









## Article

# Identification of LLDPE Constitutive Material Model for Energy Absorption in Impact Applications

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**Abstract:** Current industrial trends bring new challenges in energy absorbing systems. Polymer materials as the traditional packaging material seem to be promising due to their low weight, structure and production price. Based on the review, the linear low-density polyethylene material was identified as the most promising material for absorbing impact energy. The current paper addresses the identification of the material parameters and the development of a Constitutive material model to be used in future design by virtual prototyping. The paper deals with the experimental measurement of the stress-strain relations of the linear low-density polyethylene under static and dynamic loading. The quasi-static measurement is realized in two perpendicular principal directions and is supplemented by a test measurement in the 45 degrees direction, i.e. exactly between the principal directions. The quasi-static stress-strain curves are analyzed as an initial step for dynamic strain rate dependent material behavior. The dynamic response is tested in the drop tower using a spherical impactor hitting the flat material multi-layered specimen at two different energy levels. The strain rate dependent material model is identified by optimizing the static material response obtained in the dynamic experiments. The material model is validated by the virtual reconstruction of the experiments and by comparing the numerical results to the experimental ones.

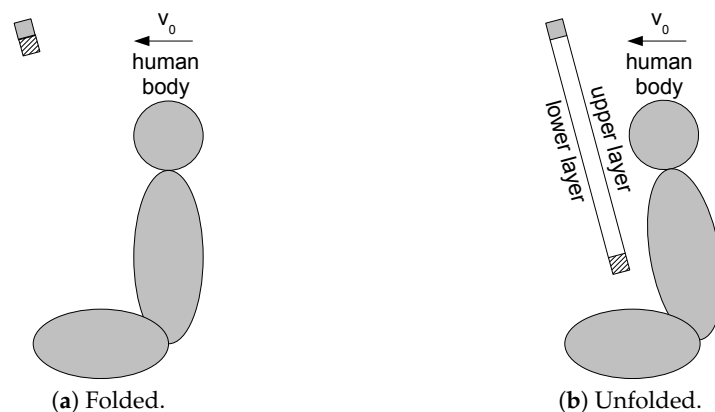
**Keywords:** LLDPE; quasi-static and dynamic experimental tests, impact energy absorption; material parameter identification; constitutive material model; validation; simulation

## 1. Introduction

Thin-layered polymer materials are traditionally used for packaging goods to protect them during transportation. Therefore the major desired properties relate to thickness, density (which relates to weight), strengths, elongation, puncture resistance and stretching level, see Table 1. On the other hand, preliminary experimental tests show also a good performance of such materials in energy absorption.

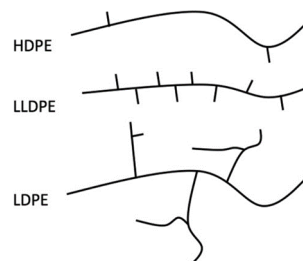
Current trends in the automotive industry regarding future mobility bring new challenges for energy-absorbing safety systems. Non-traditional seating configurations in autonomous vehicles and complex crash scenarios including multi-directional loading are to be considered hand in hand with the advanced materials for energy absorption. Špička *et al.* (2019) used a numerical simulation approach in his study to assess the new safety system (see Figure 1) patented by Hanuliak (2018). The system is based on two layers of a multi-layered membrane injected from the roof between the windscreen and the frontal seats, catching the driver and the passenger during the accident, in a similar manner as airbag performs. The advantage of the approach over the airbag is a simple implementation for multi-directional impact loading and addressing the out-of-position seating issue.

36 As virtual prototyping plays an important role in the design of new products  
 37 nowadays, the paper aims to identify the linear low-density polyethylene (LLDPE)  
 38 material parameters for both static and dynamic loading, to implement them into a  
 39 constitutive material model and to verify the material model by numerical simulations  
 40 representing the experiments. As the static tests are represented by quasi-static loading  
 41 conditions, the dynamic tests represent the scenario close to the one schematically  
 42 described in Figure 1.



**Figure 1.** Scheme of a new safety system for absorbing impact energy: **a)** Folded. **b)** Unfolded

43 LLDPE films have been identified as the most promising material in cases, where  
 44 the impact loading is assumed, because of their higher average peak force and the  
 45 energy to peak force when compared to LDPE [3]. LLDPE is a linear polyethylene with a  
 46 significant number of short branches (see Figure 2) commonly made by copolymerization  
 47 of ethylene and another longer olefin, which is incorporated to improve properties such  
 48 as tensile strength or resistance to harsh environments. The structure of LLDPE leads to  
 49 its heterogeneous non-linear behavior.



**Figure 2.** Chain structures of HDPE, LLDPE, and LDPE [4].

50 LLDPE is very flexible, elongates under stress, absorbs a high level of impact  
 51 energy and thus is suitable to make thin and ultra-thin films [5–8]. The mechanical  
 52 properties of polyethylene depend on its complex structure [9], which leads to a non-  
 53 linear heterogeneous behavior during mechanical and numerical tests. This behavior was  
 54 explained e.g. by [Jordan et al. \(2016\)](#), [Ragaert et al. \(2016\)](#) and [Zhang et al. \(2004\)](#), where  
 55 the differences in the chain structures among the HDPE (high-density polyethylene),  
 56 LLDPE, and LDPE (low-density polyethylene) are described. [Ren et al. \(2019\)](#) finds out  
 57 that LLDPE film MD tear strength is dependent on the utilized comonomers (higher for  
 58 hexene and octene-based resins whilst lower for butene-based resins). [Dogru et al. \(2018\)](#)  
 59 and [Dorigato et al. \(2010\)](#) defines the Poisson ratio as  $\nu = 0.44$  for LLDPE.

60 The main mechanical characteristics of polyethylene are the yield stress and the  
 61 yield strain, corresponding to the point where the plastic non-recoverable deformation  
 62 due to permanent changes in polymer chains starts. The yield stress and the yield strain  
 63 of LLDPE depend on temperature and the strain rate [5,6,15]. The yield stress increases

64 while the yield strain decreases with the increasing strain rate [9]. The double yield  
 65 point is also mentioned by Plaza *et al.* (1997). The relation between the yield stress,  
 66 the temperature and the strain rate can be described by constitutive laws [5,6,9,17].  
 67 The temperature-dependent mechanical properties of thin-layered materials are also  
 68 described by Luyt *et al.* (2021). By comparison among LDPE, LLDPE and HDPE, LLDPE  
 69 showed greater rate sensitivity than the other two materials under both static and  
 70 dynamic regions of a compression test [9].

71 The typical stress-strain relation as well as the strain-rate dependence are sketched  
 72 in Figure 3. Whilst Du *et al.* (2018) states the tensile properties dependent on the strain  
 73 rate, Omar (2013) shows, how the yield stress depends on the strain rate.

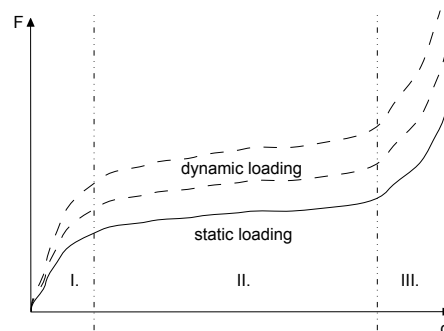


Figure 3. Typical stress-strain curve of LLDPE .

74 Durmus *et al.* (2008) shows the typical stress-strain curves of LLDPE as an initial  
 75 elastic region I. followed by yielding that is accompanied by neck propagation in region  
 76 II., see Figure 3. The third region III. is the stiffening leading to the material rupture.

77 LLDPE has anisotropic behavior due to its chain structure. The chain structure  
 78 creates the anisotropy in 2 perpendicular directions, called the machine direction (MD)  
 79 and the transversal direction (TD). The local preferential orientation of chains in LLDPE  
 80 affects the tensile strength in MD and TD [11]. In the direction of the main chain  
 81 orientation, mostly the MD, LLDPE are stiffer than in the perpendicular direction, mostly  
 82 the TD [3,5,6,15]. The tensile stress-strain relations in MD and TD play an important role  
 83 during the biaxial deformation of the impact test [3].

## 84 2. Materials and Methods

85 The material parameter identification is applied to the LLDPE thin foil produced  
 86 by Tic (2020). Table 1 summarizes its parameters presented by the producer.

Table 1: LLDPE properties as produced by Tichelmann (2020).

Physical properties	Unit	Tolerance $\pm$	Value	Testing method
Thickness	$\mu\text{m}$	2	12	Thickness gauge
Width	mm	5	500	Measuring tape
Length	-	-	5	High speed encoder
Density	$\text{g}/\text{cm}^3$	-	0.91 - 0.92	ASTM D-1505 [23]
Mechanical properties	Unit	Tolerance $\pm$	Value	Testing method
Tensile strength MD	MPa		29.2	
Tensile strength TD			14.1	
Break elongation MD	%	10	245	ASTM D-882 [23]
Break elongation TD			540	
Dart drop	g		40	ASTM D-1709 [23]
Puncture	kg		1.7	
Stretching level	-	-	110	High light tester

87 2.1. Quasi-static loading

88 The unilateral quasi-static loading test of the material sample was executed using  
 89 the testing machine 574LE2 TestResources. From the material roll (see Figure 4) provided  
 90 by the producer, the testing samples of length  $l_0 = 5$  mm and width  $w = 10$  mm were  
 91 extracted, see Figure 5 (a), where the left and right yellow sides are fixed to the testing  
 92 machine jaws. The thickness of the sample was  $h = 12$   $\mu$ m. The samples were fixed in  
 93 the testing machine jaws (see Figure 5 (b)) and stretched in 2 major orthotropic directions  
 94 (MD and TD). MD is in the direction, in which the material is wound up on the roll,  
 95 whilst TD is perpendicular to MD, see Figure 4.

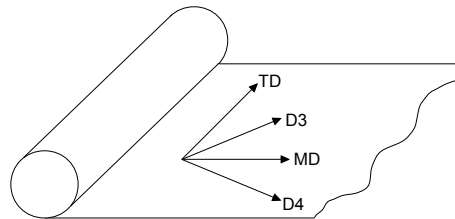
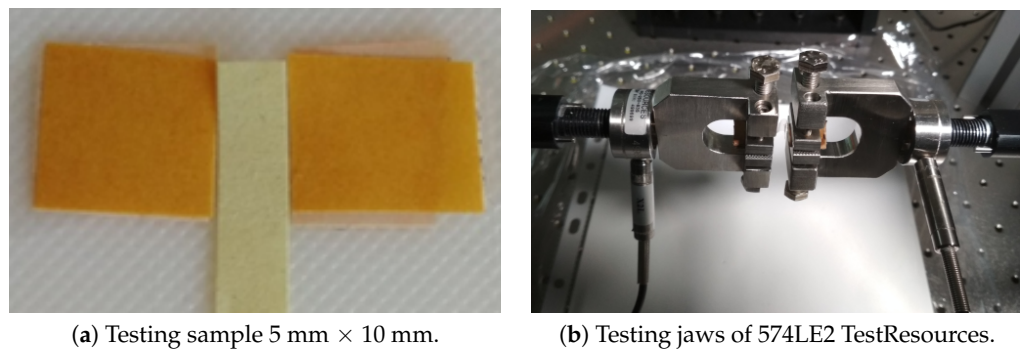


Figure 4. Sketch of the material roll.



(a) Testing sample 5 mm  $\times$  10 mm.

(b) Testing jaws of 574LE2 TestResources.

Figure 5. Quasi-static test setup: (a) Testing sample. (b) Testing jaws.

96 Several samples were tested in each direction by three different stretching velocities  
 97  $v$ , namely 0.0002 m/s, 0.02 m/s and 0.2 m/s, half per each side of the jaws. Additional  
 98 tests in the directions between MD and TD (labeled as D3 and D4, see Figure 4) were  
 99 done to check the material behavior in the skewed (45 degrees) direction. Table 2  
 100 summarizes all the quasi-static tests.  $N = 6$  samples were measured in each direction  
 101 for each velocity except the  $v = 0.2$  m/s, where D4 was already not necessary to be  
 102 measured. The particular test finished when the sample ruptured.

Table 2: Quasi-static tests matrix.

Stretching velocity $v$ [m/s]	Directions	Number of samples $N$
0.0002	MD	6
	TD	6
	D3	6
	D4	6
0.02	MD	6
	TD	6
	D3	6
	D4	6
0.2	MD	6
	TD	6
	D3	6

103 During the sample stretching, force  $F$  versus displacement  $d$  was recorded. Based  
 104 on the sample size with the sample initial cross-sectional area  $A_0 = hw$ , the engineering  
 105 stress  $\sigma$  versus engineering strain  $\epsilon$  curves were calculated as

$$\sigma = \frac{F}{A_0}, \quad \epsilon = \frac{d}{l_0}. \quad (1)$$

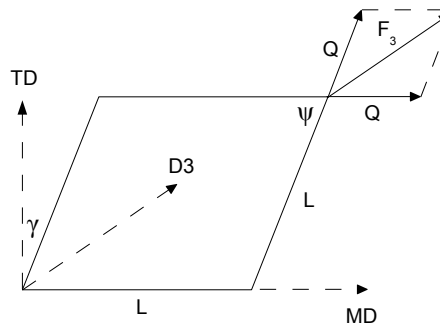
106 The constant Young modulus  $E$  was also identified as the slope of the initial elastic  
 107 region as

$$\sigma = \frac{F}{A_0} = E \frac{d}{l_0} \Rightarrow E = \frac{Fl_0}{dA_0}. \quad (2)$$

108 Fulfilling the aim of this study, the quasi-static tests were reproduced by numerical  
 109 simulations. The simulation was realized in Virtual Performance Solution (VPS by ESI  
 110 Group), version 2020. Following the structure of LLDPE in Figure 3 (2 mutually perpen-  
 111 dicular sets of fibers), the material model 151 Fabric Membrane Element with Nonlinear  
 112 Fibers [24] from the ESI constitutive material model database was proposed. According  
 113 to the membrane theory, the stress resultant curves were calculated by multiplying the  
 114 engineering stress by the membrane thickness as

$$\sigma_h = \sigma h \quad (3)$$

115 in both MD and TD. The resulting material curves taken as the average curves from  
 116 the quasi-static test measurements in particular directions serve as the constitutive  
 117 data to feed the material model 151. The model concerns 2 sets of fibers, whose stress  
 118 versus strain relation is defined by engineering stress resultant versus engineering strain  
 119 curve. The angle between the sets of fibers is  $90^\circ$ . The shear stress resultant necessary  
 120 to complete the membrane material model was calculated using the measurement in  
 121 direction D3 as shown in Figure 6.



**Figure 6.** Evaluating the shear stress resultant.

122 Supposing a square sample, the shear force  $Q$  and shear angle  $\gamma$  are calculated  
 123 through the following formulas.

$$Q = \frac{F_3}{2 \cos \frac{\psi}{2}}, \quad (4)$$

124 where the shear angle

$$\gamma = \frac{\pi}{2} - \psi \quad (5)$$

125 is calculated based on the deformed sample angle  $\psi$  as

$$\cos \frac{\psi}{2} = \frac{\sqrt{2L + d}}{2L}, \quad (6)$$

126 where  $L$  is the side of the square sample,  $d$  is the displacement in direction D3 and  $F_3$  is  
 127 the force recorded in direction D3. Therefore the shear stress can be calculated as

$$\tau = \frac{Q}{Lh} \quad (7)$$

128 and the shear stress resultant is

$$\tau_h = \tau h = \frac{Q}{L}. \quad (8)$$

129 The thickness of the material is  $h = 12 \mu\text{m}$  as defined by the producer [22]. The  
 130 energy absorption was calculated from the dynamic experimental measurements and  
 131 other numerical parameters feeding the material model were used as proposed by the  
 132 VPS manual [24]. A single 4-nodal membrane element model was loaded by stretching  
 133 both sides of the element by the 3 different loading velocities  $v$ , namely 0.0002 m/s, 0.02  
 134 m/s and 0.2 m/s, half per each side of the jaws, see Figure 7.

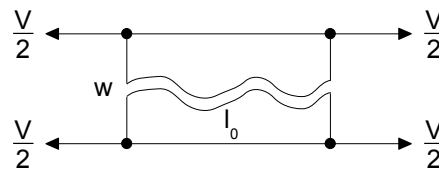


Figure 7. Single element quasi-static stretching simulation setup.

135 The element section force leading to the stress resultant was recorded during the  
 136 simulation to be compared to the experimental data.

### 137 2.2. Dynamic loading

138 The dynamic tests, realized to reproduce the scenario from Figure 1, were in the  
 139 form of drop tests of a spherical impactor falling with a given velocity on a multi-layered  
 140 material sample. A special drop tower was designed for this purpose, see Figure 8.

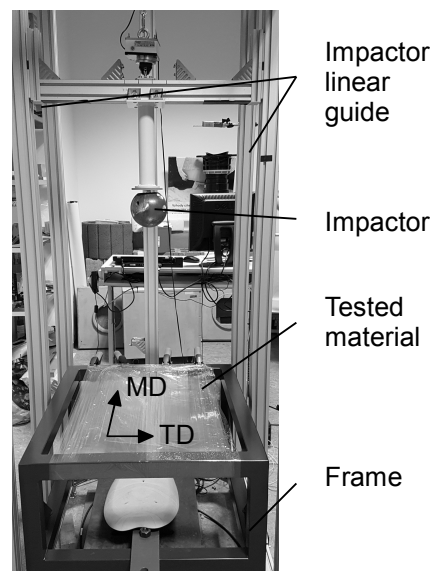


Figure 8. Drop tower.

141 According to Figure 1, the drop test was used to simulate the collision scenario  
 142 similar to the impact of the human head into the safety layers during the frontal crash.  
 143 Typical impacts for testing safety systems are designed for the velocities  $v_0$  equal to  
 144 30 km/h and 50 km/h, corresponding to those used in sled tests. As the mass of the

145 human head is approximately  $m = 4.5$  kg and the mass of the testing impactor is  
 146  $M = 10.72$  kg, the drop test height  $H$  was calculated from the energy balance equation

$$\frac{1}{2}mv_0^2 = MgH \quad (9)$$

147 using the gravity acceleration  $g = 9.81$  m/s<sup>2</sup>. Equation (9) yields the drop heights  
 148 equal to 1.49 m and 4.13 m for the velocities 30 km/h and 50 km/h, respectively. Due  
 149 to the design limitations (limited maximum height of the drop tower), only the height  
 150  $H = 1.5$  m corresponding to the velocity  $v_0 = 30$  km/h was considered. To include  
 151 different impact velocities for optimizing the constitutive material model, additional  
 152 tests at the height  $H = 1$  m corresponding to the velocity  $v_0 = 25$  km/h were realized.  
 153 Relating the energy balance with the head impactor mass  $M = 10.72$  kg, the impact  
 154 velocities corresponded to 4.43 m/s and 5.43 m/s for  $H = 1$  m and 1.5 m, respectively.

155 As the dynamic impact loading was aggressive, the target material was wind up on  
 156 the frame in several layers, see Figure 9. Preliminary experiments showed the sufficient  
 157 number of layers  $n$  to be 8, 9 and 10, so the matrix of experiments contained 2 drop  
 158 heights (10 dm and 15 dm)  $\times$  3 sets of layers (8, 9, and 10).

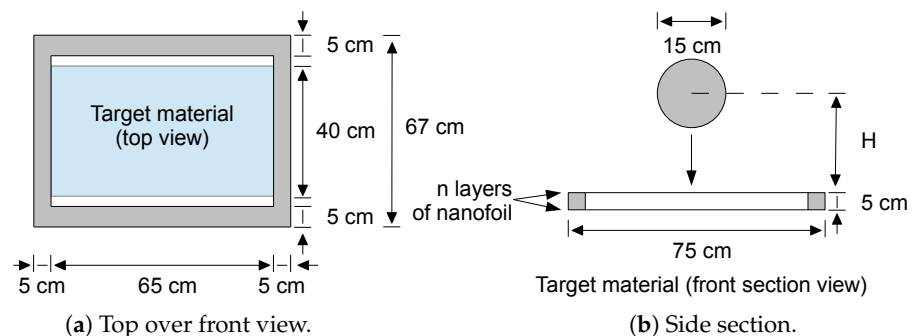


Figure 9. Drop tower scheme: (a) Top over front view. (b) Side section.

159 Table 3 summarizing the drop tests shows that finally only 5 experimental drop  
 160 tests were used for the optimization procedure, as the most aggressive one, meaning the  
 161 fall from the highest height  $H = 15$  dm to the lowest number of layers  $n = 8$ , ruptured  
 162 the target material layers. The last column of Table 3 designates the identification of the  
 163 particular drop tests in the following figures and analyses.

Table 3: Drop tests matrix.

Drop height $H$ [dm]	Number of layers $n$	Optimization	Designation
10	8	✓	1008
	9	✓	1009
	10	✓	1010
15	8	✗ (material ruptured)	
	9	✓	1509
	10	✓	1510

164 The acceleration was measured using a uniaxial piezoelectric accelerometer Kistler  
 165 8742A5 fixed to the impactor, with the axis of the measurement parallel to the axis of  
 166 the impactor. The impactor was held by an electromagnet and the free-fall motion was  
 167 controlled by a linear guide, see Figure 9. Additionally, the deflection of the impactor was  
 168 measured with the laser measuring system Micro-Epsilon optoNCDT 2300-50 connected  
 169 to the voltage input module NI 9215 in NI cDAQ-9178 Chassis. The final time-correlated  
 170 signals were recorded by the NI Signal Express Software. The measured acceleration  
 171 signal was filtered by the CFC 1000 filter [25]. From the physical principle, the piezoelec-

172 tric accelerometer cannot measure a free-fall gravity acceleration [26]. The experimental  
 173 acceleration curve decreases to minus  $g$  just after releasing and reaches the equilibrium  
 174  $0\text{ g}$  during the free fall, so the experimental acceleration curve needs to be adjusted to be  
 175 comparable to the simulation results.

176 As the measured displacement was limited by the range of the laser measuring  
 177 system, the double integration of the acceleration signal was used to extend the displace-  
 178 ment in the whole time interval of the loading and unloading phases of the impact. Using  
 179 the updated acceleration and displacement signals, the time-dependent total energy  
 180 composed by the kinetic energy, the potential energy and the work done, respectively,  
 181 were monitored as

$$E(t) = E_k + E_p + W = \frac{1}{2}Mv(t)^2 + Mgd(t) + \int_0^{d(t)} Ma(t)ds \quad (10)$$

182 to check the correctness of the calculations and to identify the energy absorption. Here  
 183  $a(t)$  is the updated measured time-dependent impactor acceleration and the impactor  
 184 velocity  $v(t)$  and the impactor displacements  $d(t)$  are calculated by the first and the  
 185 second integration, respectively, of the acceleration signal  $a(t)$ . The gravity acceleration  
 186  $g$  is subtracted from the impactor acceleration to subtract the work done by the potential  
 187 energy. Marking  $E_{kp}(t) = E_k(t) + E_p(t)$  as the time-dependent sum of the kinetic and  
 188 total energies, the energy absorption was calculated as the energy loss

$$D = 1 - \frac{\Delta E_u}{\Delta E_l}, \quad (11)$$

189 where  $\Delta E_l = \max E_{kp}(t) - \min E_{kp}(t)|_{\text{loading}}$  is the energy difference during the loading  
 190 phase and  $\Delta E_u = \max E_{kp}(t) - \min E_{kp}(t)|_{\text{unloading}}$  is the energy difference during the  
 191 unloading phase, where the resting energy is absorbed by the material work in order to  
 192 have the constant total energy  $E(t)$  from Equation (10).

### 193 2.3. Dynamic material parameters identification

194 As the material properties of LLDPE are strain-rate dependent [21], the constitutive  
 195 material curves achieved by the quasi-static experimental measurements were used as  
 196 the initial optimization step for dynamic material parameters optimization. The opti-  
 197 mization was done using the numerical simulation reproducing the drop test experiment.  
 198 The strain-rate dependent curves from the first optimization ( $H = 10\text{ dm}$  and  $n = 8$   
 199 layers) were used as the initial curves for the other optimization runs to speed up the  
 200 optimization process.

201 The standard MATLAB function *fminsearch* was adopted to optimize values for the  
 202 stiffness and the yield stress in the two directions MD and TD towards the expected  
 203 values. According to Figure 3, the stiffness of region I. and the yield stress were optimized.  
 204 The MD and TD curves in region I. were updated as

$$\sigma_h^{\text{MD}} = k_y \sigma_h^{\text{MD}} \left( \frac{\epsilon_I^{\text{MD}}}{k_1 k_e} \right), \quad k_1 = \begin{cases} \bar{k} & \forall \epsilon_I^{\text{MD}} \in [0, \epsilon_{y1}^{\text{MD}}] \\ 1 & \forall \epsilon_I^{\text{MD}} \in (\epsilon_{y1}^{\text{MD}}, \epsilon_{y2}^{\text{MD}}] \end{cases} \quad (12)$$

$$\sigma_h^{\text{TD}} = k_y \sigma_h^{\text{TD}} \left( \frac{\epsilon_I^{\text{TD}}}{k_1 k_e} \right), \quad k_1 = \begin{cases} \bar{k} & \forall \epsilon_I^{\text{TD}} \in [0, \epsilon_{y1}^{\text{TD}}] \\ 1 & \forall \epsilon_I^{\text{TD}} \in (\epsilon_{y1}^{\text{TD}}, \epsilon_{y2}^{\text{TD}}] \end{cases} \quad (13)$$

205 by multiplying by dimensionless coefficients  $k_y$ ,  $\frac{1}{k_1}$  and  $\frac{1}{k_e}$  during the optimization  
 206 process. As the strain is divided by  $k_1 k_e$ , the coefficient  $\frac{1}{k_1 k_e}$  is the stiffness multiplier in  
 207 the region I. Dividing the region I.  $[0, \epsilon_{y2}^{\text{MD,TD}}]$  by the first yield point  $\epsilon_{y1}^{\text{MD,TD}}$ ,  $k_1$  stiffens

only the first part of the region I. by a constant  $\bar{k}$  till the first yield point is reached. Coefficient  $k_y$  is the yield stress multiplier.

The optimization process was run in a loop controlled by a MATLAB script updating the constitutive material curves in MD and TD according to Equations (12) and (13). The cost function in the optimization measured the relative acceleration error  $E_a$  defined as

$$E_a = \frac{\|a_s(t) - a_e(t)\|}{\|a_e(t)\|} \Big|_{t \in [t_1, t_2]} \quad (14)$$

noindent where  $a_e(t)$  is the time-dependent acceleration signal measured from the experiment,  $a_s(t)$  is the time-dependent acceleration response calculated by the numerical simulation and  $t$  is the time in the error calculation interval  $[t_1, t_2]$ . As well as the experimental acceleration signal, the calculated acceleration signal was also filtered by the CFC 1000 filter [25]. Figure 10 shows the simulation setup for optimization runs. The initial prestrain of the material wound on the frame was estimated based on preliminary numerical simulations to be 10 %, i.e.  $\epsilon_0 = 0.1$  in MD. The displacement error  $E_d$  was calculated similarly to the acceleration error as

$$E_d = \frac{\|d_s(t) - d_e(t)\|}{\|d_e(t)\|} \Big|_{t \in [t_1, t_2]} \quad (15)$$

where  $d_e(t)$  is the time-dependent displacement signal obtained by the double integration of the acceleration signal and  $d_s(y)$  is the time-dependent displacement response calculated by the numerical simulation.

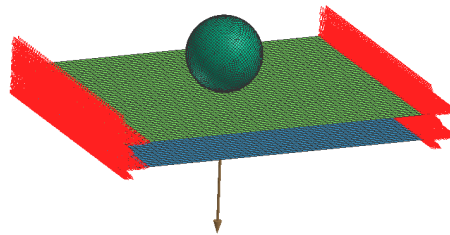


Figure 10. Drop test simulation setup.

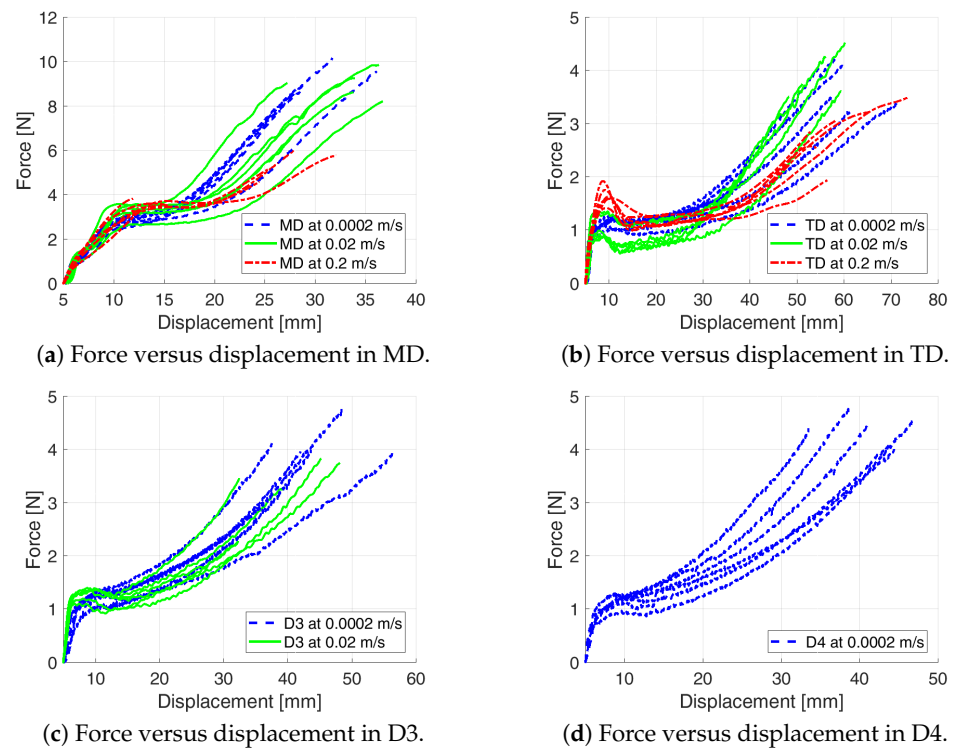
For each testing scenario with  $n$  layers, the material is modeled by a single-layered membrane elements, where the number of upper and lower layers of the model is specified using the membrane material thickness defined by multiplying the single-layer thickness  $h$  by the number of layers  $n$  meaning also, that the stress resultant curves in Equations (3) and (8) are also multiplied by  $n$  for the particular model. Both sides of the layers are fixed by boundary conditions representing the attachment to the frame. The spherical impactor is modeled as a rigid body situated just above the upper layer and loaded by the initial velocity  $v$  corresponding to the particular height. The vertical acceleration and the vertical displacement are stored and compared to the experimental data.

### 3. Results

The following figures and tables summarize the results from the quasi-static tests as well as the identification of LLDPE parameters under dynamic loading.

#### 3.1. Quasi-static loading

The quasi-static experiments prove that the typical stress versus strain curve for LLDPE is composed of three regions [21], see Figure 3. The summary of all results obtained by static experimental measurements under different quasi-static loading velocities using a single material layer is displayed in Figure 11. The curves are cut at positions of sample ruptures.



**Figure 11.** Material response in all directions: (a) MD. (b) TD. (c) D3. (d) D4.

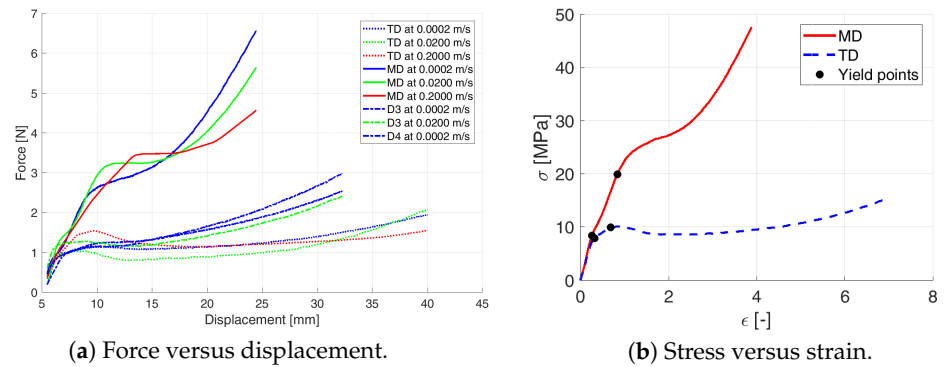
243 Table 4 compares the measured experimental properties to those defined by [Tic \(](#)  
 244 [2020](#)[\)](#).

Table 4: Material properties.

Direction	MD		TD	
Variable	Tensile stress [MPa]	Break elongation [%]	Tensile stress [MPa]	Break elongation [%]
Data sheet <a href="#">[22]</a>	29.2	245	14.1	540
Experiment	29.3	139	16.2	701
Error [%]	0.5	-43	15	30

245 As the quasi-static tests in all three stretching velocities show similar performance,  
 246 the curves for each direction were averaged as shown in Figure 12. It can be seen that  
 247 the stretching responses in the skewed directions D3 and D4 fit between the MD and  
 248 TD curves, so no unpredictable behaviour during the multi-directional loading should  
 249 be expected. Therefore the skewed direction D3 was also used to identify the shear  
 250 behavior according to Equations (4) – (8).

251 Whilst Figure 12 (a) shows the force dependent on the displacement averaged per  
 252 the direction and per the stretching velocity, Figure 12 (b) shows the total average of the  
 253 calculated stress versus strain curves in MD and TD calculated using Equation (1) for  
 254 each quasi/static test measurement.



**Figure 12.** Averaged constitutive material model curves: (a) Force versus displacement averaged per direction in all stretching velocities. (b) Stress versus strain averaged per direction.

255 By detailed analysis of the measured data in Figure 12 (b), the double yield point [16]  
 256 is observed in both directions. In MD, the first point appears at the stress  $\sigma_{y1}^{MD} = 8.4$   
 257 MPa, which corresponds to the strain  $\epsilon_{y1}^{MD} = 0.26$ . The second yield point appears by  
 258 reaching the stress  $\sigma_{y2}^{MD} = 20$  MPa, which corresponds to the strain  $\epsilon_{y2}^{MD} = 0.84$ . In  
 259 TD, the first yield point appears at the stress  $\sigma_{y1}^{TD} = 8$  MPa, which corresponds to the  
 260 strain  $\epsilon_{y1}^{TD} = 0.33$ . The second yield point appears before reaching the maximum stress  
 261 in region I. at the stress  $\sigma_{y2}^{TD} = 10$  MPa corresponding to the strain  $\epsilon_{y2}^{TD} = 0.69$ . Table 5  
 262 summarizes the yield points.

Table 5: Yield points ( $\sigma_{hy}$  means the yield stress resultant).

Yield point	MD			TD		
	$\epsilon_y$ [-]	$\sigma_y$ [MPa]	$\sigma_{hy}$ [N/mm]	$\epsilon_y$ [-]	$\sigma_y$ [MPa]	$\sigma_{hy}$ [N/mm]
1	0.26	8.4	0.1	0.33	8	0.1
2	0.84	20	0.24	0.69	10	0.12

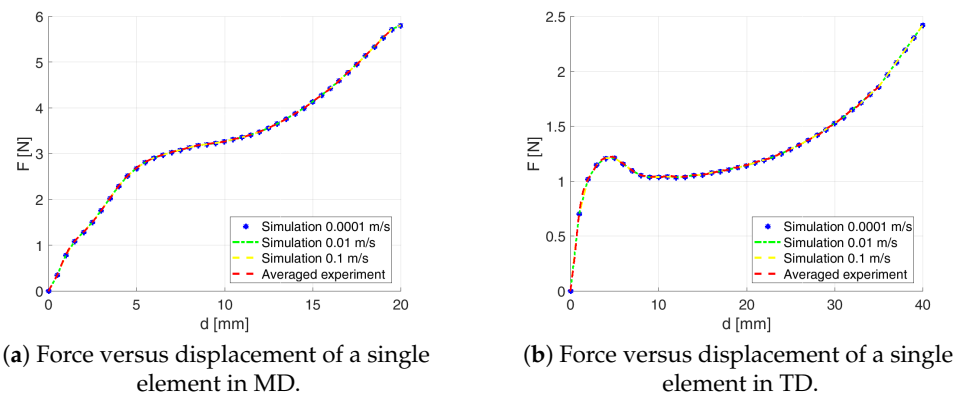
263 Taking into account the elastic region, the Young modulus  $E = 50$  MPa is identified  
 264 using Equation (2) by averaging the slopes of the elastic regions of all curves, see Table 6.  
 265 The average was calculated for the particular directions and stretching velocities firstly  
 266 leading to the global average. Both directions MD and TD are averaged as they exhibit  
 267 similar stiffness in the first region.

Table 6: Young modulus.

Direction	MD						TD							
	Stretching velocity $v$ [m/s]		0.0002		0.02		0.2		0.0002		0.02		0.2	
Young modulus E [Mpa]			44	63	63	67	36	42	54	76	55	41	35	26
			57	76	63	47	37	50	39	53	64	47	31	30
			63	41	81	70	31	30	55	44	49	58	27	38
Young modulus E [Mpa]			57		65		38		53		52		31	
					53						46			
E [Mpa]	(50)													

268 Finally, the constitutive material stress resultant curves developed using Equations  
 269 (3) and (8) for a single layer of LLDPE were calculated for the quasi-static loading to feed

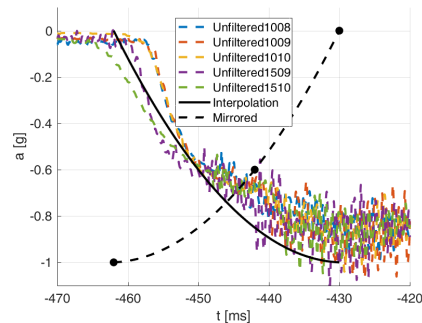
270 the constitutive material model, see Figure 21. A single-element numerical simulation to  
 271 reproduce the stretching was run. Figure 13 shows perfect fit to the experimental curves.



**Figure 13.** Performance of a single element model: (a) MD. (b) TD.

### 272 3.2. Dynamic loading

273 The acceleration decrease interval from 0 g to minus g approximately within the  
 274 first 32 ms was used as the approximated parabolic acceleration ramp (see Figure 14)  
 275 after the first contact of the impactor, where the mirrored signal from minus g to 0 g was  
 276 added to the measured acceleration at the first 50 ms after the first contact between the  
 277 impactor and the material, see Figures 15 – 19. By this the inability of the acceleration  
 278 sensor to measure the free fall acceleration was mitigated.



**Figure 14.** Gravity acceleration ramp.

279 The time of the first contact of the impactor to the material as well as the impact  
 280 velocity were estimated from the ideal free fall from the height  $H$  after releasing the  
 281 electromagnet. Due to uncertainty in the frame versus impactor linear guide friction, the  
 282 related actual impact velocity and time of contact, an iterative process starting from the  
 283 free fall assumptions was used to determine the actual moment of impact and the impact  
 284 velocity, based on comparing the doubly integrated accelerations to the displacements  
 285 obtained by the laser measuring system.

286 Such process led to a perfect fit in both measured and calculated displacements  
 287 (both shown in Figures 15 – 19) identifying also the real impact velocity (see Table 7).  
 288 The only exception was scenario 1509, where the displacement measurement failed. So  
 289 the impact velocity was estimated to fit the remaining part of the displacement curve.

290 The dynamic loading proved the strain-rate dependency of LLDPE. LLDPE also  
 291 exhibits strong energy absorption. The energy absorption was calculated by Equation (11)  
 292 and was identified as being similar for all five drop test scenarios and averaged per  
 293 drop height to obtain the final average  $D = 88.96\%$  (see Table 7) used for the constitutive  
 294 material model.

Table 7: Energy absorption.

Drop height $H$ [dm]	10			15	
Number of layers $n$	8	9	10	9	10
Impact velocity $v$ [m/s]	4.16	4.14	4.18	4.94	5.01
Energy absorption $D$ [%]	90.03	88.26	87.61	89.13	89.43
Energy absorption $\bar{D}$ [%]		88.63		89.28	
			(88.96)		

### 295 3.3. Dynamic material parameters identification

296 Several approaches to optimize the strain-rate dependent constitutive material  
 297 curves were used and finally, the same stiffening ratio in MD and TD was proposed.  
 298 The optimization process controlled by a MATLAB script involved running a series  
 299 of simulations for updating the constitutive material model curves. The quasi-static  
 300 response was taken as the initial guess for the optimization.

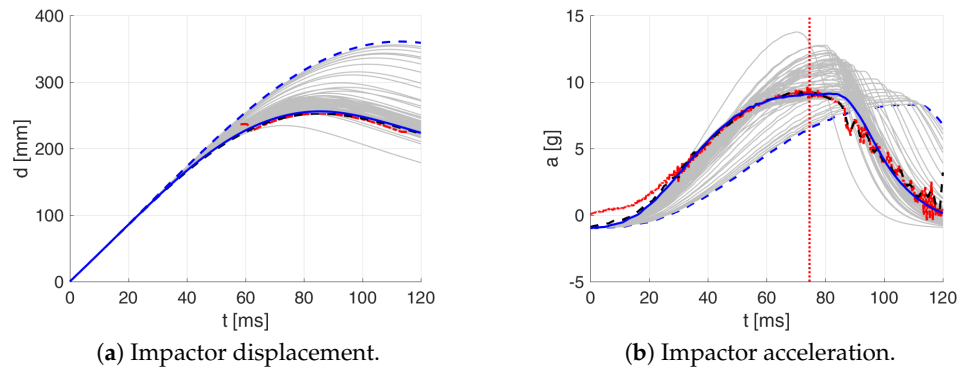
301 Table 8 shows the coefficients coming from the optimization process. Table 8 also  
 302 shows the number of iterations leading to the optimized constitutive material curves as  
 303 well as the errors from the cost function calculated by Equation (14) and the error in the  
 304 displacement calculated by Equation (15).

Table 8: Optimised coefficients.

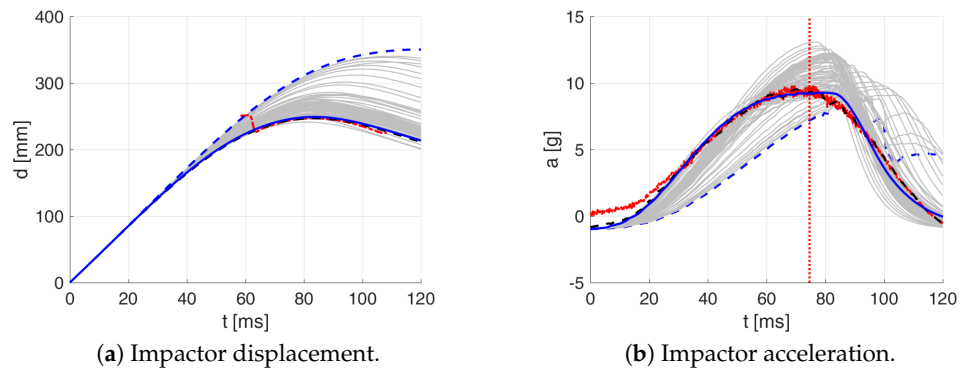
Drop height $H$ [dm]	10			15	
Number of layers $n$	8	9	10	9	10
Number of iterations	278	152	205	234	221
First part stiffness multiplier $k_1$ [-]	2.75	2.89	2.99	3.14	3.07
Stiffness multiplier $k_e$ [-]	3.41	3.47	3.55	1.69	2.29
Yield stress multiplier $k_y$ [-]	1.00	0.91	0.88	1.16	1.04
Acceleration error $E_s$ [%]	3	3	2	2	3
Displacement error $E_d$ [%]	1	1	1	0	1

305 The intervals for calculating acceleration error are limited in Figures 15 – 19 by red  
 306 dotted vertical lines to consider only the loading, where the iterative processes for the  
 307 particular drop heights and the particular number of layers are shown.

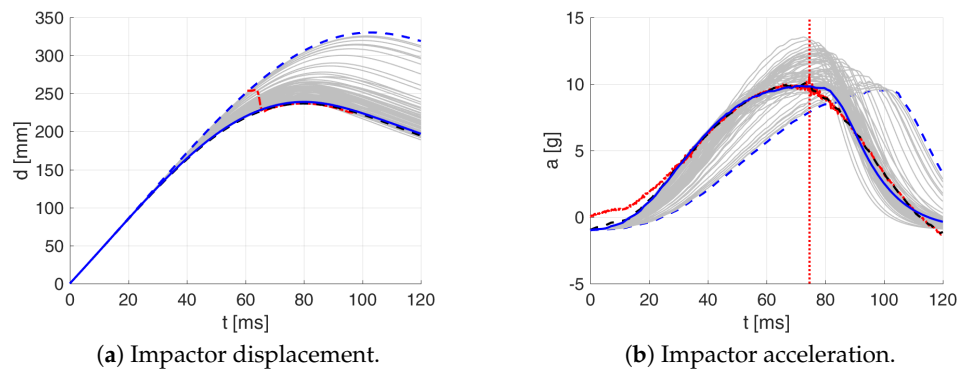
308 The original experimental curves are in red dashed lines, the updated target curves  
 309 (displacement obtained by integration and acceleration updated by the gravity) are  
 310 shown in black dashed color, the initial curves (using the static constitutive material  
 311 model) for optimization iterations are shown in blue dashed lines, the optimized curves  
 312 are shown in blue solid lines and the iterative process is shown by solid grey curves.



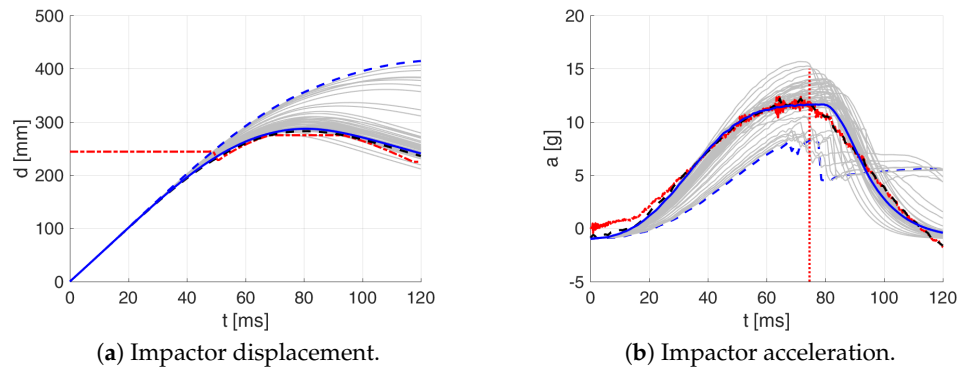
**Figure 15.** Optimisation iterations for the drop height  $H = 10$  dm and  $n = 8$  layers: (a) Displacement. (b) Acceleration.



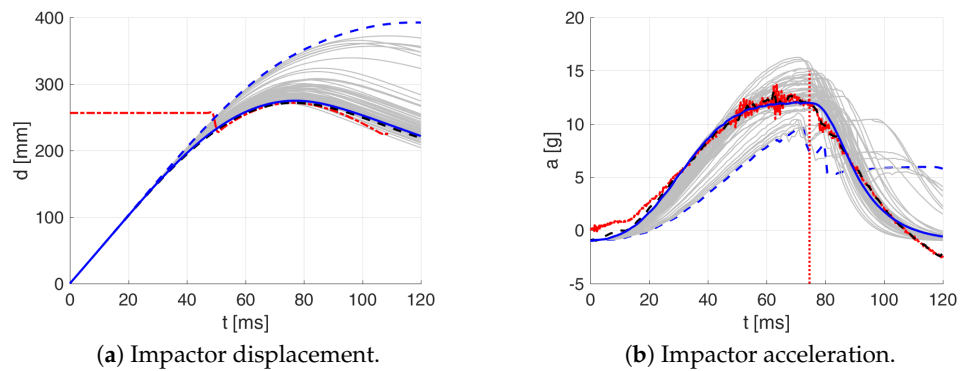
**Figure 16.** Optimisation iterations for the drop height  $H = 10$  dm and  $n = 9$  layers: (a) Displacement. (b) Acceleration.



**Figure 17.** Optimisation iterations for the drop height  $H = 10$  dm and  $n = 10$  layers: (a) Displacement. (b) Acceleration.

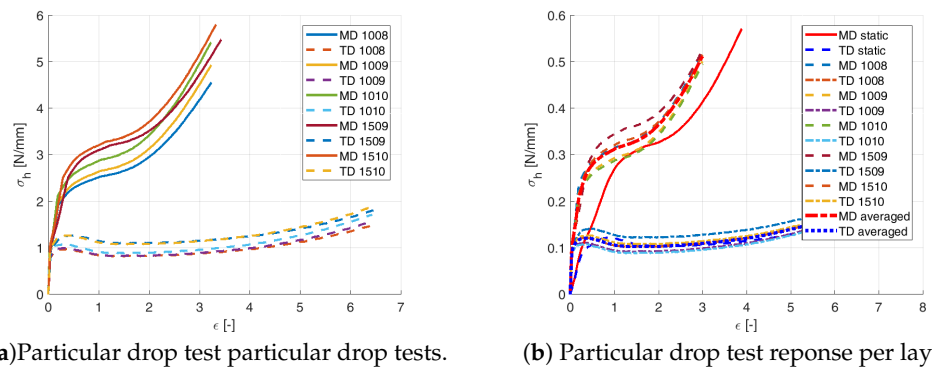


**Figure 18.** Optimisation iterations for the drop height  $H = 15$  dm and  $n = 9$  layers: (a) Displacement. (b) Acceleration.



**Figure 19.** Optimisation iterations for the drop height  $H = 15$  dm and  $n = 10$  layers: (a) Displacement. (b) Acceleration.

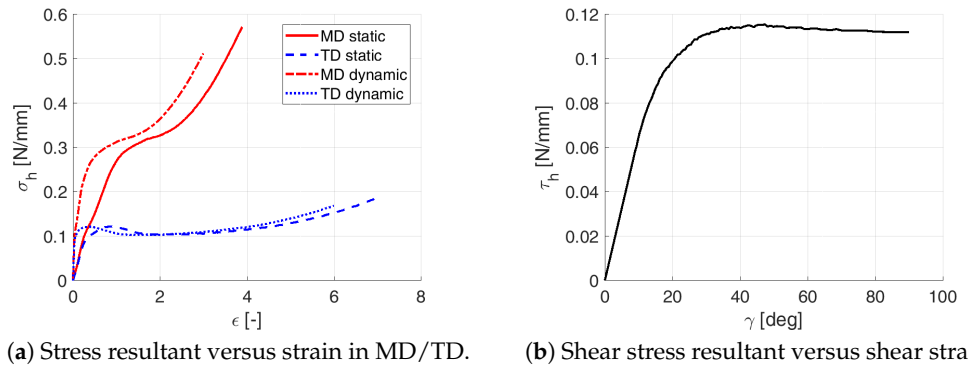
313 All the identified strain-rate dependent engineering stress resultant versus engi-  
 314 neering strain constitutive material curves in both MD and TD are show in Figure 20 (a).  
 315 Due to the two different drop heights and three different sets of multi-layers, each drop  
 316 scenario provides a different strain-rate dependent response, so all tests were normalized  
 317 by the number of layers, which leads to similar strain-rate constitutive material curves  
 318 for a single layer is both MD and TD, see Figure 20 (b).



**Figure 20.** Strain-rate dependent constitutive material model curves: (a) For particular tests. (b) Per layer.

319 As the difference between the curves corresponds to the difference during the  
 320 experimental measurement, the constitutive material curves in MD and TD are identified  
 321 by averaging the drop tests, see Figure 21 (a). Figure 21 (b) shows the shear stress versus  
 322 shear strain as calculated by Equations (3) – (8) for a single layer.

323 Using the identified averaged constitutive material curves in both MD and TD  
 324 and the average energy absorption, all drop tests were reconstructed by numerical  
 325 simulations. The results are shown in Figures 22 – 26.



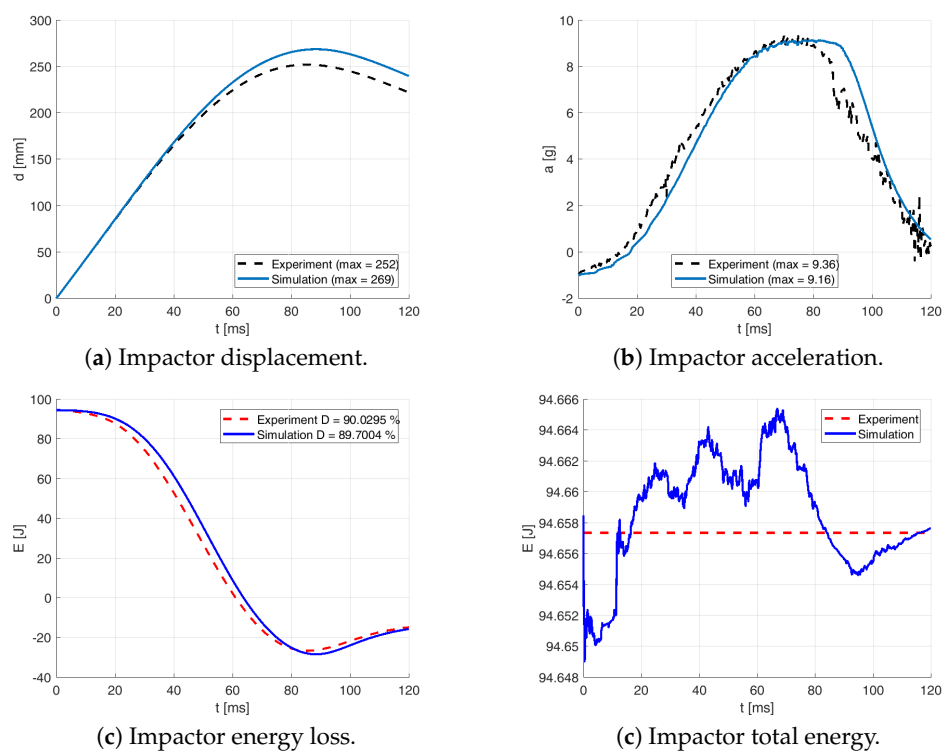
(a) Stress resultant versus strain in MD/TD. (b) Shear stress resultant versus shear strain.  
**Figure 21.** Averaged strain-rate dependent constitutive material model curves: (a) Major directions.  
 (b) Shear.

326 Table 9 shows the agreement in acceleration and displacement for all the drop tests  
 327 using the averaged constitutive material curves.

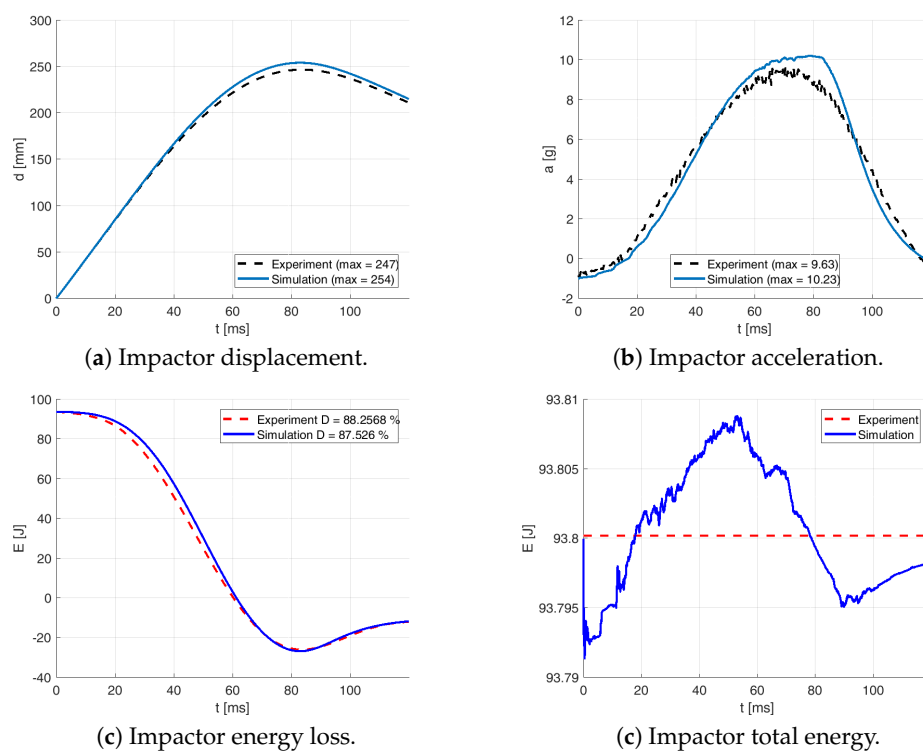
Table 9: Final results.

Drop height $H$ [dm]	10			15	
Number of layers $n$	8	9	10	9	10
Acceleration error $E_s$ [%]	6	8	10	6	3
Displacement error $E_d$ [%]	5	2	2	3	2

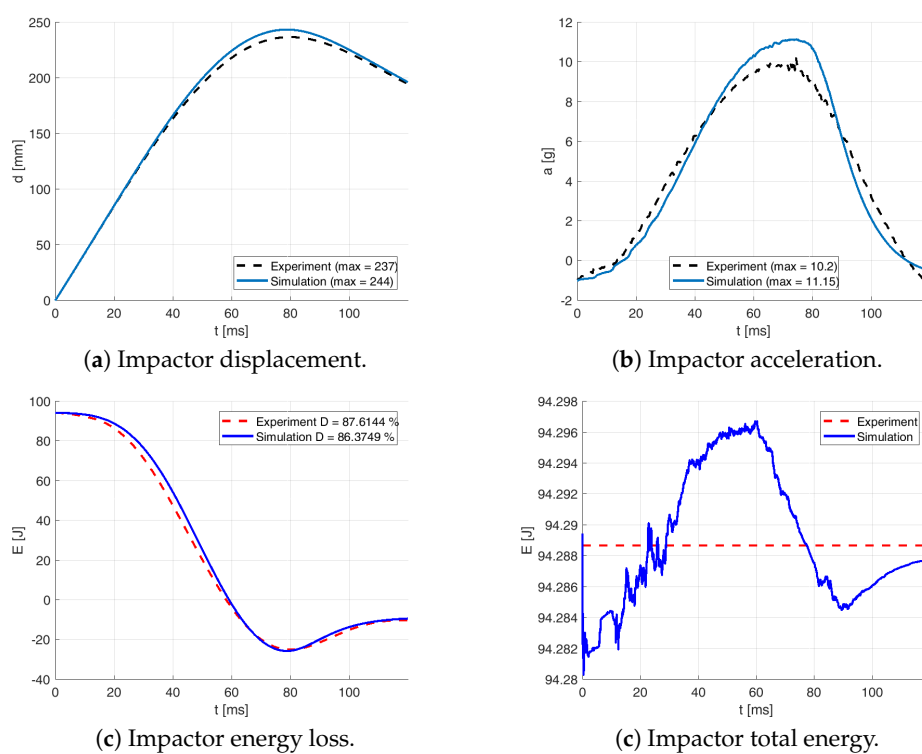
328 Figure 22 compares the simulation to the experimental drop test for the drop height  
 329  $H = 10$  dm and the number of layers  $n = 8$ . Figure 23 compares the simulation to the  
 330 experimental drop test for the drop height  $H = 10$  dm and the number of layers  $n = 9$ .  
 331 Figure 24 compares the simulation to the experimental drop test for the drop height  
 332  $H = 10$  dm and the number of layers  $n = 10$ . Figure 25 compares the simulation to the  
 333 experimental drop test for the drop height  $H = 15$  dm and the number of layers  $n = 9$ .  
 334 Figure 26 compares the simulation to the experimental drop test for the drop height  
 335  $H = 15$  dm and the number of layers  $n = 10$ .



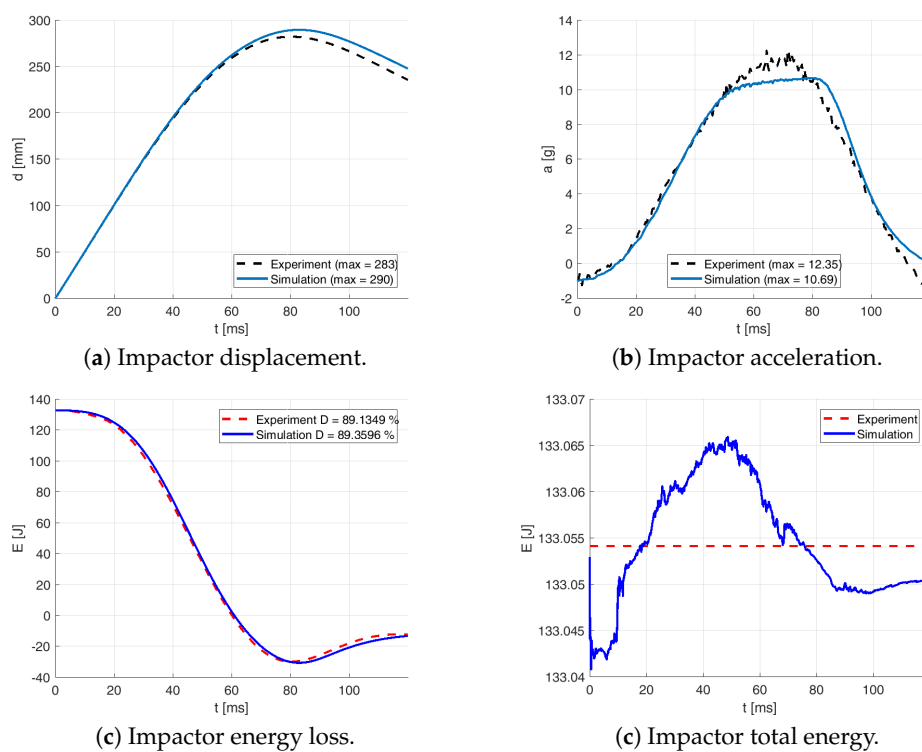
**Figure 22.** Comparison of drop test simulation to experiment for the drop height  $H = 10$  dm and  $n = 8$  layers: (a) Displacement (b) Acceleration. (c) Energy loss. (d) Total energy.



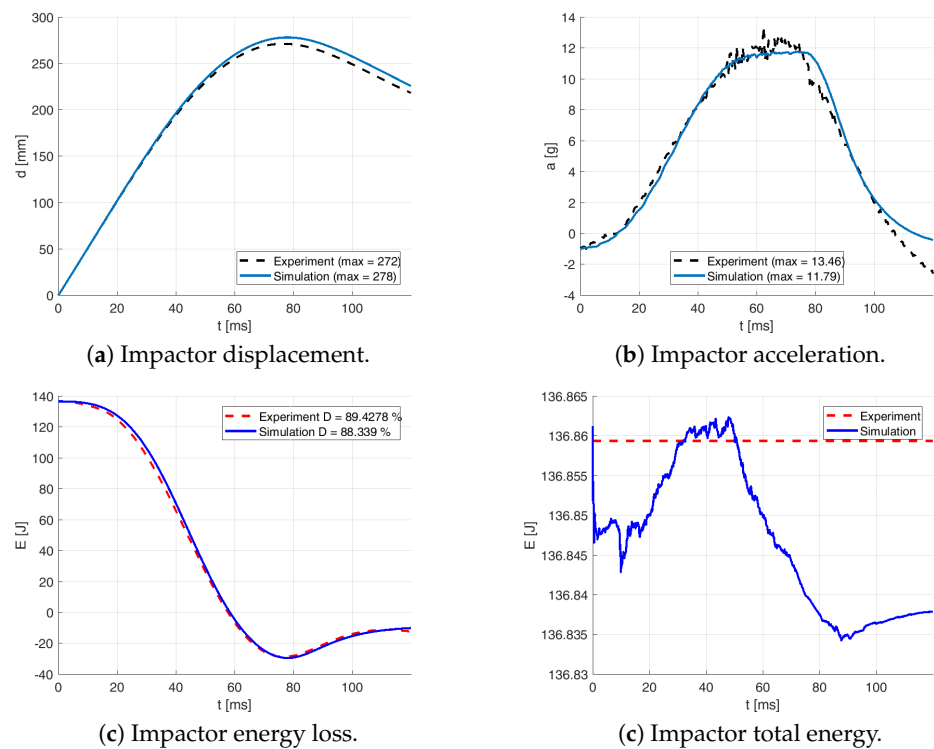
**Figure 23.** Comparison of drop test simulation to experiment for the drop height  $H = 10$  dm and  $n = 9$  layers: (a) Displacement (b) Acceleration. (c) Energy loss. (d) Total energy.



**Figure 24.** Comparison of drop test simulation to experiment for the drop height  $H = 10$  dm and  $n = 10$  layers: (a) Displacement (b) Acceleration. (c) Energy loss. (d) Total energy.



**Figure 25.** Comparison of drop test simulation to experiment for the drop height  $H = 15$  dm and  $n = 9$  layers: (a) Displacement (b) Acceleration. (c) Energy loss. (d) Total energy.



**Figure 26.** Comparison of drop test simulation to experiment for the drop height  $H = 15$  dm and  $n = 10$  layers: (a) Displacement (b) Acceleration. (c) Energy loss. (d) Total energy.

#### 336 4. Discussion

337 The quasi-static experiments were performed in 2 perpendicular directions sup-  
 338 ported by measurements in 2 skewed directions. Although MD and TD exhibited  
 339 different loading behavior, the measurements in the skewed directions support that  
 340 there is not any unexpected behavior during loading in any auxiliary direction.

341 Table 4 shows a good agreement with factory data of the quasi-static experimental  
 342 test regarding the tensile stress in both directions. The break elongation is 30 % higher  
 343 in TD and 43 % lower in MD comparing to the material data sheet in Table 1, which  
 344 might be caused by the laboratory conditions and influenced by the specimen size. The  
 345 experimental measurements also confirms the previous studies that LLDPE is stiffer in  
 346 MD comparing to TD [3,5,6,15].

347 Table 5 summarizes the yield stresses  $\sigma_y^{\text{MD}} = 8.4$  MPa and  $\sigma_y^{\text{TD}} = 8$  MPa as well  
 348 as the yield strains  $\epsilon_y^{\text{MD}} = 0.26$  and  $\epsilon_y^{\text{TD}} = 0.33$ , which are comparable to the values  
 349 presented by Durmus *et al.* (2008), who states the yield stress  $\sigma_y = 9.9$  MPa and the yield  
 350 strain  $\epsilon = 0.33$ . However, Durmus *et al.* (2008) measured the elongation at break equal  
 351 to 1045 %, which is higher to those measured and stated by the material data sheet. The  
 352 Young modulus in Table 6  $E = 50$  MPa shows also a comparable value to that published  
 353 by Durmus *et al.* (2008), who identifies the Young modulus experimentally as  $E = 64$   
 354 MPa.

355 The drop test experimental measurements prove considerable energy absorption  
 356 summarized in Table 7, which was used in the constitutive material model for the  
 357 dynamic response. To keep a stable optimization in MD and TD, the same multipliers  
 358 were supposed for developing the dynamic constitutive material model in MD and TD.  
 359 The optimized multipliers as well as optimization process errors are stated in Table  
 360 8. The optimization process leads to the stiffening about 3.5 times for the drop height  
 361  $H = 10$  dm as the stiffening is about 2 for the drop height  $H = 15$  dm. The yield stress  
 362 balances around the measured quasi-static value.

363 The acceleration error was calculated just during the loading phase, because of the  
364 complex unloading behavior and because of the fact that the constitutive material model  
365 is being developed for the energy absorption during the loading.

366 The dynamic response exhibits similar values for both drop heights, so single dy-  
367 namic constitutive material curves were developed by averaging the particular response  
368 curves in MD and TD. The averaged constitutive material curves in MD and TD were  
369 then used to recalculate all the drop test again with the error shown in Table 9. The  
370 developed constitutive material model describes well the LLDPE foil behavior to be  
371 used for energy absorption during the impact.

372 Even though the identified constitutive material model describes the expected  
373 scenario for the energy absorption, future work considers the identification of dynamic  
374 constitutive material curves for different loading patterns and different drop energy,  
375 which is also the limitation of the current study. Such future development would enable  
376 using the constitutive material model to be implemented for wider spectra of impact  
377 scenarios with energy absorption.

## 378 5. Conclusions

379 The paper contributes to the field of virtual testing by developing the material  
380 model and identifying its constitutive parameters. The target material was LLDPE, a  
381 material traditionally used for packaging goods to protect them during transportation.  
382 The paper proves a high energy absorption of the material suitable for impact protection,  
383 also due by its low weight. Both quasi-static and dynamic responses of the material were  
384 considered in the constitutive material model.

385 Besides the constitutive material parameter identification for both quasi-static and  
386 dynamic responses, the paper provides a complex description of the experimental  
387 measurements. While the quasi-static response is measured using a unilateral stretch  
388 measurement in MD and TD, the dynamic tests employ a sphere impact using the drop  
389 tower.

390 The quasi-static response is analyzed and evaluated based on the measurement  
391 of several samples providing the final curves describing the stress resultant dependent  
392 on the strain in MD and TD. Those quasi-static curves serve initial values for the dy-  
393 namic response, which is optimized using aligning the experimental and calculated  
394 accelerations of the impactor.

395 A good agreement of experimental and model results was achieved and reported,  
396 providing the linear low-density polyethylene material model for virtual testing.

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398 and R.C.; funding acquisition, L.H.; investigation, P.K., T.B., J.Š., S.K., R.K. and L.H.; methodology,  
399 L.H. and J.Š.; project administration, L.H.; resources, P.K.; software, J.Š., L.H., R.C. and M.P.;  
400 supervision, L.H.; validation, L.H. and J.Š.; visualization, L.H.; writing–original draft preparation,  
401 L.H.; writing–review and editing, L.H., P.K., J.Š., T.B., R.K., S.K., R.C. and M.P. All authors have  
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410 manuscript, or in the decision to publish the results’.

## 411 Abbreviations

412 The following abbreviations are used in this manuscript:

413

	LLDPE	Linear low-density polyethylene
	MD	Machine direction
414	TD	Transverse direction
	D3	Skewed direction by 45°
	D4	Skewed direction by -45°

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