

1 Article

2 **Combining living microorganisms with regenerated**  
3 **silk provides nanofibril based thin films with heat**  
4 **responsive wrinkled states for smart food packaging**

5 **Luca Valentini<sup>1,\*</sup>, Silvia Bittolo Bon<sup>1</sup> and Nicola Pugno<sup>2,3,4\*</sup>**

6 <sup>1</sup> Dipartimento di Ingegneria Civile e Ambientale, Università di Perugia, UdR INSTM, Strada di Pentima 4,  
7 05100 Terni - Italy. E-mail: luca.valentini@unipg.it

8 <sup>2</sup> Laboratory of Bio-Inspired and Graphene Nanomechanics, Department of Civil, Environmental and  
9 Mechanical Engineering, University of Trento, Trento, Italy. E-mail: nicola.pugno@unitn.it

10 <sup>3</sup> School of Engineering and Materials Science, Queen Mary University of London, Mile End Road, London,  
11 United Kingdom

12 <sup>4</sup> Ket-Lab, Edoardo Amaldi Foundation, Italian Space Agency, via del Politecnico snc, I-00133 Roma, Italy

13 \* Correspondence: luca.valentini@unipg.it; Tel.: +39-0744-492924; nicola.pugno@unitn.it; Tel.:  
14 +39-0461-282525

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16 **Abstract:** Regenerated silk (RS) is a protein-based “biopolymer” that enables the design of new  
17 materials; here we sought to “bionic” the process of regenerated silk production by fermentation  
18 assisted method. Based on yeast’s fermentation, here we produced a living hybrid composite made  
19 of regenerated silk nanofibrils and a single-cell fungi, the *Saccharomyces Cerevisiae* yeast extract, by  
20 fermentation of such microorganisms at room temperature in the dissolution bath of silkworm silk  
21 fibers. The fermentation-based processing enhances the beta-sheet content of the RS, corresponding  
22 to a reduction in water permeability and CO<sub>2</sub> diffusion through RS/yeast thin films enabling the  
23 fabrication of mechanically robust film that enhances the food storage durability. Finally a  
24 transfer print method, which consists of transferring RS and RS/yeast film layers onto self-adherent  
25 paraffin substrate, was used for the realization of heat – responsive wrinkles by exploiting the high  
26 thermal expansion of the paraffin substrate that regulates the applied strain, resulting in a  
27 switchable coating morphology from the wrinkle-free state to a wrinkled state if the food  
28 temperature overcomes a designed threshold. We envision that such efficient and smart coatings  
29 can be applied for the realization of smart packaging that through such temperature sensing  
30 mechanism can be used to control the food storage conditions.

31 **Keywords:** bionic composites; thin films; mechanical properties.

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33 **1. Introduction**

34 Living microorganisms have long been used in food preservation [1,2]; such microorganisms  
35 form living surfaces that provide an attractive platform to the development of functional materials.  
36 At present, biotech companies uses fungi to produce valuable products [3], thus combining the  
37 fermentation mechanism of some microorganisms with biomaterials could give a raise to bionic  
38 composites with novel properties.

39 Between such novel products, innovative packaging solutions to increase the shelf-life of fresh  
40 fruits by slowing down their metabolism so they remain fresh and appetizing for longer and the  
41 development of sensors to monitor if perishable food in the cold chain is maintained in the desired  
42 temperature range to prevent the growth of pathogens and spoilage microorganisms, are still  
43 challenging.

44      Concerning the sensing issue, spontaneous generation of wrinkles induced by the buckling of a  
45      thin skin due to thermal contraction of the underlying substrate may be used as diffraction gratings  
46      in optical sensors, and as the basis for monitoring temperature changes in food cold chain. In this  
47      regard, micro/nanoscale surface patterns obtained by coupling a stiff skin to a soft substrate have  
48      been used to create reversible patterns that are responsive to temperature and provided unique  
49      surface morphology to sense temperature changes [4-7].

50      From the material point of view, silk fibroin is an ideal candidate for the packaging applications  
51      since it is a biocompatible structural protein that can be processed to obtain films which recently have  
52      been used as biodegradable and edible sensors to monitor food degradation [8-10]. The  
53      polymorphism of silk fibroin (i.e. random coil, silk I, and silk II structures) can be also tailored by  
54      controlling the content of  $\beta$ -sheet crystals that enables the correct gas exchange and water vapour  
55      permeability through silk-based membranes [11-13]. Between the different fabrication methods,  
56      transfer printing is the most known method for interfacing silk on soft substrates [14]. In this regard,  
57      the hidden strength and stiffness of natural honeycomb walls constructed from recycled silk and  
58      wax secreted by worker bees [15-17] is reminiscent of modern fiber-reinforced composite laminates.

59      Taking inspiration from the honeybee comb cell wall, a self-adhesive and soft thermoplastic  
60      paraffin wax, can be used to stick RS film to produce a bilayer system [18]. Being the paraffin wax a  
61      material with high thermal expansion coefficient, wrinkles occur to minimize the total energy of  
62      such bilayer system when the compressive strain, caused by the thermal expansion coefficients and  
63      rigidities mismatches between the skin layer and the substrate induced by thermal stimulus.

64      Inspired by our previous work [19] on the production of beer's yeast cells/carbon nanotube  
65      composite directly by fermentation of the yeast extract in presence of carbon nanotube aqueous  
66      dispersion, we therefore designed a robust composite coating combining inert (RS) and living parts  
67      (brewer's yeast). It was observed that once *Saccharomyces Cerevisiae* yeast cells were fermented by  
68      nutrient addition into a silk fibroin solution, the regenerated silk shows a higher content of  
69      beta-sheet structures. Moreover, the microorganism growth increased the cell density and reduced  
70      the porosity of the RS membrane limiting the exchange of water and gas diffusion. As conceptual  
71      proof, we demonstrated as an example that the deposition of such living coating on fruits helps the  
72      preservation of their shelf-life. Finally, we demonstrate that RS based film layers can be laminated  
73      onto a paraffin wax substrate for the realization of temperature – responsive bilayer systems.

## 74      2. Materials and Methods

75      For the preparation of RS film, commercial *B. mori* silk cocoons were boiled for 1 h in distillated  
76      water solution of 0.025 wt % NaHCO<sub>3</sub>, rinsing with distilled water every 30 min. to remove the  
77      sericin. According to the method adopted by Kaplan et al. [20], the degummed silk (i.e. 0.2 g) was  
78      then added to a CaCl<sub>2</sub> (i.e. 0.14 g) - formic acid (i.e. 20 ml) solution and stirred over the night at 40°C  
79      yielding an 1 wt % solution. Water solution (50 mg/ml) of *Saccharomyces Cerevisiae* based commercial  
80      beer yeast extract was prepared separately by mechanical stirring at 30 °C for 1 h. After that, sucrose  
81      was added to start the fermentation. The amount of sugar added is usually between 3 and 5 times  
82      the weight of the medium. Water solution of fermenting yeast was then added to the silk fibroin  
83      solution. RS/yeast films were prepared by leaving the silk-yeast solution to evaporate for 12 h in a  
84      polystyrene Petri dish (diameter 15 cm). The growth of yeast cells was monitored by the optical  
85      density (OD) method, measuring the absorbance at wavelength 600 nm and temperature 30°C of the  
86      yeast and RS/yeast solutions in sucrose growth medium. The morphology of the films was  
87      investigated by optical and field emission scanning electron microscopy (FESEM). Fourier transform  
88      infrared (FTIR) analysis was performed in a Jasco FTIR FT/IR-615 spectrometer, equipped with an  
89      ATR mode in the wave number range from 400 to 4000 cm<sup>-1</sup>. X-ray diffraction was performed using a  
90      Bruker D8 Advance diffractometer with a radiation source of CuK $\alpha$  and wavelength  $\lambda=0.154$  nm  
91      operated at 40 kV and 40 mA. The incidence angle ( $2\theta$ ) was varied between 2° and 60° and the scan  
92      rate was 0.02°/s. The tensile properties of films were measured using a universal tensile testing  
93      machine (Lloyd Instr. LR30K) with a 50 N static load cell. The film samples were cut into strips (30  
94      mm  $\times$  12 mm). The gauge length was 20 mm, and the extension rate was set at 2 mm/min.

95 The effect of different types of coatings on bananas freshness was evaluated by monitoring the  
96 colour change through time-lapse photography. The water permeability was determined after  
97 soaking a sponge in water and subsequently dip coating the sponge in RS and RS/yeast solutions.  
98 The variation of the weight was monitored at different hours with a standard laboratory balance  
99 (Mettler Toledo AB135-S/FACT). The weight variation was calculated as an average of three  
100 measurements for each coating. The respiration rate of bananas was evaluated by monitoring the  
101  $\text{CO}_2$  production. In brief, bananas were placed in a sealed FTIR chamber and the production of  $\text{CO}_2$   
102 was monitored by measuring the evolution of the  $\text{CO}_2$  absorption peak over period of 7 days (see  
103 supplementary material Fig. S1). This measurement takes into account the initial background  
104 performed in air to remove the initial contribution of the carbon dioxide moisture of the air.

105 For the adopted transfer print process to realize the bilayer system, regenerated silk was  
106 transferred to Parafilm film (Parafilm M®, Pechiney Plastic Packaging Company) through direct  
107 transfer process, which consists of placing RS and RS/yeast free standing films on the receiving  
108 Parafilm substrate while applying with a hot press a pressure of 2kPa at 60°C for 15 min.; the obtained bilayer  
109 systems were heated at 60°C and cooled to room temperature (i.e.  $\Delta T \approx 40^\circ\text{C}$ ) and AFM was carried  
110 out to measure wrinkle morphology in tapping mode (easyScan DFM system).

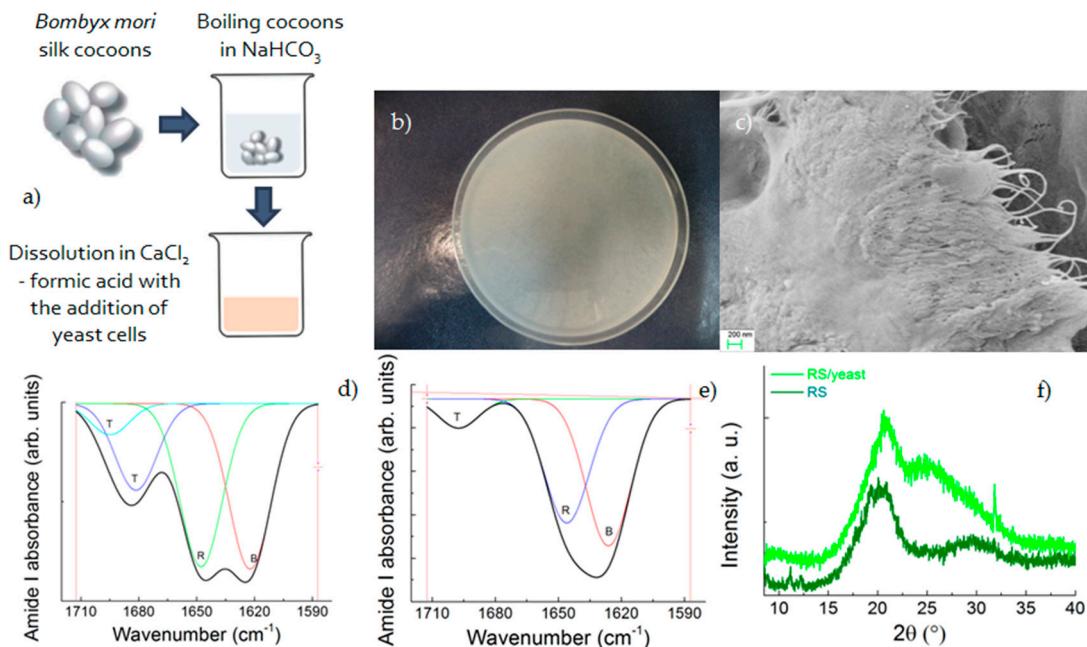
### 112 3. Results and discussion

113 The production method adopted for the realization of thin films with the aid of living  
114 microorganisms is schematically reported in Fig. 1a. The yeast fermentation was implemented into a  
115  $\text{CaCl}_2$ -formic acid dissolution system, which can be used to produce large films (Fig. 1b) with a  
116 nano-fibrillar structure (Fig. 1c).

117 Changes in structure of the films deposited after the fermentation assisted silk dissolution were  
118 detected by FTIR and X Ray Diffraction (XRD). The  $\beta$ -sheet (crystalline) content was determined by  
119 the deconvolution of the amide I region ( $1580 - 1720 \text{ cm}^{-1}$ ) estimating the ratio between the peak area  
120 in the wavenumber region of  $1622\text{--}1637 \text{ cm}^{-1}$ , which is the main absorbance region of  $\beta$ -sheet crystal  
121 in amide I [21], and the whole area of the amide I comprising the peaks of the structural components  
122 including turns (T) and random coil (R). The deconvolution of the amide I band provides an  
123 estimation of  $\beta$ -sheet structure in the RS and RS/yeast of 37% and 44%, respectively (Figs. 1d-e).

124 The XRD data in Fig. 1f show the RS/yeast film is characterized by diffraction peaks at  $2\Theta$  values of  
125  $20.4^\circ$  and  $25.4^\circ$ , corresponding to silk I structure [22,23]. RS film also showed silk I and silk II crystal  
126 structures, having diffraction peaks at  $20.4^\circ$  and  $29.0^\circ$ . Compared with RS/yeast, the silk I peak at  
127  $25.0^\circ$  disappeared, indicating more silk II formation.

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130       Figure 1. (a) Scheme of silk fibroin production using regenerated silk. (b) Visual appearance of the  
 131       RS/yeast film and (c) FESEM image of silk nanofibrils. (d-e) FTIR spectra of regenerated silk and RS/yeast  
 132       films, respectively. The coloured lines represent the components of the amide I band and are indicated as  
 133       β-sheets (B), random coil (R) and turns (T). (f) XRD results of regenerated silk and RS/yeast films.

134       The cell division of yeast occurs by budding in which a daughter is initiated from the mother  
 135       cell, followed by nuclear division and finally cell separation. The yeast cell growth reported in Fig. 2  
 136       shows the three main phases: the lag phase where the individual cells are activated in preparation  
 137       for division, the exponential phase once the cell starts actively metabolizing shortly after the cells  
 138       divide and finally the stationary phase when metabolism slows and the cells stop rapid cell division  
 139       [24]. The factors that cause cells to enter stationary phase are related to change in the environment  
 140       typically caused by high cell density. The data reported in Fig. 2a state the stability of the yeast cells  
 141       to proliferate also with the presence of formic acid in the nutrient broth. The effects of RS addition  
 142       during the growth curve are demonstrated in Fig. 2a by measuring the OD during the cell growth.  
 143       The OD curve of the yeast cells is substantially altered by the RS addition, with the effect on the lag  
 144       time and final cell yield being particularly pronounced: RS/yeast cells have a lag phase of ~ 2-3 hrs,  
 145       after which they proliferate rapidly, in comparison, the neat yeast cells show an increased lag  
 146       phase of ~ 10 hrs. The morphology of the stationary phase for the RS/yeast system observed by  
 147       means of FESEM and optical microscopy (Figs. 2b-c) indicates the cell proliferation for such culture.  
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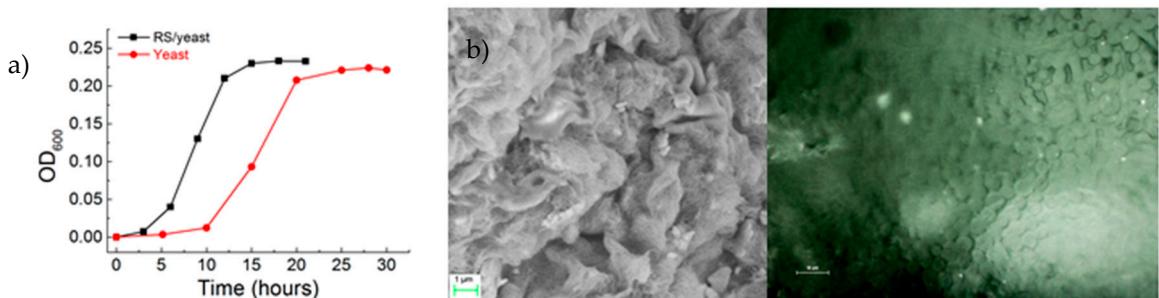


Figure 2. (a) OD measured at 600 nm wavelength during growth of neat *Saccharomyces Cerevisiae* and *Saccharomyces Cerevisiae* in RS with sucrose until stationary was reached. (b) Appearance under FESEM and optical microscope of the RS/yeast cell growth during the stationary phase.

In order to demonstrate potential application of such living coating in consumer exposed food the mechanical robustness of the films is required to withstand, for example, handling procedures.

Figure 3a represents typical stress/strain curves obtained from testing of yeast, RS and RS/yeast samples. The maximum average toughness (i.e. the area underlying the stress-strain curves) obtained from RS/yeast sample was 0.14 MPa (Fig. 3b) as well as the highest average strength (i.e. the stress at the ultimate strain) obtained was 1.26 MPa, with a maximum elastic modulus of 37.2 MPa recorded. The improved mechanical properties with fermentation-based dissolution of silkworm silk fibers agree with our previous studies reported for the assisted-fermentation synthesis of bionic composites [19,25]. In these studies, addition of carbon nanotubes (CNTs) and/or graphene in the fermentation broth, resulted in composites with higher toughness value. In our studies, tensile tests on dried composite films were rationalized in terms of a CNT cell bridging mechanism where the strongly enhanced strength of the composite is governed by the adhesion energy between the bridging CNTs and the matrix. The presence of glucose can plasticize regenerated silks and increase the ultimate strain leading to toughness values (see Table 1) that are comparable to those measured for traditional biopolymers used for food packaging [26,27] being in our case as added value the edibility of the coating.

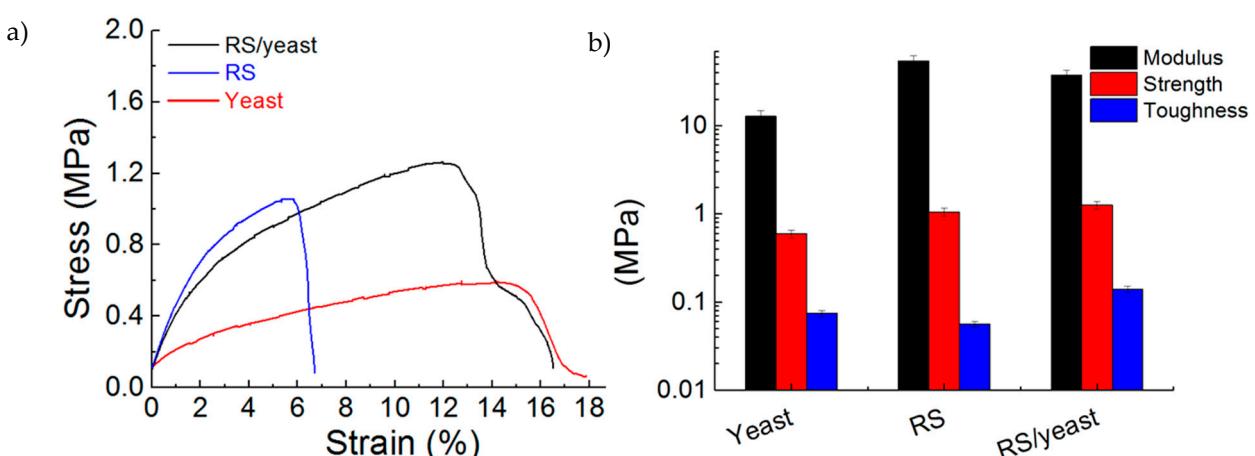


Figure 3. (a) Stress-strain curves from testing of yeast, RS and RS/yeast samples. The curves represent analysis performed on three different samples. (b) Effect of yeast fermentation on modulus, strength, and toughness of RS film. The modulus, strength, and toughness of neat yeast film have been reported for comparative purposes.

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183Table 1. Mechanical properties of various biopolymers for food packaging [27].  $\sigma^*$  — specific tensile strength,  $E^*$  — specific tensile modulus,  $\epsilon$  — ultimate strain.

Type of biopolymer	Toughness (MPa)	$\sigma^*$ (Nm/g)	$E^*$ (kNm/g)	$\epsilon$ (%)
RS/yeast (this work)	0.14	1.1	0.032	15.4
PLA	0.26	16.8	0.3	2.5
PLLA	0.23	40.0	2.2	3.0
PDLLA	0.28	22.1	0.8	2.0
PDLLA/PGA50/50	0.41	30.9	0.8	2.0

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In general, coatings for food packaging beyond the mechanical robustness should exert low permeability to water vapors; fruit dehydration is, in general, an indicator of the breakdown of the protective skin, which results in loss of turgor and water evaporation. We observe that the increase in beta-sheet content yields a less water permeability through the RS/yeast membrane as indicated by the less variation of the initial weight of soaked sponges coated with different types of coatings, as reported in Fig. 4a (see supplementary material Fig. S2). These results are in agreement with those obtained by Omenetto et al.<sup>10</sup> who showed that when the silk fibroin beta-sheet content is in the range of 36%–58%, the water vapour permeability is five times smaller than that measured for the film with a lower beta-sheet content.

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Many fresh fruits have high metabolic activity and due to microbial attack they result in short conservation time, colour change (Fig. 4b), and off-flavour. The change in colour of fresh fruit, in particular, is associated to ethylene production and cell respiration. To evaluate the exploitation of RS/yeast film as barrier coating, the change in color of coated and non-coated fresh bananas was evaluated (Fig. 4b). Time-lapse photography shows that RS/yeast coating decreases the fruit degradation, when compared to uncoated or RS coated fruit at day 7. During the continuing metabolism of the fresh fruit, oxygen is transformed into carbon dioxide; thus gas permeation through a coating film plays a crucial role in fruit storage. To prevent the spoilage of fresh fruits it is necessary to reduce the breathing process. In our case, the higher beta-sheet content of the RS/yeast film decreases the production of CO<sub>2</sub>, which indicated a decrease in the respiration rate of the fruits (Fig. 4c).

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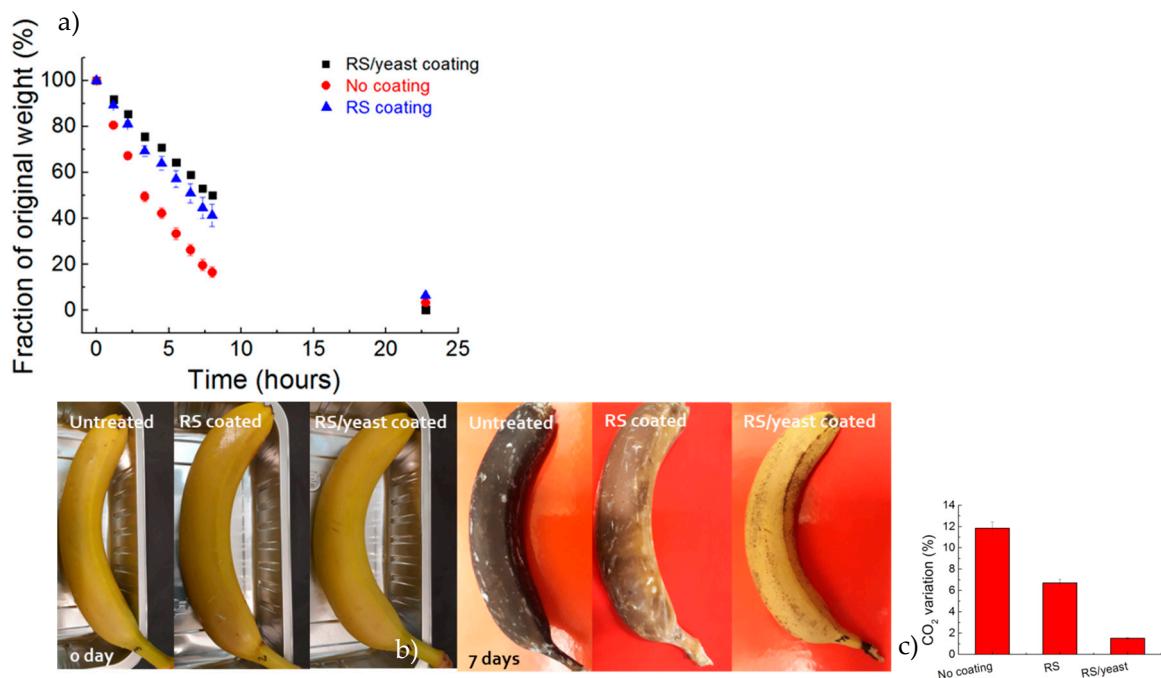
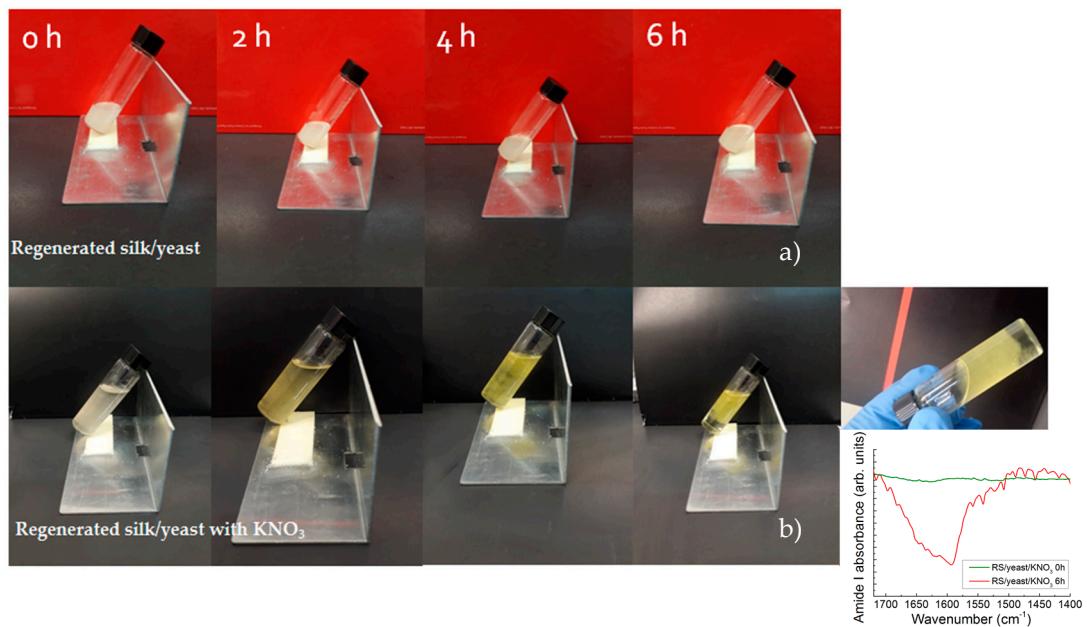


Figure 4. (a) Weight loss of sponges soaked in water and stored for up to 25 hours at 23 °C and 32% RH. Sponges were stored without coating or after dip coating with RS and RS/yeast suspensions, respectively. (b) Evaluation of bananas degradation. Fruits were stored at 23 °C and 32% RH as received and after coating with silk/yeast film. Time lapse photography of banana degradation, indicating that silk/yeast coating reduces the degradation rate. (c) CO<sub>2</sub> variation of uncoated, RS coated and RS/yeast coated bananas, respectively.

Another very interesting property of such silk nanofibrils relies on their ability to self-assemble giving rise to sol-gel transition with rapid gelation time induced by the presence of salts [28,29]. Pregelation occurs when a fresh solution has a  $\beta$ -sheet content of about 20% with negligible intermolecular bindings; gelation is then induced by interchain interactions that become irreversible with the formation of  $\beta$ -sheet intermolecular structures of the gel phase [29]. In our case, the gelation was observed when KNO<sub>3</sub> salt was added to silk nanofibrils/yeast/formic acid solution (Figure 5). Without salt, the RS/yeast retains with time a sol characteristic (Fig. 5a). In comparison with the sol RS/yeast solution, the RS/yeast with salt transform into a semi-solid gel within several hours, together with the appearance of a strong infrared absorption peak at 1626 cm<sup>-1</sup> due to the formation of strong  $\beta$ -sheet structures (Fig. 5b).

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230231 Figure 5. Dynamic optical morphology of the (a) RS/yeast and (b) RS/yeast/KNO<sub>3</sub> solutions  
232 resulting at different times (temperature 37°C). The inset of panel (b) shows the evolution of the  
233 amide I FTIR band of the RS/yeast/KNO<sub>3</sub> solution with time and gelation of the RS/yeast/KNO<sub>3</sub>  
234 solution.

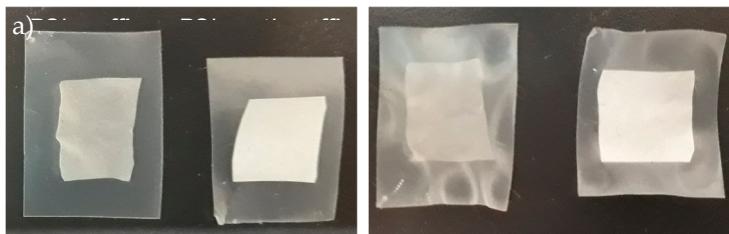
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236 Considering the application of controlling the food cold chain, it is essential to design a bilayer  
237 packaging system where the mechanical properties of the top skin layer when laminated onto a soft  
238 substrate will be beneficial for the creation of temperature-driven surface patterning. Figure 6a  
239 shows the RS and RS/yeast films laminated onto the paraffin wax substrate. Wrinkle formation is  
240 typically connected to the high thermal expansion coefficient of the substrate used for the transfer.  
241 Parafilm is worldwide used in research laboratories as self-adhesive and sealant plastic foil, it is soft  
242 (tensile strength ≈ 2.0 MPa) with a high thermal expansion coefficient (i.e. 0.89\*10<sup>-3</sup> K<sup>-1</sup>) and due to its  
243 low melting point (≈ 60°C) it becomes adhesive applying heat during lamination, sticking strongly to  
244 the receiving material. The formation of wrinkles occurs to minimize the total potential energy of the  
245 skin layer and the substrate induced by thermal expansion. The strategy for the realization of the  
246 heat-driven wrinkle patterns is illustrated in Figure 6b; once heated, the paraffin wax upon cooling  
247 to room temperature, generates a compressive stress at the interface of the bilayer sample, owing to  
248 the considerable mismatch between the modulus and thermal expansion ratio of the substrate and  
249 the stiff top layer made of RS or RS/yeast [30]. AFM analysis reported in Figure 6c shows that a  
250 smooth surface converts into wrinkled state through heating and then cooling down to room  
251 temperature.252  
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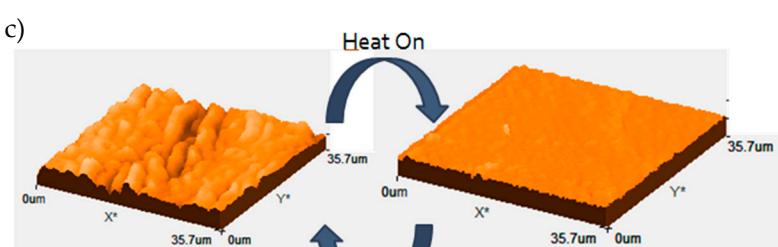
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Figure 6. (a) Photographs of RS/yeast/paraffin and RS/paraffin bilayer systems before (left) and after (right) the wrinkle activation. (b) Schematic illustration of temperature driven wrinkling. The bilayer system is flat at room temperature, the heat induces the temperature increasing and thus the thermal expansion of the paraffin substrate, resulting in the increase in the compressive strain of the bilayer systems once cooled down to room temperature with the appearance of the wrinkles. Once the wrinkled state was activated, the heat/temperature can again induce the thermal expansion of the paraffin substrate, resulting in the reversibility of the temperature-driven wrinkles. (c) AFM images showing the reversibility of the temperature-driven wrinkles. (d) AFM images and related profile images of RS (left) and RS/yeast (right) film layers.

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The amplitude (A) and the wavelength ( $\lambda$ ) of the wrinkles depend on the thickness and mechanical properties of the skin layer and the substrate and according to linear buckling theory [31-35] can be expressed as:

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$$A = h_f ((\sigma_0 - \sigma_c) / \sigma_c)^{1/2} / a \quad (1)$$

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and

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$$\lambda = 4.36 h_f (E_f (1 - v_s^2) / (E_s (1 - v_f^2)))^{1/3} - 2A(1-a)/a \quad (2)$$

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where  $\sigma_c$  refers to the critical strain to wrinkle formation and is given as<sup>33</sup>

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$$\sigma_c = (1/4) (E_f)^{1/3} (E_s)^{2/3} \quad (3)$$

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and  $\sigma_0$  is the compressive stress of the film layer at temperature below the heating temperature given by the equation 4 [37]:

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$$\sigma_0 = (E_f (\alpha_s - \alpha_f) \Delta T) / (1 - v_f) \quad (4)$$

282 where  $E' = E/(1-v^2)$  and subscript f and s refer to the film layer and the substrate of the bilayer system  
 283 and  $E'$ ,  $E$  and  $v$  are in plane modulus, Young's modulus and Poisson's ratio, respectively.  $h_f$  (equal to  
 284  $1\mu\text{m}$ ) is the film thickness and  $0 \leq a \leq 1$  is an adhesion parameter that we have introduced (ideally  $a=1$ )  
 285 for accounting the nonideal bonding between film and substrate (imposing the film inextensibility  
 286 and simply assuming squared shape wrinkles, i.e.  $2A+\lambda=\text{cost}$ ). The Young's modulus for RS,  
 287 RS/yeast and paraffin are 54 MPa, 37 MPa, 1.4 MPa [18], respectively, and  $v=0.5$ . The applied strain  $\epsilon$   
 288 when the bilayer system is heated is calculated as  $\epsilon=(\alpha_s-\alpha_f)*\Delta T$  where  $\alpha_s$  and  $\alpha_f$  are the thermal  
 289 expansion coefficients with  $\alpha_s > \alpha_f$ . Finally, the theoretical wavelength and amplitude values  
 290 obtained from Eq. 1 and Eq. 2 are reported in Tab. 2 and compared with the experimental findings  
 291 by fitting the single parameter  $a$ .

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Table 2. Theoretical and experimental (estimated by AFM images and related profiles reported in Fig. 6d) amplitude and wavelength of the different skin layers, with  $a=0.4$ .

Sample	$A_{\text{Theor.}}$ ( $\mu\text{m}$ )	$\lambda_{\text{Theor.}}$ ( $\mu\text{m}$ )	$A_{\text{Exp.}}$ ( $\mu\text{m}$ )	$\lambda_{\text{Exp.}}$ ( $\mu\text{m}$ )
RS/paraffin	2.72	1.83	1.23	2.93
RS/yeast/paraffin	2.20	2.21	1.04	2.14

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## 297 5. Conclusions

298 Here we described how incorporating microorganisms in bio-based structural proteins, by  
 299 providing an opportune nutrient for their growth, can results in a composite material with novel  
 300 properties. We observed that the fermentation of yeast cells activated in the dissolution bath of silk  
 301 fibers is critical in determining the  $\beta$ -sheet content of RS and thus the mechanical properties. In  
 302 particulat the yeast fermentation increases the crystalline content of regenerated silk fibroin, such  
 303 bionic coating reduces the water permeability and acts as an effective gas barrier, e.g. to increase the  
 304 shelf-life of perishable food. Then we reported a method which consists of transfer printing the  
 305 prepared freestanding RS and RS/yeast layer films onto on self-adherent Parafilm substrate. It is  
 306 reported how the mechanical characteristics of the film layers as well as the high thermal expansion  
 307 coefficient of the paraffin substrate can be used to fabricate a sensor with the surface morphology  
 308 that changes between the wrinkled state and the wrinkle-free state under an external thermal  
 309 stimulus. This naturally derived material is thus a promising solution for different problems, e.g. a  
 310 valid alternative for efficient and smart food packaging with respect to traditional polymer-based  
 311 coatings. Finally, the proposed model can help in the real thus nonideal design of such coatings and  
 312 the degree of adhesion also emerges as designing parameter for tuning the wrinkle morphologies.

313 **Supplementary Materials:** The following are available online, Figure S1: Evolution of the  $\text{CO}_2$  absorption peak  
 314 over period of 7 days for (left) RS/yeast and (right) RS coating, respectively; Figure S2: Water soaked sponges  
 315 with different types of coatings.

316 **Author Contributions:** Conceptualization, L.V. and N.M.P.; Methodology, L.V.; Investigation, S.B.B.; Data  
317 Curation, S.B.B.; Writing-Original Draft Preparation, L.V. and N.M.P.; Writing-Review & Editing, L.V. and  
318 N.M.P.

319 **Funding:** N.M.P. is supported by the European Commission under the Graphene Flagship Core 2 grant No.  
320 785219 (WP14 "Polymer Composites") and FET Proactive "Neurofibres" grant No. 732344 as well as by the  
321 Italian Ministry of Education, University and Research (MIUR) under the "Departments of Excellence" grant  
322 L.232/2016. L.V. is supported by the Italian Ministry of Education, University and Research (MIUR) under the  
323 "Departments of Excellence" grant L.232/2016.

324 **Acknowledgments:** Prof. Miguel-Angle Lopez-Manchado (Instituto de Ciencia y Tecnología de Polímeros,  
325 ICTP-CSIC, Madrid) is kindly acknowledged for the XRD analysis.

326 **Conflicts of Interest:** The authors declare no conflict of interest.

## 327 **References**

- 328 1. Cleveland, J.; Montville, T. J.; Nes I. F.; Chikindas, M. L. Bacteriocins: safe, natural antimicrobials for food  
329 preservation. *Int. J. Food Microbiol.* **2001**, *71*, 1–20.
- 330 2. Paul Ross, R.; Morgan S.; Hill, C. Preservation and fermentation: past, present and future. *Int. J. Food  
331 Microbiol.* **2002**, *79*, 3–16.
- 332 3. Synowiecki J.; Al-Khateeb, N. A. Production, properties, and some new applications of chitin and its  
333 derivatives. *Crit. Rev. Food Sci. Nutr.* **2003**, *43*, 145–171.
- 334 4. Ohzono, T.; Monobe, H.; Yamaguchi, R.; Shimizu, Y.; Yokoyama, H. Dynamics of surface memory effect in  
335 liquid crystal alignment on reconfigurable microwrinkles. *Appl. Phys. Lett.* **2009**, *95*, 14101.
- 336 5. Kim, P.; Hu, Y.; Alvarenga, J.; Kolle, M.; Suo, Z.; Aizenberg, J. Rational Design of Mechano-Responsive  
337 Optical Materials by Fine Tuning the Evolution of Strain-Dependent Wrinkling Patterns. *Adv. Opt. Mater.*  
338 **2013**, *1*, 381 – 388.
- 339 6. Bowden, N.; Huck, W. T. S.; Paul, K. E.; Whitesides, G. M. The controlled formation of ordered, sinusoidal  
340 structures by plasma oxidation of an elastomeric polymer. *Appl. Phys. Lett.* **1999**, *75*, 2557 – 2559.
- 341 7. Ohzono, T.; Suzuki, K.; Yamaguchi, T.; Fukuda, N. Tunable optical diffuser based on deformable wrinkles.  
342 *Adv. Opt. Mater.* **2013**, *1*, 374 – 380.
- 343 8. Kim, D. H.; Viventi, J.; Amsden, J. J.; Xiao, J.; Vigeland, L.; Kim, Y. S.; Blanco, J. A.; Panilaitis, B.; Frechette,  
344 E. S.; Contreras, D.; Kaplan, D. L.; Omenetto, F. G.; Huang, Y.; Hwang, K. C.; Zakin, M. R.; Litt B.; Rogers, J.  
345 A. Dissolvable films of silk fibroin for ultrathin conformal bio-integrated electronics. *Nature Materials* **2010**,  
346 *9*, 511–517.
- 347 9. Tao, H.; Brenckle, M. A.; Yang, M.; Zhang, J.; Liu, M.; Siebert, S. M.; Averitt, R. D.; Mannoor, M. S.;  
348 McAlpine, M. C.; Rogers, J. A.; Kaplan D. L.; Omenetto, F. G. Silk-based conformal, adhesive, edible food  
349 sensors. *Advanced Materials* **2012**, *24*, 1067–1072.
- 350 10. Marelli, B.; Brenckle, M. A.; Kaplan D. L.; Omenetto, F. G. Silk fibroin as edible coating for perishable food  
351 preservation. *Sci. Rep.* **2016**, *6*, 25263.
- 352 11. Lawrence, B. D.; Pan, Z.; Liu, A.; Kaplan D. L.; Rosenblatt, M. I. Human corneal limbal epithelial cell  
353 response to varying silk film geometric topography in vitro. *Acta Biomaterialia* **2012**, *8*, 3732–3743.
- 354 12. Lawrence, B. D.; Marchant, J. K.; Pindrus, M. A.; Omenetto F. G.; Kaplan, D. L. Silk film biomaterials for  
355 cornea tissue engineering. *Biomaterials* **2009**, *30*, 1299–1308.
- 356 13. Lawrence, B. D.; Wharram, S.; Kluge, J. A.; Leisk, G. G.; Omenetto, F. G.; Rosenblatt, M. L.; Kaplan, D. L.  
357 Effect of hydration on silk film material properties. *Macromolecular Bioscience* **2010**, *10*, 393–403.
- 358 14. Brenckle, A. M.; Kaplan, D. L.; Omenetto, F. G. Direct Transfer Printing of Water Hydrolyzable Metals  
359 onto Silk Fibroin Substrates through Thermal-Reflow-Based Adhesion. *Adv. Mater. Interfaces* **2016**, *3*,  
360 1600094.
- 361 15. Jay, S. C. The cocoon of the honey bee, *Apis mellifera*. *Can. Entomol.* **1964**, *96*, 784–792.
- 362 16. Hepburn, H. R.; Kurstjens, S. P. The combs of honeybees as composite materials. *Apidologie* **1988**, *19*, 25–36.
- 363 17. Dorset, D. L. The crystal structure of waxes. *Acta Crystallogr. B* **1995**, *51*, 1021–1028.
- 364 18. Valentini, L.; Bittolo Bon, S.; Lopez-Manchado, M.-A.; Mussolin, L.; Pugno, N. Development of conductive  
365 paraffin/graphene films laminated on fluoroelastomers with high strain recovery and anti-corrosive  
366 properties. *Composites Science and Technology* **2017**, *149*, 254–261.

367 19. Valentini, L.; Bittolo Bon, S.; Signetti, S.; Tripathi, M.; Iacob, E.; Pugno, N. M. Fermentation based carbon  
368 nanotube multifunctional bionic composites. *Sci. Rep.* **2016**, *6*, 27031.

369 20. Ling, S.; Zhang, Q.; Kaplan, D. L.; Omenetto, F.; Buehler, M. J.; Qin, Z. Printing of stretchable silk  
370 membranes for strain measurements. *Lab Chip* **2016**, *16*, 2459-2466.

371 21. Hu, X.; Kaplan D.; Cebe, P. Determining Beta-Sheet Crystallinity in Fibrous Proteins by Thermal Analysis  
372 and Infrared Spectroscopy. *Macromolecules* **2006**, *39*, 6161-6170.

373 22. Q. Lu, X. Hu, X. Wang, J. A. Kluge, S. Lu, P. Cebe and D. L. Kaplan, Water-insoluble silk films with silk I  
374 structure. *Acta Biomaterialia* **2010**, *6*, 1380–1387.

375 23. J. P. Anderson, Morphology and crystal structure of a recombinant silk-like molecule, SLP4. *Biopolymers*  
376 **1998**, *45*, 307–321.

377 24. J. Warringer, E. Ericson, L. Fernandez, O. Nerman and A. Blomberg, High-resolution yeast phenomics  
378 resolves different physiological features in the saline response. *PNAS* **2003**, *100*, 15724–15729.

379 25. L. Valentini, S. Bittolo Bon, S. Signetti and N. M. Pugno, Graphene-Based Bionic Composites with  
380 Multifunctional and Repairing Properties. *ACS Applied Materials & Interfaces* **2016**, *8*, 7607-7612.

381 26. A. Seidel, O. Liivak, S. Calve, J Adaska, G. Ji, Z. Yang, D. Grubb, D. B. Zax and L. W. Jelinski, Regenerated  
382 Spider Silk: Processing, Properties, and Structure. *Macromolecules* **2000**, *33*, 775-780.

383 27. K. Van de Velde, P. Kiekens, Biopolymers: overview of several properties and consequences on their  
384 applications. *Polym. Test.* **2002**, *21*, 433–442.

385 28. D. S. Im, M. H. Kim, Y. I. Yoon, W. H. Park. Gelation behaviors and mechanism of silk fibroin according to  
386 the addition of nitrate salts. *Int. J. Mol. Sci.* **2016**, *17*, 1697.

387 29. A. Matsumoto, J. Chen, A. L. Collette, U. J. Kim, G. H. Altman, P. Cebe and D. L. Kaplan, Mechanisms of  
388 silk fibroin sol-gel transitions. *J. Phys. Chem. B* **2006**, *110*, 21630 – 21638.

389 30. Hou, H.; Yin, J.; Jiang, X. Reversible diels-alder reaction to control wrinkle patterns: from dynamic  
390 chemistry to dynamic patterns. *Adv. Mater.* **2016**, *28*, 9126 – 9132.

391 31. Rodríguez-Hernández, J. Wrinkled interfaces: taking advantage of surface instabilities to pattern polymer  
392 surfaces. *Prog. Polym. Sci.* **2015**, *42*, 1 – 41.

393 32. Chen, C.-M.; Yang, S. Wrinkling instability in polymer films and their applications. *Polym. Int.* **2012**, *61*,  
394 1041 – 1047.

395 33. N. Bowden, S. Brittain, A. G. Evans, J. W. Hutchinson, G. M. Whitesides, Spontaneous formation of  
396 ordered structures in thin films of metals supported on an elastomeric polymer. *Nature* **1998**, *393*, 146 –  
397 149.

398 34. C. M. Stafford, C. Harrison, K. L. Beers, A. Karim, E. J. Amis, M. R. VanLandingham, H.-C. Kim, W.  
399 Volksen, R. D. Miller, E. E. Simonyi, A buckling-based metrology for measuring the elastic moduli of  
400 polymeric thin films. *Nat. Mater.* **2004**, *3*, 545 – 550.

401 35. Wang, Q.; Zhao, X.; Beyond wrinkles: Multimodal surface instabilities for multifunctional patterning.  
402 *MRS Bull.* **2016**, *41*, 115 – 122.

403 36. Allen, H. G. Analysis and Design of Structural Sandwich Panels (Pergamon, New York, 1969).

404 37. Ohring, M. The Material Science of Thin Films (Academic, San Diego, 1992).