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Bioelectrochemical Changes During the Early Stages of Chalcopyrite Interaction with *Acidithiobacillus thiooxidans* and *Leptospirillum* sp.

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Abstract: A bioelectrochemical study of charge transfer in the biofilm/chalcopyrite interface was performed to investigate the effect of surficial sulfur reduced species (SRS), as non-stoichiometric compounds or polysulfides (S_n^{2-}), and elemental sulfur (S^0) on a biofilm structure during the earliest stages (1, 12 and 24 h) of chalcopyrite biooxidation by *A. thiooxidans* alone and adding *Leptospirillum* sp. The surface of massive chalcopyrite electrodes was exposed to the bacteria, which were analyzed electrochemically, spectroscopically, and microscopically. At the studied earlier times, charge transfer and significant differences in the biofilm structure were detected, depending on the presence of *Leptospirillum* sp. acting on *A. thiooxidans* biofilms. Such differences were a consequence of a continuous chalcopyrite pitting and promoting changes in biofilm hydrophobicity. *A. thiooxidans* modifies the reactive properties of SRS and favors an acidic dissolution, which shifts into ferric when *Leptospirillum* sp. is present. *A. thiooxidans* allows H^+ and Fe^{3+} diffusion, and *Leptospirillum* sp. allows surpassing the charge transfer (reactivity) barrier between the mineral interface and the ions. The observed changes of hydrophobicity on the interface are associated to ions and electrons activity and transfer. Finally, a model of S^0 biooxidation by *A. thiooxidans* alone or with *Leptospirillum* sp., is proposed.

Keywords: Sulfur reduced species; Biofilms; Chalcopyrite; Biooxidation; Hydrophobicity

1. Introduction

Chalcopyrite ($CuFeS_2$) is the most important source of copper in the world. For this reason, this Cu sulfide is intensively studied to gain fundamental knowledge at improving copper extraction. Bioleaching is a growing technology to recover Cu during chalcopyrite dissolution; thus, the fate of chalcopyrite oxidation depends on enhanced biooxidation, which is determined by the evolution of the mineral and by the interaction of chemolithoautotroph microorganisms (i.e., *Acidithiobacillus thiooxidans*, *A. ferrooxidans*). Most studies have focused on analyzing biooxidation processes at long time intervals (> 120 h), even though initial stages are a key factor at determining the evolution of biooxidation processes.

According to Crundwell [1], the slow rate of chalcopyrite dissolution is explained by its electronic structure, as a semiconductor. The electrochemical oxidizing mechanism of chalcopyrite implies that mineral dissolution is a surficial phenomenon, where anodic and cathodic reactions occur simultaneously at the whole mineral surface. Thus, charge transfer between the mineral surface

and ferric ions (Fe^{3+}) in the solution determines the rate of mineral dissolution of both chemical and bacterial oxidation [2]. This gives rise to multiple phenomena, by which chalcopyrite is further oxidized by bioleaching microorganisms that enhance H^+ and $\text{Fe}^{2+}/\text{Fe}^{3+}$ availability. Consequently, charge transfer and other simultaneous phenomena must be carefully analyzed on the biofilm/mineral/electrolyte interface during the first stages of biooxidation process. Bevilaqua et al. [3] have used cyclic voltammetry (CV), chronoamperometry (CA), electrochemical noise (EN), and electrochemical impedance spectroscopy (EIS) to analyze extensively the surface properties to better describe the electrochemical behavior of biofilm/mineral interfaces. Because diverse secondary compounds appear on mineral surfaces and interact with biofilm-forming bacteria, the use of electrochemical techniques together with surface analysis techniques, by scanning electronic microscopy (SEM) and confocal laser scanning microscopy (CLSM), yield information about the structure of such interface, as well as about the mechanisms of electrochemical reactions.

It is currently accepted that bacteria attachment to the surface influences the rate of dissolution of mineral, since complex surface phenomena occur (mixed potential decreases, kinetics, and mass-transport phenomena) [2]; these changes also influence the structural characteristics of the formed biofilms on the surface, mainly the composition of the extracellular polymeric substances (EPS) of the matrix wherein microorganisms are embedded. García-Meza et al. [4] correlated the surface speciation (reactivity of Sn^{2-} , S_0 , CuS), EPS hydrophobicity, and biofilm thickness during chalcopyrite biooxidation by the sulfur-oxidizing microorganism (SOM) *A. thiooxidans*, while Florian et al. [5] discussed that biofilms contribute to change the surficial properties, enhancing the microbiological activity and mineral dissolution, being more significant if iron-oxidizing microorganisms (IOM; *Leptospirillum* spp.) are present in mixed biofilms. Also, it has been suggested that SRS formed during bioleaching influence the availability of Fe and consequently, bioleaching rates [6]. However, a comprehensive assessment of surface processes produced by incipient *A. thiooxidans* and *Leptospirillum* sp. attachment is still needed.

The aim of this work is to account for the relationships between chalcopyrite biooxidation and complex-surface processes taking place at the biofilm/mineral interface during chalcopyrite biooxidation, first only with the SOM *A. thiooxidans* and then adding *Leptospirillum* sp. (SOM+IOM) during the first stages (1, 12 and 24 h) of the bacteria-mineral interaction. A combined methodology was used to obtain closer details to describe complex biofilms, utilizing electrochemical (CV, CA, EN, and EIS), spectroscopic (Raman), and microscopic (SEM and CLSM) techniques. The present study could contribute to better understand the initial stages of chalcopyrite biooxidation by mixed SOM+IOM and its relationship with surface properties (reactivity, secondary compounds, dissolution), thus, favoring strategies to enhanced chalcopyrite dissolution for larger times of bioleaching.

2. Materials and Methods

2.1. Analysis and obtainment of SRS on chalcopyrite

Chalcopyrite samples were acquired from San Luis Potosí (Mexico). The samples were selected as high-purity crystals. Mineral identity, composition and purity were verified by X-ray diffraction patterns (XRD Rigaku 22002, $\theta = 0.02$, 10 to 90° , $\text{Cu-K}\alpha$ radiation), and SEM-EDS (Philips XL-30); accordingly, the mineral sample comprised: 99.6% wt chalcopyrite, 0.2% wt pyrite (FeS_2), and 0.2% wt quartz (SiO_2).

The selected chalcopyrite crystals were used to construct massive chalcopyrite electrodes (MCE). We used MCE instead of carbon-chalcopyrite paste electrodes to analyze the same surface electrochemically, spectroscopically, and microscopically, and to avoid the capacitive contribution of carbon paste.

The MCE were potentiostatically modified as indicated previously [4] (Appendix A), using a potentiostat (EPSILON BAS 2.10.73). The electrochemical oxidation of MCE surfaces allows a rapid,

significant and quasi-homogeneous generation of sulfur reduced species (SRS; $S_{n^{2-}}/S^0$) to sustain metabolic activity of *A. thiooxidans*. Such SRS were generated by applying an anodic pulse (Ean) from 0.36 to 1.015 V *vs.* SHE (the standard hydrogen electrode). The occurrence of SRS was verified by Raman spectroscopy for each electrochemically modified MCE (referred as MCE* surface throughout the text). Since SRS change the mineral reactivity, a different $S_{n^{2-}}$ -to- S^0 ratio was obtained on each specific MCE* surface. On the first one, a $S_{n^{2-}}$ -surface enrichment (MCE*- $S_{n^{2-}}$) was observed, whereas on the second one, mainly S^0 (MCE*- S^0) was detected. This is important considering that formation of SRS comprises several stages, including mainly $S_{n^{2-}}$ and S^0 , in which different responses of bacterial attachment are expected. Thus, the proportion of such SRS as well as of EPS modifies the hydrophathy of the initial biofilm-mineral interface.

2.2. Biofilm formation

The strain *A. thiooxidans* ATCC-19377 (SOM) was aerobically cultured in an ATCC-125 (M1) medium (pH 2.0), which contained per one liter of distilled water: 0.4 g of $(NH_4)_2SO_4$, 0.5 g of $MgSO_4 \cdot 7H_2O$, 0.25 g of $CaCl_2$, 3 g of KH_2PO_4 , 0.005 g of $FeSO_4 \cdot 7H_2O$, and 5 g of S^0 (Baker); the pH was fixed at 2.0 with concentrated H_2SO_4 . A wild strain of *Leptospirillum* sp. was aerobically cultured in an ATCC-882, M2) medium, which in turn contained the following salts per one liter of distilled water: 0.132 g of $(NH_4)_2SO_4$, 0.5 g of $MgCl_2 \cdot 6H_2O$, 0.027 g of KH_2PO_4 , 0.147 g of $CaCl_2 \cdot 2H_2O$, and 20 g of $FeSO_4 \cdot 7H_2O$; the pH was also fixed at 2.0 with concentrated H_2SO_4 . The mixed SOM+IOM culture of *A. thiooxidans* and *Leptospirillum* sp. was made by adding SOM and IOM biomass in a ratio of 1:1, aerobically cultured in a previously mixed medium (1:1, M1:M2), to set a 1:10 cells:medium ratio. All cultures were aerobically incubated at 30 °C under orbital agitation (120 rpm).

The biotic experiments were carried out using separately MCE*- $S_{n^{2-}}$ or MCE*- S^0 surfaces, which were settled inside an Erlenmeyer flask with 50 mL of M1 or M1+M2 inoculated with *ca.* 3×10^7 cells/mL of SOM or SOM+IOM, respectively. All the cultures were incubated at 30 °C and 120 rpm during 1, 12 and 24 h. All the experiments were performed in triplicate; abiotic controls were also carried out in triplicate to compare between chemical and biological oxidation of $S_{n^{2-}}$ or S^0 . Immediately afterwards, the biooxidized MCE*-SRS (MCE*- $S_{n^{2-}}$ and MCE*- S^0) surfaces were dried with direct nitrogen current for spectroscopic and microscopic analyses.

2.3. Electrochemical assessment of the MCE*-SRS reactivity

Mineral reactivity is a measure of oxidation capacity, which can be used to assess variations of this parameter caused by external perturbations, as secondary compounds, emerging/dissolution of passive phases; such parameter can be systematically studied using cyclic voltammetry. Accordingly, these analyses were performed using a potentiostat (PAR versaSTAT 3F). For all the electrochemical analyses, the MCE*-SRS (MCE*- $S_{n^{2-}}$ or MCE*- S^0) was used as working electrode, a saturated Hg/Hg_2SO_4 (0.615 V *vs.* SHE) as a reference, and a graphite rod (Alfa Aesar, 99.9995% purity) as counter-electrode; the specific culture medium for SOM (M1) or SOM+IOM (M1-M2) was used as electrolyte. Immediately before (0 h) and 1, 12 and 24 h after the MCE*-SRS (bio)oxidation assays, the surficial reactivity was assessed using positive-going scans at 20 mV/s.

Electrochemical potential noise (EPN) and electrochemical current noise (ECN) analyses were also acquired for a complete description of the MCE*-SRS surface reactivity, using a zero-resistance amperemeter (ZRA) and an acquisition rate of 1024 points. These analyses were performed assuming that no-aliasing effects were produced by the electrochemical system itself (*i. e.*, parasitic noise due to electrical devices). Therefore, the obtained EPN and ECN were used as an indirect approach to

elucidate the global oxidation behavior after chemical and biological MCE treatment. The data analysis was carried out with MATLAB software.

An electrochemical impedance spectroscopy (EIS) analysis was also performed to describe further bacteria and chalcopyrite interactions using a FRA Z module coupled to a PAR Versastat 3F potentiostat.

Before each EIS analysis, the stabilization time of OCP was taken, and EIS OCP-*vs.*-time measurements were made at 7 points per decade and using a range of frequency from 10 kHz to 1 mHz, respectively. To fit the experimental data, equivalent electric circuits (cee) were proposed. Finally, the diffusivity coefficients of H⁺, Fe²⁺ and Fe³⁺ (DH⁺, DFe²⁺ and DFe³⁺) (Eq. 1) were calculated according to He et al. [7] and Jin et al. [8]:

$$D = \frac{RT}{n^2 F^2 A \sqrt{2}} \left[\frac{1}{D_{ox}^{1/2} C_{ox}^0} + \frac{1}{D_{red}^{1/2} C_{red}^0} \right] \quad (1)$$

Where, R: The gas constant; n: Number of transferred electrons; F: Faraday constant; A: Electroactive area (m²); C: Bulk concentration of diffusing species (mol/m³).

2.4. Surficial analyses after the biooxidation of MCE*-SRS

The electrodes MCE*-S_n²⁻ and MCE*-S⁰ were analyzed using Raman spectroscopy and SEM-EDS. Also, environmental scanning electron microscopy (ESEM) was conducted with a FEI Quanta 200. Previously, the biooxidized MCE*-SRS surfaces were fixed with a volume of 3% v/v glutaraldehyde at 4°C; then, the biofilms were rinsed three times with a phosphate buffer solution (pH 7.2), dehydrated in 30%, 50%, 70%, and 100% ethanol, and critical-CO₂ dried in a Samdri-PVT-3D (Tousimis). The dry samples were mounted and coated with gold particles with a Cressington sputter (108 auto).

Forming biofilms on MCE* surfaces comprise an organic fraction (cells and EPS) and an inorganic fraction of the S_n²⁻ and S⁰, previously electrogenerated (S1). The proportion of these fractions modifies the hydrophathy of the initial biofilm-mineral interface. Thus, the biooxidized MCE* surfaces were stained to analyze their hydrophathy by means of confocal laser scanning microscopy (CLSM; Leica DMI4000B with argon laser and ×63 immersion objective). This analysis was done with an apochromatic plan 20X on a surface of 3.4 × 10⁻⁵ μm², after the micrographics were 3D reconstructed with LAS AF 2.4.1-version software. The images were generated from the 3D reconstruction at a same angle of a stack in the Z direction (depth 153 μm).

For the exopolysaccharides (α-mannose and α-glucose), the lectin *Canavalia ensiformis* (Con-A, tetramethylrhodamine conjugated; Molecular Probes, Eugene, OR, USA) was used for hydrophilic compounds; an excited signal at 554 nm was used and the maxima of emitted signal was detected using a band-pass filter I3 and N2.1) for emission at 576 nm. A Red Nile (NR; Sigma) fluorochrome was used to stain hydrophobic domains [9]; an excited signal at 552 nm emission and a band-pass filter for emission at 636 nm were used. These analyses were performed to describe the changes in the biochemical composition (proportion of hydrophobic and hydrophilic domains) associated with biofilm/SRS.

Raman spectra were recorded with a spectrometer (T64000 Jobin Yvon) coupled to a confocal-imaging module (Olympus BH2-UMA), which used a laser beam (λ = 532 nm). For calibration

purposes, a Si wafer disc (521 cm^{-1}) was used. The vibrational range was from 100 to 750 cm^{-1} , as the S_n^{2-} , the S^0 , and main copper-bearing compounds show their main active modes within this interval [10,11]. At least 10 Raman spectra were collected for each sample.

After biotic and abiotic assays (batch systems), leachates were recovered by vacuum filtration, and the extracted solutions were immediately analyzed for pH and oxidation-reduction potential (ORP; Thermo Electron pH-meter), total Fe (AAS-GF, Perkin Elmer 3100 atomic absorption spectrometer), Fe^{2+} (titration with $0.01\text{ N K}_2\text{Cr}_2\text{O}_7$) and Fe^{3+} ions (difference between the total Fe and Fe^{2+} concentrations). Also, the non-attached bacteria were counted microscopically in a Neubauer chamber; the number of attached bacteria was calculated by subtracting the non-attached bacteria number from the initially added, 3×10^6 cells/mL, to determine the planktonic/colonizing ratio.

3. Results

3.1. SOM biofilms and biooxidation of S_n^{2-} and S^0

The $\text{MCE}^*\text{-S}_n^{2-}$ displayed a higher surficial Q (up to 0.77 mC) than the $\text{MCE}^*\text{-S}^0$ (up to 0.44 mC) after 12 and 24 h of SOM biooxidation activity (Figure 1a), due to the lability of -S_n^{2-} compared to -S^0 , as it was also indicated by the EN analyses: Lower R_n (up to $2.15\text{ m}\Omega/\text{cm}^2$) during the S_n^{2-} biooxidation; this process is clearly microbiological, since the R_n of the corresponding abiotic control increased significantly ($14.5\text{ m}\Omega/\text{cm}^2$, $p < 0.05$) (Figure 1b). The fast biooxidation of S_n^{2-} is also evidenced by the Raman spectra, which only showed chalcopyrite on the $\text{MCE}^*\text{-S}_n^{2-}$ at 12 and 24 h (Figure 2a). Finally, the $\text{MCE}^*\text{-S}_n^{2-}$ was poorly colonized by bacteria (Figures 3a and b), and a hydrophilic surface persisted (Figures 3a' and b'). The previous results suggest that S_n^{2-} favors incipient attachment of bacteria (scant biofilm formation), and that biofilms are formed once S_n^{2-} has been oxidized to S^0 , in direct relationship with SRS reactivity (Figure 1).

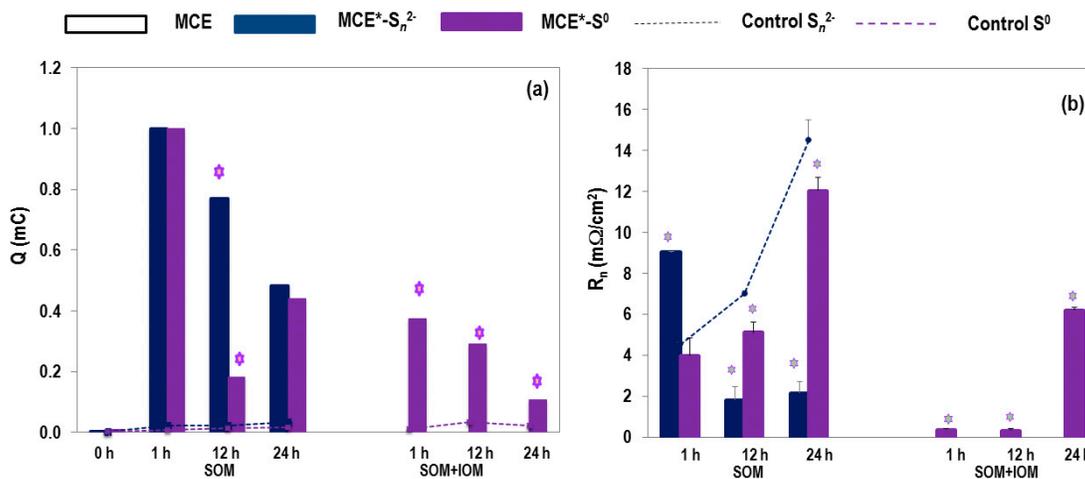


Figure 1. Changes the interfacial charge transfer or Q (a), and resistance to noise, R_n (b) in $\text{MCE}^*\text{-S}_n^{2-}$ (blue) and $\text{MCE}^*\text{-S}^0$ (purple) surfaces after biooxidation assays with SOM (left) and SOM+IOM (right) at the assayed times (0 h or initial condition, 1, 12 and 24 h). Data: Average values ($n = 3$) and standard deviation (error bars). *: Values significantly different ($p < 0.05$) for both, control and surficial SRS (-S_n^{2-} or -S^0).

In contrast, the R_n of the $\text{MCE}^*\text{-S}^0$ exposed to SOM during 1 and 12 h did not show significant differences (Figure 1b). At such times, the S^0 remained at the MCE^* surface (Figure 2c). The bacteria

were extensively attached on the MCE*-S⁰ surface from the first hour of assays (Figure 3c), indicating a rapid colonization that continued throughout 24 h. After 24 h, a biofilm with EPS was detected (Figure 3d) as well as a hydrophobic MCE*-S⁰ surface (Figure 3d'). Also, the non-attached biomass increased (Figure S2c) due to the SRS biooxidation.

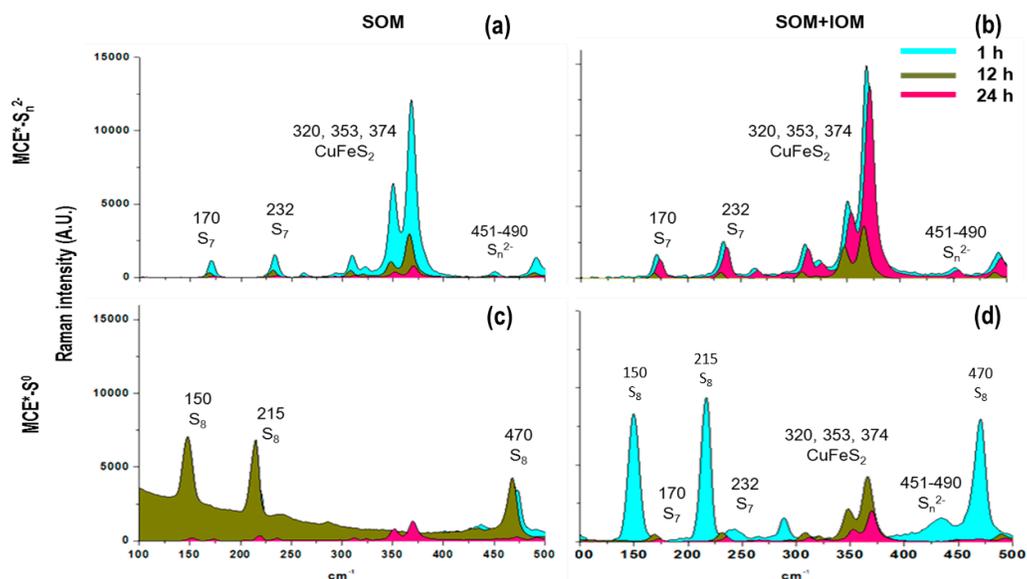


Figure 2. Identified Raman transitions peaks in terms of main surficial SRS on MCE*-S_n²⁻ interacting with SOM (a) and on MCE*-S⁰ with SOM and SOM+IOM (b and c) after the assayed times: 1 (blue), 12 (red) and 24 h (green). Abiotic controls (d).

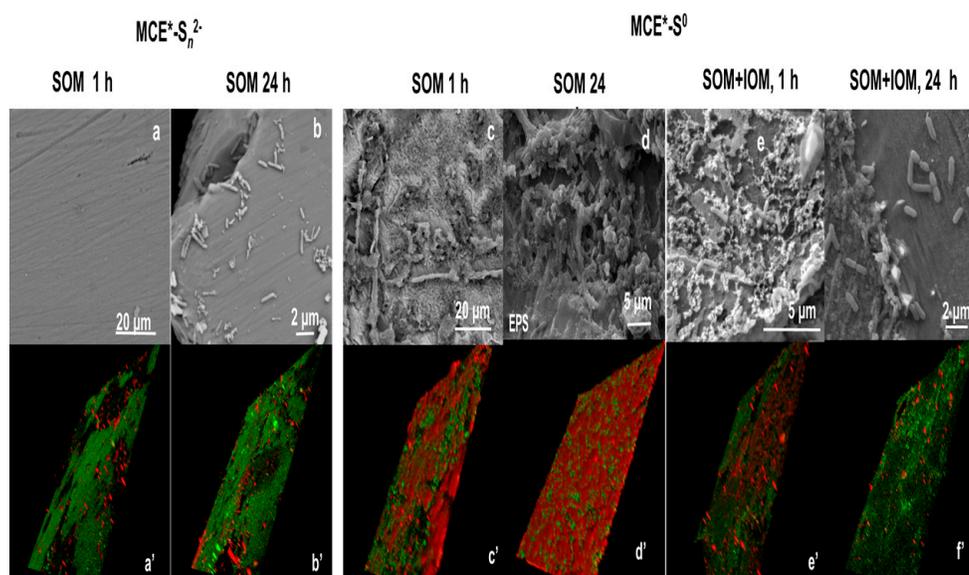


Figure 3. ESEM and CLSM images of the MCE*-S_n²⁻ with SOM (a-b and a'-b'), and of the MCE*-S⁰ with SOM (c-d and c'-d') and SOM+IOM (e-f and e'-f') after 1 and 24 h. Epifluorescence (CLSM) of hydrophilic compounds as exopolysaccharides is shown in green, and that of hydrophobic compounds is shown in red.

3.2. Biooxidation of MCE*-S⁰ by SOM and IOM

Because of the SOM biooxidation of the MCE*-S_n²⁻, we only analyzed the electrochemical behavior of the MCE*-S⁰ exposed to the mixed SOM+IOM and its abiotic control. In the MCE*-S⁰ exposed to SOM+IOM, the interfacial Q decreased from 0.38 to 0.11 mC (Figure 1a). However, comparing the SOM and the SOM+IOM assays, the R_n is the lowest using both species: 3.6 mΩ/cm² in average after 1 and 12 h, and 6.2 mΩ/cm² at 24 h (Figure 1b). The lowest Q and R_n confirm that the simultaneous activity of SOM and IOM improves S⁰ biooxidation. These facts indicate that the fate of SRS produced in assays, including IOM (affecting SOM activity), results from continuous surface pitting (Figures 3a to 3f) associated to IOM and chalcopyrite interactions.

The CLSM and SEM results showed that the number of attached SOM+IOM cells on the MCE*-S⁰ decreased after 24 h (Figures 3e and f), and the interface became hydrophilic (Figure 3f'). The number (per mL) of non-attached SOM+IOM was high at 1 and 24 h, but decreased significantly after 12 h (Figure S2c).

The acidity and ORP of the bulk media significantly increased after 24 h of assays with SOM+IOM; significant differences ($p < 0.05$) were found between the biotic and abiotic controls (Figure S2). The bioleaching of Cu and Fe initiated since the first hour of exposure of the MCE*-S⁰ to the microorganisms, but the Fe bioleached first and after 12 h, the Cu bioleaching initiated.

3.3. EIS analyses of MCE*-S⁰ with SOM or SOM+IOM biofilms

Figure 4 shows experimental and fitted Nyquist spectra obtained for all type of MCE surfaces in M1 and M1+M2. As observed, a semi-circle with larger real components, compared to the imaginary ones, was recorded for low frequencies and is followed by a diffusive component (W or Warburg type). The diagrams collected for all type of samples were fitted to an electrical circuit denoting two times constant in the EIS measurements (Figure 4):

$$R_{sol}*(C_{layer} - R_{layer}*(C_{dl} R_{ct} * Z(\omega)))$$

where R_{sol} is the resistance of the solution (electrolyte), C_{layer} and R_{layer} are the effective capacitance activity and resistance in the MCE*-S⁰/biofilm interface, respectively. Specifically, R_{layer} is associated to the interface thickness. C_{dl} and R_{ct} represent the charge redistribution and resistance on the interface, respectively, and thus, the (bio)oxidative activity on the MCE* without S⁰. Finally, Z(ω) (or W) is the Warburg impedance that indicated the infinite diffusion process or mass transport.

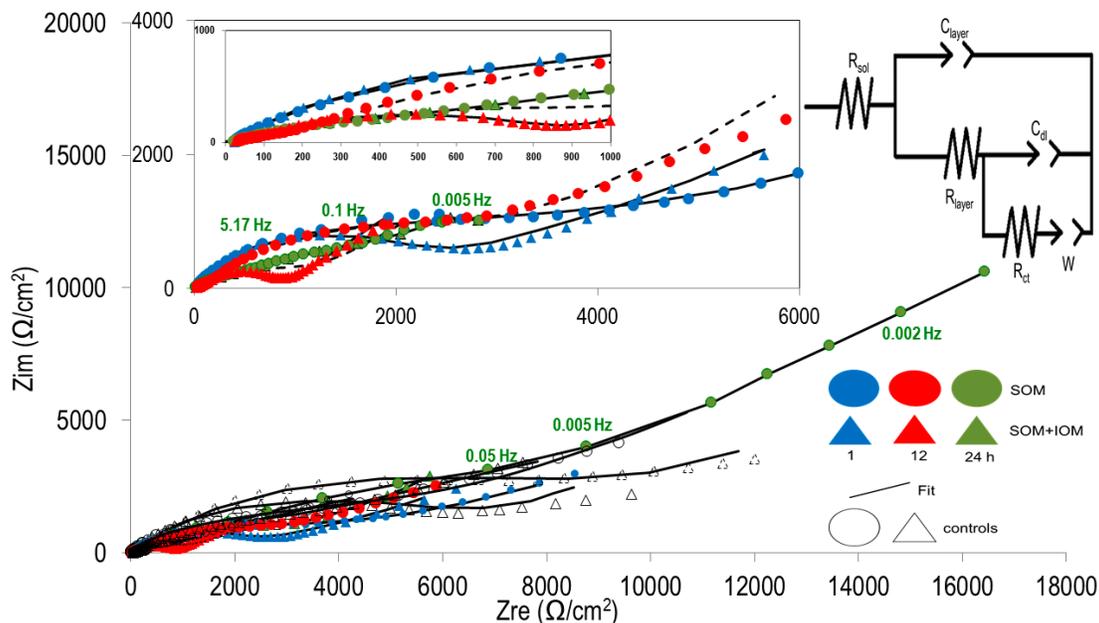


Figure 4. Nyquist diagram for MCE*-S⁰ in presence of SOM (triangle) and SOM+IOM (circle) after the assayed times: 1 (blue), 12 (red) and 24 h (green). The equivalent circuit is also presented. R_{sol} : Resistance of the solution (electrolyte); C_{layer} and R_{layer} : Effective capacitance activity and resistance on the MCE*-S⁰/biofilm interface, respectively; C_{dl} and R_{ct} : Charge redistribution and resistance on the interface, respectively; W : Warburg impedance.

In general, the activity of soluble ions (R_{sol} values) was lower (15.3 to 20 Ω/cm^2) in biotic assays than in abiotic controls (up to 37.2 Ω/cm^2) and tended to decrease (Table 1). Similarly, the increments of C_{layer} and R_{layer} at 24 h were significantly different comparing biotic and abiotic assays, and between the initial (1 h) and final (24 h) assayed times (Table 1). Consequently, the final R_{ct} , C_{dl} , and $Z(\omega)$ values are lower, mainly after SOM interaction with the MCE*-S⁰ surface (Table 1). Thus, all the MCE* surfaces show the same Nyquist contributions (capacitance C_{layer} , and Warburg impedance, W) (Figure 4). However, in presence of SOM biofilms, the diffusive parameter is enhanced, suggesting a more important activity of mass transport due to the porosity of this SRS (Figure 5b). The decrease of W for SOM+SOM assays suggests a lower diffusion process occurring at the interface, which seems to be related to a rapid access of IOM interacting with surface, in agreement with the change of hydrophathy previously determined by CLSM (Figures. 3a' to 3f') and EN analyses (Figure 1b). Note that Figure 4 and Table 1 allow the identification of the main stages in biooxidation mechanisms, in conjunction with EN, CLSM, Raman and SEM analyses.

Table 1. EIS parameters after adjusting the equivalent circuits for biotic assays of SOM or SOM+IOM interaction with MCE*-S⁰ and their corresponding abiotic control. Data: Average, n = 3.

	time (h)	R _{sol} (Ω/cm ²)	R _{layer} (Ω/cm ²)	C _{layer} (1x10 ⁻⁶)	R _{ct} (Ω/cm ²)	C _{dl} (1x10 ⁻⁶)	Z(ω) or W (kΩ/cm ² s ^{0.5})
SOM (control)	1	20 (12.3)	480 (650)	5.14 (0.002)	12453 (9445)	6200 (4.6)	1.6 (0.7)
	12	20 (9.3)	195 (18.5)	2.1 (1.63)	3110 (1298)	0.93 (1.7)	2.0 (5.2)
	24	17.0 (22.7)	3607 (650)	18.4 (27)	4930 (12930)	160 (39000)	0.7 (5.3)
SOM+IOM (control)	1	20 (5.0)	2114 (129)	11.8 (0.1)	6830 (7307)	22000 (0.13)	2.0 (1.6)
	12	16.1 (18.4)	141 (396)	2.8 (1.04)	701 (9942)	0.19 (0.2)	7.0 (1.2)
	24	15.3 (37.2)	758 (286)	28.4 (10.6)	1815 (6368)	830 (0.5)	3.5 (0.3)

Table 2. Diffusivity values (D) for soluble H⁺, Fe²⁺ and Fe³⁺ (Equation (1)) of assays with MCE*-S⁰ subjected to SOM or SOM+IOM activity and their corresponding abiotic control. Data: Average (n = 3).

Time (h)	SOM			IOM+SOM		
	1	12	24	1	12	24
DH ⁺ mm ² /s (control)	0.006 (0.001)	0.004 (0.05)	0.01 (0.04)	0.002 (0.001)	0.34 (0.002)	0.34 (0)
DFe ²⁺ mm ² /s (control)	0 (0)	0 (0)	0 (0)	0.001 (0)	0.016 (0)	0.007 (0)
DFe ³⁺ mm ² /s (control)	0 (0)	0 (0)	0 (0)	0.37 (0.52)	6.20 (0.18)	1.50 (0)

4. Discussion

The analysis of the culture media, MCE* surface, as well as of the interface biofilm/chalcopyrite confirmed that the biooxidation of both S_n²⁻ and S⁰, previously formed on MCE by electrooxidation, was faster than the abiotic oxidation, while the electrochemical analyses allowed to disclose these processes, suggesting that the charge transfer due to the biocorrosion is mainly uniform; thus, the chalcopyrite and cells interactions are controlled by a quasi-homogeneous distribution of SRS.

According to Fowler et al. [12] and Holmes and Crundwell [13], “the S_n²⁻ layers are not inherently passivating”. Even a low Q transfer is enough to remove S_n²⁻ in the presence of SOM (Figure 1b) due to the characteristics of S_n²⁻, a partially dissolved, hydrophilic, linear and lower bond length than that of cyclic S_s (2.04 and 2.057 Å for S_n²⁻ and S⁰, respectively) [14]. The hydrophilic character of S_n²⁻ configures a hydrophilic surface (Figures 3a' and b'), in which few SOM cells attach after 24 h (Figure 3b). Meanwhile the values of Q and R_n indicate that SOM attachment to the hydrophobic S⁰ is an energy-dependent process, and the bacteria essentially

activate the S^0 (Figures 1b and c); the S^0 is prone to biooxidation. Thus, the SOM attachment and S^0 activation are the limiting steps. In other words, the greater hydrophobic surface overridden mainly by S^0 compounds, the greater effect on bacteria attachment [15,16].

With both microorganisms, the SOM *A. thiooxidans* and the IOM *Leptospirillum* sp., exposed to the MCE*- S^0 , the decay of R_{sol} and the increase of C_{dl} (Table 1) were more notable; the SRS were removed since the first hour (Figure 2d), and a greater attachment of bacteria was observed (Figure 3e). That means that the C_{dl} increases because of the hydrophobic S^0 , but also probably because of charged species in the biofilms. According, we suggested that the energy is stored in the EPS of biofilm as a pseudocapacitor that does not involve Q transfer, but capacitive current. Because of the pseudocapacitance, the reaction rate diminished as indicated by the R_{sol} (Table 1).

Accordingly, the R_{layer} increased when we assayed only the SOM (0.48 to 2.8 $k\Omega/cm^2$; Table 1), because the active microorganisms gradually removed the SRS, shaping a thick resistivity interface with SRS agglomerates (Figure 5b) that limits the H^+ diffusion (Table 2) and charge and mass transfer. Consequently, the final R_{ct} , C_{dl} , and W values were lower in the SOM and MCE*- S^0 assays (Table 1) in absence of *Leptospirillum* sp. that regenerates Fe^{3+} . Specifically, the lowest R_{ct} in SOM+IOM assays demonstrated that the IOM *Leptospirillum* sp. favors the biooxidation of MCE*.

Significant differences of SOM attachment have been reported as a consequence of the presence of IOM in a bi- or multi-specific cultures [5,17,18], since its presence shapes a progressive reactive surface due to the regenerated Fe^{3+} , which accelerates SRS biooxidation and chalcopyrite dissolution [19]. In this work, the EIS also confirms that the SOM+IOM biofilm is electrochemically more active and that the removal of SRS increases the mixed potential [13]. With both bacteria the biooxidation of S^0 initiates since the first hour, when the Q transfer (Figure 1b) and the W values increase (Table 1). Therefore, the surface is less resistive (Figure 1c) and modifies the semiconductor character of the MCE*.

The higher W value of the SOM+IOM assays (up to 7.0 $K\Omega/cm^2s^{0.5}$; Table 1) showed that the overall biooxidation rate of the MCE*- S^0 is controlled by mass transfer due to SOM activity. Additionally, the lower pH in the MCE* exposed to SOM+IOM may result as a consequence of a higher diffusivity of H^+ to the interface that may alter the pH in the formed biofilm [12,13]. Thus, the removal of reactive SRS enhances the diffusion of chemical species through its porous and spongy structure (Figure 5c) that improved the Cu bioleaching kinetics by diffusion process [20,21]. This porous and spongy SRS are characteristic of a catalytic process and may be formed because of vacancies or point defects during the biooxidation [22].

The electrochemical analyses indicated that S^0 and EPS hindered the charge (as pseudocapacitor), as it was observed by Bevilaqua et al. [23]. Although both SOM and SOM+IOM biofilms were formed on MCE*- S^0 , the different electrochemical dynamics between them are associated to the role of Fe^{3+} ions into the biofilms, as Rohwerder and Sand [24] have indicated: The Fe^{3+} acts as an electron acceptor or electrophile and may form Fe-EPS complexes that enhance the biooxidation. This mechanism explains the performance of IOM, since the Fe-EPS complexes could be considered the main intermediate adsorbent on S^0 for charge transfer, as it is proposed in the model shown in Figure 5a. The SOM acts over the interfacial SRS, permitting the Fe^{3+} diffusion, while the overall SOM+IOM activity modifies the bacterial density, the biofilm thickness, and the surficial hydrophobicity; all these phenomena occur within the first 24 h of interaction. (1) In SOM cultures exposed to MCE*- S^0 , the biofilm/mineral interface has mainly hydrophilic compounds, like exopolysaccharides that are essential for the initial attachment of cells on the surface [24]. (2) The behavior of SOM and SOM+IOM biofilms on the MCE*- S^0 surface seems to be controlled by hydrophobic residues (organic and inorganic); the hydrophobic and less reactive S^0 enhances the bacterial attachment via hydrophobic interactions. (3) Only in SOM+IOM assays, a

thinner (ca. 37 mm) biofilm/mineral interface shifts from hydrophobic to hydrophilic character when S^0 was completely activated to linear SRS as S_n^{2-} , by the Fe^{3+} regenerated by IOM (Figure 5a). Hence, we would like to emphasize that the hydrophathy character of the interface “determines the free energy of the adhesion process” [16]. All these mechanisms occur by (i) allowing SOM colonization and (ii) by IOM affecting SOM and setting up a more accessible hydrophilic surface to induce corrosion pittings.

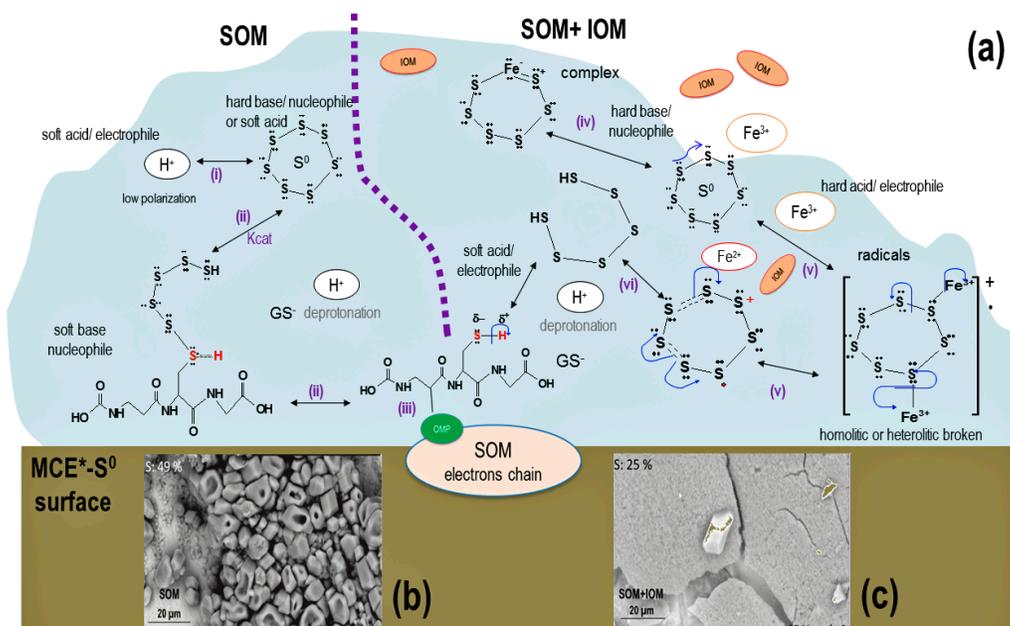


Figure 5. Proposed model of S^0 (bio)oxidation by *A. thiooxidans* alone or with the IOM, *Leptospirillum* sp. (a): (i) Cyclic S^0 behaves as a hard base; thus, the presence of H^+ , is not sufficient to favor a reaction affinity. (ii) Although the deprotonated glutathione (due to the acid medium) GS^- acts as a soft and nucleophilic base and may form $GS-S_nH$, it has been reported that K_{cat} for this reaction is very low. (iii) $GS-S_nH$ is transported to the SOM periplasm via the outer membrane protein (OMP). GSH catalyst is regenerated. In presence of IOM, S^0 as a hard-base, shows affinity for the produced Fe^{3+} that is also a polarizing agent. Thus, Fe^{3+} may induce a S^0-Fe^{3+} complex formation and/or (iv to vi) charge delocalization, until rise to linear S_n^{2-} that interacts with the GS^- to form $GS-S_nH$, available for SOM, as already mentioned (iii). Agglomerated SRS (b) and thin and spongy layer (c) after SOM and SOM+IOM biooxidation of MCE*, respectively [15,29,30,32,33].

Currently, there is a limited research on electron transfer between Fe^{3+} and SRS. According to Rohwerder et al. [25] and Yin et al. [26], the activation of S^0 from cyclic to linear SRS proceeds via a nucleophilic attack of glutathione (GSH) and sulfonate (-SH) groups found in the outer-membrane protein of SOM. We hypothesize that Fe ions also favor such activation of the linear structures (Figure 5a), as the results of this and other works suggested, as well as the Hard-Soft Acid Base Theory studies on solid sulfur allotropes and sulfur oxidation [15,26 - 33]. This hypothesis should be proved and it is the subject of our future work.

The described effects due to the initial S_n^{2-} or S^0 and the presence of IOM, occurred just during the first 24 h. Afterwards, an attachment-detachment-attachment cycle is expected, as Echeverría-Vega and Demergasso [16] demonstrated to be regulated by a feedback control. The presented results

have an industrial interest, since the addition of S^0 increases the Cu bioleaching rate and produces an acidity [34,35] that modifies the interfacial pH [36].

5. Conclusions

This study on charge and mass transfer in the biofilm/mineral interface demonstrates that SRS and chalcopyrite biooxidation by *A. thiooxidans* (SOM) and *Leptospirillum* sp. (IOM) initiates in early stages of the interaction (1 to 24 h), and it is controlled by (1) the diffusion and adsorption of ions into EPS and (2) the transfer of ions and electrons from the mineral interface to the solution.

At the studied early times, the SOM modifies the reactive properties of surficial sulfurs and favors an acidic dissolution, which shifts into a ferric dissolution when IOM is presented. *Leptospirillum* sp. enhances the chalcopyrite dissolution and allows surpassing the charge-transfer barrier (reactivity) between the mineral interface and ferric ions. In such context, the observed changes of hydrophathy on the different interfaces are associated with the activity and transfer of ions and electrons.

Bacteria attachment is due to self-organization by a particular bioelectrochemical evolution on the interface: Changes in the mixed potential are triggered from the beginning of the bacteria/chalcopyrite interaction (1 h), when the biofilm is not constituted; the bacteria/chalcopyrite interaction depends on the surficial characteristics that pivoted the bacterial attachment to the mineral. As the chalcopyrite dissolution progresses, a surficial biofilm is formed and is dynamically reconfigured: On the hydrophilic $MCE^*-S_n^{2-}$ a less extended bacterial attachment was observed, compared with that on MCE^*-S^0 , where the SOM+IOM presence shifts the surface from hydrophobic to hydrophilic as a response of the electrochemical changes while the chalcopyrite is biooxidized.

Supplementary Materials: The following are available online, **Figure S1:** Voltammograms in positive (a) and negative (b) potential sweep of pristine MCE (dotted line), and electrooxidized EMC at 695 mV, $MCE^*-S_n^{2-}$ (gray line), and at 915 mV, MCE^*-S^0 (black line), in ATCC-125 medium pH 2. Scan rate: 20 mV/s with stirring; the potential scan was initiated at the OCP., **Figure S2:** Certain characteristics of the media as electrolyte (after 1, 12 and 24 h of the exposure of $MCE^*-S_n^{2-}$ (blue) and MCE^*-S^0 (purple) to SOM or SOM+IOM media (abiotic controls; dotted lines) and cultures (biotic; columns): pH (a); ORP (b), soluble Fe^{3+}/Fe^{2+} (c), and biomass of non-attached bacteria (d). Data: Average values ($n=3$) and standard deviation (error bars). *: Values significantly different from both, controls and surficial SRS ($-S_n^{2-}$ or $-S^0$).

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Conflicts of Interest: The authors declare no conflict of interest.

Appendix A

Mineral coupons of 1.0 cm² were coated with Cu via electrolytic deposit using a CuSO₄ solution, to improve the current distribution; a Cu wire was added with a silver solder to enhance the electrical

contact of the MCE. Finally, the electrode was imbibed in epoxy resin and the exposed MCE surface was polished with a water sandpaper until it reached a mirror-like surface condition. The MCE were maintained in desiccators under anaerobic conditions until their use.

The potentiostatic modification was conducted with an EPSILON BAS 2.10.73 potentiostat at 25 °C, in a typical electrochemical, three-electrodes device; the MCE was used as working electrode, the reference electrode was a saturated sulfate electrode, Hg/Hg₂SO₄ (0.615 V *vs.* SHE, the standard hydrogen electrode), and a graphite rod (Alfa Aesar, 99.9995% purity) was used as counter-electrode. These MCE* were then achieved by application of anodic pulse (E_{an} , 3600 s); ATCC-125 culture at pH 2.0 was the electrolyte, hence emerging S⁰ and S_n²⁻ compounds (sulfur-reduced species, SRS), as a function of the applied E_{an} , from 0.36 to 1.015 V *vs.* SHE.

The Raman spectra were recorded with a triple subtractive monochromator (T64000 Jobin Yvon spectrometer) equipped with a confocal microscope, Olympus BH2-UMA ($\lambda = 514$ nm). At least 10 spectra were recorded for each MCE* surface. Calibration was done using a Si wafer, which showed a single peak at 521 cm⁻¹. The noise/signal ratio was better than 100.

After the CV and Raman analysis of the MCE, anodic E_{an} of 695 mV and 915 mV were chosen to electrogenerated two different SRS, since at these E_{an} was observed minor electrooxidation and low activation current (anodic peak a1 at the open circuit potential, OCP; Fig. S1a and b). Raman peaks for these MCE* indicate the predominance of S_n²⁻ and heptagonal sulfur S₇ for $E_{an} \leq 695$ mV; such electrode was referred as MCE*-S_n²⁻. Octagonal sulfur S₈ was detected at $E_{an} \geq 915$ mV, for MCE*-S⁰. The reactivity j_{act} and of the MCE*-S_n²⁻ was significantly lower than of the MCE*-S⁰ (Fig. S1 c and d); thus, more energy is necessary to (bio)oxidize the MCE*-S⁰ than for MCE*-S_n²⁻ surface.

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