

1 Article

2 Efficiency of True-Green Light Emitting Diodes: 3 Non-Uniformity and Temperature Effects

4 Ilya E. Titkov^{1,3,*}, Sergey Yu. Karpov², Amit Yadav³, Denis Mamedov¹, Vera L. Zerova^{3,4}
5 and Edik Rafailov³

6 ¹ Ostendo Technologies Inc., 6185 Paseo del Norte, Carlsbad, CA 92011, USA; ilya.titkov@ostendo.com

7 ² STR Group – Soft-Impact, Ltd., P.O.Box 83, 27 Engels ave., St.Petersburg, 194156, Russia;

8 sergey.karpov@str-soft.com

9 ³ Optoelectronics and Biomedical Photonics Group, Aston Institute of Photonic Technologies,

10 Aston University, Birmingham, B4 7ET, United Kingdom; a.yadav1@aston.ac.uk

11 ⁴ Nanoscale Physics Research Laboratory, School of Physics and Astronomy, University of Birmingham,

12 Birmingham, B15 2TT, United Kingdom; vzerova@gmail.com

13 * Correspondence: ilya.titkov@ostendo.com; Tel.: +1-760-710-3042

14 **Abstract:** External quantum efficiency of industrial-grade green InGaN light-emitting diodes
15 (LEDs) has been measured in a wide range of operating currents at various temperatures from 13 K
16 to 300 K. Unlike blue LEDs, the efficiency as a function of current is found to have a multi-peak
17 character, which could not be fitted by a simple ABC-model. This observation correlated with split-
18 ting of LED emission spectra into two peaks at certain currents. The characterization data are inter-
19 preted in terms of non-uniformity of the LED active region, which is tentatively attributed to ex-
20 tended defects like V-pits. We suggest a new approach to evaluation of temperature-dependent
21 light extraction and internal quantum efficiencies taking into account the active region non-
22 uniformity. As a result, the temperature dependence of light extraction and internal quantum effi-
23 ciencies have been evaluated in the temperature range mentioned above and compared with those
24 of blue LEDs.

25 **Keywords:** InGaN green LEDs; active region non-uniformity; temperature-dependent electrolumi-
26 nescence; internal quantum efficiency; light extraction efficiency; extended defects; modeling.

27

28 1. Introduction

29 Semiconductor LEDs invented almost 100 years ago [1] have now become key components in
30 numerous applications: solid-state lighting, traffic lights, brake lights, various indicators, and signs
31 [2-3]. For many cases and, especially, for phosphor-free solid state lighting, high-efficiency green
32 LEDs are of primary importance. Nowadays, a new level of wall-plug-efficiency (WPE) of ~35% is
33 achieved from mass produced commercial green LEDs. This progress is a result of both bandgap and
34 strain engineering of the active multiple-quantum-well (MQW) regions, and optimized carrier
35 transport across the LED structures and using advanced chip designs.

36 Nevertheless, the EQE of green LEDs is still remarkably lower than that of blue LEDs, which is
37 now over 70% at the emission wavelengths of 440-450 nm. This fact is a manifestation of the so-
38 called "green gap" problem, i.e. substantial efficiency reduction of InGaN-based LEDs from blue to-
39 wards green/orange spectral range. Among the mechanisms that may be responsible for the "green
40 gap", the ones that are frequently referred are: degradation of InGaN crystal quality at high indium
41 content; polarization field built in the InGaN quantum wells (QWs) and carrier localization due to
42 composition fluctuations in InGaN (see more detailed discussion given in Sec.4.2). To better under-
43 stand the role of each factor plays in this problem, examination of temperature-dependent efficiency
44 is quite constructive.

45 This paper is a natural continuation of our previous study [4] in which temperature dependence
46 of EQE, internal quantum efficiency (IQE) and light extraction efficiency (LEE) of an industrial-grade

47 blue LED emitting at 440-450 nm has been determined in a wide range of temperatures from 13 to
48 440 K. The study [4] was largely based on approximating the dome-like dependence of EQE on cur-
49 rent/output optical power within a simple ABC recombination model. The subject of the present
50 study is a green LED emitting in the range of 530-550 nm (so-called “true-green”). Recently, this
51 method with minor modification has been extended to single-quantum well (SQW) LEDs operating
52 at longer emission wavelengths [5]. However, EQEs of the green-emitting multiple quantum wells
53 reported in [6-7] for a wide temperature range of 4-300 K did not exhibit a dome-like dependence on
54 current, demonstrating more complex behavior that could not be fitted by a simple ABC-model. So,
55 more elaborate approach is required for experimental estimation of IQE and LEE from those or simi-
56 lar characterization data. Development of such an approach to evaluation of temperature-dependent
57 efficiency of true-green LEDs is one of the goals of our study.

58 Generally, blue LEDs exhibit a nearly evenly symmetrical dome-like EQE dependence on cur-
59 rent/output power with respect to its value corresponding to the EQE maximum. In contrast, the
60 EQE dependence of green LEDs frequently becomes asymmetric with a more extended low-current
61 wing [8]. To explain that behavior, various mechanisms have been invoked: electron leakage into p-
62 side of the LED structure [9]; delocalization of carriers captured by InGaN composition fluctuations
63 [10]; imbalance between the electron and hole injection into InGaN QWs caused by non-equilibrium
64 QW population [11] and suppression of non-radiative recombination at threading dislocations via
65 carrier localization by composition fluctuations in InGaN alloys [12]. In addition, contribution of de-
66 vice self-heating at high current was not reliably excluded in many experimental studies. Therefore,
67 understanding the nature of the particular efficiency behavior of green LEDs is another goal of this
68 study.

69 Our data obtained with commercial true-green LEDs demonstrate non-ordinary variation of the
70 emission wavelength with current and considerable deviation of EQE from a simple ABC-model.
71 The low temperature measurements enable interpretation of the data in terms of the active region
72 non-uniformity. The ABC-model modified for non-uniform QWs allow us to estimate the tempera-
73 ture-dependent LEE.

74 2. Experimental

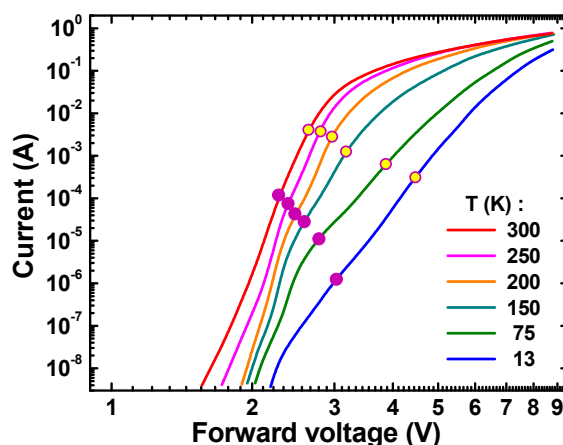
75 2.1. Samples and characterization techniques

76 The LED samples and characterization techniques utilized herein were quite similar to those
77 used in our previous research [4]. The main difference was a higher, more than 20%, indium content
78 in InGaN QWs, providing green light emission. The LED structures were grown by metal-organic
79 chemical vapor deposition on C-plane sapphire and consisted (from bottom to top) of an undoped
80 GaN layer followed by a Si-doped n-GaN contact layer, undoped InGaN/GaN MQW active region, a
81 p-AlGaN electron blocking layer, and Mg-doped p-GaN contact layer. The structures were pro-
82 cessed with Osram’s thin-film LED technology with highly reflective metallic electrodes on the p-
83 contact layer. The chip size was 1.1×1.1 mm². After removing the sapphire substrate the back (emit-
84 ting) surface of the n-GaN contact layer was textured to increase LEE. The chips were mounted onto
85 the Dragon packages without any molding, which is suitable for temperature-dependent electrolu-
86 minescence measurements.

87 A Labsphere spectrometer CDS-600, Keithley 2400 source-meter and Janis CCS-450 helium
88 closed-cycle cryostat were used for temperature- and current-dependent electroluminescence (T-I
89 DEL) technique. In order to determine EQE, we measured electroluminescence (EL) over a wide
90 range of operating current, from 20 nA to 800 mA. To cover the corresponding range of the output
91 optical power, from 1 nW to 600 mW, we varied spectrometer exposure time from 1 ms to 5 s and
92 used neutral ND1-4 filters. For calibration of EQE values, EL was measured both in an integrating
93 sphere and in the cryostat at room temperature (RT). Then optical alignment was not changed at
94 lower temperatures. The current-voltage (I-V) characteristics were measured with Keithley 4200
95 semiconductor characterization system that provided extended current range.

96 2.2. Current-voltage characteristics

97 Current-voltage characteristics (I-V curves) of the green LED measured at various temperatures
 98 (13-300 K) are shown in Figure 1. As expected, every high temperature curve can be seen in two
 99 parts: the first one below ~ 3 Volts has a higher slope and can be associated with the carrier injection
 100 into the active region; the second one, above ~ 3 V, is controlled by the series resistance. No defect-
 101 mediated shoulders can be distinguished in the high-temperature I-V curves in contrast to those ob-
 102 served for blue LED [4]. However, one can see some traces of such shoulders in the low-temperature
 103 curves, looking like waving at the lower voltages (see also discussion on this issue in Sec.4.3).



104
 105 **Figure 1.** Current-voltage characteristics of the green LED measured at various temperatures. Closed
 106 and open circles indicate points at which shorter-wavelength and longer-wavelength emission ap-
 107 proaches their maximum efficiency (see Figure 3 and text for more details).

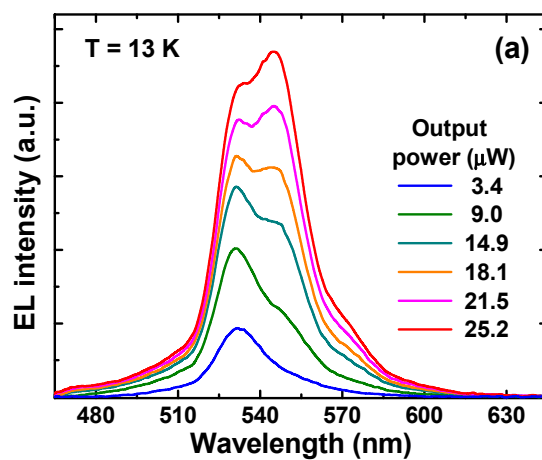
108 Fitting of the I-V curves with Shockley's diode equation accounting for the LED series resistance
 109 has shown that in the temperature range of 200-300 K the curves can be characterized by the ideality
 110 factor of ~ 3 -5 and series resistance of ~ 7.3 -7.6 Ω . For comparison, the ideality factors typical for blue
 111 LEDs [4] were about 1.7 with series resistance ~ 6.3 -6.5 Ω . The relatively high ideality factor may be
 112 the evidence for impeded carrier transport over the barriers between the QWs [13], which correlates
 113 with much deeper InGaN QWs in green LEDs compared to blue ones, or high asymmetry in the don-
 114 or and acceptor concentrations in the LED p-n junction [14].

115 2.3. Emission spectra

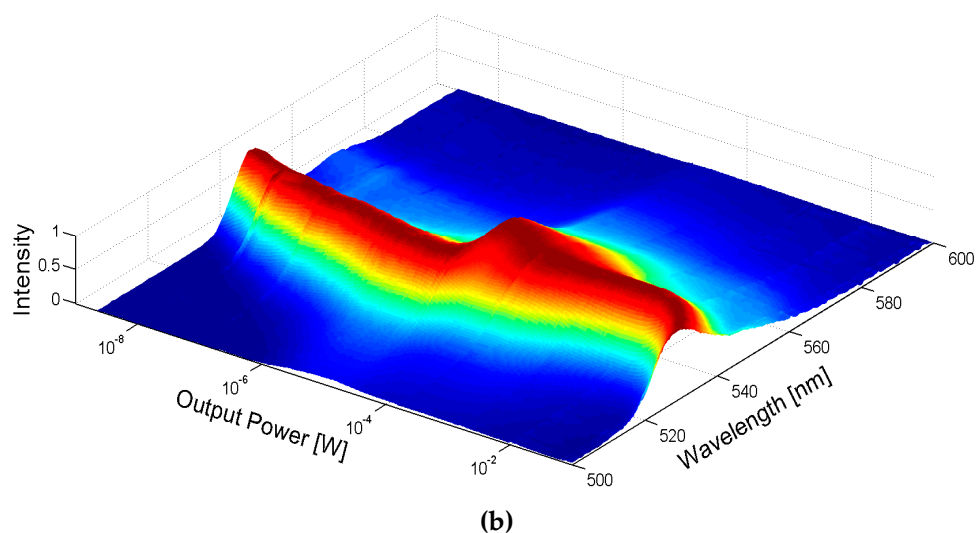
116 The EL spectra of the green LED were monitored over a wide range of currents and tempera-
 117 tures. We found the spectra to consist of two peaks (see Figure 2a). One of them centered at about
 118 535 nm and referred to hereafter as shorter-wavelength (sw) emission, dominant at low output pow-
 119 er. Another one centered at ~ 545 nm and referred to hereafter as longer-wavelength (lw) emission,
 120 became stronger at higher output power. The peaks competed with each other in a narrow range of
 121 power variation. The third low-intensity peak observed at ~ 575 -580 nm (Figure 2a) was attributed to
 122 LO phonon replica. The two-peak structure of the emission spectra was clearly seen at temperatures
 123 below 200 K and became blurred toward 300 K.

124 In order to understand better the behavior of EL spectra, we examined their normalized intensi-
 125 ty *vs.* output power. For this purpose, 3D contour plots were generated using the Matlab program-
 126 ming framework. We processed and stored the measured EL intensity I as a function of wavelength
 127 λ and corresponding emission power P for each temperature. Spectra obtained in such a way were
 128

129 then normalized to a maximum intensity at every value of P . Then we created continuous normal-
 130 ized functions $I(P, \lambda)$ using linear interpolation of the scattered data. The functions $I(P, \lambda)$ obtained
 131 for cryogenic (13 K) and room (300 K) temperatures are plotted in Figure 2b,c using a log-spaced vari-
 132 able resolution along the P axis.

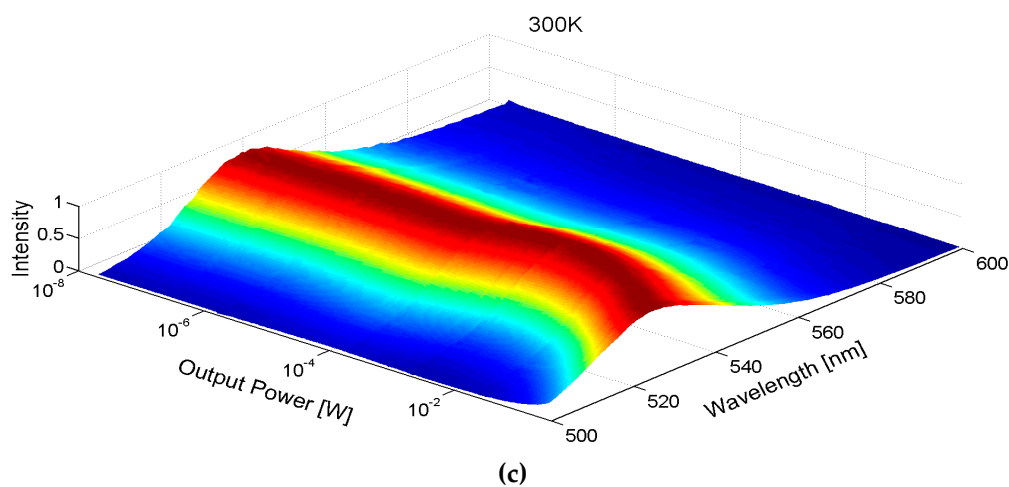


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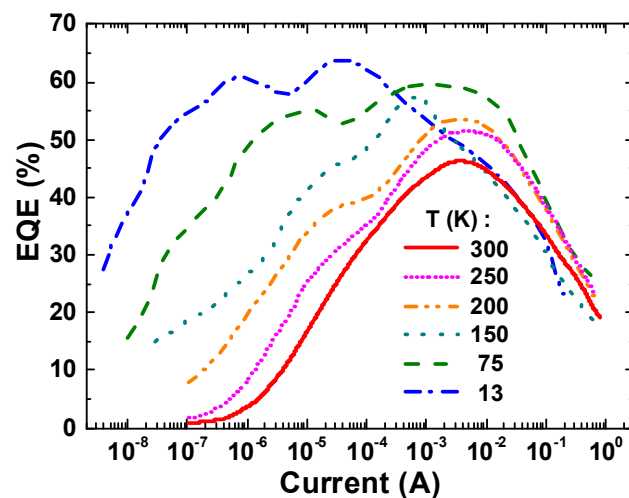
138 **Figure 2.** Emission spectra of the green LED at 13 K and various output power (a) and evolution of
 139 the normalized emission spectra vs. output power at 13 K (b) and 300 K (c).

140 Figure 2b demonstrates that transition from sw- to lw-emission peaks occurs in a very narrow
 141 range of optical power (transition region) from $\sim 6 \times 10^{-6}$ to $\sim 5 \times 10^{-5}$ W. Beyond this range both peaks
 142 exhibit blue shifts with growing P and overlap partly in the transition region. Similar evolution of
 143 the normalized emission spectra is observed at the temperatures from 13 K to 200 K. At temperatures
 144 over 200 K, however, the sw- and lw- peaks are almost indistinguishable in the normalized spectra,
 145 Figure 2c. Nevertheless, the transition region is represented via bowing of the EL peak towards
 146 longer wavelengths at $P \sim 10^{-3}$ - 10^{-2} W.

147 Summarizing the emission spectra behavior, one can see that the transition from the sw- to the
 148 lw-emission peak occurs in a narrow range of output power and has a “switching” character. Out-
 149 side the narrow transition region, single sw- or lw-emission peaks dominate at low and high cur-
 150 rents, respectively.

151 2.4. Emission efficiency

152 The output power (P) and centroid wavelength measured at various temperatures versus oper-
 153 ating current were converted to EQE and presented in Figure 3. The same EQE data plotted *vs.* P
 154 are presented separately for every single temperature at Figures 4 and 5. Looking at the low temperature
 155 EQE curves for 13 K and 75 K, one can see a distinct two-peak shape with local maxima located at
 156 notable different values of the current/output power. At higher temperatures above 75 K, the low-
 157 current peaks gradually transform to the shoulders and merge with high-current peaks. Unlike these
 158 two main peaks, the third one becomes apparent at low temperatures and low current only, $\sim 0.1 \mu\text{A}$.
 159 This behavior differs from one observed previously with blue LED which was an evenly symmet-
 160 rical dome-like function $\text{EQE}(P)$ on a log-scale.



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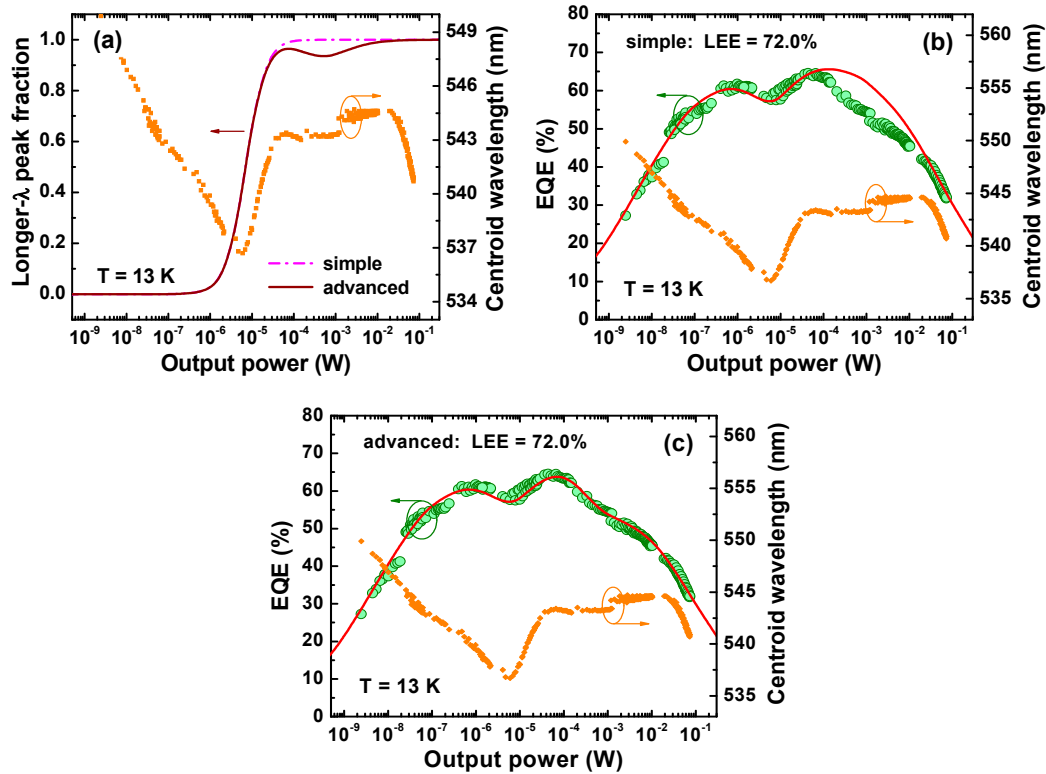
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Figure 3. EQEs of green LED versus driving current measured at various temperatures.

164 It is also important to note that the local minimum of $\text{EQE}(P)$ is placed between two local max-
 165 ima at every temperature and correlated well with transient region between two spectral peaks (see
 166 Fig.2) as well as with the red shift of centroid wavelength (Figures 4-5). This observation allows us to
 167 attribute the low-current EQE peak to the sw-emission and high-current one to the lw-emission, re-
 168 spectively.

169 Since the measured two-peak EQE dependencies shown in Figures 4 and 5 could not be approx-
 170 imated with a simple ABC-model, a new approach was developed for evaluation of IQE and LEE of
 171 the green LED.



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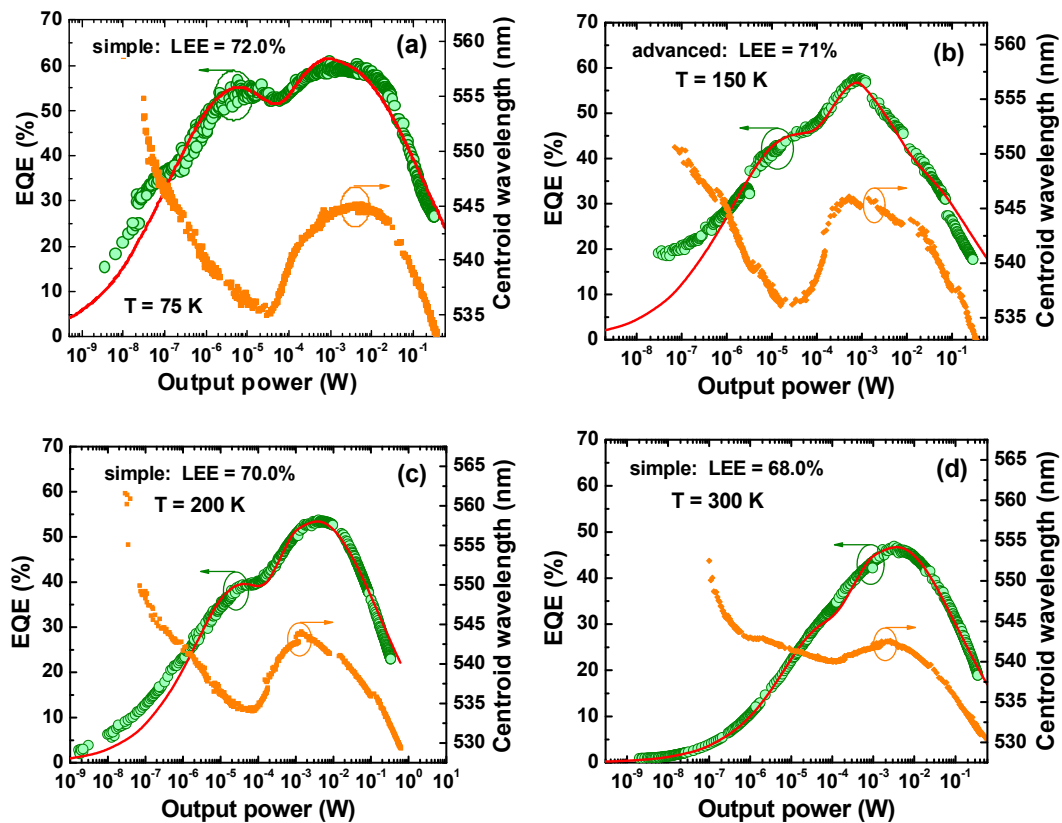
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Figure 4. Simple and advanced transition functions used in the simulations (a), EQE of the green LED at $T = 13$ K predicted by modified ABC-model (solid lines) obtained with simple (b) and advanced (c) transition functions, and emission wavelength as a function of output optical power. Symbols are experimental data points.



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Figure 5. EQE of the green LED and emission wavelength as a function of output power measured at various temperatures (symbols) and predicted by modified ABC-model (solid lines).

182 3. Modeling

183 3.1. Model

184 The experimental multi-peak structure of the EQE dependence on current/output power shown
 185 in Figure 4b,c does not allow fitting by a simple ABC-model, which predicts the efficiency to peak at
 186 a certain single value of the output power [15]. However, the model can be modified assuming the
 187 properties of InGaN QWs in the LED active region (AR) to be non-uniform. The non-uniformity may
 188 be attributed to either InGaN composition or the QW width. From the efficiency point of view, vari-
 189 ous types of non-uniformity are possible. One is the vertical non-uniformity corresponding, for in-
 190 stance, to different InGaN compositions in top and bottom QWs. In this case, the overall IQE of the
 191 MQW active region η_i can be regarded in the same manner as used for multi-color LEDs in [16]
 192 where

$$193 \eta_i = \left(\sum_{k=1}^N f_k / \eta_i^{(k)} \right)^{-1}, \quad (1)$$

194 f_k is the fraction of photons emitted by the k -th QW in the total LED emission spectrum and $\eta_i^{(k)}$ is
 195 IQE of the k -th well; summation in (1) is performed over all N QWs.

196 Another type of non-uniformity is the lateral/in-plane inhomogeneity. Here, different in-plane
 197 areas of the AR are assumed to emit light at different wavelengths and to have different IQEs, re-
 198 spectively. Assuming current density to be laterally uniform, one can come again to Eq.(1) connect-
 199 ing η_i with partial efficiencies $\eta_i^{(k)}$ and fractions of emitted photons f_k associated with the above in-
 200 plane areas. For brevity, the QWs in the case of vertical non-uniformity and in-plane areas in the case
 201 of lateral non-uniformity will be referred to hereafter as different active regions, forming altogether
 202 the whole AR of LED structure.

203 Examination of the LED emission spectra (see Sec.2.2) has revealed the existence of at least two
 204 ARs that emit at different wavelengths. As the LED operating current is increased, the sw-emission
 205 is nearly abruptly switched to the lw-emission (see Figure 2b,c). On the other hand, two efficiency
 206 peaks can be clearly seen in practically all the EQE plots vs. output optical power P shown in Figure
 207 4b,c and Figure 5. It was already mentioned above, that it is reasonable to attribute the two-peak
 208 character of EQE(P) curves to the existence of two ARs emitting at different wavelengths.

209 As it has been suggested earlier for dual-wavelength LEDs [16], IQE of every particular AR,
 210 $\eta_i^{(k)}$, can be approximated by the analytical expression

$$211 \eta_i^{(k)} = \frac{Q_k}{Q_k + (P_k / P_m^{(k)})^{1/2} + (P_k / P_m^{(k)})^{-1/2}} ; \quad P_k = f_k P, \quad (2)$$

212 where Q_k is the quality factor [15] of the k -th AR, $P_m^{(k)}$ is the optical power corresponding to the IQE
 213 maximum of the k -th AR, and P_k is the optical power produced by the k -th AR at a chosen operating
 214 current. Within the ABC-model, Q_k and $P_m^{(k)}$ are proportional to certain combinations of the recom-
 215 bination constants: A corresponding to Shockley-Read-Hall non-radiative recombination, B related
 216 to radiative recombination, and C associated with Auger recombination. So, if all f_k are known, the
 217 total IQE of LED structure η_i can be calculated using Eqs.(1)-(2) and then parameters Q_k and $P_m^{(k)}$ can
 218 be fitted to experimental data.

219 Considering just two different ARs to coexist in the whole active region of green LEDs, one can
 220 regard the only fraction f_{lw} corresponding to photons produced by AR with lw-emission; another
 221 fraction attributed to sw-emission is: $f_{sw} = 1 - f_{lw}$. As one can see from Figure 2b, switching between
 222 two spectral peaks occurs in a rather narrow range of the output power. In most of cases the switch-
 223 ing could be well approximated by a simple "transition function"

$$f_{hw} = [1 + (P_i / P)^\gamma]^{-1}, \quad (3)$$

where P_i is the critical output power corresponding to the transition between the efficiency peaks (transition power) and the parameter γ is related to the width of the transition region identified on the log-scale of the output power. We have found $\gamma = 2$ to fit well the transition from shorter to longer emission wavelength at temperatures from 13 to 250 K. At 300 K, the value $\gamma \sim 1.5-1.7$ seems to be more suitable. Nevertheless, we used the parameter $\gamma = 2$ for 300 K as well in order to unify approximations of the spectral transition at various temperatures.

The approximation based on Eq.(3) will be called hereafter as "simple". In some cases, non-monotonous transition functions should be applied to provide more accurate fitting of the characterization data. The latter type of the transition function will be referred to as "advanced".

3.2. Model application

Assuming the light extraction efficiency (LEE) of LED to be the same for the close spectral peaks, we have fitted EQE(P) curves with Eqs.(1)-(3). For this purpose, parameters Q_k , $P_m^{(k)}$ ($k = sw, lw$), and P_i were first fitted to provide the best correlation with the multiple-peak dependence of LED efficiency on the output power. Then IQE was calculated using Eq.(1). Finally, LEE was estimated as a ratio of experimental EQE and calculated IQE.

The dash-dotted line in Figure 4a shows the simple switching function used for fitting of the EQE(P) measured at $T = 13$ K. The transition power P_i , being the only adjustable parameter of the transition function, was determined from the observed "switching" in the LED emission spectra from sw- to lw-emission shown in Fig.2b. The use of the simple switching function with different Q_k and $P_m^{(k)}$ corresponding to sw ($k = sw$) and lw ($k = lw$) peaks provided the theoretical curve shown by solid line in Figure 4b at LEE = 72%. The curve fits well the general shape of the EQE(P) dependence but fails in reproducing some of its details at output power between 10^{-4} and 10^{-2} W. On the other hand, the spectral evolution of the emission spectra shown in Fig.2b is more complex than it is predicted by the simple transition function. Therefore, an advanced transition function (solid line in Figure 4a) has been applied. This enabled much better fitting of the experimental results, as one can see from Figure 4c.

Figure 5 presents results of fitting EQE(P) measured at other temperatures. Except for that of 150 K, simple switching functions (4) were sufficient to get a reasonable fitting of the experimental points. As one can see from the figure, noticeable discrepancy between the fitting and the data is still observed at low power and $T = 75-200$ K. This may be the evidence for additional AR non-uniformity, manifesting itself just at extremely low currents. However, we will ignore this possibility in the further discussion, as no evidence for existence of one more distinct spectral peak was found.

4. Discussion

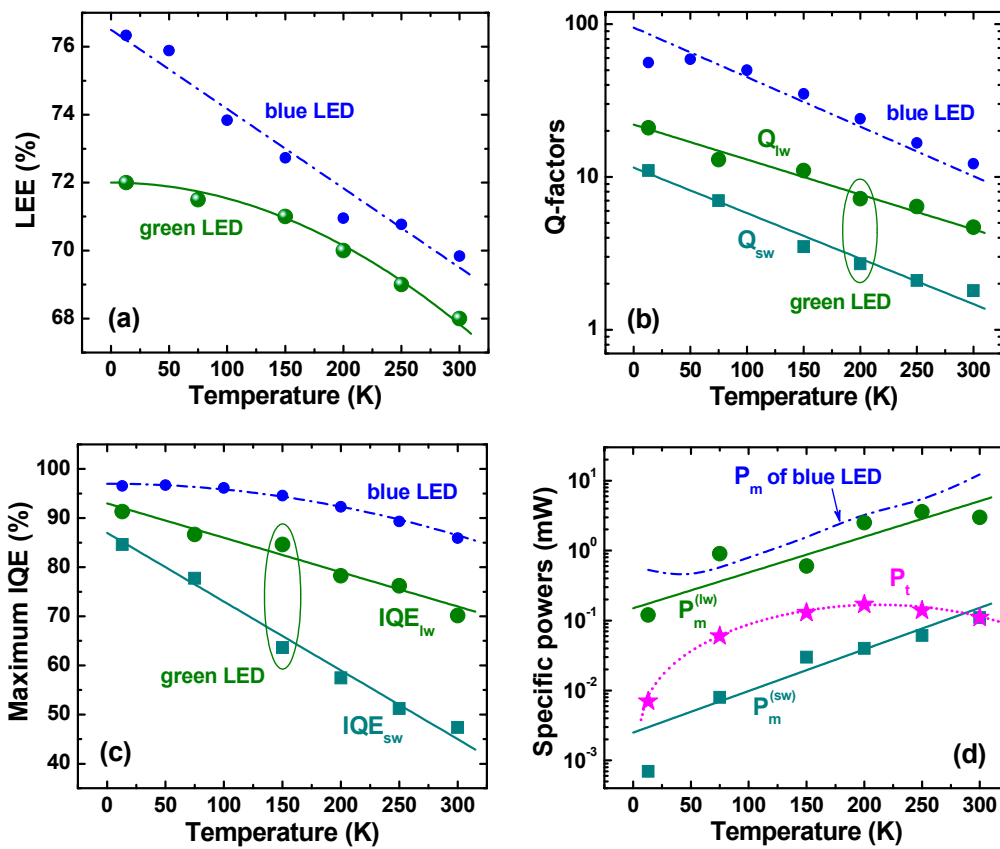
This section discusses general trends in variation of important LED characteristics and parameters with temperature. In order to interpret the trends, we will compare the results obtained for true-green LED with those reported earlier for blue LED [4].

4.1. Light extraction efficiency

Figure 6a compares the temperature-dependent efficiencies of light extraction from the LED dice to air obtained for green and blue LEDs. As one can see from Figure 6, LEE of the green LED is lower than that of the blue one at all the temperatures. This finding disagrees with the increase in RT LEE from blue to green spectral range observed on SQW-based LEDs [17]. The LEE increasing with wavelength was attributed in [17] to temperature dependent optical properties of silver-based p -electrode providing dominant optical losses of the emitted light via its incomplete reflection from the electrode. On the other hand, contribution of internal optical losses in the LED die was also found to

269 be considerable. Nevertheless, in LED structures of similar designs, the optical losses have been
 270 shown not to change the general trend of LEE to rise with the emission wavelength [17].

271



272

273 **Figure 6.** Maximum IQE, light extraction efficiency, quality factors corresponding to lower-
 274 wavelength (Q_{lw}) and shorter-wavelength (Q_{sw}) spectral peaks, and specific optical power (see text
 275 for more detail) as a function of temperature (big symbols). Dash-dotted lines and small circles show
 276 for comparison similar data reported earlier for blue LED [4]. Solid and dotted lines in the plots are
 277 drawn for eye.

278 The AR of the green LED differs from those studied in [17] in two aspects. First, it consists of
 279 five InGaN QWs, which makes band-to-band light absorption quite critical for achieving high LEE.
 280 Secondly, the AR non-uniformity of the green LEDs results in two different spectral peaks (Sec.2). At
 281 that, the sw-emission may be effectively absorbed by the AR producing the lw-emission, which may
 282 increase additionally the band-to-band optical losses. As the sw-emission is more pronounced at low
 283 temperatures ($T < 100-120$ K), this should lead to a greater difference between LEEs of blue and
 284 green LEDs in this temperature range, in accordance with the data shown in Fig.6b. At high temper-
 285 atures ($T \sim 250-300$ K) the sw-emission is suppressed and the difference between LEEs of blue and
 286 green LEDs in this temperature range (less than $\sim 2\%$) may be attributed to a difference in the band-
 287 to-band light absorption in the MQW active region.

288 Figure 5d shows that the RT EQE dependence on the output power plotted in log scale has a
 289 nearly dome-like character with asymmetric low-power and high-power wings. In this case, the pro-
 290 cedure of the LEE extraction from this dependence based on approximation of the high-power wing
 291 with the ABC-model [17] becomes applicable, providing the LEE value of about 66%. The method
 292 used in our study, which implies two ARs to co-exist, gives the value of 68% which is quite close to
 293 the above cruder estimate.

294

295 4.2. Quality factors and internal quantum efficiency

296 It has been shown in Sec.3.2 that dependence of IQE of the green LED on output optical power
 297 can be interpreted in terms of two co-existing ARs having different quality factors, Q_{sw} and Q_{lw} , the
 298 temperature dependences of which is given in Figure 6b. One can see from the figure that Q -factors
 299 of the green LED are much lower than those reported in [4] for the blue one. This is manifestation of
 300 the so-called “green gap” problem, i.e. remarkable decline of the LED efficiency towards longer
 301 emission wavelength. Two origins of the “green gap” are commonly discussed: quantum-confined
 302 Stark effect leading to spatial separation of electron and hole wave functions inside the InGaN QWs
 303 (see, for example, [2,3] and references therein) and degradation of materials quality in InGaN alloys
 304 with high indium content [18-19]. Recently, implication of composition fluctuations in InGaN to the
 305 LED efficiency reduction in the “green gap” has been also demonstrated [20-22].

306 Figure 6b shows that Q -factor corresponding to sw-emission from the green LED is a few times
 307 lower than that corresponding to the lw-emission. Since the Q -factors determine the maximum IQE
 308 values through the relationship $IQE_k = Q_k / (Q_k + 2)$ ($k = sw, lw$), the lower Q_{sw} results immediately in
 309 a lower IQE values of the sw-emission. The temperature dependences of maximum IQEs for sw- and
 310 lw-emission of the green LED are plotted in Figure 6c. One can see from the figure that the efficiency
 311 of the sw-emission declines with temperature much faster than that of the lw-emission, resulting in
 312 suppression of the sw-peak in the emission spectrum at RT. In turn, IQE corresponding to the lw-
 313 emission of the green LED declines faster than the efficiency of blue LED. This observation is in line
 314 with the conclusion made in [21] that contribution of Auger recombination to the temperature-
 315 dependent IQE reduction is much stronger in green LEDs than in blue ones.

316 Figure 6c shows also that absolute maximum of IQE is controlled by the lw-emission irrespec-
 317 tive of temperature. The maximum IQE tends to the value of ~92% at zero temperature, i. e. the non-
 318 radiative recombination does not vanish completely at cryogenic temperatures. This makes doubtful
 319 estimations of the RT IQE based on comparison of the LED emission intensities measured at cryo-
 320 genic and room temperatures.

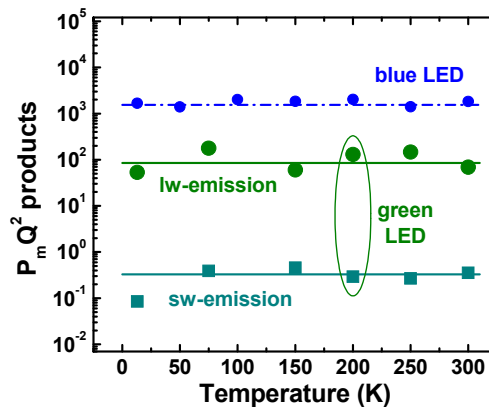
321 4.3. Specific powers and recombination volumes

322 Output power corresponding to maximum IQEs of the sw- ($P_m^{(sw)}$) and lw-emission ($P_m^{(lw)}$) of
 323 green LED are plotted in Figure 6d versus temperature along with the transition power P_t . The only
 324 temperature-dependent output power P_m of blue LED from [4] is also shown in the figure by the
 325 dash-dotted line just for comparison. One can see that $P_m^{(lw)}$ and P_m of blue LED are rather close to
 326 each other in the whole temperature range of study. In contrast, $P_m^{(sw)}$ is about two orders of magni-
 327 tude lower than $P_m^{(lw)}$ and P_m . The reason for this is discussed below in more detail.

328 The transition power P_t is situated between $P_m^{(sw)}$ and $P_m^{(lw)}$ at $T < 200$ K and approaches $P_m^{(sw)}$ at
 329 RT. The latter corresponds to merging of the sw- and lw-emission peaks at $T = 300$ K, as it can be
 330 seen from Figure 2c.

331 In order to get more information from the data obtained, we have plotted in Figure 7 the $P_m Q^2$
 332 products corresponding to sw- and lw-emission peaks *vs.* temperature. Since $P_m Q^2 \propto E_{ph} V_r B^3 / C^2$
 333 where E_{ph} is the energy of emitted photon, V_r is the recombination volume, and B and C are the radi-
 334 ative and Auger recombination coefficients, respectively (see, for example, [15]), the $P_m Q^2$ product
 335 does not include the Shockley-Read-Hall coefficient A responsible for carrier recombination at point
 336 and extended defects. As one can see from Figure 7, the $P_m Q^2$ products of both sw- and lw-emission
 337 peaks are found to be nearly independent of temperature, similar to the case of blue LED [4]. This is
 338 evidence for the fact that both radiative and Auger recombination coefficients in the green LED, like
 339 in the blue one, have qualitatively similar temperature dependence: either ascending or descending.
 340 This result is in qualitative agreement with the recently observed anomalous (ascending) tempera-
 341 ture variation of the radiative recombination coefficient [21], which was attributed to strong hole

342 localization by composition fluctuations in InGaN alloys. At that, the Auger recombination coefficients
 343 were found in [21] to increase with temperature in both blue and green LEDs.



344

345 **Figure 7.** $P_m Q^2$ product as a function of temperature plotted for sw- and lw-emission of green LEDs.
 346 Big symbols are experimental points, solid lines are drawn for eyes. Data for blue LED reported in [4]
 347 are also shown for comparison by small circles and dash-dotted line.

348 Using the values of the B - and C -coefficients at 450 and 540 nm reported in [23] for RT, we have
 349 estimated the ratio of the recombination volumes of blue LED ($V_r^{(b)}$) and that corresponding to lw-
 350 emission of green LED ($V_r^{(lw)}$). The obtained ratio $V_r^{(b)} / V_r^{(lw)} \approx 0.85$ indicates that the recombination
 351 volumes are comparable with each other. As the active region of the blue LED was specially opti-
 352 mized to provide a uniform carrier injection in all five QWs in the active region, the latter fact en-
 353 ables attributing the lw-emission of green LED to operation of its active region as a whole.

354 A similar estimation provides $V_r^{(b)} / V_r^{(sw)} \approx 220$, which demonstrates the recombination vol-
 355 ume of the AR responsible for sw-emission to be about two orders of magnitude smaller than that of
 356 the AR producing lw-emission. Such a big difference cannot be explained by dominant carrier injec-
 357 tion in one of the five QWs. Therefore, a natural interpretation of the result implies the sw-emission
 358 comes from the local lateral areas distributed within the QWs.

359 4.4. Possible origins of active region non-uniformity

360 Any interpretation of the above results should account for and/or explain four main points: (i)
 361 the difference between the wavelengths of sw- and lw-emission, (ii) the sequence of sw- and lw-
 362 peaks appearance in the emission spectrum with the operating current, (iii) the difference in the re-
 363 combination volumes associated with sw- and lw-emission, and (iv) the difference in the Q -factors or
 364 IQEs of the sw- and lw-emission. We will consider below a number of scenarios for the active region
 365 non-uniformity, addressing these points.

366 The difference in the emission wavelength can be attributed to variation of either composition
 367 or width of InGaN QWs in the LED active region. Simulations of InGaN SQW LED structures oper-
 368 ating at the current density of 20 A/cm² carried out with the commercial SiLENSe 5.10 package [24]
 369 have shown that the observed difference between the sw- and lw-emission wavelengths may be as-
 370 sociated with either ~2% variation of the indium content in the InGaN alloy or ~0.4 nm (i. e. ~1.5
 371 monolayer) variation of the QW width. Both versions seem to be realistic for green LEDs examined
 372 here.

373 A possible mechanism producing compositional non-uniformity of the active region is partial
 374 stress relaxation in the LED structure. Indeed, the critical thickness of (0001) InGaN/GaN layer is
 375 comparable with the QW widths in the green spectral range [25]. If the total width of all QWs in the
 376 LED active region exceeds the critical thickness, stress relaxation becomes possible, resulting, in par-
 377 ticular, in an increase of indium content just by a few percent in the top partially relaxed QWs [26].
 378 The appearance of the sw-emission at low currents and of lw-emission at high currents may be then

379 explained, assuming dominant hole injection into the bottom (unrelaxed) QWs at low currents. This
 380 assumption does not agree, however, with the data of experiments carried out on LEDs with color-
 381 coded blue/cyan QWs in the active region [27]. In addition, the above scenario cannot explain a lower
 382 IQE of the sw-emission, as compared to lw-emission, and the substantial difference in their re-
 383 combination volumes.

384 A similar conclusion can be made about another mechanism, which implies the lw-emission to
 385 come from the In-rich clusters more or less uniformly distributed over all the QWs. Here lw-
 386 emission is expected to appear first at low currents and its recombination volume should be smaller
 387 than that of the sw-emission, in contrary to observations. Therefore, other mechanisms are necessary
 388 to invoke in order to interpret the characterization data of our green LEDs.

389 Here, we suggest the following qualitative model for the active region non-uniformity. First of
 390 all, we assume rarely distributed regions with lower indium content responsible for sw-emission to
 391 be embedded into the matrix of InGaN with higher indium content, which is responsible for lw-
 392 emission. These low-indium regions have a lower height of the potential barriers formed at the
 393 InGaN/GaN interfaces for both electrons and holes. This is due to lower band offsets in both conduc-
 394 tion and valence bands and lower polarization charges induced at the interfaces. At low LED operat-
 395 ing currents, the carrier injection into the InGaN QWs is limited by thermionic emission over the
 396 barriers. Therefore pinching of the current is expected to occur in such a way, as to produce domi-
 397 nant pumping of the low-indium regions and, eventually, the sw-emission of photons. At higher
 398 currents, pumping of the high-indium matrix starts to occur, resulting in the lw-emission, which be-
 399 comes quickly dominant partly due to a much larger recombination volume.

400 Such a scenario is consistent with most of observations discussed above but it does not yet ex-
 401 plain the difference in IQEs (Q -factors) of the sw- and lw-emission regions. The explanation can be
 402 given, assuming the low-indium regions to be formed around extended defects like threading dislo-
 403 cations and V-pits. Indeed, less effective indium incorporation is expected next to the dislocation
 404 cores because of excess elastic energy related to dislocation-mediated strain. Formation of the InGaN
 405 QWs with larger bandgap on the side walls of V-pits has been directly demonstrated in [28]. Exist-
 406 ence of dislocation cores serving as non-radiative recombination centers may explain a lower IQE of
 407 the sw-emission regions.

408 In order to assess whether the small recombination volume $V_r^{(sw)}$ may be attributed to extended
 409 defects, we assume the sw-emission to originate from the carriers collected from the area of about
 410 $L_d^2 = D_a \tau_d$ around each of the defects, where L_d is the carrier diffusion length, D_a is the ambipolar
 411 diffusion coefficient, and τ_d is the differential carrier life time. Using experimental values $D_a \sim 0.25$
 412 cm^2/s [29] and $\tau_d \sim 20$ ns [21] typical for green InGaN LEDs operating at low currents, we obtain
 413 $L_d^2 \sim 5 \times 10^{-9}$ cm^2 . In this case, the density of the extended defects necessary to provide the ratio
 414 $V_r^{(b)} / V_r^{(sw)} \sim 220$ is $\sim 10^6$ cm^{-2} , which may be tentatively associated with the density of V-pits. Indeed,
 415 the carrier injection into semi-polar QWs formed at the side walls of V-pits is not hindered by high
 416 potential barriers typically induced at the {0001}-interfaces of polar QWs. Therefore, the current flow
 417 through the V-pits may dominate under low-current conditions [30]. This conclusion is indirectly
 418 supported by the experimental correlation between the waving observed in I-V curves, which is the
 419 evidence for carrier leakage through extended defects, and the values of the current corresponding
 420 to maxima of the sw-emission efficiency (closed circles in Fig.1). Of course, the V-pit density of $\sim 10^6$
 421 cm^{-2} is just the lower-limit estimate, which does not account for the complex mechanism of current
 422 pinching around this type of defect [30] and dispersion of their dimensions affecting the current flow
 423 as well.

424 It is interesting that two-peak character of the low-temperature efficiency dependence on cur-
 425 rent has been reported earlier for both blue and green LEDs [6]. Moreover, the efficiency maxima of
 426 green LEDs were shifted remarkably to lower currents in green LEDs compared to blue ones, in line
 427 with our observations. Those data may also be interpreted in terms of competition between the cur-
 428 rent flow through the V-pits and through the {0001}-interfaces of QWs, assuming the efficiency of the
 429 main QWs to be somewhat lower than that of the semi-polar QWs on the side walls of the pits. At

430 RT, the low-current efficiency peaks quenched and the efficiency dependence on current gains con-
431 ventional dome-like shape [6].

432 Generally, the above mechanism considering V-pits as the origin of AR non-uniformity is ex-
433 pected to be especially pronounced in green LEDs, as higher indium content in the QWs is known to
434 favor the V-pit formation.

435 5. Conclusions

436 In this study, we have observed mutually correlated non-ordinary evolution of the LED emis-
437 sion spectra and efficiency which, to our best knowledge, had not been reported before. The observa-
438 tion could be interpreted in terms of the active region non-uniformity, assuming co-existence of, at
439 least, two sub-regions emitting at different wavelengths and having different radiative efficiencies.
440 The use of the ABC-model extended to the case of non-uniform active region enabled estimating the
441 recombination volumes corresponding to these sub-regions, which were found to differ by a factor
442 of ~220. To explain such a big difference, as well as the sequence of appearance of the sw- and lw-
443 emission in the integral spectra, one of the sub-regions was associated with extended defects, like V-
444 pits, whereas another sub-region was attributed to the main part of the LED active region. As higher
445 indium content in InGaN QWs favors the V-pit formation during LED structure growth, the above
446 observations should be regarded as those specific for green LEDs and much less typical for blue
447 ones.

448 Characterization of the green LED in a wide temperature range from 13 K to 300 K allowed
449 evaluating temperature-dependent LEE and IQE. The estimated efficiency of light extraction, 68% at
450 RT, was slightly lower than that reported earlier for blue LEDs but it tended to saturate at low tem-
451 peratures at the value of 72%, which was tentatively related to re-absorption of the sw-emission by
452 the lw-emission sub-region. Maximum IQE of the green LED was found to decrease from ~91% at 13
453 K to ~70% at 300 K and controlled by the lw-emission from the main part of the active region. Re-
454 markable deviation of the IQE maximum from 100% at cryogenic temperatures indicates that non-
455 radiative recombination does not vanish in the green LED, in contrast to blue one where it was ~97%
456 [4]. This makes doubtful the method of IQE determination based on comparison of EL intensity at
457 cryogenic and room temperatures.

458 Defect-mediated sw-emission approaches its maximum efficiency at very low currents, produc-
459 ing additional peak/shoulder in the EQE dependence on current. It looks like in the previous studies,
460 IQE of the defect-mediated emission exceeded that of the main active region. This resulted in a
461 strong shift of the total IQE/EQE maximum of green LEDs to lower currents, as compared to the case
462 of blue LEDs. In our study, the lw-emission from the main active region dominated at all tempera-
463 tures. Therefore, the IQE/EQE maximum of the green LED is approached at nearly the same currents
464 as in the case of blue LED examined in [4].

465 Processing of the characterization data has shown that the B^3 / C^2 ratio, where B is the radiative
466 and C is the Auger recombination coefficients, is nearly independent of temperature in both cases of
467 sw- and lw-emission of green and blue LEDs. Accounting for rather strong temperature dependence
468 of single B and C coefficients reported in [21], this fact implies existence of a universal relationship
469 between the recombination coefficients in InGaN QWs irrespective of their particular composition.
470 The physics behind this observation is one of the still open questions to be addressed by future stud-
471 ies.

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476 concept, analytical model and largely contributed to the manuscript preparation; D. Mamedov contributed to
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480 Abbreviations

481 The following abbreviations are used in this manuscript:

482	AR	Active region
483	EQE	External quantum efficiency
484	EL	Electroluminescence
485	IQE	Internal quantum efficiency
486	LED	Three letter acronym
487	LEE	Light extraction efficiency
488	LO	Longitudinal optical
489	MQW	Multiple quantum well
490	ND	Neutral density
491	QW	Quantum well
492	RT	Room temperature
493	SQW	Single quantum well
494	WPE	Wall plug efficiency

495 References

- 496 1. Zheludev, N. The life and times of the LED a 100-year history. *Nat. Photonics* **2007**, *1*, 189-129,
497 doi:10.1038/nphoton.2007.34.
- 498 2. Crawford, M. H. LEDs for Solid-State Lighting: Performance Challenges and Recent Advances. *IEEE J. Sel.*
499 *Top. Quantum Electron.*; **2009**, *15*, 1028-1040, doi: 10.1109/JSTQE.2009.2013476.
- 500 3. Weisbuch, C.; Piccardo, M.; Martinelli, L.; Iveland, J.; Peretti, J.; and Speck, J.S. The efficiency challenge of
501 nitride light-emitting diodes for lighting. *Phys. Status Solidi A* **2015**, *212*, 899-913,
502 doi:10.1002/pssa.201431868.
- 503 4. Titkov, I.E.; Karpov, S.Yu.; Yadav, A.; Zerova, V.L.; Zulonas, M.; Galler, B.; Strassburg, M.; Pietzonka, I.;
504 Lugauer, H.-J.; and Rafailov, E.U. Temperature-Dependent Internal Quantum Efficiency of Blue High-
505 Brightness Light-Emitting Diodes. *IEEE J. Quantum Electron.* **2014**, *50*, 189-129, doi:
506 10.1109/JQE.2014.2359958.
- 507 5. Karpov, S.Yu.; Binder, M.; Galler, B.; and Schiavon, D. Spectral dependence of light extraction efficiency of
508 high-power III-nitride light-emitting diodes. *Phys. Status Solidi RRL* **2015**, *1*, 189-129, doi:
509 10.1002/pssr.201510073.
- 510 6. Peter, M.; Laubsch, A.; Bergbauer, W.; Meyer, T.; Sabathil, M.; Baur, J.; and Hahn, B. New developments in
511 green LEDs. *Phys. Status Solidi A* **2007**, *1*, 189-129, doi: 10.1002/pssa.200880926.
- 512 7. Shin, D.-S.; Han D.-P.; Oh, J.-Y and Shim, J.-I Study of droop phenomena in InGaN-based blue and green
513 light-emitting diodes by temperature-dependent electroluminescence. *Appl. Phys. Lett.* **2012**, *100*, 153506,
514 doi: 10.1063/1.3703313.
- 515 8. Dai, Q.; Shan, Q.; Cho, J.; Schubert, E.F.; Crawford, M.H.; Koleske, D.D.; Kim, M.-H; and Park, Y. On the
516 symmetry of efficiency-versus-carrier-concentration curves in GaInN/GaN light-emitting diodes and rela-
517 tion to droop-causing mechanisms. *Appl. Phys. Lett.* **2011**, *98*, 033506, doi: 10.1063/1.3544584.
- 518 9. Lin, G.-B.; Meyaard, D.; Cho, J.; Schubert, E.F.; Shim, H. and Sone, C. Analytic model for the efficiency
519 droop in semiconductors with asymmetric carrier-transport properties based on drift-induced reduction of
520 in-junction efficiency. *Appl. Phys. Lett.* **2012**, *100*, 161106, doi: 10.1063/1.4704366.
- 521 10. Wang, J.; Wang, L.; Wang, L.; Hao, Z.; Luo, Y.; Dempewolf, A.; Müller, M.; Bertram, F.; and Christen, J.
522 An improved carrier rate model to evaluate internal quantum efficiency and analyze efficiency droop
523 origin of InGaN based light-emitting diodes. *Appl. Phys. Lett.* **2012**, *102*, 023107, doi: 10.1063/1.4736591.
- 524 11. Kisin, M.V. and El-Ghoroury, H.S. Inhomogeneous injection in III-nitride light emitters with deep multiple
525 quantum wells. *J. Comput. Electron.* **2015**, *14*, 432-443, doi: 10.1007/s10825-015-0673-5.
- 526 12. Karpov, S.Yu. Effect of localized states on internal quantum efficiency of III-nitride LEDs. *Phys. Status*
527 *Solidi RRL* **2010**, *4*, 320-322, doi: 10.1002/pssr.201004325.
- 528 13. Shah, J. M.; Li, Y.-L.; Gessmann, Th. and Schubert, E.F. Experimental analysis and theoretical model for
529 anomalously high ideality factors ($n_2 > 2.0$) in Al-GaN/GaN p-n junction diodes. *J. Appl. Phys.* **2003**, *94*,
530 2627-2630, doi: 10.1063/1.1593218.
- 531 14. Masui, H. Diode ideality factor in modern light-emitting diodes. *Semicond. Sci. Technol.* **2011**, *26*, 075011,
532 doi:10.1088/0268-1242/26/7/075011.

- 533 15. Karpov, S. Yu. ABC-model for interpretation of internal quantum efficiency and its droop in III-nitride
534 LEDs: a review. *Opt. Quant. Electron.* **2015**, *47*, 1293-1303, doi:10.1007/s11082-014-0042-9.
- 535 16. Karpov, S.Yu.; Cherkashin, N.A.; Lundin, W.V.; Nikolaev, A.E.; Sakharov, A.V.; Sinitsyn, M.A.; Usov, S.O.;
536 Zavarin, E.E. and Tsatsulnikov, A.F. Multi-color monolithic III-nitride light-emitting diodes: Factors con-
537 trolling emission spectra and efficiency. *Phys. Status Solidi A* **2016**, *213*, 19-29, doi: 10.1002/pssa.201532491.
- 538 17. Karpov, S. Yu.; Binder, M.; Galler, B. and Schiavon, D. Spectral dependence of light extraction efficiency of
539 high-power III-nitride light-emitting diodes. *Phys. Stat. Solidi RRL* **2015**, *9*, 312-316,
540 doi: 10.1002/pssr.201510073.
- 541 18. Langer, T.; Jönen, H.; Kruse, A.; Bremers, H.; Rossow, U. and Hangleiter, A. Strain-induced defects as non-
542 radiative recombination centers in green-emitting GaInN/GaN quantum well structures. *Appl. Phys. Lett.*
543 **2013**, *103*, 022108, doi: 10.1063/1.4813446.
- 544 19. Hammersley, S.; Kappers, M.J.; Massabuau, F.C.-P.; Sahonta, S.-L.; Dawson, P.; Oliver, R. A. and Hum-
545 phreys, C. J. Effects of quantum well growth temperature on the recombination efficiency of InGaN/GaN
546 multiple quantum wells that emit in the green and blue spectral regions. *Appl. Phys. Lett.* **2015**, *107*,
547 132106, doi: 10.1063/1.4932200.
- 548 20. Auf der Maur, M.; Pecchia, A.; Penazzi, G.; Rodrigues, W. and Di Carlo, A. Efficiency Drop in Green
549 InGaN/GaN Light Emitting Diodes: The Role of Random Alloy Fluctuations. *Phys. Rev. Lett.* **2016**, *116*,
550 027401, doi: 10.1103/PhysRevLett.116.027401.
- 551 21. Nippert, F.; Karpov, S. Yu.; Callsen, G.; Galler, B.; Kure, T.; Nenstiel, C.; Wagner, M. R.; Straßburg, M.;
552 Lugauer, H.-J. and Hoffmann, A. Temperature-dependent recombination coefficients in InGaN light-
553 emitting diodes: Hole localization, Auger processes, and the green gap. *Appl. Phys. Lett.* **2016**, *109*, 61103,
554 doi: 10.1063/1.4965298.
- 555 22. Karpov, S.Yu. Carrier localization in InGaN by composition fluctuations: implication to “green gap”. *Pho-
556 ton. Res.* **2017**, *5*, A7-A12, doi: 10.1364/PRJ.5.0000A7.
- 557 23. Schiavon, D.; Binder, M.; Peter, M.; Galler, B.; Drechsel, P. and Scholz, F. Wave-length-dependent determi-
558 nation of the recombination rate coefficients in single-quantum-well GaInN/GaN light emitting diodes.
559 *Phys. Status Solidi B* **2013**, *250*, 283-290, doi: 10.1002/pssb.201248286.
- 560 24. SiLENSe—software tool for light emitting diode (LED) bandgap engineering. Available online:
561 <http://www.str-soft.com/products/SiLENSe/index.htm> (accessed on 03 July 2017).
- 562 25. Lobanova, A. V.; Kolesnikova, A. L.; Romanov, A. E.; Karpov, S. Yu.; Rudinsky, M. E. and Yakovlev, E. V.
563 Mechanism of stress relaxation in (0001)InGaN/GaN via formation of V-shaped dislocation half-loops.
564 *Appl. Phys. Lett.* **2013**, *103*, 152106, doi: 10.1063/1.4824835.
- 565 26. Sakharov, A.V.; Lundin, W.V.; Zavarin, E.E.; Sinitsyn, M.A.; Nikolaev, A.E.; Usov, S.O.; Sizov, V.S.; Mi-
566 khailovsky, G.A.; Cherkashin, N.A.; Hytch, M.; Hue, F.; Yakovlev, E.V.; Lobanova, A.V. and Tsatsulnikov,
567 A.F. Effect of strain relaxation on formation of active region of InGaN/(Al)GaN heterostructures for green
568 range LEDs. *Semiconductors* **2009**, *43*, 812-817, doi: 10.1134/S1063782609060232.
- 569 27. Galler, B.; Laubsch, A.; Wojcik, A.; Lugauer, H.; Gomez-Iglesias, A.; Sabathil, M. and Hahn, B. Investiga-
570 tion of the carrier distribution in InGaN-based multi-quantum-well structures. *Phys. Stat. Solidi C* **2011**, *8*,
571 2372-2374, doi: 10.1002/pssc.201001075.
- 572 28. Hangleiter, A.; Hitzel, F.; Netzel, C.; Fuhrmann, D.; Rossow, U.; Ade, G. and Hinze, P. Suppression of
573 Nonradiative Recombination by V-Shaped Pits in GaInN/GaN Quantum Wells Produces a Large Increase
574 in the Light Emission Efficiency. *Phys. Rev. Lett.* **2005**, *95*, 127402, doi: 10.1103/PhysRevLett.95.127402.
- 575 29. Danhof, J.; Solowan, H.-M.; Schwarz, U. T.; Kaneta, A.; Kawakami, Y.; Schiavon, D.; Meyer, T. and Peter,
576 M. Lateral charge carrier diffusion in InGaN quantum wells. *Phys. Stat. Solidi B* **2012**, *249*, 480-484,
577 doi: 10.1002/pssb.201100476.
- 578 30. Li, C.-K.; Wu, C.-K.; Hsu, Ch.-Ch.; Lu, L.-Sh.; Li, H.; Lu, T.-Ch. and Wu, Y.-R. 3D numerical modeling of
579 the carrier transport and radiative efficiency for InGaN/GaN light emitting diodes with V-shaped pits. *AIP
580 Advances* **2016**, *6*, 055208. doi: 10.1063/1.4950771.