample was prepared from starting materials with a purity of 99.99%, arc melted, annealed for 46 4 days at 1100°C and quenched in water. To relax the occurring strains, the sample was annealed at 47 1000°C for one day. The cylindrical sample has a height of approximately 20 mm and a diameter of 48 about 10 mm. Atom probe tomography revealed an averaged chemical composition of $c_{\rm re} = 17.8$ and 49 $c_{\rm cr} = 82.2$ at.%. The concentration of impurities from the production process is approximately 0.01 at.% 50 and thus neglectable. The phase diagram of the Fe-Cr-system is given by Burke et al. [11]. Based on 51 the found chemical composition $c_{\rm Fe}$ and $c_{\rm cr}$ the material is supposed to show no magnetic order and to 52 transfer directly from the paramagnetic regime to the spin glass phase. 53 The ac susceptibility measurements were performed using a Quantum Design physical property

measurement system (PPMS). An excitation amplitude of 1 mT with an excitation frequency between
10 Hz and 10 kHz was used.

⁵⁷ The relaxation measurements were carried out on the neutron (resonance) spin echo spectrometers

⁵⁸ RESEDA [12] and JNSE [13] at the neutron source FRM II. The wavelength chosen was 5.5 Å for ⁵⁹ RESEDA and 5 Å for JNSE. At RESEDA, a range of spin echo times τ from approx. 0.05 ns to 1 ns was ⁶⁰ accessible. With JNSE the range could be extended to 10 ns. The τ -dependence of the polarization ⁶¹ *P* of the neutrons was independent of the momentum transfer *q*. Thus, to improve measurement ⁶² statistics, the experimental results are averaged for *q* values between 0.04 Å⁻¹ $\leq q \leq$ 0.08 Å⁻¹. The ⁶³ intermediate scattering function *S*(*q*, τ) was determined for temperatures *T* = 12.1, 14.5, 19.4, 25.1,

⁶⁴ 30.1 and 34.2 K. After correcting the data for depolarization by the sample and the spectrometer at

⁶⁵ the different temperatures, the data sets are normalized to the intermediate scattering function as

measured at T = 3.6 K, where the dynamics appears to be frozen on the time scales achieved by the

67 spectrometers.

⁶⁸ The atom probe tomography measurements were carried out with a Cameca LEAP 4000X HR system

⁶⁹ from Cameca located at the Karlsruhe Nano Micro Facility (KNMF). The atom probe tips were prepared

⁷⁰ by means of a focused ion beam with a Strata 400 system. The atom probe tips were measured in the

voltage mode with a pulse rate of 200 kHz and a pulse fraction of 20% at a temperature of about 55.8 K.

72 Analysis and reconstruction of the data was done with the IVAS 6.8.10 software package from Cameca.

73 3. Generalized spin glass relaxation model based on Weron

In the Weron model [5] the fractal character of the spin glass is described by the exponent β , whereas *k* is a measure for the contribution of hierarchical parallel and thus independent relaxation processes. Combined with the relaxation time τ , the generalized relaxation function based on the Weron model is given by

$$\varphi(t) = [1 + k(t/\tau)^{\beta}]^{-1/k}.$$
(1)

- In the Tsallis model [6] the sum of the entropy of two independent processes is larger or smaller 74 than the direct sum of their entropies. The difference is scaled with the non-extensivity parameter 75 $q_{\rm T}$. Brouers and Sotolongo-Costa [14] showed, that there is a direct relation between the Tsallis and 76 the Weron model when k is replaced by $k = (q_T - 1)/(2 - q_T)$. The application to spin glasses will be 77 shortly described in the following and is explained in greater detail by Pickup et al [4]. Generally, the 78 dynamics in a system slows down with decreasing temperature. This statement also applies for spin 79 glasses. Thus, the lifetime of relaxation processes, τ , must increase with decreasing temperature. For 80 the temperature limit $T \to 0$, $\tau \to \infty$ must apply. Above the spin glass transition temperature, T_{sG} , spin-spin interactions are dominant. In the 82 Weron model, this fact is expressed by a relatively large value of β ($\beta_{max} = 1$). Because the inter-cluster 83 interactions are suppressed in the regime above T_{sG} , the value of k will be small. For high temperatures,
- where relaxation processes are dominated by thermal activation, $\beta = 1$ and $k \to 0$, i.e. the Weron 85
- model describes a pure exponential decay. For decreasing temperatures the spins are slowed down 86
- (until they are frozen) and the contribution of parallel, independent relaxation processes increases: The 87
- value of k increases, while β decreases. If all relaxation processes occur in single spin clusters, which 88
- do not interact with each other, *k* converges to infinity. In this limit $q_T = 2$ applies. 89

4. Results 90

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The ac susceptibility measurements shown in figure 1 demonstrate that the sample exhibits a 91 temperature driven spin glass (SG) phase transition near $T_{\rm SG} \approx 25$ K. This value is consistent with 92 results shown by Burke et al. for an iron concentration of $c_{\rm Fe} \sim 17.8$ at.% [11]. As expected from the 93 chemical composition there is no clear phase transition from the paramagnetic (PM) to a magnetically 94

ordered phase for temperatures between 50 and 200 K in the ac susceptibility measurements.

- Figure 2 a) shows the reconstructed APT with the elements Fe and Cr. The black box represents the 96
- area $(120 \times 30 \text{ mm}^2, \text{length} \times \text{height})$ in which the 2D concentration maps of Fe and Cr are shown after 97
- integration of the concentration along the depth (2.5 mm) of the sample. The resulting 2D concentration 98
- maps of Cr, Fe and Cr+Fe, as analyzed by the software IVAS are shown in figures 2 b-d). Impurities in 99
- the material, such as N and C from the production process and Ga from the FIB cut of the atom probe 100 tip are locally far less than 0.05 at.%. 101
- Figure 3 shows the intermediate scattering function versus the relaxation time as measured at RESEDA 102 and JNSE together with fits to the Weron model given by equation 1. It is clearly seen that the spin 103 fluctuations freeze with decreasing temperature T. For the fitting procedure the following two obvious 104 assumptions were made: i) $0 \le \beta \le 1$. Of course, $\beta = 1$ above T_{sG} in the PM phase. Therefore, β can 105 only decrease with decreasing T. ii) τ has to monotonically increase with decreasing T due to freezing If the parameter k was not restricted, it increased with decreasing T, as one would expect because more 107 and more independent relaxation processes can contribute to the dynamics. However, τ decreased 108 which contradicts the physical intuition (dashed lines in Figure 4). Meaningful fits, i.e. a decreasing τ 109 was only obtained if k was constrained to a constant value for all $T \leq T_{sc}$. The exact value was hard to 110 determine and was found to be $k \le 4$. All values $k \le 4$ led to a physically meaningful decrease of τ with decreasing temperature. 112

Of course, the exact values of the fit parameters depend strongly on the chosen value of k, thus 113 exact values of the fit parameters cannot be extracted. The following discussion is based on fit results 114 obtained with a constant k = 2.5. Figure 4 shows the resulting temperature dependence for k, β 115 and τ with the restriction of k = 2.5 (solid lines and closed symbols) compared to fit results with no 116 117 boundary conditions except $0 \le \beta \le 1$ (dashed lines with open symbols). Table 1 shows all values of the intermediate scattering function in dependence of the spin echo times for all temperatures, so that 118

the fit values might be reproduced by interested readers. 119

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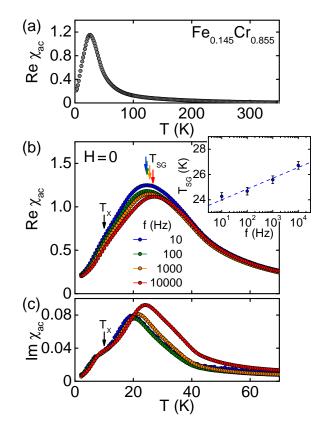


Figure 1. Temperature dependence of the ac susceptibility of the studied sample. (a) Real part of the ac susceptibility between 2 K and 350 K measured at 911 Hz. The absolute value is large, when compared with a typical antiferromagnetic compound. (b) Real part of the ac susceptibility, Re χ_{ac} , as measured at low temperatures for excitation frequencies between 10 Hz and 10 kHz. As depicted in the inset and characteristic for a spin glass, the position of the maximum at T_{SG} shifts clearly as a function of the excitation frequency. (c) Imaginary part of the ac susceptibility, Im χ_{ac} , at low temperatures. A change of slope at $T_x \sim 10$ K in Re χ_{ac} and, in particular, in Im χ_{ac} suggests the presence of an additional transition, which is not captured in the neutron data and will not be considered further.

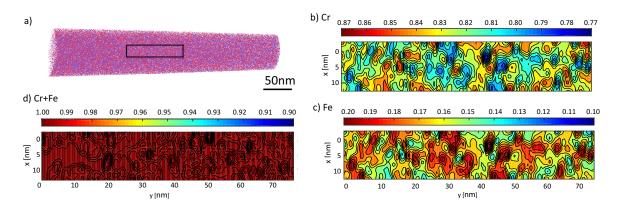


Figure 2. a) Atom probe reconstruction of the Fe-Cr alloy, measured at 50 K. The black box marks the area were the data for the 2D concentration maps b) and c) are taken from. Clearly visible are small clusters of Cr and Fe, respectively. The concentration distribution is integrated over a sample slice with a thickness of 2.5 nm. d) shows the summation of b) and c), which should sum up to 100%. The small differences of less than 1% are caused by impurities such as P, N, and Ga.

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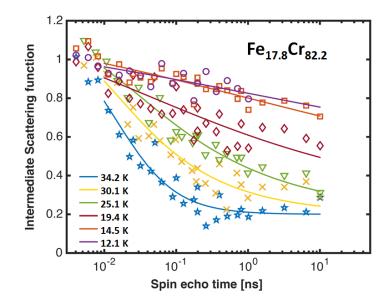


Figure 3. Intermediate scattering function as measured by neutron (resonance) spin echo at RESEDA and at JNSE for $q \simeq 0.06$ Å⁻¹. The solid lines show the restricted Weron model fitted to the experimental data (symbols) assuming k = 2.5 for $T \leq T_{sc}$. The parameters of the fit are shown in figure 4.

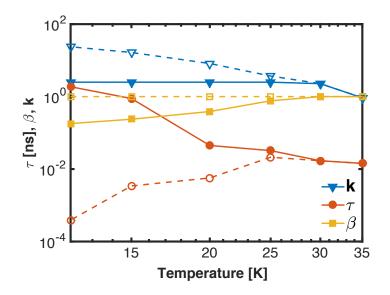


Figure 4. Parameters τ , β and k analysed by a fit of the Weron function to the intermediate scattering functions observed for different temperatures by N(R)SE measurements with no restrictions except $0 \le \beta \le 1$ (dashed lines and open symbols) and with restrictions i) $0 \le \beta \le 1$, ii) τ constantly increasing with decreasing T and iii) k = 2.5 for $T \le 25$ K (solid lines and closed symbols).

120 5. Discussion

The spin glass transition temperatures $T_{\rm sc} \sim 25 \, K$ as measured by means of ac susceptibility and 121 neutron scattering are consistent with each other. For N(R)SE, the transition might be identified due 122 to the observation that the time dependence becomes more exponential with increasing temperature. 123 $T_{SG} \sim 25 K$ is also consistent with the value given by Burke et al. [11] for an iron concentration of 124 $c_{\rm Fe} = 17.8$ at.%. The susceptibility measurements show that there is no other phase transition for 125 $T > T_{sG}$, indicating that there is no magnetically ordered phase in the material (figure 1). A change of 126 slope at $T_x \sim 10$ K in Re χ_{ac} and in particular in Im χ_{ac} suggests the presence of an additional transition, 127 which shows no signature in the neutron scattering data and will not be considered further. 128

The atom probe tomography studies clearly show, that Fe rich regions coexist with Cr rich regions in the material (figure 2). In those zones, the concentration of Fe might be high or low enough, that ferromagnetic (FM) or antiferromagnetic (AFM) ordering will be observed. According to Burke et al. [11] an Fe concentration of about 19 at% for FM ordering and of 16 at% for AFM ordering is necessary for temperatures lower than ~ 50 K. Figure 2 shows that those values are reached locally over a length scale of approximately 2–5 nm.

According to Burke et al. [11] and Fincher and Shapiro et al. [15,16], the Néel temperature for 135 concentrations of $c_{\rm Fe}$ of 14 at.% is supposed to be $T_N \sim 60$ K. The Curie temperature for $c_{\rm Fe} = 20$ at.% is 136 $T_{\rm C} \sim 80$ K. Thus, with decreasing T a change in polarization and intensity of the neutron signal around 137 $q = 0.06 \text{ Å}^{-1}$ is expected as follows: Approaching T_C from high temperatures will initiate the formation 138 of ferromagnetic clusters, which reduces the polarization of the scattered neutrons. Decreasing Tfurther towards T_N leads to the formation of antiferromagnetic ordering in regions surrounding the 140 ferromagnetically ordered regions. Thus the size of the ferromagnetic clusters decreases, resulting in 141 an increase in polarization. Lowering T further favors the spin glass phase thus shrinking the regions 142 of both, ferromagnetic and antiferromagnetic phase further thus leading to a continuing increase of 143 intensity and polarization with decreasing *T*.

Indeed these trends can be qualitatively observed in the neutron scattering data. Figure 5 shows the neutron polarization and the intensity of the neutrons at q = 0.06 Å⁻¹ with decreasing *T*. The polarization decreases with decreasing *T*, reaches a minimum around 120 K and increases again. The deviation from the expected minimum around 80 K may be due to the volume averaging and *q*-weighting by the neutrons. Overall, an increase in neutron intensity is observed as expected.

To put our results on a more quantitative basis we estimate the size of the ferromagnetic clusters 150 using the expression for the polarization of the neutrons $P = P_0 \exp(-\frac{1}{3}\gamma^2 \langle B^2 \rangle \frac{\delta}{r^2} d_{\text{eff}})$ derived by 151 Halpern and Holstein [17] in the limit of small fields and domains. P is a function of the gyromagnetic 152 ratio $\gamma = 2.916 \text{ kHz/Gauss}$, the magnetic field *B*, the domain size of the ferromagnetic clusters δ , the 153 effective sample thickness for iron d_{eff} , and the velocity of the neutrons v. The velocity of neutrons with a wavelength of $\lambda = 5.5$ Å is $v \approx 720$ m/s. For the magnetization of the Fe clusters we assumed 155 the bulk magnetization of Fe $\mu_{\rm M} = 1.74$ T [?]. With $d_{\rm eff} = d_{\rm bulk}c_{\rm Fe} = 10$ mm $\cdot 0.145 \sim 1.38$ mm an 156 approximate cluster size of ~ 200 A for temperatures between 250 and 120 K is obtained. For the final 157 polarization measured in our experiment (about 28%) at 8K this assumption corresponds to a size 158 of the ferromagnetic clusters of $\delta \sim 140$ Å. This is still considerably larger than the regions of high 159 Fe concentrations in the APT analysis, which scale to 2-5 nm. However, the conclusion, what can 160 be drawn from APT and N(R)SE is similar: In regions of high Fe concentration the ferromagnetic 161 order might still be present, even if those regions are surrounded by pure spin glass phase. The same 162 holds true for the regions of high Cr concentration, where the antiferromagnetic ordering might still be 163 present. Both phenomena lead to islands of magnetic ordering in the spin glass phase. 164

Based on the above discussion the results of the N(R)SE experiments can be interpreted as follows: For $T > T_{sG}$ the sample is magnetically ordered in locally restricted areas as indicated by the ATP-data, leading to a maximum value $\beta = 1$ and a minimal value of k, as seen in figure 4. Regions of magnetic fluctuations are present but their parallel hierarchical relaxation processes have only a small weight when compared to the spin-fluctuations within the ordered clusters. With decreasing T, the volume

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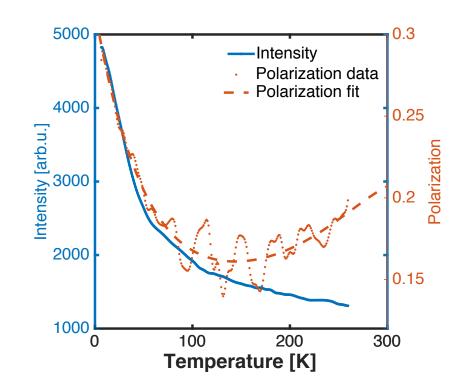


Figure 5. Temperature dependence of the intensity (blue) and the polarization (orange) as measured at $q = 0.06 \text{ Å}^{-1}$ at RESEDA. The data are recorded during the course of cooling down the sample. Data recorded during heating up the sample give similar results (not shown), thus hysteresis effects do not play a role.

fraction of the sample, where the spin fluctuations slow down increases and thus the contribution of the parallel relaxation processes increases due to the decreasing size of magnetically ordered clusters. The value of *k* increases, while β decreases. In diluted spin glasses *k* is supposed to diverge to infinity $(q_T \rightarrow 2)$, because relaxation processes on all time scales occur. In our sample, however, the value of *k* converges to a constant value (figure 4). The value of *k* is constrained because of the limited number of parallel processes due to the remaining clusters, which are directly observed by ATP and indicated by the finite polarization of the neutrons at low *T*.

Figure 6 shows a comparison of the results obtained from the diluted spin glass systems $Au_{1-x}Fe_x$ 177 and $Cu_{1-x}Mn_x$ as studied by Pickup [4] and the results of our work. The major difference between 178 the two data sets below T_{sc} reflects the different mechanisms taking place during the spin freezing. 179 Whereas in diluted spin glasses all spins are frozen in random orientations, in the Fe-Cr system 180 ferromagnetic and antiferromagnetic clusters coexist within a sea of disordered spin glass phase. 181 Therefore, the hierarchical order of relaxation processes changes from only parallel relaxation processes 182 to spin-spin and parallel relaxation processes. Since our study already showed the complementation 183 of scattering methods and atom probe tomography, the further investigations must concentrate on 184 the study of the short range order of Fe-Cr alloys in dependence of the found clusters. Here, special 185 emphasis will be laid on the influence of the heat treatment of the material on both the phase transition 186 and the short range order in the spin glass phase, because Fe-Cr alloys are known for their clustering 187 and spinodal decomposition [18]. Also the phase transition at $T \sim 10$ K needs further attention to be 188 fully identified. 189

190 6. Conclusions

By combining atomically resolved atomic probe tomography (APT) with quasielastic neutron scattering using N(R)SE we have characterized the evolution of the spin glass phase in the presence

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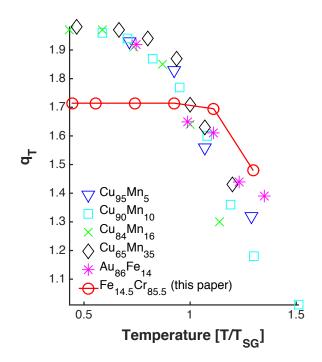


Figure 6. Comparison between data published by Pickup [4] on diluted spin glasses $Au_{1-x}Fe_x$ and $Cu_{1-x}Mn_x$ and the results observed in the present study on the cluster spin glass $Fe_{17.8}Cr_{82.2}$.

of ferro- and antiferromagnetic clusters in the cluster spin glass Fe_{17.8}Cr_{82.2}. The measured data can

¹⁹⁴ be interpreted in terms of the Weron model that assumes the decay of spin fluctuations in parallel

¹⁹⁵ processes in the presence of magnetically ordered clusters. Our results may also be relevant to other

¹⁹⁶ cluster-type spin glasses.

Acknowledgments: We thank the FRM II for providing beam time at the neutron resonance spin echo spectrometer
 RESEDA and the JCNS for providing beam time at the spectrometer JNSE. The work at Brookhaven was supported
 by the Office of Basic Energy Sciences, Division of Materials Science and Engineering, U.S. Department of Energy
 (DOE) under Contract No. DEAC02-98CH10886.

Author Contributions: J. N. W. leading scientist for experiments (neutrons and APT), analysis and interpretation
 of the data; W. H. and O. H. instrument responsible of RESEDA and NRSE, respectively, supported before, during
 and after the neutron experiments, discussed analysis and interpretation of the data; A. B. conducted the ac
 susceptibility measurements and interpreted them; S. M. S. provided the sample material; P. B. supported all steps
 of the experiment, the interpretation of the data and improved the paper. All authors reviewed the manuscript.

Conflicts of Interest: The authors declare no conflicts of interest. The founding sponsors had no role in the design
 of the study; in the collection, analyses, or interpretation of data; in the writing of the manuscript, and in the
 decision to publish the results.

209 Abbreviations

²¹⁰ The following abbreviations are used in this manuscript:

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	APT	atom probe tomography
	FWHM	full width at half maximum
	ISF	intermediate scattering function
	KNMF	Karlsruhe Nano Micro Facility
	N(R)SE	neutron (resonance) spin echo
12	NSE	neutron spin echo
	PM	paramagnetic
	PPMS	physical property measurement system
	RF	radio frequency
	SET	spin echo time
	SG	spin glass

213 References

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Table 1. Spin echo time (SET) and values of the intermediate scattering function (ISF) for all measured	
temperatures (12.1 $\leq T \leq$ 34.2 K).	

SET [ns] ISF 12.1K error 12.1K		ISF 14.5K	error 14.5K	SET [ns]	ISF 19.4K	error 19.4K
0.00409 1.02276 0.06006	0.00409	1.05768	0.05938	0.00409	0.98686	0.05903
0.00602 1.00878 0.05935	0.00602	1.09378	0.06067	0.00602	1.06622	0.06179
0.00813 0.96591 0.05791	0.00813	1.01485	0.05781	0.00813	0.95992	0.05795
0.01145 0.92834 0.0538	0.01145	0.92272	0.05197	0.01145	0.82503	0.05063
0.01617 0.91736 0.05504	0.01617	0.97386	0.05513	0.01617	0.88892	0.05423
0.02253 0.93979 0.05484	0.02253	0.88035	0.05109	0.02253	0.79933	0.05037
0.03181 0.91483 0.05615	0.03181	0.90611	0.05406	0.03181	0.77239	0.05158
0.04465 0.85836 0.05449	0.04465	0.88378	0.0536	0.04465	0.74521	0.05102
0.06279 0.97755 0.06212	0.06279	0.95468	0.05939	0.06279	0.71707	0.05341
0.08888 0.87447 0.05982	0.08888	0.87917	0.0582 0.06025	0.08888	0.72044	0.05503
0.12548 0.92691 0.06301 0.17724 0.78653 0.05843	0.12548 0.17724	0.90804 0.76687	0.06025	0.12548 0.17724	0.7405 0.5812	0.05691 0.05268
	0.17724 0.19732			0.17724 0.19732	0.5812	0.05268
0.19732 0.87647 0.06422	0.19732	0.90613	0.0629	0.19732		0.05615
0.26199 0.93792 0.06647 0.36492 0.8335 0.0552	0.26199	0.84559 0.84718	0.06081 0.05369	0.26199	0.66925 0.62548	0.03729 0.04892
	0.51595			0.51595	0.62348 0.52864	0.04892 0.048
		0.8401	0.05497			0.048
0.71752 0.85023 0.07396 1.01391 0.79711 0.07074	0.71752	0.79749 0.73849	0.06927	0.71752 1.01391	0.55169 0.54046	0.06368 0.06272
1.01391 0.79711 0.07074	0.02529		0.06605 0.0217	0.02543	0.54046 0.91805	0.06272 0.02341
	0.02529	0.92578 0.90893	0.0217 0.02173	0.02543	0.91805 0.90867	0.02341 0.02357
	0.04978	0.90893	0.02173 0.02124	0.04989	0.90867 0.8605	0.02357 0.02246
	0.0998	0.92581 0.8752	0.02124 0.01941	0.1003	0.8605	0.02246 0.0202
	0.19972	0.8752	0.01941 0.01845	0.19945	0.74776 0.71885	0.0202 0.0191
	0.39854	0.86228	0.01845 0.01753	0.39976	0.71885 0.66614	0.0191 0.01803
	1.59816					
	3.19855	0.7778 0.7861	0.01495 0.01559	1.59894 3.19747	0.64925 0.64314	0.01571 0.0161
	6.39774	0.76103	0.01559	6.39856	0.59164	0.0161
	9.9985	0.70518	0.02048	9.99848	0.55466	0.01050
SET [ns] ISF 25.1K error 25.1K	SET [ns]	ISF 30.1K	20 1V	CET [ma]	ISF 34.2K	24 2 V
			error 30.1K	SET [ns]		error 34.2K
0.00526 1.09754 0.06667	0.00525	0.96771	0.0617	0.00409	1.01164	0.06198
	0.00765	1 000 20		0.00602	0 00/17	0.05747
0.00754 1.03817 0.06788	0.00765	1.00928	0.06715	0.00602	0.88442	0.05747
0.01079 0.99185 0.06757	0.01082	0.92648	0.06523	0.00813	0.89517	0.05781
0.01079 0.99185 0.06757 0.01512 0.88966 0.06278	0.01082 0.01508	0.92648 0.88101	0.06523 0.06266	0.00813 0.01145	0.89517 0.73806	0.05781 0.04983
0.010790.991850.067570.015120.889660.062780.021250.802960.05852	0.01082 0.01508 0.02125	0.92648 0.88101 0.66394	0.06523 0.06266 0.05382	0.00813 0.01145 0.01617	0.89517 0.73806 0.61188	0.05781 0.04983 0.04774
0.010790.991850.067570.015120.889660.062780.021250.802960.058520.030020.855920.06467	0.01082 0.01508 0.02125 0.02987	0.92648 0.88101 0.66394 0.59078	0.06523 0.06266 0.05382 0.05516	0.00813 0.01145 0.01617 0.02253	0.89517 0.73806 0.61188 0.49692	0.05781 0.04983 0.04774 0.04387
0.010790.991850.067570.015120.889660.062780.021250.802960.058520.030020.855920.064670.042160.674080.05884	0.01082 0.01508 0.02125 0.02987 0.04217	0.92648 0.88101 0.66394 0.59078 0.60437	0.06523 0.06266 0.05382 0.05516 0.0567	0.00813 0.01145 0.01617 0.02253 0.03181	0.89517 0.73806 0.61188 0.49692 0.44948	0.05781 0.04983 0.04774 0.04387 0.04482
0.010790.991850.067570.015120.889660.062780.021250.802960.058520.030020.855920.064670.042160.674080.058840.059650.80160.06767	0.01082 0.01508 0.02125 0.02987 0.04217 0.05955	0.92648 0.88101 0.66394 0.59078 0.60437 0.57911	0.06523 0.06266 0.05382 0.05516 0.0567 0.05997	0.00813 0.01145 0.01617 0.02253 0.03181 0.04465	0.89517 0.73806 0.61188 0.49692 0.44948 0.42374	0.05781 0.04983 0.04774 0.04387 0.04482 0.04469
0.010790.991850.067570.015120.889660.062780.021250.802960.058520.030020.855920.064670.042160.674080.058840.059650.80160.067670.084040.583130.06082	0.01082 0.01508 0.02125 0.02987 0.04217 0.05955 0.08394	0.92648 0.88101 0.66394 0.59078 0.60437 0.57911 0.50844	0.06523 0.06266 0.05382 0.05516 0.0567 0.05997 0.05899	0.00813 0.01145 0.01617 0.02253 0.03181 0.04465 0.06279	0.89517 0.73806 0.61188 0.49692 0.44948 0.42374 0.36484	0.05781 0.04983 0.04774 0.04387 0.04482 0.04469 0.04685
0.010790.991850.067570.015120.889660.062780.021250.802960.058520.030020.855920.064670.042160.674080.058840.059650.80160.067670.084040.583130.060820.118730.600050.06782	0.01082 0.01508 0.02125 0.02987 0.04217 0.05955 0.08394 0.11872	0.92648 0.88101 0.66394 0.59078 0.60437 0.57911 0.50844 0.4726	0.06523 0.06266 0.05382 0.05516 0.0567 0.05997 0.05899 0.06416	0.00813 0.01145 0.01617 0.02253 0.03181 0.04465 0.06279 0.08888	0.89517 0.73806 0.61188 0.49692 0.44948 0.42374 0.36484 0.28966	0.05781 0.04983 0.04774 0.04387 0.04482 0.04469 0.04685 0.04716
0.010790.991850.067570.015120.889660.062780.021250.802960.058520.030020.855920.064670.042160.674080.058840.059650.80160.067670.084040.583130.060820.118730.600050.067820.168030.571630.08267	0.01082 0.01508 0.02125 0.02987 0.04217 0.05955 0.08394 0.11872 0.16771	0.92648 0.88101 0.66394 0.59078 0.60437 0.57911 0.50844 0.4726 0.42479	0.06523 0.06266 0.05382 0.05516 0.0567 0.05997 0.05899 0.06416 0.0778	0.00813 0.01145 0.01617 0.02253 0.03181 0.04465 0.06279 0.08888 0.12548	0.89517 0.73806 0.61188 0.49692 0.44948 0.42374 0.36484 0.28966 0.25518	0.05781 0.04983 0.04774 0.04387 0.04482 0.04469 0.04685 0.04716 0.04778
0.010790.991850.067570.015120.889660.062780.021250.802960.058520.030020.855920.064670.042160.674080.058840.059650.80160.067670.084040.583130.060820.118730.600050.067820.168030.571630.082670.191880.607910.09113	0.01082 0.01508 0.02125 0.02987 0.04217 0.05955 0.08394 0.11872 0.16771 0.19179	0.92648 0.88101 0.66394 0.59078 0.60437 0.57911 0.50844 0.4726 0.42479 0.37019	0.06523 0.06266 0.05382 0.05516 0.0567 0.05997 0.05899 0.06416 0.0778 0.0829	0.00813 0.01145 0.01617 0.02253 0.03181 0.04465 0.06279 0.08888 0.12548 0.17724	$\begin{array}{c} 0.89517\\ 0.73806\\ 0.61188\\ 0.49692\\ 0.44948\\ 0.42374\\ 0.36484\\ 0.28966\\ 0.25518\\ 0.27402 \end{array}$	0.05781 0.04983 0.04774 0.04387 0.04482 0.04469 0.04685 0.04716 0.04778 0.04849
0.010790.991850.067570.015120.889660.062780.021250.802960.058520.030020.855920.064670.042160.674080.058840.059650.80160.067670.084040.583130.060820.118730.600050.067820.168030.571630.082670.191880.607910.091130.254980.506250.07413	0.01082 0.01508 0.02125 0.02987 0.04217 0.05955 0.08394 0.11872 0.16771 0.19179 0.25486	0.92648 0.88101 0.66394 0.59078 0.60437 0.57911 0.50844 0.4726 0.42479 0.37019 0.36084	0.06523 0.06266 0.05382 0.05516 0.0567 0.05997 0.05899 0.06416 0.0778 0.0829 0.07005	0.00813 0.01145 0.01617 0.02253 0.03181 0.04465 0.06279 0.08888 0.12548 0.17724 0.19732	$\begin{array}{c} 0.89517\\ 0.73806\\ 0.61188\\ 0.49692\\ 0.44948\\ 0.42374\\ 0.36484\\ 0.28966\\ 0.25518\\ 0.27402\\ 0.21112\\ \end{array}$	0.05781 0.04983 0.04774 0.04387 0.04482 0.04469 0.04685 0.04716 0.04778 0.04849 0.05007
$\begin{array}{llllllllllllllllllllllllllllllllllll$	0.01082 0.01508 0.02125 0.02987 0.04217 0.05955 0.08394 0.11872 0.16771 0.19179 0.25486 0.35483	0.92648 0.88101 0.66394 0.59078 0.60437 0.57911 0.50844 0.4726 0.42479 0.37019 0.36084 0.30882	0.06523 0.06266 0.05382 0.05516 0.0567 0.05997 0.05899 0.06416 0.0778 0.0829 0.07005 0.059	0.00813 0.01145 0.01617 0.02253 0.03181 0.04465 0.06279 0.08888 0.12548 0.17724 0.19732 0.26199	$\begin{array}{c} 0.89517\\ 0.73806\\ 0.61188\\ 0.49692\\ 0.44948\\ 0.42374\\ 0.36484\\ 0.28966\\ 0.25518\\ 0.27402\\ 0.21112\\ 0.14018 \end{array}$	0.05781 0.04983 0.04774 0.04387 0.04482 0.04469 0.04685 0.04716 0.04778 0.04849 0.05007 0.04911
$\begin{array}{llllllllllllllllllllllllllllllllllll$	0.01082 0.01508 0.02125 0.02987 0.04217 0.05955 0.08394 0.11872 0.16771 0.19179 0.25486 0.35483 0.50191	0.92648 0.88101 0.66394 0.59078 0.60437 0.57911 0.50844 0.4726 0.42479 0.37019 0.36084 0.30882 0.22345	0.06523 0.06266 0.05382 0.05516 0.0567 0.05997 0.05899 0.06416 0.0778 0.0829 0.07005 0.059 0.06644	0.00813 0.01145 0.01617 0.02253 0.03181 0.04465 0.06279 0.08888 0.12548 0.17724 0.19732 0.26199 0.36492	$\begin{array}{c} 0.89517\\ 0.73806\\ 0.61188\\ 0.49692\\ 0.44948\\ 0.42374\\ 0.36484\\ 0.28966\\ 0.25518\\ 0.27402\\ 0.21112\\ 0.14018\\ 0.1709 \end{array}$	0.05781 0.04983 0.04774 0.04387 0.04482 0.04469 0.04685 0.04716 0.04778 0.04849 0.05007 0.04911 0.04189
$\begin{array}{llllllllllllllllllllllllllllllllllll$	0.01082 0.01508 0.02125 0.02987 0.04217 0.05955 0.08394 0.11872 0.16771 0.19179 0.25486 0.35483 0.50191 0.70109	0.92648 0.88101 0.66394 0.59078 0.60437 0.57911 0.50844 0.4726 0.42479 0.37019 0.36084 0.30882 0.22345 0.41314	0.06523 0.06266 0.05382 0.05516 0.0567 0.05997 0.05899 0.06416 0.0778 0.0829 0.07005 0.059 0.06644 0.09901	0.00813 0.01145 0.01617 0.02253 0.03181 0.04465 0.06279 0.08888 0.12548 0.17724 0.19732 0.26199 0.36492 0.51595	$\begin{array}{c} 0.89517\\ 0.73806\\ 0.61188\\ 0.49692\\ 0.44948\\ 0.42374\\ 0.36484\\ 0.28966\\ 0.25518\\ 0.27402\\ 0.21112\\ 0.21112\\ 0.14018\\ 0.1709\\ 0.19155 \end{array}$	0.05781 0.04983 0.04774 0.04387 0.04482 0.04469 0.04685 0.04716 0.04778 0.04849 0.05007 0.04911 0.04189 0.04417
$\begin{array}{llllllllllllllllllllllllllllllllllll$	0.01082 0.01508 0.02125 0.02987 0.04217 0.05955 0.08394 0.11872 0.16771 0.19179 0.25486 0.35483 0.50191 0.70109 0.99062	0.92648 0.88101 0.66394 0.59078 0.60437 0.57911 0.50844 0.4726 0.42479 0.37019 0.36084 0.30882 0.22345 0.41314 0.28469	0.06523 0.06266 0.05382 0.05516 0.0567 0.05997 0.05899 0.06416 0.0778 0.0829 0.07005 0.059 0.06644 0.09901 0.09488	0.00813 0.01145 0.01617 0.02253 0.03181 0.04465 0.06279 0.08888 0.12548 0.17724 0.19732 0.26199 0.36492 0.51595 0.71752	$\begin{array}{c} 0.89517\\ 0.73806\\ 0.61188\\ 0.49692\\ 0.44948\\ 0.42374\\ 0.36484\\ 0.28966\\ 0.25518\\ 0.27402\\ 0.21112\\ 0.21112\\ 0.14018\\ 0.1709\\ 0.19155\\ 0.19639 \end{array}$	0.05781 0.04983 0.04774 0.04387 0.04482 0.04469 0.04685 0.04716 0.04778 0.04849 0.05007 0.04911 0.04189 0.04417 0.05956
$\begin{array}{llllllllllllllllllllllllllllllllllll$	0.01082 0.01508 0.02125 0.02987 0.04217 0.05955 0.08394 0.11872 0.16771 0.19179 0.25486 0.35483 0.50191 0.70109 0.99062 0.02534	0.92648 0.88101 0.66394 0.59078 0.60437 0.57911 0.50844 0.4726 0.42479 0.37019 0.36084 0.30882 0.22345 0.41314 0.28469 0.75225	0.06523 0.06266 0.05382 0.05516 0.0597 0.05997 0.05899 0.06416 0.0778 0.0829 0.07005 0.07905 0.059 0.06644 0.09901 0.09488 0.02896	0.00813 0.01145 0.01617 0.02253 0.03181 0.04465 0.06279 0.08888 0.12548 0.17724 0.19732 0.26199 0.36492 0.51595 0.71752 1.01391	$\begin{array}{c} 0.89517\\ 0.73806\\ 0.61188\\ 0.49692\\ 0.44948\\ 0.42374\\ 0.36484\\ 0.28966\\ 0.25518\\ 0.27402\\ 0.21112\\ 0.14018\\ 0.1709\\ 0.19155\\ 0.19639\\ 0.18747 \end{array}$	0.05781 0.04983 0.04774 0.04387 0.04482 0.04469 0.04685 0.04716 0.04778 0.04778 0.04849 0.05007 0.04911 0.04189 0.04417 0.05956 0.05861
0.010790.991850.067570.015120.889660.062780.021250.802960.058520.030020.855920.064670.042160.674080.058840.059650.80160.067670.084040.583130.060820.118730.600050.067820.168030.571630.082670.191880.607910.091130.254980.506250.074130.3550.561780.065610.502150.515150.073450.701430.493820.100950.99110.412380.098010.099930.628050.023970.200280.610870.02242	0.01082 0.01508 0.02125 0.02987 0.04217 0.05955 0.08394 0.11872 0.16771 0.19179 0.25486 0.35483 0.50191 0.70109 0.99062 0.02534 0.04913	0.92648 0.88101 0.66394 0.59078 0.60437 0.57911 0.50844 0.4726 0.42479 0.37019 0.36084 0.30882 0.22345 0.41314 0.28469 0.75225 0.66412	0.06523 0.06266 0.05382 0.05516 0.0567 0.05997 0.05899 0.06416 0.0778 0.0829 0.07005 0.07005 0.059 0.06644 0.09901 0.09488 0.02896 0.02883	0.00813 0.01145 0.01617 0.02253 0.03181 0.04465 0.06279 0.08888 0.12548 0.17724 0.19732 0.26199 0.36492 0.51595 0.71752 1.01391 0.02533	$\begin{array}{c} 0.89517\\ 0.73806\\ 0.61188\\ 0.49692\\ 0.44948\\ 0.42374\\ 0.36484\\ 0.28966\\ 0.25518\\ 0.27402\\ 0.21112\\ 0.14018\\ 0.1709\\ 0.19155\\ 0.19639\\ 0.18747\\ 0.58292 \end{array}$	0.05781 0.04983 0.04774 0.04387 0.04482 0.04469 0.04685 0.04716 0.04778 0.04849 0.05007 0.04911 0.04189 0.04417 0.05956 0.05861 0.03211
0.010790.991850.067570.015120.889660.062780.021250.802960.058520.030020.855920.064670.042160.674080.058840.059650.80160.067670.084040.583130.060820.118730.600050.067820.168030.571630.082670.191880.607910.091130.254980.506250.074130.3550.561780.065610.502150.515150.073450.701430.493820.100950.99110.412380.098010.099930.628050.023970.200280.610870.02087	0.01082 0.01508 0.02125 0.02987 0.04217 0.05955 0.08394 0.11872 0.16771 0.19179 0.25486 0.35483 0.50191 0.70109 0.99062 0.02534 0.04913 0.09842	0.92648 0.88101 0.66394 0.59078 0.60437 0.57911 0.50844 0.4726 0.42479 0.37019 0.36084 0.30882 0.22345 0.41314 0.28469 0.75225 0.66412 0.58733	0.06523 0.06266 0.05382 0.05516 0.0567 0.05997 0.05899 0.06416 0.0778 0.0829 0.07005 0.07005 0.059 0.06644 0.09901 0.09488 0.02896 0.02883 0.02762	0.00813 0.01145 0.01617 0.02253 0.03181 0.04465 0.06279 0.08888 0.12548 0.17724 0.19732 0.26199 0.36492 0.51595 0.71752 1.01391 0.02533 0.05043	0.89517 0.73806 0.61188 0.49692 0.44948 0.42374 0.36484 0.28966 0.25518 0.27402 0.21112 0.14018 0.1709 0.19155 0.19639 0.18747 0.58292 0.56945	0.05781 0.04983 0.04774 0.04387 0.04482 0.04469 0.04685 0.04716 0.04778 0.04849 0.05007 0.04911 0.04189 0.04417 0.05956 0.05861 0.03211 0.03201
0.010790.991850.067570.015120.889660.062780.021250.802960.058520.030020.855920.064670.042160.674080.058840.059650.80160.067670.084040.583130.060820.118730.600050.067820.168030.571630.082670.191880.607910.091130.254980.506250.074130.3550.561780.065610.502150.515150.073450.701430.493820.100950.99110.412380.098010.099930.628050.023970.200280.610870.022420.399780.482470.2003	0.01082 0.01508 0.02125 0.02987 0.04217 0.05955 0.08394 0.11872 0.16771 0.19179 0.25486 0.35483 0.50191 0.70109 0.99062 0.02534 0.04913 0.09842 0.19907	0.92648 0.88101 0.66394 0.59078 0.60437 0.57911 0.50844 0.4726 0.42479 0.37019 0.36084 0.30882 0.22345 0.41314 0.28469 0.75225 0.66412 0.58733 0.43724	0.06523 0.0526 0.05382 0.05516 0.0567 0.05997 0.05899 0.06416 0.0778 0.0829 0.07005 0.07005 0.059 0.06644 0.09901 0.09488 0.02896 0.02883 0.02762 0.02542	0.00813 0.01145 0.01617 0.02253 0.03181 0.04465 0.06279 0.08888 0.12548 0.17724 0.19732 0.26199 0.36492 0.51595 0.71752 1.01391 0.02533 0.05043 0.09976	0.89517 0.73806 0.61188 0.49692 0.44948 0.42374 0.36484 0.28966 0.25518 0.27402 0.21112 0.14018 0.1709 0.19155 0.19639 0.18747 0.58292 0.56945 0.38865	0.05781 0.04983 0.04774 0.04387 0.04482 0.04469 0.04685 0.04716 0.04778 0.04849 0.05007 0.04911 0.04189 0.04417 0.05956 0.05861 0.03211 0.03201 0.03089
0.010790.991850.067570.015120.889660.062780.021250.802960.058520.030020.855920.064670.042160.674080.058840.059650.80160.067670.084040.583130.060820.118730.600050.067820.168030.571630.082670.191880.607910.091130.254980.506250.074130.3550.561780.065610.502150.515150.073450.701430.493820.100950.99110.412380.098010.099930.628050.023970.200280.610870.022420.399780.482470.020870.799250.494320.020031.598640.445970.01732	0.01082 0.01508 0.02125 0.02987 0.04217 0.05955 0.08394 0.11872 0.16771 0.19179 0.25486 0.35483 0.50191 0.70109 0.99062 0.02534 0.04913 0.09842 0.19907 0.40046	0.92648 0.88101 0.66394 0.59078 0.60437 0.57911 0.50844 0.4726 0.42479 0.37019 0.36084 0.30882 0.22345 0.41314 0.28469 0.75225 0.66412 0.58733 0.43724 0.45993	0.06523 0.0526 0.05382 0.05516 0.0567 0.05997 0.05899 0.06416 0.0778 0.0829 0.07005 0.07005 0.059 0.06644 0.09901 0.09488 0.02896 0.02883 0.02762 0.02542 0.02417	0.00813 0.01145 0.01617 0.02253 0.03181 0.04465 0.06279 0.08888 0.12548 0.17724 0.19732 0.26199 0.36492 0.51595 0.71752 1.01391 0.02533 0.05043 0.09976 0.19893	0.89517 0.73806 0.61188 0.49692 0.44948 0.42374 0.36484 0.28966 0.25518 0.27402 0.21112 0.14018 0.1709 0.19155 0.19639 0.18747 0.58292 0.56945 0.38865 0.34034	0.05781 0.04983 0.04774 0.04387 0.04482 0.04469 0.04685 0.04716 0.04778 0.04849 0.05007 0.04911 0.04189 0.04417 0.05956 0.05861 0.03211 0.03201 0.03089 0.02876
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