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Article

# Green Manufacturing of Rutile (TiO<sub>2</sub>) Welding Electrodes with Blast Furnace Slag

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# **Abstract**

This study develops a sustainable welding approach by incorporating 35–50% blast furnace slag (BFS), a byproduct of the steel industry, into rutile-type electrode coatings. Electrodes were fabricated by dry mixing BFS with fluxes, adding potassium silicate binder to form a paste, pressing at 150 bar onto a 3.25 mm core wire, and heat treatment at 150°C for two hours. Weld quality and performance were evaluated through visual inspections, microstructure and XRD analyses, hardness, tensile, and impact tests. Visual inspections confirmed weld quality comparable to commercial standards, with stable arc and minimal spatter. Microstructure analysis revealed a ferrite-dominated weld metal with TiO<sub>2</sub> and FeTiO<sub>3</sub> phases in the slag layer, enhancing strength and toughness. Electrodes with 35–40% BFS achieved yield strength of 477–482 MPa, tensile strength of 570–573 MPa, and impact energy of 58–59 J at 0°C, complying with ISO 2560:2020. BFS integration reduced CO<sub>2</sub> emissions by 0.28–0.4 kg per kg of coating and diverted 200–600 kg of slag per ton of steel from landfills. Coating and raw material costs decreased by 33–48% and 15–25%, respectively, aligning with the EU Green Deal's circular economy goals and enhancing weld quality and sustainability.

**Keywords:** green manufacturing; blast furnace slag; Rutile (TiO<sub>2</sub>) electrode coatings; sustainable welding; circular economy

# 1. Introduction

The steel industry underpins contemporary global development, driving advancements in infrastructure, manufacturing, and sustainable material innovation. With an annual production of approximately 2 billion tons, steel remains vital for construction and transportation [1]. However, blast furnace refining generates 200–600 kg of blast furnace slag (BFS)—primarily silicates and oxides—per ton of crude iron [2]. These slags contribute to land degradation, greenhouse gas emissions, and lost economic value, necessitating innovative recycling strategies.

BFS is valued for its chemical composition and versatile applications, supporting sustainable industrial practices. It is repurposed in construction (e.g., bricks, Portland cement, road aggregates) and infrastructure (e.g., railway ballast, port fill) [3]. BFS also functions as a pH stabilizer in bio-oxidation for acid mine drainage [4–6], and as a precursor in advanced materials such as geopolymers and composite binders [7–10]. For instance, slag-based geopolymer matrices are used to repair damaged concrete [11]. Nonetheless, significant quantities remain underutilized—Türkiye alone discards ~55,000 tons annually [12]. Environmental concerns, including radiation emissions from SMAW welding [13], further emphasize the need for recycling innovations aligned with the EU Green Deal [14].

Integrating BFS into rutile-type coated electrodes offers a novel, underexplored route to sustainable welding. Rutile electrodes—known for arc stability, weld quality, and slag removal—depend on flux components (SiO<sub>2</sub>, CaO, MgO, TiO<sub>2</sub>, Al<sub>2</sub>O<sub>3</sub>, MnO) that are abundant in BFS [15–20]. Prior studies on dissimilar welds, structural steel joints, and hardfacing layers show that tailored coatings enhance mechanical properties [21–24]; for example, buttering improves wear resistance in low-carbon steel [23], and optimized parameters strengthen armor steel joints [24]. BFS's

compatibility with flux materials suggests strong potential in electrode coatings, with studies reporting improved durability [25–27]. However, BFS use in rutile electrodes remains limited due to compositional variability, processing challenges, and proprietary constraints [25,26,28–33].

This study investigates the feasibility of incorporating 35–50% BFS as a flux component in rutile-type coated electrodes to promote green manufacturing. By reducing virgin flux material consumption, it supports the EU Green Deal's climate neutrality goal by 2050 [14]. Welding performance, microstructure, phase composition, chemical content, and mechanical properties are analyzed to evaluate the effectiveness of BFS-containing coatings. Building on prior research in welding and sustainable materials [29–35], this work contributes to slag recycling, waste valorization, and eco-friendly welding for sustainable manufacturing. Recent studies have demonstrated the potential of biobased magnetic composites [36], foam glass structures derived from silicate-rich residues [37], and geopolymer matrices enhanced with recycled additives [38], reinforcing the viability of circular economy principles in materials engineering.

#### 2. Materials and Methods

# 2.1. Raw Materials Used in the Rutile-Type Coating

The primary raw material in this study was blast furnace slag (BFS), a by-product sourced from Iskenderun Iron and Steel Inc., Türkiye, and supplied by Ciner Construction and Transportation Ltd. Co. The slag, provided in both rock and granulated forms, was ground to a particle size of -100 to +50 microns for integration into the rutile-type coating formulation. Visual representations of the BFS forms are shown in Figure 1.

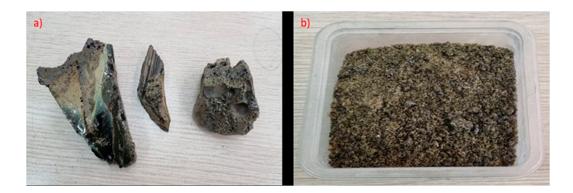


Figure 1. Blast furnace slag sample.

Preliminary tests indicated that the slag's morphology—rock-like or granular—had a negligible impact on the coating production process or the mechanical properties of the resulting weld metal. The average component ratios of the BFS are presented in Table 1.

Table 1. Chemical composition of blast furnace slag used in rutile-type electrode coatings.

Material	SiO <sub>2</sub>	CaO	$Al_2O_3$	MgO	TiO <sub>2</sub>	$Na_2O + K_2O$	Fe <sub>2</sub> O <sub>3</sub>	MnO	S <sup>2-</sup>	PO <sub>4</sub> 3-	Rest
BFS	39	37.5	12.4	F (+0.1)	1.6	1.1	0.0 (+0.1)	0.4 (+0.5)	0.06	0.05	Othors
DFS	$(\pm 0.5)$	$(\pm 0.4)$	$(\pm 0.2)$	3 (±0.1)	$(\pm 0.2)$	1.1 (±0.05)	0.9 (±0.1)	0.4 (±0.5)	(±0.01)	(±0.01)	Otners

Note: Trace elements include minor components not individually listed.

In addition to BFS, the rutile-type coating formulation included essential raw materials aligned with industry standards [16–20]:

• Slag formers: Feldspar (KAlSi<sub>3</sub>O<sub>8</sub>) or quartz (SiO<sub>2</sub>), rutile (TiO<sub>2</sub>) or ilmenite (FeTiO<sub>3</sub>), enhancing coating flux properties.

- Gas formers: Calcite (CaCO<sub>3</sub>) or dolomite (CaMg(CO<sub>3</sub>)<sub>2</sub>), and cellulose (C<sub>6</sub>H<sub>10</sub>O<sub>5</sub>), generating a protective gaseous atmosphere during welding.
- Deoxidizer and alloying agent: Low-carbon ferromanganese (FeMn, 75% Mn, 0.1–0.5% C), refining and improving weld metal properties.
- Binders: Sodium silicate (Na<sub>2</sub>(SiO<sub>3</sub>)<sub>n</sub>O) or potassium silicate (K<sub>2</sub>SiO<sub>3</sub>), ensuring robust adhesion of the coating to the core wire.
- Iron powder: 99% pure iron powder, enhancing coating efficiency and weld deposition.

The core wires, 3.25 mm in diameter and 350 mm in length, were produced via cold rolling from wire rod, in accordance with DIN 17145 standards. The chemical composition of these wires, supplied by Tosyalı Holding (Türkiye), is detailed in Table 2. For welding tests and coating performance evaluation, SAE 1020 steel plates ( $20 \times 100 \times 3$  mm) were used as the substrate material, with their chemical composition also provided in Table 2.

Table 2. Chemical composition of core wire and SAE 1020 substrate.

Chemical Composition (wt.%)									
Material	C	Si	Cu	Mn	P	S	Ni	Cr	Fe
Core wire	0.08	0.022	0.089	0.544	0.009	0.011	0.046	0.041	Bal.
SAE 1020	0.20	0.001	-	0.400	0.026	0.022	-	0.064	Bal.

#### 2.2. Preparation of Blast Furnace Slags for Coating Integration

The BFS samples were ground and preheated at 1200 °C for 2 hours to optimize their integration into the rutile-type coating formulation. This preprocessing step aimed to reduce sulfur and phosphorus content, which—if excessive—could impair weld ductility and compromise adherence to industrial coating standards. Preheating enhanced the mechanical properties of the weld metal, including tensile strength and impact resistance, by promoting the evaporation, oxidation, and decomposition of sulfur and phosphorus. This minimized weld metal contamination and ensured consistent electrode coating performance [39–42].

#### 2.3. Production of Rutile-Type Coated Electrodes

Rutile-type coated electrodes are widely used in shielded metal arc welding due to their versatility and ability to deliver consistent weld quality. The rutile-based coating works in conjunction with the core wire to provide arc stability, alloying elements, and protection to the weld pool. In this study, a mold-pressing method was selected for electrode production due to its costeffectiveness, precision, and suitability for laboratory-scale fabrication, compared to conventional industrial methods such as extrusion and dipping [16–20,43–45]. Electrodes were produced with a 3.25 mm core wire diameter, 35 cm length, 5.17 mm total diameter, and 1.92 mm coating thickness, weighing 23 g for the core wire and 12 g for the coating, totaling 35 g per electrode. While extrusion and dipping are common in industrial applications, mold-pressing was adopted here for its practicality and alignment with extrusion-based techniques. The dipping method, which relies on surface tension for adhesion, often results in non-uniform coating thickness and potential separation of heavy metals in the liquid phase. Achieving thicker coatings via dipping requires multiple cycles, increasing production time and cost. In contrast, the mold-pressing method uses a paste-like coating formulation, allowing precise control of coating thickness through mold adjustments and delivering homogeneous coatings in a single cycle. Controlled pressure within the molds ensured uniform thickness and dimensional accuracy. The core wire conducts welding current, facilitates arc formation, and melts to form the weld metal, while both the wire and coating contribute alloying elements to the weld deposit. Electrode coating production began with the homogenization of dry materials in a Z-bladed rotary mixer for 15 minutes. A binder (20–30% by weight) and a small amount of water were added, followed by 15 minutes of wet mixing to achieve a paste-like consistency. The coating paste and core wires were loaded into a mold-pressing machine lined with transparent plastic

sheets to prevent adhesion. Pressing was conducted at 150 bar to achieve the desired thickness. Excess material was removed, and pressing/cleaning steps were repeated three times for uniformity. Coated electrodes were air-dried at room temperature for 24 hours, followed by heat treatment at 150 °C for two hours in a Magma Therm laboratory furnace to enhance mechanical integrity and coating stability. The final flux ratios of the coated electrodes are presented in Table 3, and the production flowchart is illustrated in Figure 2. Incorporating 35–50% BFS by weight represents the highest feasible proportion for rutile-type coated electrodes, ensuring coating integrity and optimal welding performance.

			, 1		
		Flux ratios	s (wt.%)		
Electrode Code	Blast	Rutile	Other flux	Binder	Water
(Rutile)	furnace slag	(TiO <sub>2</sub> )	components	(K <sub>2</sub> SiO <sub>3</sub> )	water
R1	50	35-50	0-15	20-25	1-2
R2	45	35-50	0-15	20-25	1-2
R3	40	35-50	0-15	20-25	1-2
R4	35	35-50	0-15	20-25	1-2

**Table 3.** Formulation ratios of rutile-type coated electrodes.

Note: Other flux components include silicates ( $SiO_2$  or feldspars), carbonates ( $CaCO_3$ ), cellulose ( $C_6H_{10}O_5$ ), iron powder, and deoxidizer (FeMn). Binder and water are added as percentages after the total mixture is prepared as 100%.

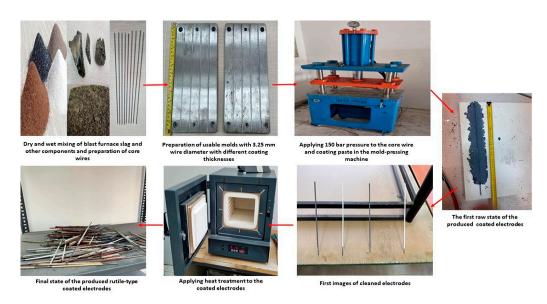


Figure 2. Production process of rutile-type coated electrodes.

# 2.4. Welding Tests of Rutile-Type Coated Electrodes Containing Blast Furnace Slag

The rutile-type coated electrodes, produced via the mold-pressing method, were tested on  $20 \times 100 \times 3$  mm SAE 1020 steel plates to evaluate welding and coating performance. Welding tests were conducted using a Magmaweld electric arc welding machine, employing the bead-on-plate technique for initial characterization. This method enabled visual inspection of weld quality and slag behavior derived from the coating. Bead-on-plate samples were used for all characterization processes except tensile and Izod V-notch impact tests, which utilized specially prepared butt weld samples following established methodologies [46,47]. These prior studies validated the suitability of butt weld samples for mechanical testing. Welding parameters and variables affecting coating performance were systematically controlled, with details summarized in Table 4. Preliminary tests identified electrodes with optimal performance, assessed through visual inspection and mechanical analyses. Selected electrodes underwent comprehensive testing to confirm their practical

applicability. The dual approach—bead-on-plate for initial assessments and butt weld samples for mechanical evaluations—minimized external variability and enabled robust characterization of coating performance.

**Table 4.** Welding test parameters of rutile-type coated electrodes.

Parameters	Properties
Electrode Type	Rutile (Thick coating)
Electrode outer diameter (mm)	5.17 mm
Electrode Core Wire Diameter (mm)	3.25 mm
Electrode Core Wire weight (g)	23 g
Electrode coating thickness (mm)	1.92 mm
Electrode coating weight g)	12 g
Coating/Core wire (%)	59
Total electrode length (cm)	35 cm
Welding Machine (type)	Shielded metal arc welding-SMAW
Current (A)	70–100
Welding Angle (°)	45
Pole (±)	(+)
Substrate	SAE 1020
Substrate Dimensions (mm)	20x100x3 mm
Pre-heat for substrate	No
Pre-heat for electrodes	No

# 2.5. Microstructure, XRD, and Chemical Composition Analyses

Microstructural examinations were performed on weld samples (originally  $20 \times 100 \times 3$  mm), cut into  $20 \times 5 \times 5$  mm dimensions (weld metal height ~2 mm) and subjected to metallographic preparation. Sanding was conducted using progressively finer grit papers (80 to 1200), followed by polishing with 3 µm and 1 µm diamond paste. Samples were etched with 2% nital solution (98% ethyl alcohol, 2% nitric acid) for 10 seconds and examined using a Tronic XJL-17AT inverted trinocular metallurgical microscope.

Chemical composition analyses of the weld metal were performed using an Oxford Instruments optical emission spectrometer for elemental quantification, and a Joel JMS-7001F SEM (15 kV) with EDX capabilities for semi-quantitative analysis of trace elements and distribution patterns. XRD phase analysis was conducted using a Rigaku RadB-Dmax2 device with  $\text{CuK}\alpha$  radiation at room temperature to identify crystallographic phases in the weld metal and slag, providing insights into structural and chemical characteristics.

# 2.6. Hardness Measurements

Hardness measurements were conducted on the same  $20 \times 5 \times 5$  mm samples used for microstructural analysis. A Tronic DHV-1000 Digital Vickers Hardness Tester applied a standardized 9.8 N load for consistent evaluation. Measurements were taken from five distinct regions of the weld metal, including the centerline and boundary zones (Figure 3), to account for localized variations. Averaged results provided a comprehensive assessment of hardness distribution, supporting the mechanical evaluation of rutile-type coated electrodes containing BFS.

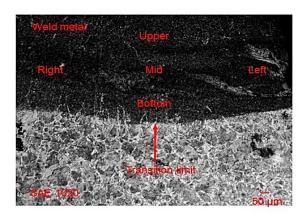


Figure 3. Hardness measurement points on the weld metal.

# 2.7. Tensile and Impact Tests

Mechanical properties of the weld metal—including yield strength, tensile strength, and elongation (A5)—were assessed using a Shimadzu-Autograph AG-X tensile testing machine (50 kN capacity). Butt weld samples were machined to ISO 9018:2015 specifications [48]. Impact resistance was evaluated via Izod V-notch tests in accordance with ISO 148-1:2016 and ASTM E23:2023 standards [49,50]. Samples (<75 mm length, 10 mm thickness) were tested using an Instron/Ceast 9350 impact machine (0.59–757 J capacity) at 0 °C and 20 °C to simulate standard and low-temperature conditions. Results were compared with ISO 2560:2020 requirements for rutile electrodes [51], providing insights into the reliability and adaptability of BFS-containing coatings. The combined tensile and impact tests enabled comprehensive mechanical characterization, correlating with microstructure, hardness, and chemical composition analyses.

# 2.8. Environmental and Economic Impact Assessment

Environmental and economic impacts of incorporating 35–50% BFS into rutile-type electrode coatings were assessed to quantify sustainability benefits. CO<sub>2</sub> emission reductions from substituting virgin flux materials (e.g., SiO<sub>2</sub>, CaO, Al<sub>2</sub>O<sub>3</sub>, MgO, TiO<sub>2</sub>) with BFS were calculated using slag recycling and viscosity analysis methods [46,52–54]. Landfill diversion was evaluated using steel industry waste metrics [55,56], and economic savings were estimated based on regional disposal fees [49,56].

Cost reductions from BFS substitution were determined by comparing market prices of virgin flux materials and energy savings benchmarked against embodied energy reductions in similar recycling processes [57]. These assessments align with the EU Green Deal's goals for climate neutrality and circular manufacturing [14].

# 3. Results and Discussion

#### 3.1. Visual Inspection Results of the Produced Rutile-Type Coated Electrodes

The performance of rutile-type coated electrodes, incorporating BFS, was evaluated through a detailed visual inspection of welds and coating-derived slag properties formed during and after welding. These evaluations were systematically scored based on performance criteria outlined in Table 5(a), focusing on critical parameters such as arc stability, burning characteristics of the core wire, spatter generation, smoothness and uniformity of the coating-derived slag layer on the weld metal, spontaneous slag removal, seam appearance, odor, smoke generation, and pore presence in the weld seam. Each coated electrode's performance was scored on a 1–10 scale, with thresholds defined as follows:

- 1-4: Evaluated as "unsuccessful."
- 5–6: Evaluated as "improvable."



# 7-10: Evaluated as "successful."

This scoring system provided a quantitative framework to assess the suitability of the coated electrodes for practical coating applications, capturing both welding performance (e.g., arc stability, burning behavior) and weld quality influenced by the coating (e.g., seam appearance, slag removal). Figures 4–7 illustrate welding test results for electrodes R1 (50% slag, Figure 4), R2 (45% slag, Figure 5), R3 (40% slag, Figure 6), and R4 (35% slag, Figure 7), showcasing visual appearance, coating-derived slag layers, weld metal characteristics, and post-welding electrode condition. These images complemented the scoring process, offering visual insights into coating performance. The electrode application performance results, summarized in Table 5(b), revealed significant differences in coating behavior and quality, with the scoring system effectively distinguishing superior, moderate, and poor performance based on visual criteria. Electrodes scoring 7–10 demonstrated consistent arc stability, minimal spatter, smooth coating-derived slag formation, and spontaneous slag removal, marking them as highly suitable for practical welding and coating applications.

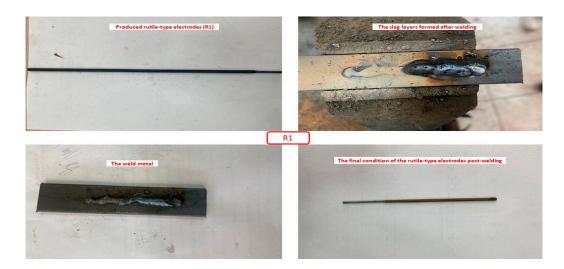


Figure 4. Weld appearance and test results for R1 rutile-type coated electrode (50% slag).

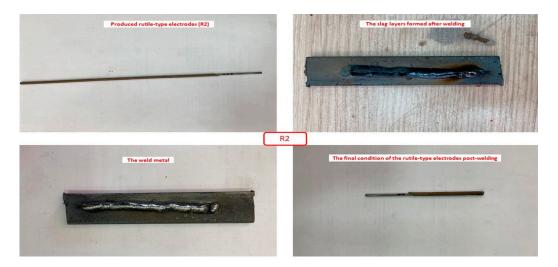


Figure 5. Weld appearance and test results for R2 rutile-type coated electrode (45% slag).

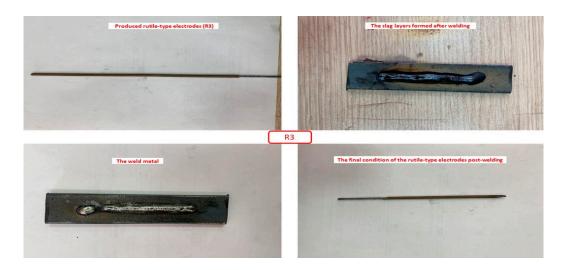


Figure 6. Weld appearance and test results for R3 rutile-type coated electrode (40% slag).



Figure 7. Weld appearance and test results for R4 rutile-type coated electrode (35% slag).

In contrast, electrodes scoring below 5 exhibited deficiencies such as inconsistent arc behavior, excessive spatter, and poor seam appearance, highlighting areas for coating optimization. This visual inspection process provided a comprehensive understanding of the welding behavior and quality of rutile-type coated electrodes, laying a foundation for further development of coating formulations containing BFS. The visual inspection analysis, detailed in Table 5(b), confirmed the successful performance of rutile-type coated electrodes incorporating BFS. During welding tests, arc stability was consistently rated "very good" or "excellent," and the electrodes exhibited "very easy" burning performance, indicating high usability. Spatter formation was minimal, and coating-derived slag smoothness, except for electrode R1, was rated "very good." The slag was spontaneously removed, enhancing operational efficiency. Weld seams displayed a "glossy" to "very glossy" appearance, with low odor and smoke levels during welding. Partial porosity was observed in the coating-derived slag of electrodes R1 and R2, but no pores were detected in other electrodes or weld metals. Scores ranged from 7 to 10, indicating overall success for electrodes R1-R4, with R3 and R4 (35-40% slag) showing superior slag smoothness and minimal porosity compared to R1 and R2. These results align with commercial market standards and literature findings on rutile-type coatings [16-20,43-45,51]. Despite these positive outcomes, certain limitations were noted. Porosity and reduced slag smoothness in electrodes R1 and R2 highlighted areas for coating improvement. The success is attributed to the optimized chemical composition of the coating formulation, ensuring compatibility with slag properties and weld metal characteristics. However, challenges such as operator

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inexperience, which affected electrode coating application techniques and slag formation, electrode eccentricity from the mold-pressing method causing minor uniformity issues, deoxidizer deficiencies in the coating contributing to porosity, and moisture presence impacting combustion and slag behavior, were identified as factors influencing weld quality [16–20,43–45,51]. These findings underscore the balance between optimized coating composition and technical challenges in coated electrode production. Nevertheless, this study validates the feasibility of incorporating BFS into rutile-type coating manufacturing, establishing a basis for further refinement and optimization.

	I u v	c or r crioring	arice criterie	turia visuar	inspection re	ours or rum	type couted	cicciioacs	
			(a) Perfor	mance criteria	a of rutile-type	covered electr	odes		
Rat	ing	Arc	Burning		Slag	Slag	Seam	Odor	Seam or
Po	int	stability	Durining	Spatter	uniformity	removal	appearance	and smoke	slag porosity
1-	-2	Very Bad	Very difficult	Excess	Very poor	Very difficult	Matt	Excess	Very high
3-	-4	Bad	Difficult	Increased	Poor	Difficult	Low gloss	Less	High
5-	-6	Good	Medium	Moderate	Average	Easy	Medium	Medium	Medium
7-	-8	Very good	Easy	Minimal	Good	Very easy	Glossy	Minimal	Low
9-	10	Excellent	Very easy	None	Very good	Spontane	High gloss	None	None
			(b) Scores	received by	the rutile-type	covered electro	odes		
	trode lbers	Arc stability	Burning	Spatter	Slag uniformity	Slag removal	Seam appearance	Odor and smoke	Seam porosity
R1		8	9	7	7	9	7	8	7
R2	Doint	10	10	8	10	10	10	8	8
R3	Point	10	10	8	10	10	10	8	10

Table 5. Performance criteria and visual inspection results of rutile-type coated electrodes.

## 3.2. Microstructure, XRD, and Chemical Composition Analysis Results

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Welding tests using rutile-type coated electrodes developed in this study revealed a predominant ferrite+pearlite phase mixture in the weld metal, primarily composed of ferrite ( $\alpha$ -iron), influenced by the electrode coating formulation. This microstructure, characteristic of weld metals produced under similar conditions, balances mechanical strength and ductility, aligning with industry benchmarks for coating performance [15]. The observed microstructures were validated through microstructure photographs (Figure 8), weld metal XRD analyses (Figure 9), and chemical composition analyses (Table 6), consistent with literature and commercial coated electrode standards [16–20,43–45,51]. The ferrite phase, forming the weld metal matrix, contributed to softness and ductility, while the pearlite phase, with its lamellar ferrite-cementite structure, enhanced strength and hardness. Uniform phase distribution, achieved through controlled welding parameters, minimized segregation and ensured coating-driven homogeneity.

Microstructure photographs (Figure 8) displayed distinct phase boundaries and consistent thermal conditions, while XRD analysis (Figure 9) confirmed dominant ferrite with trace pearlite phases, evidenced by  $\alpha$ -iron peaks. The weld metal microstructure, comprising approximately 88.1%  $\alpha$ -iron and 11.9% Fe<sub>3</sub>C, corroborated XRD results and aligned with established metallurgical data [58]. These phase analyses underscored the reliability of the coating process in delivering predictable structural properties [16–20,43–45]. Chemical composition analysis (Table 6) revealed a predominantly iron (Fe) weld metal with trace amounts of carbon (C), manganese (Mn), silicon (Si), sulfur (S), phosphorus (P), and other alloying elements, influenced by the rutile-type coating. Sulfur (0.023–0.026 wt%) and phosphorus (0.027–0.030 wt%) levels increased slightly with higher BFS content

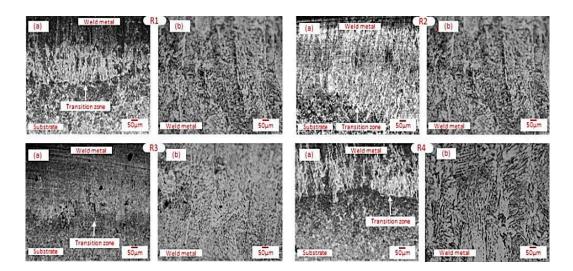


Figure 8. Microstructures of weld metal.

in the coating, due to element transfer from the slag to the weld pool. Conversely, manganese (1.06-1.19 wt%) and silicon (0.25-0.47 wt%) contents rose as slag content decreased, reflecting enhanced transfer efficiency from the core wire and coating formulation [16–20,43–45]. Carbon content, critical for steel, ranged from 0.11-0.13 wt%, decreasing slightly with reduced slag content due to carbon-oxygen reactions forming  $CO_2$  [39,40].

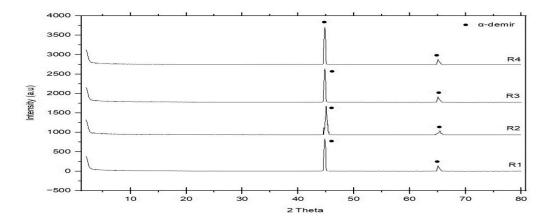


Figure 9. XRD patterns of weld metals for R1-R4 rutile-type coated electrodes.

Measurements were conducted using an optical emission spectrometer, ensuring high accuracy. SEM-EDX analysis provided semi-quantitative oxygen content data, showing a decline in oxygen levels as slag content in the electrode coating decreased. This reduction minimized oxide and inclusion formation, enhancing weld metal ductility and toughness, critical for coating performance [31–33]. Nitrogen measurements, consistent across electrodes, were deemed insignificant and unrelated to coating flux content. Optimizing flux and slag content in the coating formulation proved essential for superior weld metal quality [16–20,43–45].

The results were driven by the chemical similarity between the core wire and substrate, ensuring uniform weld metal properties, and optimized welding conditions promoting a homogeneous ferrite-pearlite microstructure. Limited filler metal transfer reinforced the core wire and substrate's role in weld metal composition, aligning with literature and commercial coating benchmarks [16–20,43–45].

XRD analysis of coating-derived slags revealed crystalline phases, including FeTiO $_3$  (nigrine), MgFe $_2$ O $_4$ , MgTi $_2$ O, CaMn $^2$ Si $_2$ O, Fe $_2$ MnTi $_3$ O $_{10}$ , and TiO $_2$  (rutile) (Figure 10), influenced by the electrode coating's chemical composition and thermal conditions [59,60]. Some of these phases, such as

MgTi<sub>2</sub>O, CaMn<sup>2</sup>Si<sub>2</sub>O, and Fe<sub>2</sub>MnTi<sub>3</sub>O<sub>10</sub>, were tentatively identified based on peak positions and intensity ratios. Acidic phases (SiO<sub>2</sub>, TiO<sub>2</sub>, FeTiO<sub>3</sub>) enhanced coating-derived slag formation and self-lifting behavior, improving weld metal cleanliness and efficiency, while basic phases impaired these properties [16–20,43–45]. Surface tension, thermal expansion coefficients, and viscosity of these phases governed slag detachment. High-density nigrine and rutile phases promoted spontaneous slag lifting, confirmed by inspection evaluations, whereas spinel (AB<sub>2</sub>O<sub>4</sub>) and perovskite (CaTiO<sub>3</sub>) phases, with high melting points, increased slag density and viscosity, hindering removal [15,46].

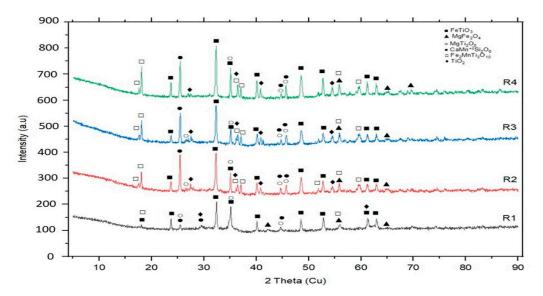


Figure 10. XRD patterns of post-weld slags from rutile-type coated electrodes.

Controlling these phases during electrode coating design is critical for optimizing slag behavior. Slag phase composition varied with raw material proportions in the coating formulation, aligning with industry standards and commercial coated electrode specifications [16–20,43–45]. These findings validate the optimized coating formulations and their suitability for practical welding applications.

			Chemi	cal Compo	sition (wt	:.%)			
Electrode	С	Mn	Si	S	Cr	P	O	N	Fe
R1	0.12	1.06	0.25	0.026	0.05	0.030	1.18	0.04	Bal.
R2	0.13	1.08	0.31	0.027	0.05	0.028	1.06	0.04	Bal.
R3	0.11	1.17	0.40	0.024	0.05	0.027	0.93	0.04	Bal.
R4	0.11	1.19	0.47	0.023	0.04	0.027	0.90	0.04	Bal.

**Table 6.** Chemical composition of weld metals from rutile-type coated electrodes.

Note: Oxygen (O), with elevated values (0.90–1.18%), and nitrogen (N) are semi-quantitative by SEM-EDX for comparison only. Other analyses use optical emission spectrometry.

# 3.3. Hardness Measurement Results

Hardness measurements conducted on weld metal samples revealed consistent mechanical behavior across electrodes containing 35–50% BFS. Vickers hardness values, obtained using a 9.8 N load, were averaged from five distinct regions of each weld metal sample to account for localized variations. As shown in Table 7 and Figure 11, hardness values ranged from 168 to 182 HV, with electrodes R3 and R4 (35–40% BFS) exhibiting slightly higher hardness due to optimized slag composition and reduced porosity.

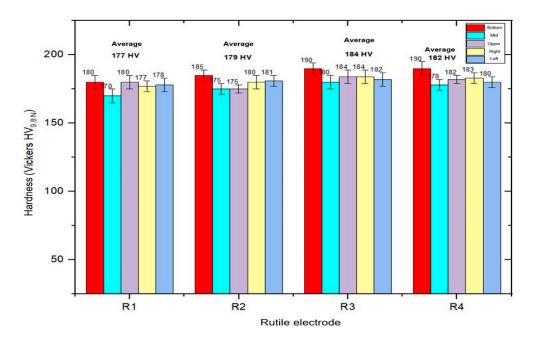


Figure 11. Vickers hardness measurements of weld metals for R1-R4 rutile-type coated electrodes.

The weld centerline consistently exhibited the highest hardness, attributed to thermal concentration and accumulation of alloying elements during welding. Boundary regions displayed marginally lower values, reflecting thermal gradients and dilution effects. The uniform distribution of hardness across samples confirmed the homogeneity of the ferrite-pearlite microstructure and effective flux-metal interaction. These results align with literature benchmarks for rutile-type coated electrodes and validate the mechanical integrity of BFS-containing coatings [16–20,43–45].

Electrode numbers	Yield strength	Tensile strength	Elongation	Izod V-notch in (Jou	•
	(MPa)	(MPa)	A5 (%)	20°C	0°C
R1	470	560	24	79	48
R2	474	562	25	83	53
R3	477	570	26	86	58
R4	482	573	26	86	59
Commercial E6013	400-500	500-560	23-28	70-100	50-60

**Table 7.** Tensile and impact test results of rutile-type coated electrodes.

Note: R1–R4 values were obtained from single tests in this study, conducted per ISO 9018:2015, ISO 148-1:2016, and ASTM E23:2023. Commercial E6013 values are from Magmaweld (2025).

# 3.4. Tensile and Impact Test Results

Tensile tests performed on butt weld samples demonstrated that electrodes containing 35–40% BFS achieved yield strengths of 477–482 MPa, tensile strengths of 570–573 MPa, and elongation ratios (A5) of 24–26%, as summarized in Table 8. These values meet the ISO 2560:2020 standard requirements for rutile-type electrodes, confirming their suitability for structural welding applications [51].

Table 8. Mechanical property requirements according to ISO 2560:2020.

Minimum yield strength	Tensile strength range	Minimum elongation	
(MPa)	(MPa)	A5 (%)	
355	440-570	22	
380	470-600	20	
420	500-640	20	
460	530-680	20	
500	560-720	18	

Izod V-notch impact tests conducted at 0 °C and 20 °C revealed impact energies ranging from 58 to 59 J for R3 and R4 electrodes, indicating excellent toughness under standard and low-temperature conditions. Electrodes R1 and R2 (45–50% BFS) showed slightly reduced impact performance, attributed to increased porosity and slag viscosity.

The combined tensile and impact results confirm the mechanical reliability of BFS-containing rutile-type coatings. The optimized flux composition, controlled welding parameters, and effective slag behavior contributed to enhanced strength, ductility, and toughness. These findings support the feasibility of BFS integration into coated electrode manufacturing and its alignment with sustainable welding practices.

#### 3.5. Environmental and Economic Impact Results

Incorporating 35–50% BFS into rutile-type electrodes (R1–R4) yielded quantifiable environmental and economic benefits. Environmentally, slag recycling diverted 200–600 kg of BFS per ton of steel, reducing landfill usage by 10–20% and minimizing the consumption of virgin flux materials such as SiO<sub>2</sub>, CaO, Al<sub>2</sub>O<sub>3</sub>, MgO, and TiO<sub>2</sub> [55,56]. CO<sub>2</sub> emissions decreased by 0.28–0.4 kg per kg of coating, as calculated in Section 2.8, based on slag recycling and the avoidance of raw material production [52,53].

Economically, BFS integration reduced raw material costs by 15–25% and coating costs by 33–48%, based on comparative market prices of virgin flux materials [45]. Landfill diversion resulted in savings of 10–20 USD per ton of steel, depending on regional disposal fee structures [49,56]. Energy savings aligned with BedZED's steel recycling benchmarks, demonstrating significant reductions in embodied energy consumption [54].

These results are consistent with recent advances in sustainable materials engineering, where industrial and post-consumer wastes have been successfully transformed into functional systems such as foam glass structures [37], geopolymer matrices [38], and porous frameworks for adsorption-based separation [61], reinforcing the viability of circular economy strategies in metallurgical applications. These outcomes support the EU Green Deal's 2050 climate neutrality target, while simultaneously enhancing weld quality and promoting sustainable manufacturing practices [62].

# 4. Conclusion

This study demonstrated the feasibility of producing rutile-type coated electrodes with up to 50% BFS in four formulations, characterized through welding tests, visual inspections, microstructure and XRD analyses, chemical composition assessments, hardness, tensile, and Izod notch impact tests. Compared to ISO 2560:2020 criteria for yield strength and 47 J impact energy, all electrodes complied, with R3 and R4 (35–40% slag) showing enhanced mechanical properties [54]. Table 7 shows R3–R4 achieved yield strength (477–482 MPa), tensile strength (570–573 MPa), elongation (26%), and impact energy (58–59 J at 0 °C), comparable to commercial E6013 electrodes [20]. These results support sustainable welding practices and industrial applicability [6,63]. Incorporating slag as a flux component reduces environmental impact, aligning with EU Green Deal goals by minimizing pollution, resource use, and waste through valorization [9,14]. Future studies should use



standardized conditions such as extrusion pressing, and conduct weld tests in various configurations with replicate trials to improve result reliability and validate electrode performance across diverse welding scenarios.

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