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Article

All Screen Printed and Flexible Silicon Carbide NTC Thermistors for Temperature Sensing Applications

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Abstract: In this study, Silicon Carbide (SiC) nano-particle based serigraphic printing inks were formulated to fabricate highly sensitive and wide temperature range SiC printed thermistors. Initially, commercial silver ink was screen printed to fabricate inter-digitated electrodes (IDE's) onto flexible Kapton® substrate via screen printing. Thermistor inks with different weight ratios of SiC nano-particles dispersed in polyimide resin matrix were fabricated. The SiC-polyimide temperature sensing inks were screen printed atop the IDE structures to form fully printed thermistors and encapsulated with a adhesive backed polyimide film for humidity inhibition. The high temperature tolerance of the Kapton[®] allowed the the sensors to be tested over a wide temperature range form 25°C to 170°C. The printed SiC thermistors exhibit excellent repeatability and stability over 15 hours of continuous operation. Optimal device performance was achieved with 30 wt.% SiC-polyimide ink. We report highly sensitive devices with a temperature coefficient of Resistance (TCR) of -0.556 %/°C, a thermal coefficient of 502 K (β -index) and an activation energy of 0.08 eV which are comparable with printed thermistors previous reported. Further, the thermistor demonstrates an accuracy of $\pm 1.35^{\circ}$ C which is well within the range offered by commercially available high sensitivity thermistors. SiC thermistors exhibit a small 6.5% drift due to changes in relative humidity between 10-90 %RH and a 4.2 % drift in baseline resistance after 100 cycles of aggressive bend testing at a 40° angle. The use of commercially available low cost materials, simplicity of design and fabrication techniques coupled with the chemical inertness of the Kapton® substrate and SiC nanoparticles paves the way to use all-printed SiC thermistors towards a wide range of applications where temperature monitoring is vital for optimal system performance.

Keywords: temperature sensing; negative temperature coefficient (NTC); printed electronics; printed temperature sensors; thermistors; silicon carbide; wide band-gap semiconductor; screen printing; silver ink

1. Introduction

Temperature sensing is crucial in key industries such as automotive [1], health [2,3], aerospace [4], agriculture [5] and consumer electronics [6]. Measuring temperature is a key variable in controlling and monitoring the intended function of a system [7] to and to optimize process yields [8]. The three predominant types of printed temperature sensors are: 1. resistance temperature detectors (RTD's) [9], 2. thermocouples [10] and 3. thermistors [11]. Choosing the correct type of temperature sensor is dependant on the sensitivity, accuracy and temperature range required for the intended applications. Thermocouples are widely used in several industries due to their small form factor, low cost and wide temperature range. Depending on the materials of construction, some thermocouple can be used in ultra high temperature conditions up to 2300 °C [12]. However, when compared to thermistors; thermocouples have a lower accuracy and sensitivity as their change in response is generally only a few milli-volts [13]. For lower temperature ranges, thermistors and RTD's are used where RTD's typically exhibit lower sensitivity and have a slower response time as compared to thermistors [13]. Thermistors on the other hand, provide high sensitivity ranging between 2 to -6%/°C [14] which makes them highly desirable for sensing across a wide variety of applications [15]. Additionally, thermistors exhibit

a non-linear negative temperature coefficient (NTC) as electrical resistance decreases with increase in temperature [16]. The thermal index β of a thermistor is an indicator or its sensitivity. Devices with high β values (3000 - 5000K) are typically used for high temperature sensing applications while those with low β values (14 - 170K) are used for applications such as integrated circuit temperature compensation and random access storage memories etc [14,17]

Several fabrication techniques such as microfabricaion [18], tape casting [19] etc. have been employed in the fabrication of thermistors. With advances in the field of printed electronics over the past two decades, it has become increasingly possible to fabricate low cost, flexible temperature sensors [20]. Literature suggests a significant increase in academic articles toward printed thermistors utilizing a variety of materials, substrates and printing methods for a different applications. Printing techniques such as screen [16], inkjet [21] and aerosol jet [22] printing have been successfully employed to fabricate all printed thermistors. Polymeric sensing materials such poly(3,4-ethylenedioxythiophene)-poly(styrenesulfonate) (PEDOT:PSS) [23] and Polydimethylsiloxane (PDMS) [24] have been widely printed to fabricate low-cost thermistors. Carbon derivaties such as carbon nanotubes (CNT's) [16], graphene oxide [25] and reduced graphene oxide [26]. have also been utilized for the same. Semi-metallic graphene is an unique material that has been widely employed as printed thermistors [17] and RTD's [27]. The above mentioned sensing materials are generally restricted to relatively low temperature sensing ranges generally below 100 °C due to thermal degradation. The use of ceramics combined with higher temperature sustaining substrates such as Kapton® promises a significant increase in the temperature sensing range of the printed device.

Various ceramic materials have been employed towards fabrication of wide range temperature sensors, some having a sensing range as high as 1500 °C[28-31] using complex low and high temperature co-firing techniques (LTCC and HTCC). Transient metal oxides such as $Ni_xMn_{2-x}O_4$ [32], MnNiCuO [33], and BiFeO₃ [14] have been used to fabricate NTC thermistor. The fabrication of ceramic sensing devices usual employs ultra high sintering temperatures and inert environments which increases overall fabrication cost and complexity [34]. Wide band gap semiconducting materials such as ZnO (II-VI), GaN (III-V) and SiC (IV-IV) exhibit excellent electrical and mechanical properties along with chemical inertness, and optical transparency which make them ideal candidates for flexible electronics devices [35]. Out of these, Silicon Carbide (SiC) has been gaining significant interest in recent years towards fabricating bio-sensing devices[36] owing to its bio compatibility [37–39]. Silicon Carbide comprises of covalent bonded Si and C atoms with very short bond lengths of 1.89 Å[40] which attribute to their mechanical and chemical stability with an electronic band-gap ranging between 2.4 to 3.2 eV depending on the polytype [36]. Silicon Carbide exists in several polytypes out of which cubic 3C-SiC (β -SiC), 4H-SiC and 6H-SiC (α -SiC) are most commonly grown and used for sensing applications. However, β -SiC is widely available in high purity and relatively low cost nano particle form. Silicon Carbide has been widely used to fabricate NTC thermistors that are predominately used in ultra high temperature and harsh environments. These thermistors are generally fabricated via processes such as chemical vapour depositon (CVD) [41,42], epitaxial SiC crystal growth [43], sputter coated electrodes on SiC single crystal wafers [44] and transfer based [45] techniques.

SiC has been sparsely used in the field of printed electronics. Researchers have demonstrated direct ink writing of SiC in borosiloxane-colloidal dispersion for Microwave optics [46], inkjet printable SiC ink [47] and vat polymerization based electrically conductive SiC features [48]. SiC nano particles have been employed towards applications such as electrochemical [49,50], gas [51] and humidity [52] sensing applications. There has been very limited work done towards printed SiC nanoparticles towards temperature sensing applications. In 2022, Aljasar et al [53]. demonstrated laser sintered SiC nano particle temperature sensors via drop casting up operational to 86 °C.

In this study, we focus on fabrication of low-cost and flexible screen printed SiC thermistors for a wide temperature range between $25\,^{\circ}$ C to $+170\,^{\circ}$ C. For this purpose SiC nanoparticles are impregnated into a polymeric matrix of 4,4'-oxydianiline (polyimide) resin via simple dispersion techniques. Initially, we study the characteristics of the commercially sourced SiC nanoparticles to

assess their crystalline and composition. Next we measure electronic properties of the printed SiC sensing films and determine the optimal loading fraction of particles in the ink. The fully fabricated devices are then tested for temperature sensing performance, durability and long term stability. The dependency of the SiC thermistors on relative humidity and mechanical deformation (bend testing) will also be investigated. Lastly, we calculate key thermistor performance matrices such as its temperature coefficient of resistance (TCR), thermal coefficient (β -index) and its activation energy (eV).

2. Experimental Section

2.1. Materials

Cubic Silicon Carbide (3C-SiC, beta, 99+ %, <80 nm, cubic) nanoparticles were acquired from US Research Nanomaterials, Inc (US2022). Poly(pyromellitic dianhydride- co-4,4-oxydianiline) amic acid resin was procured from Millipore Sigma (Product: 575771). 0.005-inch thick FPC Kapton[®] was sourced from American Durafilm. Henkel supplied Loctite[®] EDAG 725A silver screen printing paste. No modifications were made to any of the materials upon receipt.

2.2. Ink Formulation

The screen-printable thermistor ink was formulated by incorporating varying quantities of SiC nanoparticles into the poly(pyromellitic dianhydride-co-4,4-oxydianiline) amic acid solution, ranging from 30 - 40 wt.% in 5 wt.% increments. To achieve homogeneous dispersion, a two-step dispersion process was employed. The SiC nanopowder was initially weighed and added in approximately four portions to the resin. Following each addition, the ink was mixed using a planetary mixer (Thinky ARE-310) for three cycles of one minute each, with one minute degassing intervals at 2,000 rpm. Subsequently, once the entire SiC powder was added, the specified amount of graphene was weighed and added, followed by six cycles of one minute each, with six one-minute degassing intervals at 2,000 rpm. This meticulous process ensured thorough blending of the nanopowder and resin, resulting in a stable dispersion with a shelf life exceeding 6 months.

2.3. Device Design and Fabrication

The presented sensor structure includes two main functional layers: 1. the printed silver interdigitated electrodes and 2. printed SiC active sensing layer. Initially, the Kapton[®] substrate was prepared by cleaning with 99.9% pure acetone (Millipore Sigma 270725). Silver paste was printed on to the Kapton[®] substrate using a KEKO P250 automatic screen printer with a 325 mesh, 0.001-inch emulsion thickness screen in an interdigitated electrodes (IDEs) format. The IDEs had a trace width and spacing of 0.5 mm, as depicted in Figure 1. The silver ink was cured in air at 300 °C for 60 minutes in a Mancorp (MC301N) reflow oven. Thermistor inks were then applied to the cured IDEs in 7.5 mm diameter circles and cured in the reflow oven for 60 minutes at 200 °C to achieve complete polymerization of the resin. Under these conditions, a polycondensation reaction occurs between a diamine and a dianhydride, resulting in the formation of polyimide [14]. Figures 2a)-c) illustrate the fabrication process of the SiC thermistors. As high resolution optical microscpe image of the printed SiC thermsitor can be seen in Figure 2d).

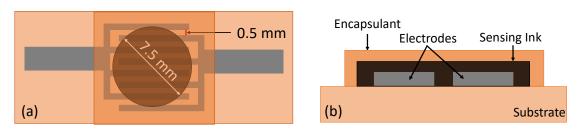


Figure 1. SiC thermistor schematic a) top view b) side view.

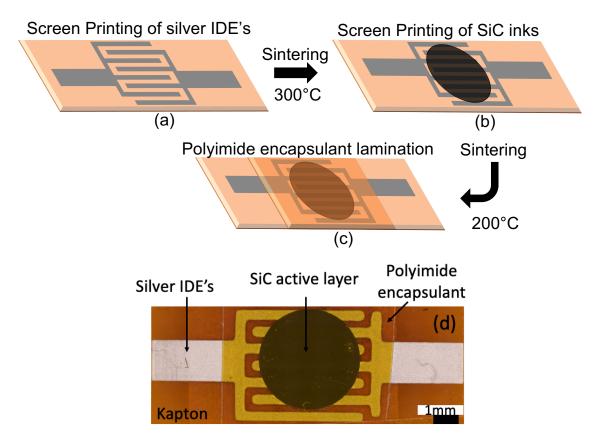


Figure 2. a)- c) SiC thermistor fabrication process d) High resolution optical microscopy image of printed SiC thersmitor.

2.4. Characterization Methods

X-ray diffraction patterns were obtained using the $XRD-X'Pert^3$ system with a cobalt source and Raman micro-spectra of the SiC nano power were obtained using the and WITec alpha300A raman microspectoscopy systems with a 532 nm green laser. Ultraviolet-visible (UV-Vis) absorption spectra of the SiC nano powder was obtained from via the Perkin Elmer, Lambda 750 system. Micrographs of the printed films is obtained using a Hitachi SU8230 scanning electron microscope (SEM) equipped with a Bruker, QUANTAX FlatQUAD EDX detector for precise elemental mapping. Transmission electron microscopy on the nano particles is performed via the Jeol, JEM-F200 multi-purpose electron microscope. A Keyence VHX7000 optical microscope was used to obtain high resolution images of the sensor. Ossilla T2001A3 four-point-probe was utilized to measure the conductivity of the printed SiC films. The printed thermistors were tested in a controlled temperature and humidity 6-channel micro probe station from Nextron® (MPS-PT6C) and the two-wire resistance readout was recorded using a 40-channel digital multi-meter from Keithley (DMM 2790-L). Temperature was varied between 25 °C and 170 °C owning to the maximum temperature limit of the Nextron[®] at a constant rate of 10 °C/minute and a constant humidity at 40% RH. A custom Matlab script was written to process the raw data obtained from both the humidity test chamber and digital multi-meter. Figure 3 illustrates the experimetal setup used. Current - voltage characteristics of the thermistors was obtained using the Keithley DAQ6510 digital multi meter and the Keithley 2400 source-measure unit. Mechanical bend testing was performed using a custom setup consisting of a liner stage attached to a ball screw stepper motor. The stage was controlled via a Arduino Uno and a custom script to control travel distance, speed and number or cycles.

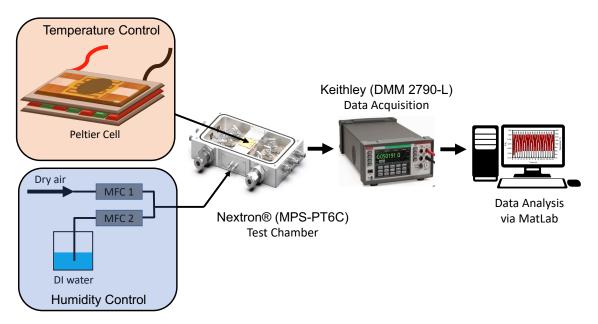


Figure 3. Experimental setup.

3. Results and Discussion

3.1. Material Characterization

The X-ray diffraction (XRD) patterns of the commercially purchased SiC particles validate their cubic crystalline nature, exhibiting well-defined and sharp diffraction peaks, as depicted in Figure 4a). The peaks observed at $2\vartheta = 39.6^{\circ}$, 42° , and 71.2° correspond to the 002, 200, and 311 crystalline planes of the $3\mathrm{C}-\mathrm{SiC}$ cubic crystal structure, consistent with the (International Centre for Diffraction Data) ICDD card: 04-002-9070 [54,55]. Additionally, the presence of $2\mathrm{H}-\mathrm{SiC}$ peaks at $2\vartheta = 45^{\circ}(015)$ and $49^{\circ}(016)$, as per the ICDD card: 04-008-2392 [56], may potentially be attributed to a blending of the two phases formed during the nanoparticle fabrication process [57]. The spectral results suggest an approximate 9:1 ratio between the two phases, respectively.

The Raman spectra of the $3\mathrm{C}-\mathrm{SiC}$ powder (Figure 4b)) confirm its cubic crystal structure. Noteworthy peaks appear at wave numbers 785, 897, 1346, 1588, 2684, and 2916 cm^{-1} . Specifically, the peaks at 785 and 897 cm^{-1} correspond to the transverse and longitudinal optical modes of $3\mathrm{C}-\mathrm{SiC}$, as documented in reference [58,59]. The 1346 cm^{-1} peak represents the D band of carbon, peak 1588 cm^{-1} corresponds to the G band, associated with the A_{1g} vibrational mode of carbon. The peak at 2684 cm^{-1} signify the second orders of the D band, referred to as the 2D band, while the 2916 cm^{-1} peak represents the D+G band, as detailed in references [60,61].

Ultraviolet visible near infrared (UV-vis-NIR) absorption spectroscopy is performed on the 3C-SiC powder (Figure 4c)) which indicates that the absorption edge is close to $400\,\mathrm{nm}$. The electronic band gap (E_g) of the nano powder is calculated by liner fitting of the Tauc plot as seen in the inset graph. It suggests the band gap of the 3C-SiC nano powder is $2.75\,\mathrm{eV}$ which is consistent with previous reports [62-64].

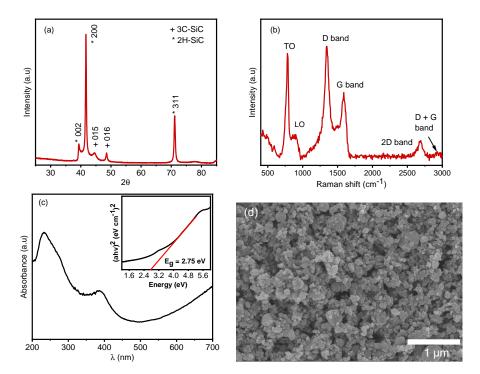


Figure 4. Characterization of commercially sourced 3C - SiC particles. a) XRD spectra, b) Raman spectra c) UV-VIS spectra with Tauc plot as inset to determine electronic band gap, d) SEM micrograph.

3.2. Morphology

The morphology of the $3\mathrm{C}-\mathrm{SiC}$ particles is investigated via scanning electron microscopy (SEM) micrograph presented in Figure 4d). We observe a densely packed film of particles with minimal porosity. Elemental mapping of the particles (Figure 5) reveals a uniform distribution of silicon and carbon elements along with the presence of elemental oxygen which could be attributed to the native oxide shell around the SiC particles.

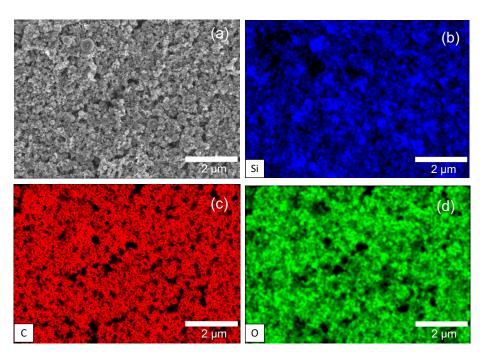


Figure 5. EDX imaging of a) SiC particles representing b) Si c) C and d) O species.

Transmission electron microscopy micrographs are shown in Figure 6a)-b). The SiC particles have an average particle size of $70\mu m$ analysed using ImageJ. We confirm the presence of an approx. 2 nm thick oxide shell (SiO_x) around the SiC particles as indicated in Figure 6b) [65]. The SAED pattern observed in Figure 6c) allows us to calculate the inter planner atomic distance to be 0.25 nm which is assigned to the (111) cubic atomic plain [66].

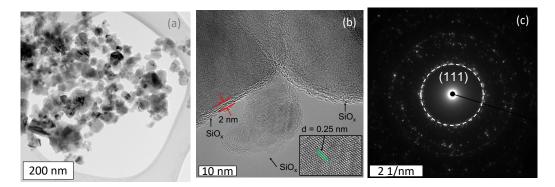


Figure 6. a), b) TEM micrographs and c) SAED pattern of SiC particles .

3.3. Printed SiC Thermistor Characterization

Once the $3\mathrm{C}-\mathrm{SiC}$ particles have been thoroughly characterized, the screen printable thermistor ink is fabricated by mixing various wt.% of $3\mathrm{C}-\mathrm{SiC}$ particles in poly (pyromellitic dianhydride-co-4,4'-oxydianiline) amic acid solution as described in the section 2.2. Initially, the SiC thermistor inks was printed directly atop the prepared Kapton® substrate to determine their electrical properties. Figure 7a) shows the current voltage characteristics of the thermistor at $25\,^{\circ}\mathrm{C}$. We observe a linear ohmic behavior between -5 V to 5 V indicating that the device functions as a resistor where the current is directly proportional to the applied voltage.

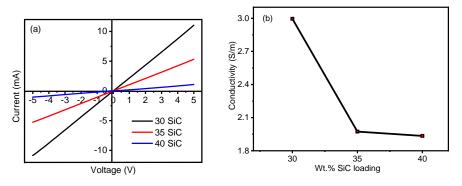


Figure 7. a) Current-voltage characteristics and b) electrical conductivity of 30 wt.%, 35 wt.%, 40 wt.% SiC thermistor inks.

Next, we measured the electrical conductivity of the three inks via four point probe. As seen in Figure 7b), the 30 wt.% loaded SiC thermistor ink formulated exhibits a maximum electrical conductivity of 2.99 ± 0.007 S/m. We observe that electrical conductivity decreases by more than 50% between the 30 wt.% and 35 wt.% loaded samples and further reduces as the loading is increased to 40 wt.%. This reduction could be attributed to the onset and growth of micro-cracks on the surface of the SiC film during the annealing process [67]. This phenomena was further investigated by acquiring SEM micrographs of films printed with the three SiC loaded inks at two different magnifications. The 30 wt.% SiC film has a dense, self levelled and uniform distribution of particles with no cracks on the surface of the film. (Figure 8a),b)). As the SiC loading is increased to 35.%, we observe the formation of surface imperfections and agglomerates as seen in Figure 8c) and the onset of micro cracks in the films which are observed at high magnifications (Figure 8d)). Lastly, the cracks are significantly

more pronounced in the 40 wt.% loaded SiC film (Figures 8e),f)). Hence, the increase in electrical resistance with increase in wt.% SiC loading can directly be attributed to film cracking due to the loss of conductive pathways. [68,69]

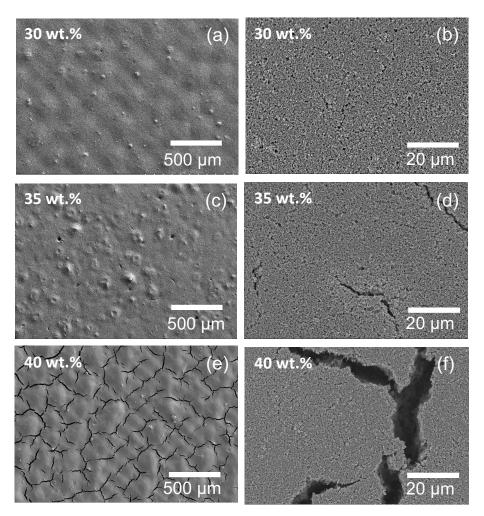


Figure 8. SEM micrographs of a) and b) 30 wt.% c) and d) 35 wt.% e) and f) 40 wt.% SiC thermistor inks.

The silver IDE's are screen printed onto the polyimide substrate as previously and the SiC inks were then screen printed atop the silver IDE's to fabricate the thermistors as described in section 2.3 and Figure 2.

Although the Kapton[®] substrate is rated to handle 350 °C and up to 400 °C for intermittent exposure [70] the performance of the printed SiC thermistor is restricted to 170 °C due to the Nextron micro-probe stations' maximum temperature limit. Thermistors of each ink formulation were thermally cycled between 25 °C and 170 °C five times and the electrical resistance was recorded. Figure 9a) shows the change in electrical resistance of all three inks with increase in temperature over five test cycles. Amongst the three ink formulations, the 30 wt.% loaded SiC thermistor showed the highest change in resistance of 41.1% going from 1267 Ω at 25 °C 988 Ω at 170 °C. The 35 wt.% and 40 wt.% SiC show a considerably low change of 20.4% and 14.7% which is less than half as compared to the 30 wt.% ink. Interestingly, the response of the three inks is identical to its change in electrical conductivity as previously discussed in Figure 7b). This further validated that the increase in SiC significantly contributes to the degradation in the thermisors performance due to film cracking. The best performing

thermistor (30 wt.% SiC) follows a second order polynomial fitting governed by Equation 1 with a \mathbb{R}^2 confidence of 0.96.

$$y = -2E - 0.5X^2 + 0.0064X + 1.0714 \tag{1}$$

Similar to conventional SiC thermistors reported in literature [41], our screen printed SiC thermistors show a reduction in electrical resistance with increase in temperature, exhibiting a negative temperature coefficient of resistance (NTCR).

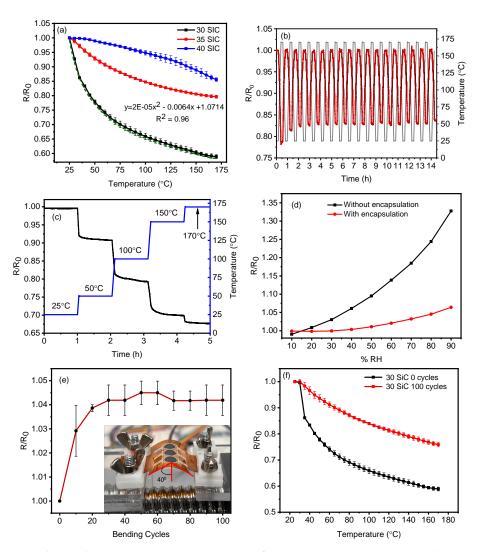


Figure 9. a) Electrical resistance versus temperature of 30 wt.%, 35 wt.%, 40 wt.% SiC thermistors b) long term cycling c) thermal stability d) humidity stability e) change in baseline resistance after bend testing and f) electrical resistance versus temperature plot after bend testing of 30 wt.% SiC thermistor.

A high performance thermistor must be stable, reliable and durable to be successfully deployed in challenging application environments [13]. To this effect, we subject the 30 wt.% SiC thermistor to various endurance tests. In Figure 9b), change in resistance of the thermsistor is recorded over 15 hours while being cycled between $25\,^{\circ}$ C and $170\,^{\circ}$ C. The device experiences minor variability during the first two thermal cycles which could be attributed to the relaxation of the SiC film and any instabilities in the temperature test chamber at the onset of the test. Beyond this, the device is extremely stable with a minor drift in baseline resistance which was corrected via background subtraction. In order to test device stability, the 30 wt.% SiC thermistor is maintained at $25\,^{\circ}$ C, $50\,^{\circ}$ C, $100\,^{\circ}$ C, $150\,^{\circ}$ C and $170\,^{\circ}$ C for 1hr each. We observe constant resistance at each fixed temperature suggesting device stability over a prolonged time duration (Figure 9c).

The influence of humidity has a significant impact on the performance of a thermistor [71]. Researches have shown that SiC exhibits good humidity response which is well suited to make humidity sensors [72,73]. To mitigate this effect, we have incorporated a polyimide film encapsulation layer. To ensure its effectiveness, we measure the electrical resistance of the 30 wt% SiC thermistor while varying the relative humidity between 10 %RH and 90 %RH at 25 °C. The change in resistance of two identical devices; one with and one without the polyimide film encapsulation are compared as seen in Figure 9d). The device without encapsulation experiences a 32.6% change in resistance at a constant temperature indicating that the SiC film is highly sensitive to changes in relative humidity. In contrast, the device with encapsulation sees a much smaller drift of 6.5% in baseline resistance under the same conditions of %RH. The laminated polyimide encapsulant provides a high degree of protection against %RH, however this could be further improved by incorporating a fully printed insulating layer in future studies.

Mechanical flexibility of the 30 wt.% SiC thermistors was tested at an aggressive angle of 40° as seen in the inset of Figure 9e). We observed a small drift in baseline resistance of 4.2 % over 100 test cycles (Figure 9e)). Interestingly, majority of the change in resistance is observed during the first 20 cycles indicating deformation in the SiC sensing film at the onset of the test. Post bending, the device was cycled five times between 25 °C and 170 °C and compared to a pristine 30 wt.% device (Figure 9f)). We notice a reduction in device response from 41.1% to 26.2%. This drop in performance is tentatively attributed to surface crack formation in the SiC film leading to loss of conductive pathways for effective change transfer with change in temperature.

3.4. Thermistor Performance

To further characterize the 30 wt.% SiC thermistor we calculate its performance parameters such as thermal index β , activation energy E_g and temperature coefficient of resistance (TCR). The electrical resistance dependence on temperature is given by the expression $R = R_o exp(\beta/T)$ where R_o is the resistance at infinite temperature and T is the absolute temperature. Thus, the thermal index can be calculated using the following [74].

$$\beta = \frac{\ln R_1 - \ln R_2}{(1/T_1) - (1/T_2)} \tag{2}$$

Here, T_1 = 248K, T_2 = 438K and R_1 and R_2 are the resistance values in Ω at the respective temperatures. The activation energy is calculated from the expression E_g = 2k β where k is the Boltzmann constant and the TCR is calculated by the expression [14].

$$TCR = -\frac{\beta}{T^2} \tag{3}$$

Using Equations 2 and 3 , the characteristics of the printed thermistors are calculated as presented in Table 1. For the 30 wt.% SiC thermistor, the thermal index observed is 502 ± 11 K, the TCR value of -0.556 ± 0.012 %/°C with an activation energy of 0.08 ± 0.001 eV.

Table 1. Electrical performance of various printed SiC thermistor inks

Ink Name	3C-SiC wt%	PI resin wt%	TCR (%/°C)	Activation energy (eV)	Thermal index (K)
30 - SiC	30	70	-0.556 ± 0.012	$\textbf{0.08} \pm \textbf{0.001}$	$\textbf{502} \pm \textbf{11}$
35 - SiC	35	65	-0.227 ± 0.008	0.035 ± 0.001	205 ± 7.4
40 - SiC	40	60	-0.157 ± 0.0068	0.025 ± 0.001	142 ± 6

Accuracy of the printed 30 wt.% SiC thermistor was determined using the Steinhart's equation (Equation 4) [75].

$$\frac{1}{T} = A + BlnR + C(lnR)^3 \tag{4}$$

Where T is the temperature in Kelvin, R is the resistance in Ω at temperature T. A,B and C are the Steinhart coefficients which are material specific and are used to calibrate the SiC thermistors accuracy. To this effect, experimentally measured resistance values at three temperatures of 25 °C (T_1 = 298 K), 100 °C (T_2 = 373 K) and 170 °C (T_3 = 443 K) are used in Equation 5.

$$\begin{bmatrix} 1 & ln(R_1) & ln^3(R_1) \\ 1 & ln(R_2) & ln^3(R_2) \\ 1 & ln(R_3) & ln^3(R_3) \end{bmatrix} \begin{bmatrix} A \\ B \\ C \end{bmatrix} = \begin{bmatrix} \frac{1}{T_1} \\ \frac{1}{T_2} \\ \frac{1}{T_3} \end{bmatrix}$$
 (5)

For the 30 wt,% SiC thermistor, the Steinhart's coefficients are $A = -8.12 \times 10^{-2}$, $B = 1.59 \times 10^{-2}$ and $C = -8.12 \times 10^{-5}$ with an accuracy of ± 1.35 °C over the entire tested temperature range (25 °C to 170 °C) (Figure 10). The accuracy of the fully printed SiC thermsitors is comparable to conventional SiC and commercially available highly accurate thermistors ranging between ± 0.05 °C to ± 1.5 °C [76–78].

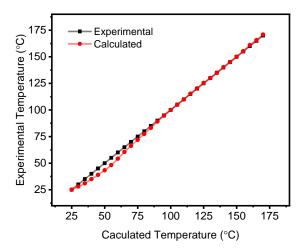


Figure 10. Accuracy of printed 30 wt.% SiC thermistor calculated via the Steinhart's equation.

Researchers have developed several SiC based thermistors in recent years that have superior performance as compared to our all printed SiC nanoparticles based thermistors (highlighted in Table 2). However these thermistors are fabricated via complex and expensive processes such as chemical vapor deposition, vacuum vapor deposition and sputtering. The over all cost of fabrication and materials used limit the use of such thermistors only to highly specialised applications. On the contrary, the SiC thermistors developed in this study employs screen printing which is an inexpensive and scalable technology. Additionally, the use of commercially sourced SiC particles, polyimide rein and Kapton® substrate makes these devices relatively extremely low cost to fabricate. When compared to other printed thermistors recently reported in literature (Table 2), the 30 % SiC thermistors exhibit a comparable TCR coefficient over such a wide operational temperature range. Both SiC particles and Kapton® are chemically inert [79], bio-compatible [80] and resistant to ultra high temperatures [81]. These properties combined with the simplicity and flexibility of our SiC thermistors allow them to accurately detect temperature variations in critical applications such as medicine, agriculture, aerospace and chemical production.

Table 2. State of the art of SiC based and printed thermistors

Sensing material	Electrodes	Fabrication Method	Temperature Range	TCR	Reference					
Silicon Carbide based thermistors										
On SiC undoped wafer	Al, Au, Pt-Pd alloy	CVD	25 °C - 400 °C	-7.9%/K	[41]					
Anodic bonding SiC on glass	Ni/Al	CVD	300K - 600K	-13,000 ppm/K at 300K -3,000 ppm/K at 600K	[45]					
n-type single crystal 21R-SiC wafer	Au-TiB _x -Ni	magnetron sputtering	77K - 450K	10 ⁵ times change in resistance	[44]					
Doped unicrystalline SiC wafer	W (Tungsten)	welding	−100 °C - 300 °C	1.9%/°C	[82]					
Polycrystalline CVD-SiC wafer	Pt-Pd alloy	Vacuum vapor deposition	25 °C - 365 °C	-5.5%/°C	[83]					
Printed Thermistors										
Carbon nanotubes	Ag	Screen printing	-40 °C - 100 °C	-0.4%/°C	[16]					
PDMS + graphene	Copper wire	3D printing	25 °C - 70 °C	0.008/°C	[84]					
graphene oxide	Ag	Inkjet printing	25 °C - 85 °C	-1.19%/°C	[85]					
$BiFeO_3 + 3.5$ wt% graphene	Ag	Screen printing	25 °C - 170 °C	-0.961%/°C	[14]					
${ m V_20_5}$	Pt	Screen printing	200K - 400K	-37,000 to -17,000 ppm/K	[86]					
3C - SiC	Ag	Screen printing	25 °C-170 °C	-0.556%/°C	This work					

4. Conclusions

In this study, we showcase a high performance, fully printed and flexible silicon carbide based thermistor for wide temperature range applications. Low cost serigraphic screen printing was used to fabricate sensors onto silver printed IDE's on flexible Kapton® substrate. SiC thermistors were cycled over a wide temperature sensing range between 25 °C and 170 °C. Three inks with different SiC ink loading's were tested and optimal device performance was achieved at 30 wt.% loading. At higher loading, we observed a reduction in device performance which was attributed to this onset and propagation of cracks with in the printed film leading to loss of conductive pathways. The 30 wt.% device exhibits a TCR of -0.556 %/°C along with a thermal index of 502K (β -index) and an activation energy of 0.08 eV. Device exhibit excellent repeatability and reliability after cycling over extended periods of time upto 15 hours. Printed thermistors shows a small variation in baseline resistance of 6.5% whiile tested over a wide relative humidity range (10 - 90 %RH). Aggressive bend testing was performed to test the thermisotrs flexibility. We observed a small 4.2% drift in baseline resistance of the device after 100 bend cycles at a 40°. Lastly, the 30 wt.% printed thermistors exhibit an accuracy of ± 1.35 °C which is at par with commercially available high accuracy thermistors. This study demonstrates that flexible, all screen printed SiC thermistors have an immense potential in the field of flexible and printed electronics. These low cost, flexible and mass producible thermistors can help improve critical user-device interactions in urgent application such as healthcare, agriculture and food monitoring.

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