

1 Article

## 2 Analysis of the process parameter influence in laser 3 cladding of 316L stainless steel

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11

12 **Abstract:** Laser Cladding is one of the leading processes within Additive Manufacturing  
13 technologies, a fact which has concentrated an important amount of effort on its development. In  
14 regard to the latter, the current study aims to summarize the influence of the most relevant process  
15 parameters in the laser cladding processing of single and compound volumes (solid forms) made  
16 from AISI 316L stainless steel powders and using a coaxial nozzle for deposition. Process speed,  
17 applied laser power and powder flow are considered to be the main variables affecting laser  
18 cladding in single clads, meanwhile overlap percentage and overlapping strategy become also  
19 relevant when dealing with multiple clads. By means of setting appropriate values of each process  
20 parameter, the main goal of this paper is to develop a processing window in which a good  
21 metallurgical bond between the delivered powder and substrate is obtained, trying simultaneously  
22 to maintain processing times in their lowest value possible. Conventional metallography  
23 techniques were performed on the cross sections of the laser tracks to measure the effective  
24 dimensions of clads for dilution analysis, height and width for the values of overlap between  
25 contiguous clads and layers, and also to analyze them for physical defects such as porosity and  
26 cracks. The resulting solid piece was 8 mm high at 800 mm/min.

27 **Keywords:** laser cladding; powder flow; 316L stainless steel

28

### 29 1. Introduction

30 Current development in Additive Manufacturing (AM) has tweaked the product design  
31 process itself, by adding to it the possibility to work on a CAD model and create solid metal parts  
32 directly from such a model [1]. Within this group, Laser Metal Deposition (LMD) or Laser Cladding  
33 (LC) stands as one of the most relevant AM processes in development and, thanks to its versatility as  
34 an AM process [2,3,4], LC technology has become a powerful process for designing and generating  
35 parts with complex structures in the industry [5,6]. This novel manufacturing process allows  
36 manufacturing parts or pieces without limitation of size, or the need of an expensive tooling and  
37 adjusting the amount of material necessary to the actual design piece by adapting the laser spot size  
38 to the geometry to be built and minimizing material waste [7,8]. In addition to the latter, when  
39 dealing with already manufactured parts, LC is often used as a mean of increasing the wear or  
40 corrosion resistance of diverse surfaces and as way of repairing damaged surfaces. Elements such as  
41 tools and dies are able to achieve longer life-times thanks to the addition of functional layers on their  
42 surface [9], by coating less noble surfaces with noble materials, thus achieving better surface  
43 properties.

44 When related with the making of new parts, LC is used to generate tridimensional geometries  
45 on top of a surface [10], like the way inserts or protrusions are placed in a mold. This type of LC

46 process starts with the slicing of CAD model components with whom a sequence of equally spaced  
47 layers are extracted to be used as an input to the process. Using this information, deposition  
48 strategies are generated in order to be able to fill the non-empty areas of each layer [11]. These  
49 strategies must comply with specific requirements of each material such as overlap percentage,  
50 energy density and powder flow to accomplish a good metallurgical bond between the deposited  
51 layer and substrate [12]

52 In this direction, for instance, dilution phenomena are very relevant results to examine when  
53 looking for an appropriate process window for these parameters. Quantitatively, dilution can be  
54 calculated as the ratio of molten substrate to molten material during a LC process. The importance of  
55 dilution control lies on the fact that, as a consequence of fusion of the substrate material, mechanical  
56 and corrosion properties of the deposited layer can be heavily degraded due to the mixture with the  
57 substrate [13]. In comparison with a higher energetic process as arc welding, dilution of the melted  
58 substrate in LC is lower but it is nonetheless difficult to find the optimization of process parameters  
59 to guarantee the efficiency of the metallurgical bond [13]. Another disadvantage is that it is  
60 considered a slow process when compared with other welding processes like arc welding process  
61 [14].

62 But besides dilution, physical defects such as porosity and cracking can also be found on clads  
63 and their appearance can lead to significant decrease in the mechanical properties of the cladded  
64 volume as a whole. These appear mostly when dealing not only with individual clads, but when  
65 overlapping one onto another horizontally or vertically. Main parameters as speed of the process,  
66 applied power and powder flow can affect in different forms the way the geometry of a cladded  
67 volume is formed at the end of a process and, therefore, defects may appear within the deposited  
68 material.

69 With this in mind, a valid focus to take is to study the LC process in a two-step iterative process:  
70 in the beginning, the main process parameters mentioned earlier can be evaluated in order to obtain  
71 those with which quality clads with appropriate dilution [15]. Afterwards, these parameters can be  
72 used to be able to start a LC process with overlapped clads, and the transient changes that must take  
73 place in the process in order to minimize the probability of obtaining defects (power decrease in  
74 time, overlapping percentage between clads, etc.) become the input variables of the studied process,  
75 the progressive optimization of a LC process, setting processing time (the less the better), quality of  
76 the clad and energetic process efficiency as output variables. Following this approach, the main goal  
77 of the current work is to present an introductory study of laser cladding, in the way to achieve a high  
78 production manufacturing process. In order to achieve this goal, the influence of each main  
79 parameter of the LC process in individual and overlapped clads will be studied as well as the  
80 strategy to manufacture solid pieces.

81 To be able to form clads, a coaxial nozzle is used for powder deposition and a fiber 6 kW IPG  
82 YLS-6000 is used as a laser power source with a 5 mm diameter spot. It is important to note that with  
83 this type of nozzle it is possible to deliver the powder flux coaxially with the laser beam, thus  
84 applying almost equally distributed amounts of powder radially. The latter makes it possible to  
85 perform the deposition in all directions of the substrate and, as a side effect; it makes it easier to  
86 control the accumulated temperature on the workpiece, so that the coaxial laser cladding process  
87 may be considered to be independent of the laser track direction [16]. As for the materials used,  
88 carbon steel S355 will serve as substrate for making clads composed of AISI 316L deposited and  
89 molten powders.

90 The first set of experiments were single deposited clads, and the optimization was performed  
91 by setting different values in the main LC process parameters (power, process speed and powder  
92 flow) to analyze the influence of each one. The second set of tests consisted in the fabrication of solid,  
93 tridimensional figures which were deposited over a plane substrate. The measured effective  
94 dimensions of single laser clads from the first set was used to calculate overlap of contiguous clads.  
95 To be able to grow layer by layer to form the figures, temperature was tried to be kept constant by  
96 regulating laser power for each subsequent layer. Moreover, the strategy of the overlap distribution  
97 was studied by proposing three approaches: Contour, Zig Zag and Parallel clads. Each one of them is

98 analyzed since they perform well in controlling the accumulated heat during the process [9, 17] and  
 99 are also appropriate to achieve enough dilution between the delivered powder and the metal  
 100 substrate.

101 Summarizing, this work is focused on a long term goal achieving a robust LC manufacturing  
 102 process for tridimensional pieces. Less processing time and good quality are the qualitative variables  
 103 taken into account. This study will allow the development of a work methodology in order to get  
 104 information about a LC process in a progressively staged mode. By this way, physical characteristics  
 105 inherent to each one of the moments of the process will be carefully selected and studied, leaving a  
 106 cleaner way to optimize the process window. Here, for instance, the geometry of the cross section of  
 107 the clad will served as the primary way of understanding individual clads paved the way to search  
 108 for the best results when manufacturing multiple clad volumes.

## 109 2. Materials and Methods

### 110 2.1. Material Preparation

111 AISI 316L (EN 1.4404) powder, with a particle size range of 44 and 106  $\mu\text{m}$ , was used as additive  
 112 material. Its chemical composition is listed in Table 1.

113 Table 1. Chemical composition (wt %) of 316L stainless steel. [1]

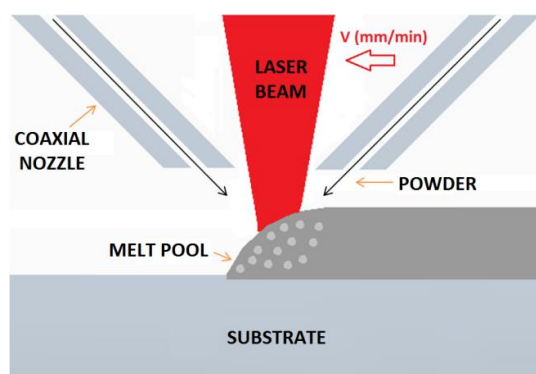
|      | C           | Cr   | Ni   | Mn       | P    | Mo  | Si  | O    | Fe      |
|------|-------------|------|------|----------|------|-----|-----|------|---------|
| wt % | $\leq 0.03$ | 17.5 | 11.5 | $\leq 2$ | 0.02 | 2.3 | 0.4 | 0.05 | balance |

114 A common carbon steel (S355) of thicknesses 8mm was used as a substrate, to study the  
 115 influence of the process parameters in a single laser cladding tracks and to manufacture bigger  
 116 coating with the overlapping of single cladding tracks.

### 117 2.2. Deposition System

118 The experimental equipment used for current study consists on a high power fiber laser, (IPG  
 119 YLS-6000) releasing radiation at a wavelength of 1070  $\mu\text{m}$ , guided to the optical head by means of a  
 120 1000  $\mu\text{m}$  core diameter optical fiber. The optical head is mounted in a 3 linear axis system. This  
 121 optical set up includes a 120 mm collimating lens and a 550 mm focusing lens generating a 5 mm  
 122 diameter spot.

123 A coaxial nozzle was used for experimental tests. Nitrogen was used as a carrier gas to deliver  
 124 the powder from the powder feeder to the melt pool and the same gas was used to generate a  
 125 protective atmosphere during the deposition process. Metallic powder particles were placed in a  
 126 powder feeder from GTV and delivered to the process zone through the nozzle. A schematic layout  
 127 of the laser cladding system is shown in Figure 1.  
 128



129  
 130 Figure 1. Schematic layout of the coaxial laser cladding system.

### 131 2.3. Experimental Procedure

132 In the first part of the experiment the main parameters involved in manufacturing of single  
 133 clads were studied. Range of levels of the main laser process parameters studied in this paper have  
 134 been between 2000-4000W for the laser power, 800-1600 mm/min for the process speed and between  
 135 12-31 g/min of powder flow. Other process parameters were kept constant and are resumed in Table  
 136 2.

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Table 2. Common process parameters.

| Laser Focus<br>Distance (mm) | Carrier gas<br>(l/min) | Shielding<br>gas (l/min) |
|------------------------------|------------------------|--------------------------|
| 550                          | 3                      | 5                        |

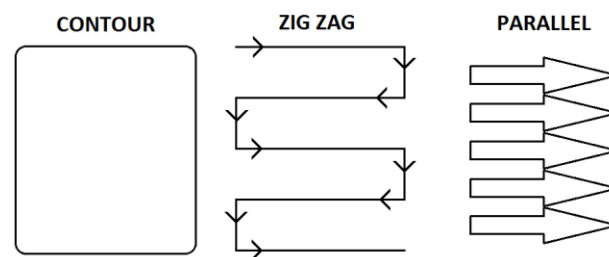
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139 In second place, multiple clad forms were manufactured by performing few overlapped  
 140 cladding tracks, increasing the coating thickness layer by layer. The purpose of building solid forms  
 141 after doing single clads is to develop a stable coating process in longer areas, seeking to minimize  
 142 lack of fusion and to maintain good mechanical properties of the deposited material. To achieve this  
 143 stability, temperature was tried to be kept constant by modulating power input for each new layer  
 144 during the coating process.

145 The value range of the main parameters was derived from the first part and, as well, the  
 146 effective dimensions measured in single clads experiments were taken into account for the  
 147 calculation of the overlapping percentage used (between 25% and 40%). In particular, the width of  
 148 the clad is used for calculate the percentage of overlap between contiguous clads and the height is  
 149 used to calculate the mean increase per each new layer (with multilayered volumes). For the latter,  
 150 the height of a layer was measured with a gauge and this value was added to the distance between  
 151 the nozzle and the workpiece for the next layer. The dimension range of the solid forms created is  
 152 between 20x20 mm<sup>2</sup> and 30x30 mm<sup>2</sup>, and their height depends on the number of layers deposited.

153 Besides the overlapping percentage, the strategy used for generating deposited layers was also  
 154 used as process parameter for solid forms. Figure 2 shows the sketches of the three types of  
 155 strategies applied to deposit the powder flow. The first strategy consists on creating a contour for the  
 156 form, by growing a wall with square form in the first place. The second draws Zig Zag tracks as the  
 157 coaxial nozzle moves back and forth parallel to the same direction all the time [18]. Finally, the third  
 158 one is called Parallel (also known as Raster strategy [14]), the laser clads are deposited in parallel,  
 159 starting and finishing at the same point.

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Figure 2. Sketches of type of strategies applied.

163 Conventional metallography techniques were performed on the cross sections of the laser  
 164 tracks to measure the effective dimensions of clads, polishing cross sections of clads and, then, they  
 165 were etched with V2A etching. Micrographs of cross sections were analyzed using an optical  
 166 microscope and a CMOS camera with 5 Megapixel sensor attach.

### 167 3. Results and Discussion

#### 168 3.1 Individual clad experiments

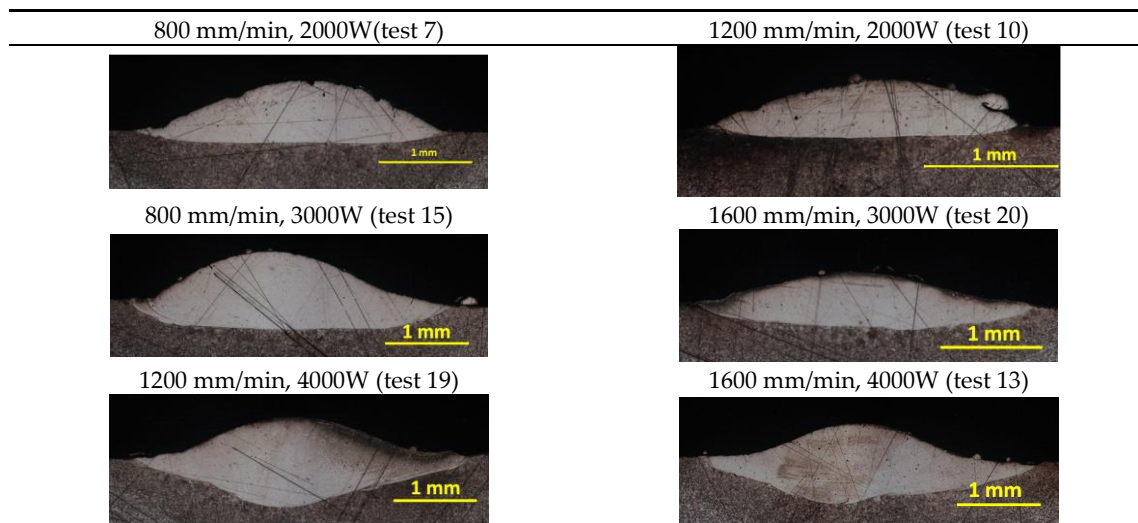
169 Results of the metallographic analysis performed in the clad specimens are represented in  
 170 the following paragraphs according to their corresponding values of powder flow. Results will be  
 171 primarily evaluated according to their resulting dilution values, which should be in the range 10 % -  
 172 20 % according to the literature studied [17,19].

173 Beginning with the case of 12 g/min of powder flow, it can be found from the results shown in  
 174 Table 3 that the mean height of clads was approximately less than 600  $\mu\text{m}$ . Dilution values range  
 175 from 20% up to more than 35 %, as observed in the tests 13, 19 and 20. The increase from the  
 176 objective values (10 % - 20 %) can be understood from the fact that there might not be enough  
 177 concentrated powder for the combination of spot size ( $\varnothing 5$  mm) and the power values used. When  
 178 this happens, a portion of the energy -which is higher than desired- affects the substrate directly  
 179 because there is not enough powder to absorb it. The energy absorption in the substrate leads to a  
 180 net heat increase, causing the surface to melt and therefore increase its absorption coefficient [20]. In  
 181 the end, this results in remelting of the substrate, making the heat affected zone surrounding the clad  
 182 to be bigger than with a more appropriate relationship between powder flow and laser power [21].

183 When checking the results with equal laser power and powder flow values, the height of the  
 184 clads was higher for the minimum value of process speed (800 mm/min) than those made the  
 185 maximum value (1600 mm/min), since going slower helped to deposit a higher amount of energy to  
 186 the LC process. This is observed in the couples of tests for 2000 W, 3000 W and 4000 W shown in  
 187 Table 3.

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Table 3. Experiments with powder flow of 12 g/min.



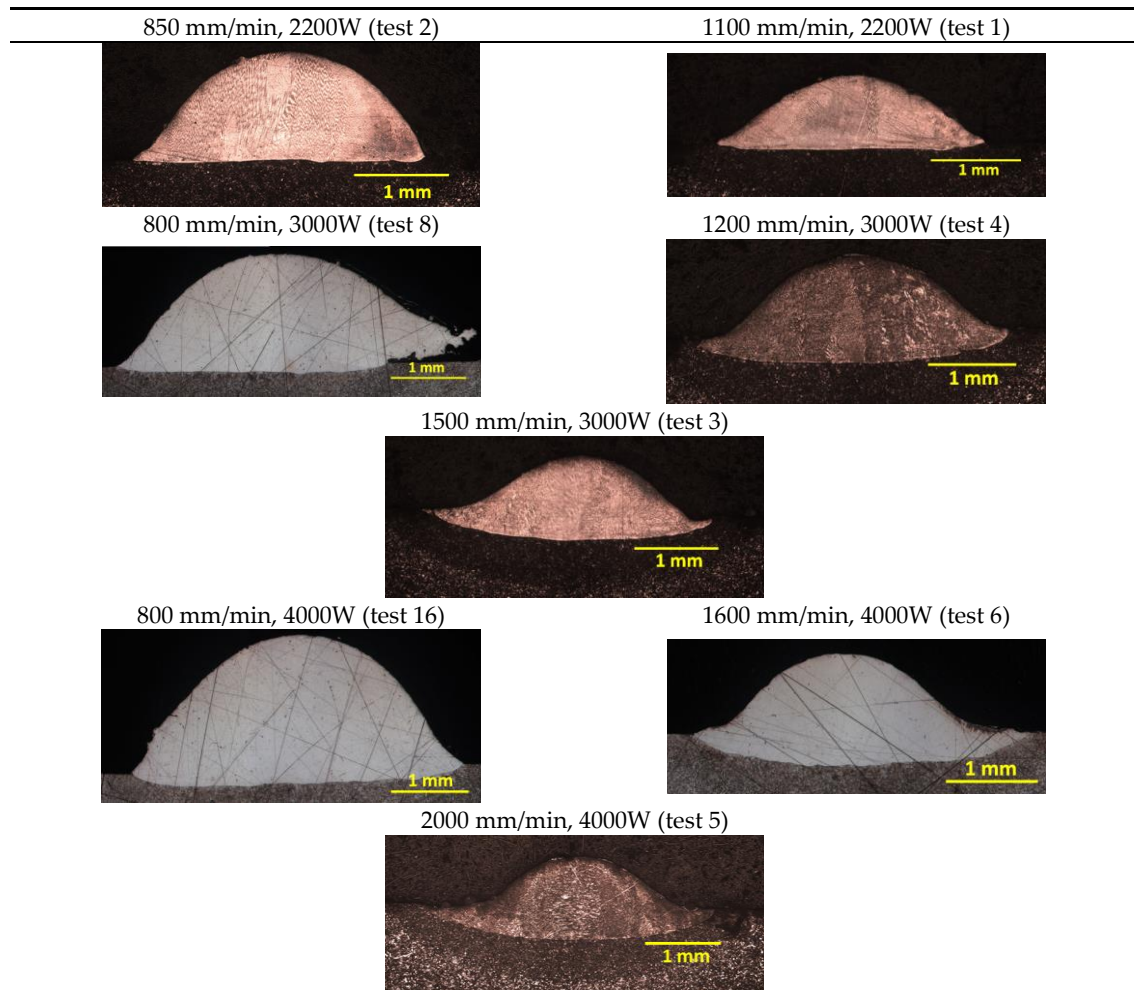
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190 The second value of powder flow to analyze is 25 g/min. In one side, for the tests 2, 4, 8 and 16  
 191 shown in Table 4, the mean percentage of dilution can be measured to be less than 20 % and the  
 192 mean clad dimension found between this tests selection is composed by a height of more than 1 mm  
 193 and a width of more than 3 mm. On the other side, the process speed values used in the tests 1, 3, 5  
 194 and 6 seem to be too fast for the power value used, achieving clads with less than 1 mm in height.  
 195 Therefore, the interaction time found at higher process speeds did not allow the powder to have  
 196 enough time to be completely molten. From this group, the test with the highest values of laser  
 197 power and process speed (test 6) show a dilution ratio higher than 20 % (not desirable).

198 It is important to highlight that test 17 (2000 W, 1200 mm/min and 25 g/min) is not shown  
 199 because there was not metallurgical bonding between laser track and the substrate, so the spallation  
 200 of the clad occurred. In this case, laser value was the lowest in the design and velocity was set at its  
 201 highest value, meaning that energy density (ratio of power to speed) reached a critical low on this  
 202 particular test.

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Table 4. Experiments with powder flow of 25 g/min.



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The last value of powder flow analyzed was 31 g/min and metallographic results are shown in Table 5. It can be observed that only test 9 (highest laser power setting and minimum process speed) was able to achieve more than 1 mm in height and more than 4 mm in width, although the dilution percentage was 30 %.

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Lower widths and heights found in the rest of the tests can be explained from the process speed values chosen for the experimental design, as they were apparently too high and therefore cooling time was left to be too short, meaning was not possible to melt all the delivered powder. Conclusions can be drawn also from the dilution percentage at different laser power value settings: when using 3000 W, dilution reached 20% and with 4000 W the dilution went up to more than 35 %.

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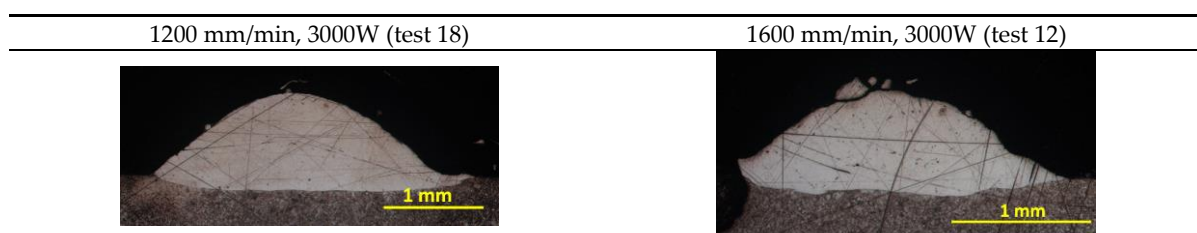
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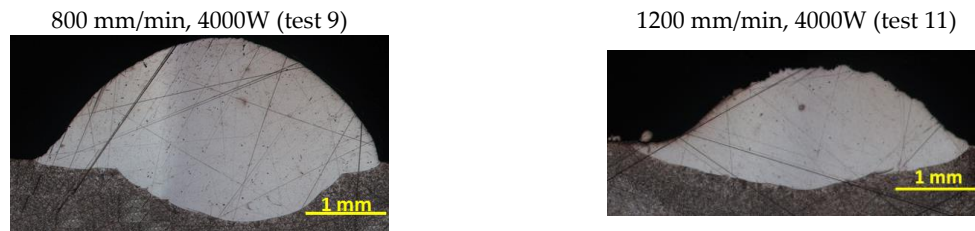
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Test 14 from Table 5 is particular because it had the same fate as test 17 in the previous powder flow level studied, so no results are shown either. This test showed no metallurgical bonding between laser track and the substrate, so again the spallation of the clad occurred. In this case, the energy carried by the laser beam was too low to melt as much powder flow, causing the powder grains to absorb most of the energy. Therefore, the substrate didn't melt and dilution did not take place, leaving an inadequately bonded clad upon the surface of the substrate [20].

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Table 5. Experiments with powder flow of 31 g/min.

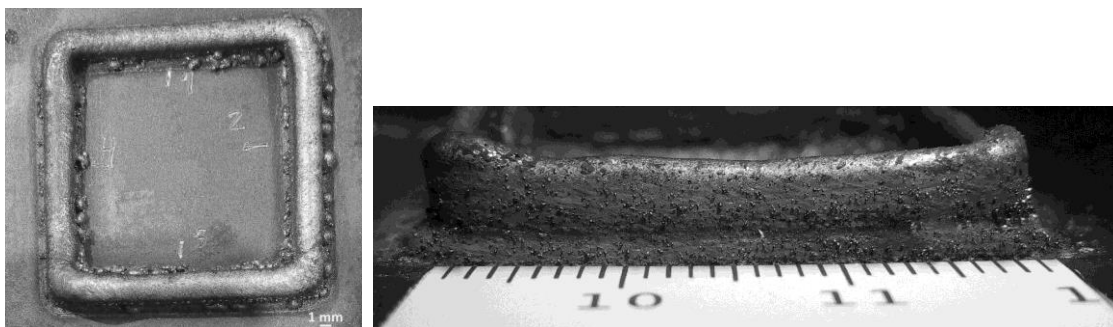




### 221 3.2 Multiple clad experiments (solid forms)

222 From the single clad experiments, it is clear that the intermediate setting of powder flow  
 223 (25 g/min) groups the best results when seeking to maximize the probability of having appropriate  
 224 dilution ratios (between 10 % and 20 %). Within this group, on the other hand, tests with maximum  
 225 height for the power levels studied were found to be those with lower processing speeds. This sets  
 226 tests 2, 8 and 16 as reference process parameters for the fabrication of solid forms. Nonetheless, in  
 227 preliminary tests it was found that power level of test 16 (4000 W) was difficult to control when  
 228 laying multiple clads side by side, taking into account that power regulation was done only after a  
 229 layer was finished. With this in mind, process speed value was set to be 800 mm/min and laser  
 230 power values were switched between 2000 W and 3000 W for the fabrication of multiple clad  
 231 volumes.

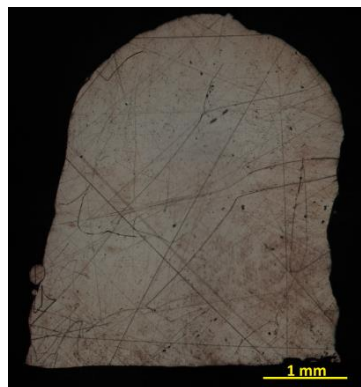
232 The first of the strategies to take into account is the Contour. The general aspect of the contour  
 233 samples are shown in Figure 3, where it can be observed that the cladding showed good aspect in  
 234 geometry. Analyzing the cross section of the wall in Figure 4, it is important to notice that the  
 235 volume is free of porosity and cracking and that there is metallurgical bonding between the  
 236 substrate and the deposited powder.



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**Figure 3.** Picture of the contour, general view and a detail of the wall.



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**Figure 4.** Micrograph of cross sections of a wall.

241 It is worth to mention that, due to the set up employed to manufacture this piece, to maintain a  
 242 homogeneous height (4 mm) in the whole contour was difficult near the turning radius (corners of  
 243 the square), because of the deceleration of the machine axis before reaching this points. This can be

244 observed in Figure 3, where there is light variation in height in the middle of the wall comparing  
245 with the corner.

246 Following with the experiments of solid forms, next experiments applied the Zig Zag and  
247 Parallel strategies. Unlike the single clads, for the study of solid forms is necessary to add the  
248 percentage of the overlap between contiguous clads as an influence parameter. Furthermore, it is  
249 necessary to study the relationship between the percentage of the overlap and the amount of powder  
250 flow to be melted. It is important to seek for an optimal combination of both parameters to guarantee  
251 that there is not lack of fusion in the manufactured part, moreover, to achieve less processing time  
252 for the same area. As indicated earlier, the range of overlap percentage studied in this work was  
253 within 20 % and 40 %.

254 Figure 5 and Figure 6 show two tests that were manufactured following the Zig Zag strategy.  
255 The first one was made with 40 % overlap (one layer), while the second was set to 30 % (two layers).  
256 The main difference that can be seen from both tests is the homogeneity of the surface. In the test 1 is  
257 observed that the delivered powder is piled up, obtaining a rough surface due to the percentage of  
258 overlap. Instead, in test 2 the edges are not very sharp, but the surface is more homogenous than the  
259 former because the percentage of overlap used. Common aspects can be observed in both tests: they  
260 are free of porosity and cracking, and there is good adhesion between the substrate and the  
261 deposited powder.



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**Figure 5.** Macrograph of a cross section following Zig Zag strategy.



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**Figure 6.** Macrograph of a cross section following Zig Zag strategy.

266 Finally, the volumes made with the Parallel strategy can be observed in Figure 7 and Figure 8.  
267 Only one layer is deposited in the case of Figure 7, allowing the study of the overlap between  
268 contiguous cladding tracks. In the results shown, good adhesion between the substrate is observed  
269 and no porosity or cracks appear.



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**Figure 7.** Macrograph of a cross section following parallel strategy.

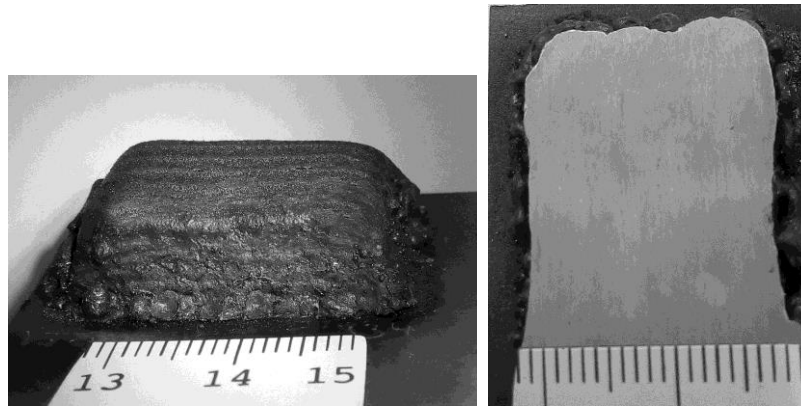
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### 3.3 Parameter reduction and building of a solid form



273 In order to test one of the strategies used in the previous section, a multiple clad, multi-layered  
274 volume was made with the process settings chosen between Parallel and Zig-Zag strategies. The  
275 Parallel strategy was found to have a more continuous height and, therefore, a more homogeneous  
276 surface and, thus, was used as an input for the solid piece proposed.

277 The multiple clad volume generated was built layer by layer in a Parallel strategy with a total of  
278 10 layers, and the result obtained is presented in the Figure 8. In this case, the surface was machined  
279 to analyze if porosity, crack or lack of material appears and the aspect was found to be similar to a  
280 bulk material (no porosity was found). Total height was measured to be around 8 mm. It is  
281 important to notice that, even though a single clad grows more than 1 mm under the process  
282 parameters chosen, continuous, multiple clads will have a different net height because of the  
283 overlapping final geometries [22].



284

285

**Figure 8.** Pictures of general views of a solid piece and its machined surface.

286 Following the steps in this work, process parameters chosen to be suitable for cladding 316L  
287 stainless steel in solid forms with the coaxial nozzle used are:

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- Power: 1700 W - 2500 W
- Velocity: 800 mm/min
- Powder Flow: 25 g/min
- Overlap percentage: 30 %
- Height increase per Layer: 0.8 mm

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#### 293 4. Conclusions

294 This paper seeks to analyze the influence of the main laser process parameters (laser power,  
295 process speed and powder flow) in the geometry of a single laser clads in order to subsequently  
296 achieve optimal process parameters to build, layer by layer, a solid piece as a first approach. The  
297 main conclusions derived from this study are the following:

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- A good metallurgical bond was achieved, thanks to the good relation between 25 g/min of powder flow and 30 % of overlap, which guaranteed a dilution value of less than 20 % in less time of process and, in the case of solid forms, without lack of fusions. In addition, it is necessary the power modulation for coating process (from 3000 W to 1700 W), in order to have the control of the process, avoiding overheating of the piece.
- Regarding the strategies studies, it can be seen that the Zig Zag strategy is better to increase the height of the deposited layer faster; and Parallel strategy is used to achieve a good finishing surface.
- Value range of the process parameters studied get valid restrictions, allowing extrapolating results more easily from simpler to more complex experiments.

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310

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312 experiments in performed the experiments, analyzing the micrographs, discussion the results, and  
313 she has been in charge of write the article.

314 MA. Montealegre. Main contribution in conceived and designed the experiments, discussing the  
315 results and supervising the paper

316 J.F. Pulido. Main contribution in performing the tests, and discussing the results.

317

318 **Conflicts of Interest:** Declare conflicts of interest or state "The authors declare no conflict of  
319 interest."

## 320 References

- 321 1. Michał Ziętała, Tomasz Durejko, Marek Polański, et al. The microstructure, mechanical properties and  
322 corrosion resistance of 316L stainless steel fabricated using laser engineered net shaping. *Materials*  
323 *Science&Engineering A*, **2016**, 677, 1-10.
- 324 2. Y. T. Pei and J. Th. M. de Hosson. Functionally graded materials produced by Laser cladding. *Acta*  
325 *Materialia*. **2000**, 48, 2617-2624.
- 326 3. Andrew J. Pinkerton. Lasers in additive manufacturing. *Optics & Laser Technology* 78, **2016**, 25-32.
- 327 4. G. Castro, J. Rodriguez, MA. Montealegre, et al. Laser additive manufacturing of high added value pieces.  
328 *Procedia Engineering* **2015**, 132, 102-109.
- 329 5. MANUFUTURE Platform. Strategic Research Agenda: Assuring the future of a competitive and  
330 sustainable manufacturing in Europe. **2006** [www.manufuture.org](http://www.manufuture.org).
- 331 6. Cortina, M.; Arrizubieta, J.I.; Calleja, A.; Ukar, E.; Alberdi, A. Design and Manufacturing of Conformal  
332 Cooling for Hot Stamping Dies Using Hybrid Process of Laser Metal Deposition (LMD) and Milling.  
333 *Preprints* **2017**, 2017120076 (doi: 10.20944/preprints201712.0076.v1).
- 334 7. Bernhard Langefeld. Additive Manufacturing. A game changer for the manufacturing industry? Roland  
335 Berger Strategy consultants. Munich, **2013**.
- 336 8. MA Montealegre, F. Vidal, S. Mann et al. Adaptive laser cladding system with variable spot sizes. **2013**  
337 *International Congress on Applications of Lasers and Electro-Optics. Congress proceedings. ICALEO*.  
338 ISBN 978-0-912035-98-7 pp 950-954.
- 339 9. Fritz Klocke; Christian Brecher; Daniel Heinen; Chris-Jörg Rosen; Tobias Breitbach. Flexible scanner-based  
340 laser surface treatment. *Physics Procedia*. **2010**, 5, 467-475.
- 341 10. Joonas Pekkarinen. Laser cladding with scanning optics Theses for the degree of Doctor of Science.  
342 University of Technology, Lappeenranta. **2014**.
- 343 11. Yanmin Li; Haiou Yang, Xin Lin; Weidong Huang; Jianguo Li; Yaohe Zhou; The influences of processing  
344 parameters on forming characterizations during laser rapid forming. **2003**, 360, Issues 1-2, 18-25.
- 345 12. A Calleja, I Tabernero, A Fernández, A Celaya, A Lamikiz. Improvement of strategies and parameters for  
346 multi-axis laser cladding operations. *Optics and Lasers in Engineering*. **2014**, 56, 113-120.  
347 (doi.org/10.1016/j.optlaseng.2013.12.017)
- 348 13. Dara Moazami Goodarzi; Joonas Pekkarinen; Antti Salminen. Effect of process parameters in laser  
349 cladding on substrate melted areas and the substrate melted shape. *Journal of Laser Applications*, **2015**, 27,  
350 S29201.
- 351 14. Donghong Ding. Process planning for robotic wire ARC additive manufacturing. University of  
352 Wollongong. **2015**.
- 353 15. Zhuqing Wang, Todd A. Palmer, Allison M. Beese. Effect of processing parameters on microstructure and  
354 tensile properties of austenitic stainless steel 304L made by directed energy deposition additive  
355 manufacturing. *Acta Materialia*, **2016**, 110, 226-235.

- 356 16. U. de Oliveira; V. Ocelík; J. Th. M. De Hosson. Analysis of coaxial laser cladding processing conditions.  
357 *Surface & Coatings Technology* **2005**, 197, 127-136.
- 358 17. Swathi Routhu; Masters Theses. 2-D path planning for direct laser deposition process. 2010.
- 359 18. S. Zanzarin, S. Bengtsson, A. Molinari. Study of dilution in laser cladding of a carbon steel substrate with  
360 Co alloy powders. *Powder Metallurgy*. **2016**, 59 -Issue 1, 85-94.
- 361 19. B. Song, T. Hussain, K.T. Voisey. Laser cladding of Ni50Cr: A parametric and dilution study. *Physics*  
362 *Procedia* **2016**, 83, 706-715.
- 363 20. Markus Wolf, O. R. Laser technologie, Dieburg. Improving the Efficiency of the DMLD Process. How  
364 particle size and laser spot size influence process quality and efficiency. [www.laser-journal.de](http://www.laser-journal.de). 2016.
- 365 21. Sörn Ocylok; Eugen Alexeev; Stefan Mann; Andreas Weisheit; Konrad Wissenbach; Ingomar Kelbassa.  
366 Correlations of melt pool geometry and process parameters during laser metal deposition by coaxial  
367 process monitoring. *Physics Procedia* **2014**, 56, 228-238.
- 368 22. Ondrej Nenadl; Václav Ocelík; Armin Palavra; J. Th. M. De Hosson. The prediction of coating geometry  
369 from main processing parameters in laser cladding. *Physics Procedia* **2014**, 56, 220–227.