

1 Article

2 Facile Fabrication of Multifunctional ZnO Urchins on 3 Surfaces

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17

18 **Abstract:** Functional ZnO nanostructured surfaces are important in a wide range of applications.
19 Here we report facile fabrication of ZnO surface structures at near room temperature with
20 morphology resembling that of sea urchins, with densely packed, μm -long, tapered nanoneedles
21 radiating from the urchin centre. The ZnO urchin structures were successfully formed on several
22 different substrates with high surface density and coverage, including silicon (Si), glass,
23 polydimethylsiloxane (PDMS), and copper (Cu) sheets, as well as Si seeded with ZnO nanocrystals.
24 Time-resolved SEM revealed growth kinetics of the ZnO nanostructures on Si, capturing the
25 emergence of “infant” urchins at the early growth stage and subsequent progressive increase in the
26 urchin nanoneedle length and density, whilst the spiky nanoneedle morphology was retained
27 throughout the growth. $\epsilon\text{-Zn(OH)}_2$ orthorhombic crystals were also observed alongside the urchins.
28 The crystal structures of the nanostructures at different growth time were confirmed by synchrotron
29 X-ray diffraction measurements. On seeded Si substrates, a two-stage growth mechanism was
30 identified, with a primary growth step of vertically aligned ZnO nanoneedle arrays preceding the
31 secondary growth of the urchins atop the nanoneedle array. The antibacterial, anti-reflective, and
32 wetting functionality of the ZnO urchins – with spiky nanoneedles and at high surface density – on
33 Si substrates was demonstrated. First, bacteria colonisation was found to be suppressed on the
34 surface after 24 h incubation in Gram-negative *E. coli* culture, in contrast to control substrates (bare
35 Si and Si sputtered with 20 nm ZnO thin film). Secondly, the ZnO urchin surface, exhibiting
36 superhydrophilic property with a water contact angle $\sim 0^\circ$, could be rendered superhydrophobic
37 with a simple silanization step, characterised by a water static contact angle θ of $159^\circ \pm 1.4^\circ$ and
38 contact angle hysteresis $\Delta\theta < 7^\circ$. The dynamic superhydrophobicity of the surface was demonstrated
39 by bouncing-off of a falling 10 μL water droplet, with a contact time of 15.3 milliseconds (ms),
40 captured using a high-speed camera. Thirdly, it was shown that the presence of dense spiky ZnO
41 nanoneedles and urchins on the seeded Si substrate exhibited a reflectance $R < 1\%$ over the

42 wavelength range $\lambda = 200\text{-}800$ nm. The ZnO urchins with unique morphology *via* a facile fabrication
43 route at room temperature, readily implementable on different substrates, may be further exploited
44 for multifunctional surfaces and product formulations.

45 **Keywords:** ZnO urchins, nanostructured surfaces, *E. coli*, superhydrophilic, superhydrophobic, anti-
46 reflective surfaces
47

48 1. Introduction

49 Nanostructured surfaces are widespread in nature and many are ingeniously multifunctional. This
50 has inspired considerable effort in fabrication of hierarchical surface nanotextures. For instance, Lotus
51 leaf inspired nanostructures on silicon, TiO₂, polymer surfaces exhibit superhydrophobic, self-
52 cleaning and antibacterial characteristics¹⁻⁵. Nanostructured surfaces on silicon, glass substrates
53 mimicking cicada/dragonfly wings or moth eye possess superhydrophobic and antimicrobial
54 characteristics, as well as gradient refractive indices for anti-reflective applications⁶⁻¹¹.

55
56 Surfaces bearing ZnO nanostructures are among most widely studied, with a wide range of
57 applications in gas and bio-sensing, field emission devices, ultra violet detectors, and photovoltaics¹²,
58 due to their thermostability¹³, low-cost production¹⁴, antimicrobial properties¹⁵⁻¹⁹, and
59 biocompatibility²⁰. Several fabrication methods of ZnO nanostructured surfaces have been
60 reported²¹⁻²⁸, such as hydrothermal synthesis, laser ablation, sputtering, thermal decomposition,
61 evaporation induced self-assembly^{29,30}, and the sol-gel technique. Various ZnO nanostructures that
62 have been reported include one-dimensional (1-D) morphologies such as nanowires³¹⁻³³, nanorods<sup>34-
63 37</sup>, and nanotubular structures³⁸⁻⁴², as well as 3-D architectures such as flowers/urchins⁴³⁻⁴⁸,
64 tetrapods/jack-like^{49,50}, and hedgehogs⁵¹. The 3-D nanostructures with enhanced surface area may be
65 used as substitutes for 1-D nanostructure arrays with enhanced functionalities, but their fabrication
66 requires either sophisticated instrumentation or elevated temperatures (thus high energy input).

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68 Here, we report facile fabrication of multifunctional 3-D ZnO urchins on silicon (Si) substrates in a
69 solution-based synthesis at near room temperature (RT) using a one-step procedure, which could
70 also be applied on Si substrates seeded with ZnO crystals and various other substrates. The
71 morphology of these ZnO surface structures resembles that of sea urchins, with tapered nanoneedles
72 of μm in length radiating from a central core. The growth kinetics of the ZnO urchins was studied by
73 examining their intermediate morphologies at different growth time intervals (0.5 - 12 h). The ZnO
74 urchin-coated surface exhibited high anti-reflectance, and superhydrophobicity after silanization as
75 investigated by contact angle, contact angle hysteresis and drop impact analyses. They were also
76 bacteriophobic against *E. coli*. The simple fabrication method for ZnO urchins could also be adapted
77 to a variety of materials such as polydimethylsiloxane (PDMS), copper sheets and glass substrates,
78 demonstrating its versatility.

79 2. Materials and Methods

80 2.1 Substrates preparation and cleaning

81 Silicon substrates were prepared by cutting a silicon wafer (100 mm diameter, P type, B dopant,
82 $\langle 100 \rangle$, 1-100 $\Omega\cdot\text{cm}$, 500 μm thick, single-sided polish, test grade, University Wafer, Inc.) into 1 cm x 1

83 cm squares. All substrates, including copper sheet, glass slide (Polar Industrial Corporation) and
84 polydimethylsiloxane (Dow Corning, Sylgard 184) were cleaned by a subsequent sonication in
85 acetone, ethanol, and deionized water (10 minutes in each) followed by drying with N₂.

86 2.2 Seeding

87 Seeding procedure was used to produce nucleation sites for the growth of ZnO nanostructures³¹.
88 Cleaned silicon substrates were dipped in a solution of zinc acetate dihydrate (Zn(CH₃COO)₂·2H₂O,
89 99%, Sigma Aldrich) in ethanol (CH₃CH₂OH, Absolute, Sigma Aldrich) for few seconds, rinsed with
90 clean ethanol and then dried with N₂. This coating step was repeated 5 times for each seeded silicon
91 substrate (see Figure S1). Subsequently, the substrates were heated to 300° C on a hot plate, and
92 annealed for 30 minutes in air to thermally decompose zinc acetate crystallites to ZnO islands with
93 (0001) planes parallel to the silicon substrate surface³¹.

94 2.3 ZnO Urchin growth

95 To grow the ZnO nanostructures, 50 mL aqueous solution of zinc nitrate hexahydrate
96 (Zn(NO₃)₂·6H₂O, 98%, Sigma Aldrich) solution was added dropwise to 50 mL of potassium hydroxide
97 (KOH, 85%, Fisher Scientific) aqueous solution under constant stirring for 30 minutes. Next, the alkali
98 solution of zincate ions was carefully poured to the glass Petri dish containing substrates (seeded Si,
99 non-seeded Si, Cu sheet, glass and polydimethylsiloxane, separately for each type) sealed with a glass
100 lid and a paraffin film, and kept at 20° ± 2°C for 12 hours. After the synthesis, samples were rinsed
101 with DI water and dried with N₂.

102 2.4 Bacterial growth conditions and sample preparation

103 Isolated single colonies of *E. coli* K-12 (MG 1655) were used to prepare the pre-inoculum⁵². The culture
104 was grown in Luria Bertani (LB) medium for 8 hrs at 37°C with constant shaking at 160 rpm. A 0.2 %
105 pre-inoculum was added into 10 mL of LB medium and it was allowed to grow until 0.3 optical
106 density (O. D.) at 600 nm⁵³. The cells were subsequently pelleted and washed with phosphate-
107 buffered saline (PBS). A cell suspension of 0.01 O. D. at 600 nm was used for studying the bacterial
108 interaction with the bare, unmodified silicon, silicon with 20 nm thin film of ZnO and ZnO nano-
109 urchin surfaces.

110 2.5 Scanning electron microscopy (SEM)

111 ZnO nanostructured surfaces were imaged using field emission scanning electron microscopy (JSM-
112 IT300 SEM (JEOL)). Dimension of the ZnO nano urchin-like structures were obtained from the SEM
113 images using ImageJ software. For bacterial sample imaging, substrates were first washed with PBS
114 to remove the loosely adhered cells. Substrates were then dipped in 2.5% of Glutaraldehyde to fix the
115 cells. After that all the substrates were allowed to dry in vacuum. Thin layer of gold (15 nm) was
116 sputtered on the samples using a Quorum sputter coater (Q150T) to avoid the charging effect while
117 doing the SEM. Samples were scanned thoroughly using SEM (sample size 1.3 cm × 1.3 cm) and the
118 imaging was done in triplicates for all the samples. Selecting three specimens from each type of a
119 sample (two controls and one ZnO Urchin surface) ensured the repeatability.

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122 2.6 Urchin dimension analysis

123 ImageJ⁵⁴ was used to obtain the morphological information of the nanoneedles comprising the ZnO
124 urchin and the vertical nanoneedles using the length and angle measurement tools.

125 2.7 X-ray Diffraction (XRD)

126 The presence of crystalline ZnO after the growth procedure was confirmed using D8 Advance
127 Bruker-AXS diffractometer (Cu K α radiation ($\lambda = 0.154178$ nm), $\theta - 2\theta$ scan, $2\theta = 10^\circ - 90^\circ$, step size
128 - 0.02 (2θ), step count - 1.5 s).

129 2.8 Grazing incidence X-ray diffraction (GIXRD)

130 GIXRD analysis of the seeded and non-seeded silicon substrates used to study the time dependence
131 growth of ZnO nanostructures was performed at Beamline BM28 at the European Synchrotron
132 Radiation Facility (France). Experimental parameters: radiation wavelength: 0.8856 Å, sample to
133 detector distance: 0.24 m, detector: MAR165, calibrant: Silver behenate and ZnO powder. The
134 diffraction patterns were reduced to one-dimensional line profiles using *pyFAI*, a pythonic library for
135 1D azimuthal / 2D radial integrations of diffraction images⁵⁵.

136 2.9 Contact angle measurement and high-speed imaging

137 Static contact angle measurements were carried out using a Krüss Drop Shape Analyser (DSA 100).
138 10 μ L water droplet was placed gently on all the substrates to measure the contact angles. A droplet
139 impact dynamics of water droplet on the superhydrophobic surface (salinized seeded silicon
140 substrate with ZnO nanostructures) was captured using a high-speed camera (Photron FastCam SA4)
141 at 10,000 fps (time resolution of 0.1 ms). The droplets were created by a micro-pipette and released

142 from a height of 7 cm (Weber number (We) = $\frac{\text{inertial force}}{\text{surface tension force}} = \frac{\rho V^2 D_w}{\gamma} = \frac{1000 \times 1.17^2 \times 0.0026}{0.072} = 49$,

143 Density of water (ρ) = $1000 \frac{\text{kg}}{\text{m}^3}$, Velocity of falling droplet (V) = $\sqrt{2gh} =$

144 $1.17 \frac{\text{m}}{\text{s}}$, Surface tension of water (γ) = $0.072 \frac{\text{N}}{\text{m}}$, and diameter of droplet (D_w) = 0.0026 m).

145 2.10 Reflectance measurements

146 To calculate the reflectance of all the surfaces, Shimadzu MPC3600 UV-VIS-NIR Spectrometer with
147 an absolute specular reflectance mode was used. Wavelengths ranging from 200 nm to 800 nm were
148 used for the reflectance measurement. D₂ light source was used for the range 200 nm – 310 nm and
149 Tungsten source was used for the range 310 nm – 800 nm. The angle of incidence and angle of
150 reflection were set to 5° throughout the experiment. The equipment used a photomultiplier tube
151 (PMT) detector. The reflectance from a surface is evaluated by its refractive index profile⁶. In general,
152 a flat surface has a high reflectance due to the discontinuous refractive index profile, whereas
153 structured surface suppresses reflection with its graded refractive index profile⁶.

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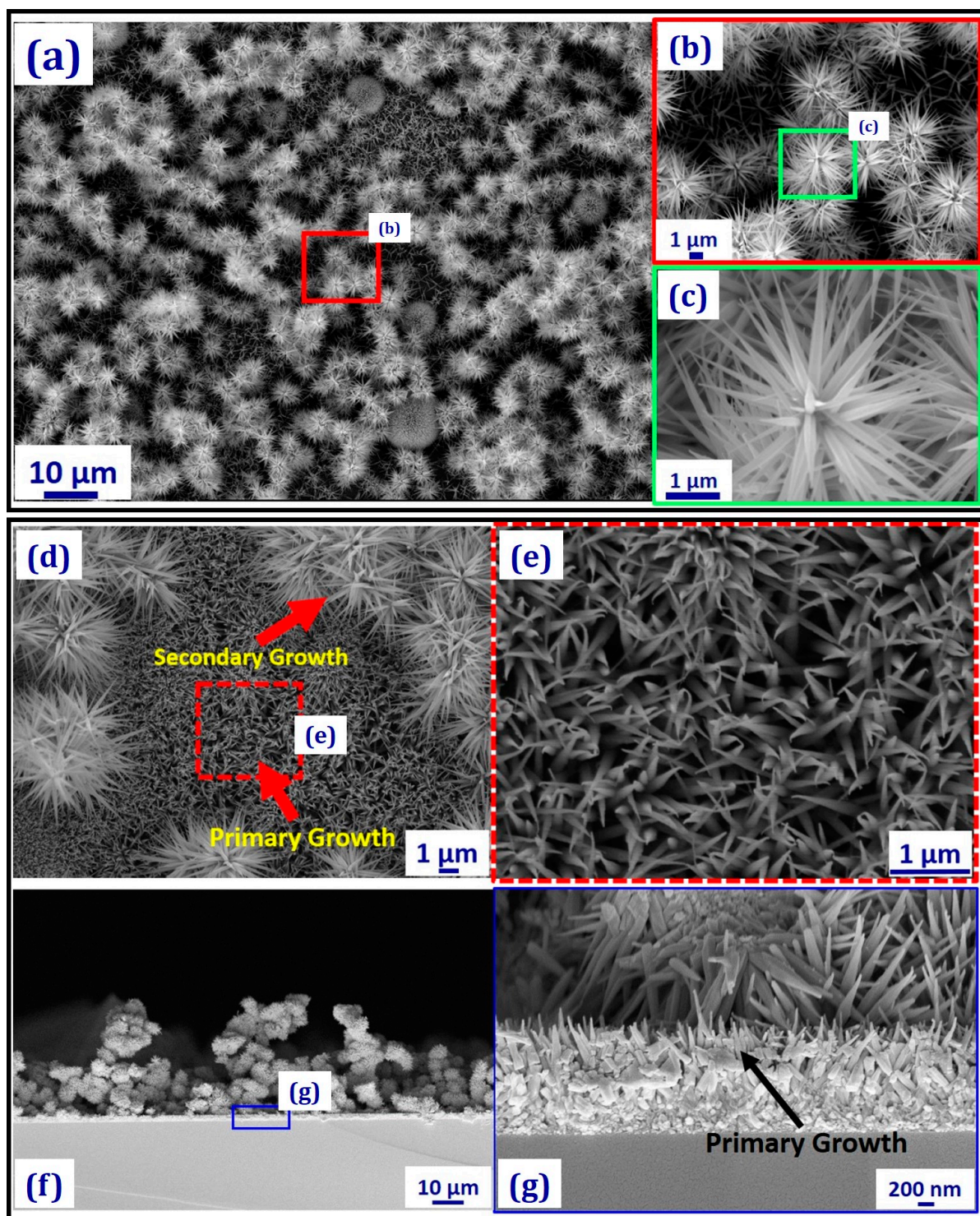
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156 3. Results and Discussion

157 **Figure 1** shows the SEM images of densely packed urchins after growth in 5 mM zincate solution on
158 a seeded silicon substrate (**Figure S2**) at $\sim 20^\circ$ C for 12 h, and the simple fabrication method is
159 described in **Figure S1**. XRD (**Figure S11**) confirmed that these urchins were ZnO. The morphology
160 of an individual urchin (**Figure 1c**) reveals sharply tapered needles radiating from the centre. The
161 length of the needles was $L \sim 1.65 \pm 0.11 \mu\text{m}$, its width at the central base $D_c \sim 156 \pm 24$ nm and at the
162 tip $D_t \sim 13 \pm 7$ nm, with a tapering angle $\theta \sim 10^\circ$ and an average tip-to-tip spacing $s \sim 504 \pm 119$ nm
163 (see **Figure S3- S5** for the size distribution analysis).

164 Previously, ZnO nano- and micro-structures bearing resemblance of the urchin morphology have
165 been reported. For instance, Elias et al.⁴⁸ prepared μm -sized hollow spheres with a ZnO nano-
166 columns coating using atom layer deposition and electrodeposition, whilst the method by Shen et
167 al.⁴⁷ used thermos-evaporation of metallic Zn powder at high temperature ($\sim 750^\circ$ C). Wahab et al.⁴³
168 fabricated ZnO nano-flowers blunt tapering via pH-controlled reactions in solution of zinc acetate
169 dihydrate and sodium hydroxide at 90° C. Using a similar approach, Gokarna et al.⁴⁴ synthesised ZnO
170 urchin-like structures with columnar nanoneedles. Hieu et al.⁴⁵ sputtered zinc onto a polystyrene-
171 sphere array and subsequent oxidation at 500° C led to columnar ZnO urchins ; and the method by
172 Taheri et al.⁴⁶ also involved depositing zinc acetate dihydrate precursor followed by calcination at
173 500° C. The solution synthesis method we report here is relatively much simpler (at RT and
174 applicable to different surfaces as we show below) compared to these previous studies, producing
175 spiky tapered morphology of the urchin needles with a very high density previously unreported. The
176 presence of high density spiky ZnO urchins endows the surface with multi-functionalities as
177 discussed below.

178 SEM images of the region not covered by the urchins reveal vertical ZnO nanoneedles (from the
179 primary growth step as referred to in **Figure 1d, g**) with an average base diameter $\sim 90 \pm 20$ nm,
180 smaller than that of the urchin needles (**Figure 1d-g**), and the cross-section SEM view confirms that
181 the ZnO urchins were formed atop a layer of nanoneedle arrays (**Figure 1a, b**). Orthorhombic ϵ -
182 $\text{Zn}(\text{OH})_2$ crystals with facet edges ~ 10 - $30 \mu\text{m}$ were also observed (**Figure S7**), with ZnO urchins (the
183 secondary growth in **Figure 1d**) decorating the facets. The tapered or spiky nanoneedle geometry
184 might be attributed to the concentration gradient of the zinc ions in the vicinity of the substrate where
185 the ZnO nanocrystal seeds provided the nucleation sites for ZnO nanoneedles growth (**Figure 1f, g**)⁵⁶.

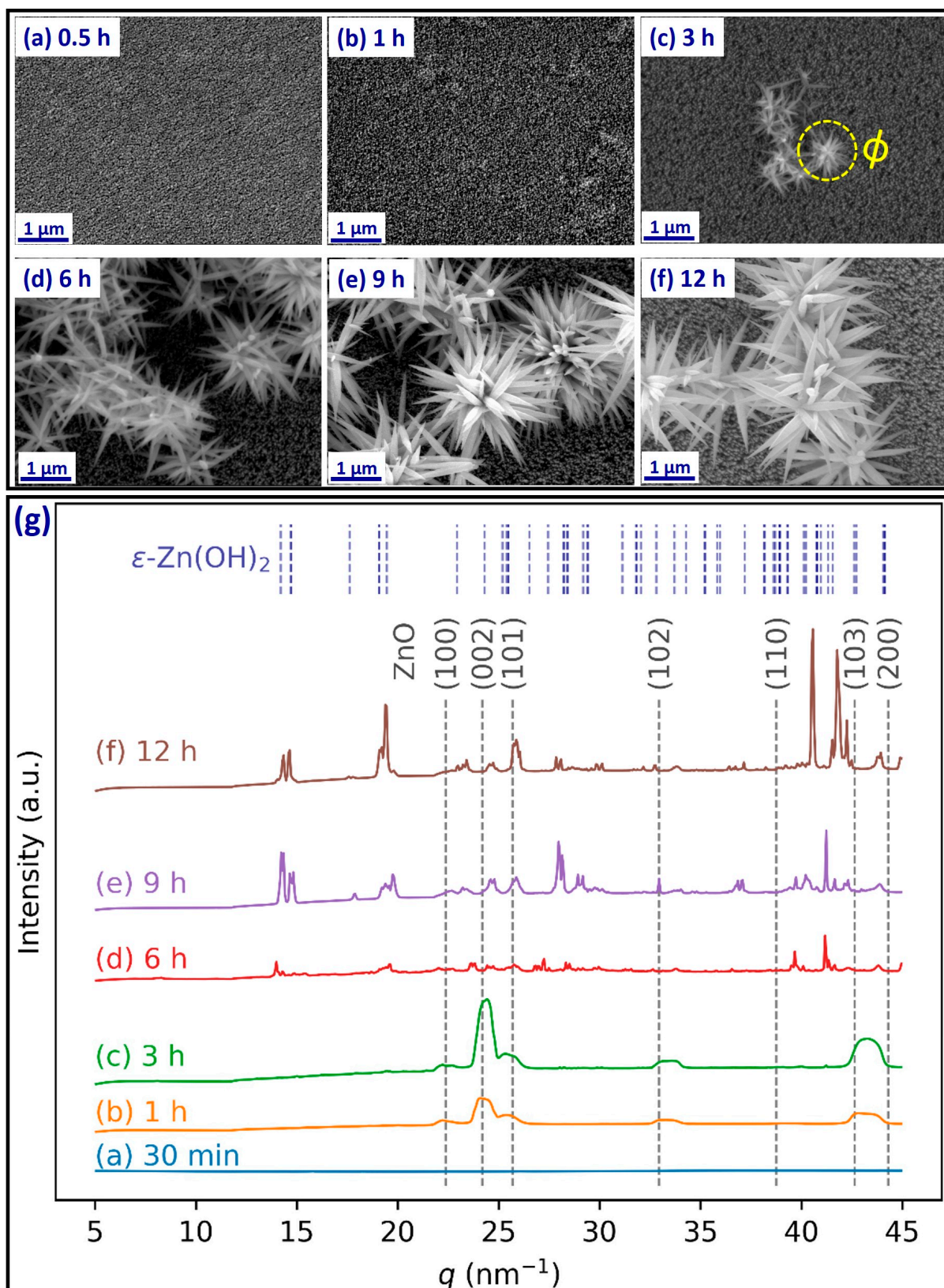


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Figure 1: (a) Scanning electron microscopy (SEM) images of ZnO urchins on a seeded silicon substrate; (b) SEM images depicting the ZnO urchin and primary vertical nanoneedle growth on the silicon substrate. SEM images show random multidirectional growth of the nanoneedles comprising the urchin (seeding concentration of zinc acetate dihydrate - 5 mM, synthesis time for ZnO urchin growth - 12 h, and temperature - 20°C).

To understand the growth kinetics of the ZnO nanostructures, SEM images (Figure 2, Figure S8 and S9) of the seeded silicon substrates were taken at different growth time intervals t . At $t = 0.5-1$ h (Figure 2a, b), only the primary growth of vertical nanoneedle structures was observed. At $t = 3$ h, the

196 secondary growth of urchin-like structures atop the nanoneedle array with a diameter $\phi \sim 1\mu\text{m}$ was
197 observed. The density and size of the urchins and length of the urchin needle L then increased with
198 time (see [Figure S5](#) for urchin needle length L vs. growth time t), whilst the tapering angle remained
199 largely constant at $\theta \sim 10^\circ$. Figure 2g presents the GIXRD data (see section II.4. in supplementary
200 information) on the seeded silicon substrates at different growth time intervals. The intense (002)
201 ZnO peak at $t < 3\text{h}$ is consistent with the primary growth of vertical ZnO nanoneedles. The $\epsilon\text{-Zn(OH)}_2$
202 phase (orthorhombic crystals) is visible for the samples with reaction time $t > 3\text{ h}$ (**Error! Reference**
203 **source not found.**). This shows that the ZnO nanoneedles started forming after 30 min and Zn(OH)₂
204 crystals after 3 h, likely together with the secondary growth of ZnO urchins ([Figure 2c-f](#)).



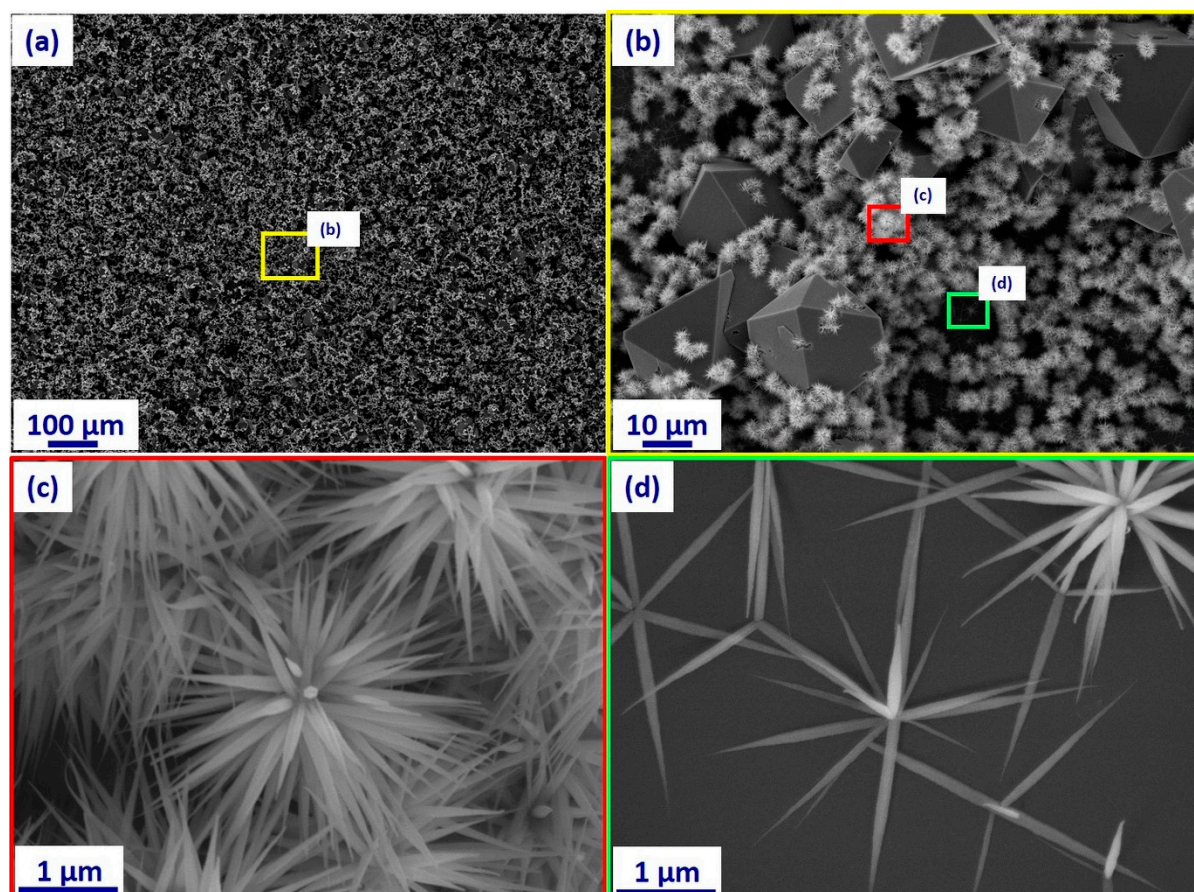
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Figure 2: (a-f) SEM images of seeded silicon substrates in 5 mM zincate growth solution at different growth time intervals $t = 0.5 - 12$ h; and (g) corresponding GIXRD data for the substrates.

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To evaluate the effect of substrate seeding on the ZnO urchin morphology, the one-step fabrication procedure was performed using unseeded silicon substrates (i.e. in 5 mM zincate solution at 20° C, and 12 h growth time). SEM image in Figure 3a shows that urchin like ZnO nanostructures were also

211 obtained with a high surface coverage (Figure 3a, c), with urchins found on orthorhombic ϵ -Zn(OH)₂
 212 crystal facets (Figure 3b). There was no significant difference in the urchin morphology (urchin size
 213 ϕ , needle length L , diameter D , and tip angle θ) compared to the seeded substrate. However, some
 214 urchins in the bare silicon area had a smaller number of nanoneedles (Figure 3d). Comparison of the
 215 XRD patterns in Figure S11 for the 12 h growth shows that, on the unseeded Si sample, the most
 216 intense peaks correspond to the ϵ -Zn(OH)₂ crystal structure, consistent with the SEM images shown
 217 in Figure 3.

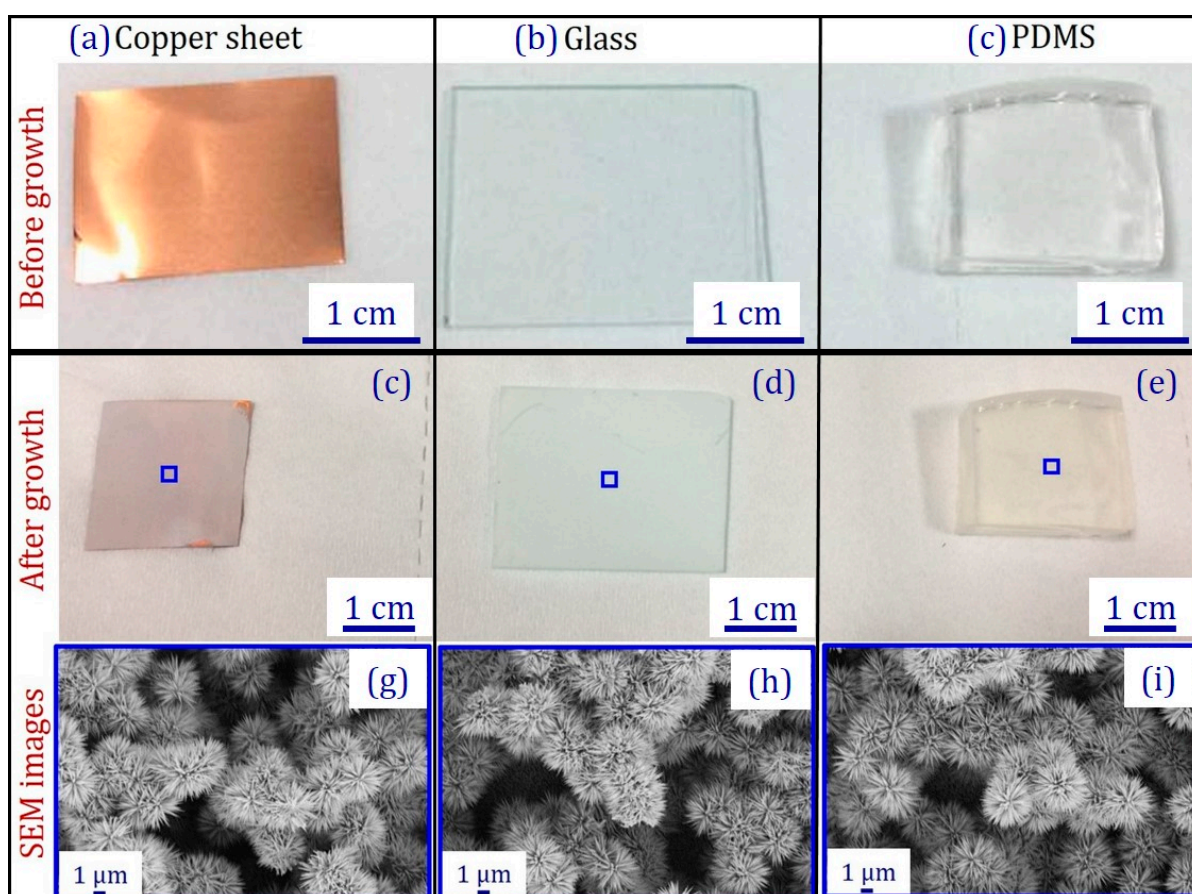


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 219 Figure 3: SEM images of ZnO nanostructures grown on the unseeded silicon substrate in the one-step
 220 synthesis (12 h, 20° C), with a high coverage of ZnO urchins (a) with morphology similar to that on
 221 seeded Si substrates (b, c). Orthorhombic ϵ -Zn(OH)₂ crystals were also observed (b) decorated with
 222 urchins. Urchins with a smaller number of needles were also observed in the bare silicon area (d).

223
 224 Time dependant growth of ZnO urchins on non-seeded silicon substrate was also studied, and SEM
 225 images at different growth time intervals are shown in Figure S10. Similar to the seeded Si substrates,
 226 ZnO urchins emerged between 1-3 h of the growth time (Figure a, e), and their dimensions and
 227 surface coverage increased with time (Figure S10), with orthorhombic ϵ -Zn(OH)₂ crystals starting to
 228 appear on the surface between 3-6 h of the synthesis (Figure S10b).

229 Figure further compares the XRD on both seeded and non-seeded silicon substrates after the
 230 growth in the zincate solution for 12 h. Peaks corresponding to both ZnO (PDF 36-1451) and ϵ -
 231 Zn(OH)₂ (PDF 38-385) were identified, confirming their formation on the surfaces. The most intense
 232 peak in the seeded silicon sample corresponds to the ZnO (002) plane, which is the preferential
 233 growth direction of ZnO nanoneedles⁵⁷. Along with the SEM images, this confirms that on seeded Si

234 the primary growth led to ZnO nanoneedles, with the ZnO urchins grown atop in the secondary
 235 growth stage. In the case of the unseeded Si sample, the most intense peaks correspond to the ϵ -
 236 $\text{Zn}(\text{OH})_2$ crystal structure. This is consistent with the SEM image (Figure 3b) which shows that the
 237 substrate was covered by ZnO urchins grown on orthorhombic $\text{Zn}(\text{OH})_2$ crystals and bare Si surface,
 238 with a significantly smaller amount of arrayed ZnO nanoneedles (Figure d).
 239 Growth of the ZnO nanoneedles and urchin structures were also trialled on copper sheets, soft PDMS
 240 and transparent glass substrates, all seeded by dipping the substrates in a solution of zinc acetate
 241 dihydrate in ethanol (Figure S1). The ZnO nanostructures grown on these surfaces (Figure g-i)
 242 exhibited similar urchin morphologies to the those on silicon substrates (Figure 1). The growth on
 243 glass was also evident, as it became translucent after the growth (Figure d and Figure S12). This
 244 demonstrates that the facile, room temperature synthesis method for ZnO nanoneedles and urchins
 245 is adaptable to various substrates.

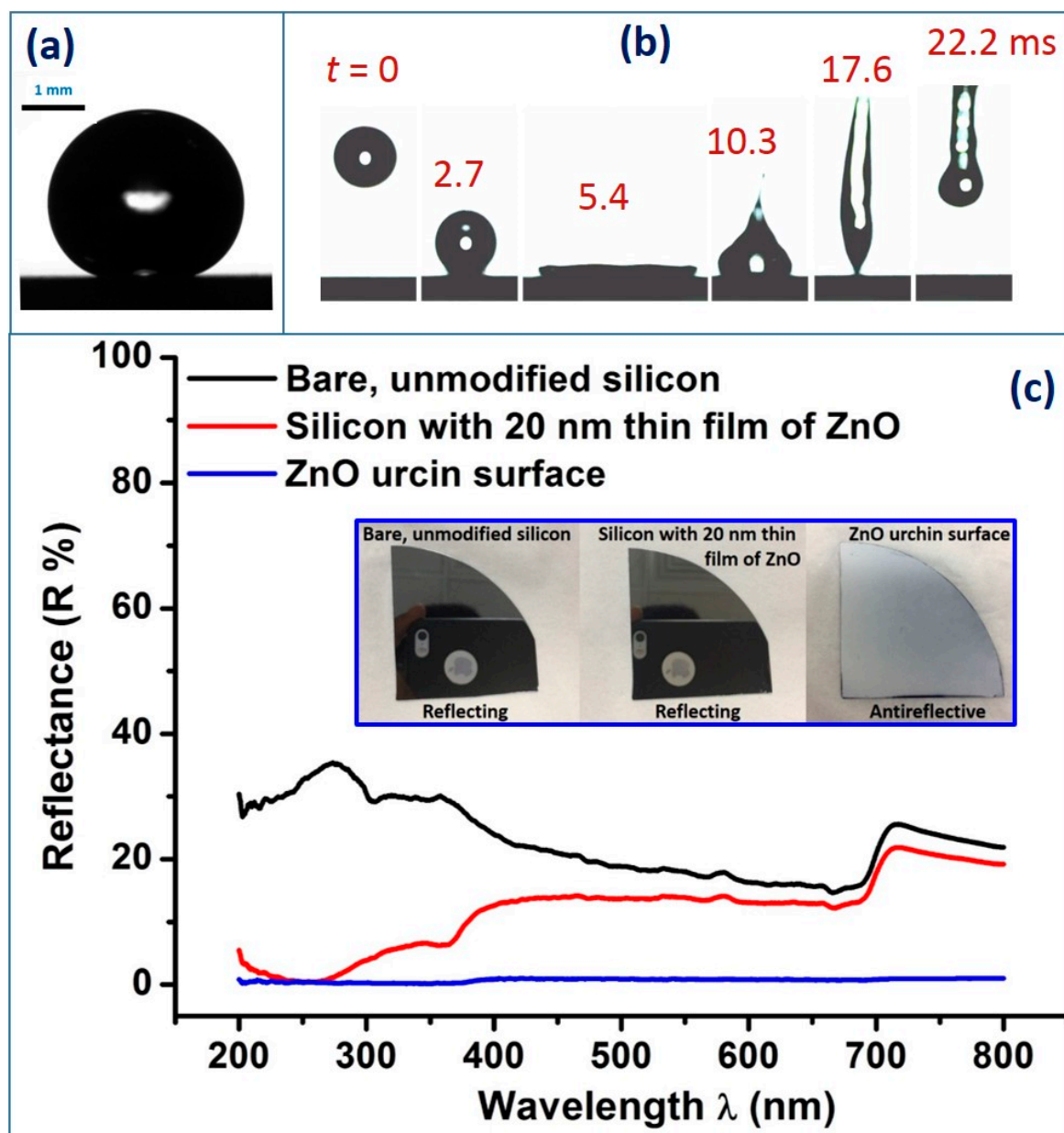


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 247 Figure 4: (a) Growth of ZnO urchin on seeded copper, glass slide and PDMS surfaces and the
 248 corresponding SEM images of the ZnO urchin formed on the substrates.

249
 250 Si surfaces with ZnO urchins (both seeded and unseeded, 12 h growth time) were superhydrophilic,
 251 with a water droplet placed on these surfaces spreading completely and showing a water contact
 252 angle $\theta \sim 0^\circ$ (Table S1). They became superhydrophobic after overnight silanization (Table S1)
 253 with a static water contact angle $\theta \sim 159^\circ \pm 1.4^\circ$ (Figure 5a) and a contact angle hysteresis $\Delta\theta < 7^\circ$
 254 was observed. For comparison, Table S1 shows that bare Si and Si coated with 20 nm ZnO thin film
 255 exhibited hydrophilic behaviour with a water static contact angle $\theta \sim 36^\circ$ and $\sim 61^\circ$, respectively.

256 To further demonstrate the superhydrophobicity in dynamic conditions, a 10 μL water droplet was
257 allowed to fall on the silanized ZnO urchin surface from a height of 7 cm with a Weber number $We =$
258 $\frac{\rho V^2 D_w}{\gamma} = 49$, where $\rho = 1 \text{ g cm}^{-3}$ is the density of water, $V = \sqrt{2gh} = 1.17 \text{ m s}^{-1}$ the impact velocity of the
259 falling droplet, $\gamma = 73.8 \text{ mN m}^{-1}$ the surface tension of water, and $D_w = 2.67 \text{ mm}$ the diameter of
260 droplet (Table S1). The dynamic process was captured using a high-speed camera with a 10,000-fps
261 capture rate. The water droplet bounced off the superhydrophobic surface completely without leaving
262 any visible residues, with a droplet-substrate contact time of 15.3 ms (Figure 5b and Video S1) which
263 is close to the theoretical contact time $(\frac{\rho D_w^3}{\gamma})^{1/2} = 16.25 \text{ ms}$ for water droplet on superhydrophobic
264 surface⁵⁸.

265 In addition, the ZnO urchin coated surface also demonstrated anti-reflective behaviour. **Error!**
266 **Reference source not found.** c shows <1% reflectance (R) over the wavelength range $\lambda = 200\text{-}800 \text{ nm}$
267 on the urchin surface, compared to two control samples (bare Si and Si with 20 nm ZnO thin film).
268 There have been previous reports on fabrication of nanostructured anti-reflective surfaces^{6,11,59-65}.
269 Different techniques such as dry etching, wet etching, metal assisted etching, nanoimprint
270 lithography, etching using O_2 plasma and inductively coupled plasma, optical lithography followed
271 by etching etc.⁶⁶ have been used to fabricate such surfaces on silicon, polymers, and glass to achieve
272 low reflectance. In comparison, the fabrication method we used to achieve the low reflectance ($R <$
273 1%) was simple with the synthesis undertaken at room temperature without the need for
274 sophisticated instrumentation.



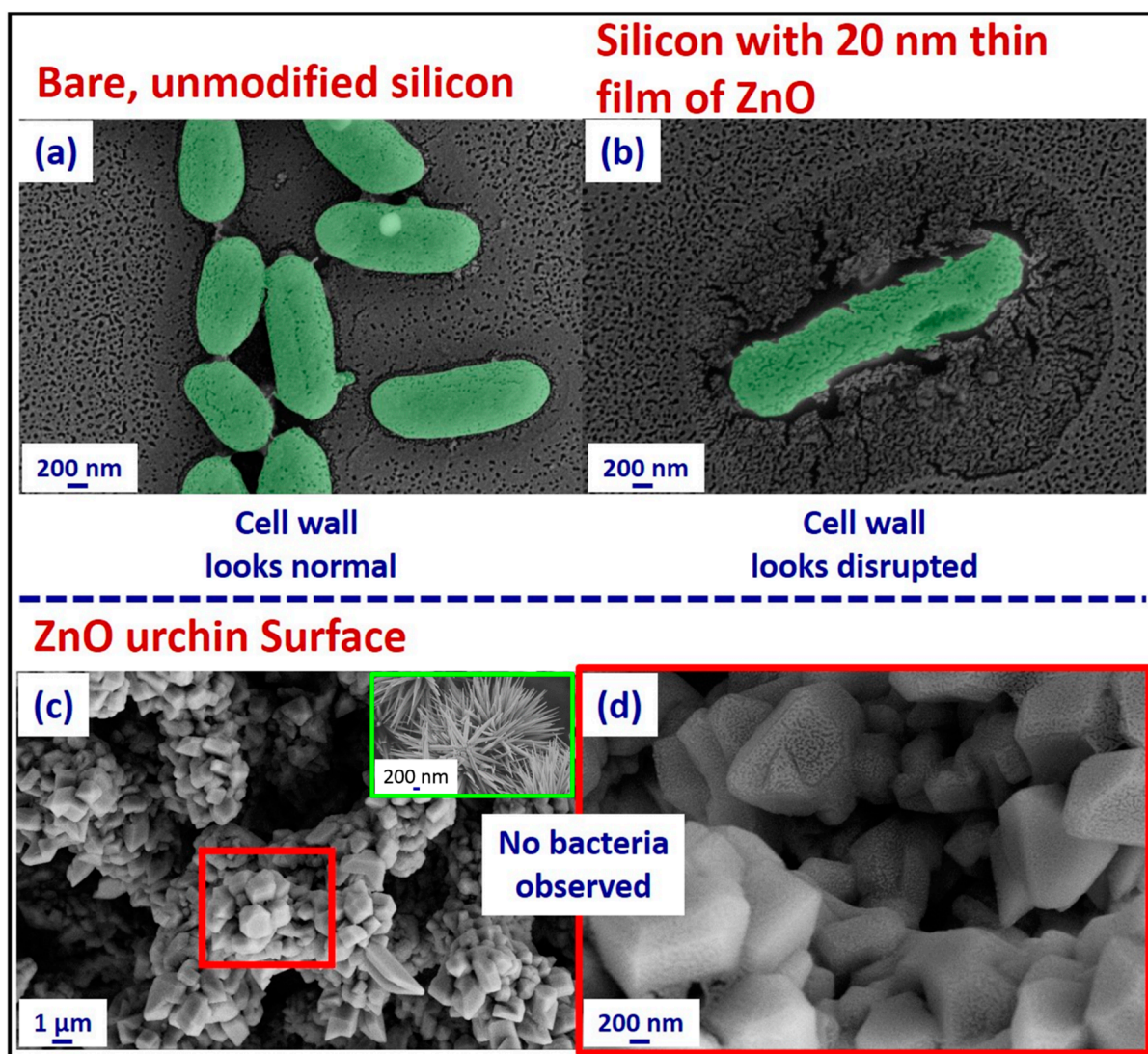
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276 Figure 5: (a) Photograph of a 10 μL water droplet on ZnO urchin-coated surface after silanization,
 277 with a contact angle $\theta \sim 159^\circ \pm 1.4^\circ$. Prior to silanization, the surface was highly hydrophilic,
 278 displaying complete wetting by water with $\theta \sim 0^\circ$ (Table S1). (b) High-speed camera images of a
 279 10 μL droplet bouncing off the superhydrophobic ZnO urchin surface, with a contact time of 15.3 ms.
 280 (c) Reflectance from the ZnO urchin surface compared to the two control surfaces in the wavelength
 281 range $\lambda = 200 - 800$ nm. ZnO urchin surface was found to be the most anti-reflective with reflectance
 282 $R < 1\%$ over the whole wavelength range. The inset shows the highly reflective bare Si and Si with 20
 283 nm ZnO film (with the reflection of the iPhone used to take the photo clearly visible), in contrast to
 284 the anti-reflective characteristic of the ZnO urchin surface.

285

286 To evaluate the bacterial interaction with the fabricated ZnO urchins on the seeded Si surface, it was
 287 submerged in 2 mL *E. coli* culture in phosphate-buffered saline (PBS) for 24 h and then imaged with
 288 SEM. This was compared with two control samples: unmodified bare Si and Si with a 20 nm sputtered
 289 ZnO film (Figure S13). SEM images in Error! Reference source not found.a (see also Table S2) show

290 bacteria growth on the control surfaces; in contrast, no bacterium was observed on the ZnO
291 nanostructured surface (Figure 6c, d). In addition, the SEM analysis revealed a change in the
292 morphology of the ZnO urchins (inset in **Error! Reference source not found.c**): after 24 h immersion
293 in the *E. coli* culture, the spiky urchin structure was transformed into the hexagonal pyramid structure
294 (see Figure S14a). This can be attributed to the formation of sodium zinc phosphate hydrate (NaZn-
295 $\text{PO}_4 \cdot \text{H}_2\text{O}$)⁶⁷ due to the reaction between ZnO urchins and the PBS (cf. XRD data in Figure S14b). ZnO
296 is known to exhibit antimicrobial efficacy¹⁵⁻¹⁹, killing bacteria and prohibiting bacterial growth on its
297 surface *via* release of Zn^{2+} ions which solicits generation of reactive oxygen species (ROS). A number
298 of naturally occurring surfaces bearing spiky nanotextures (e.g. cicada wing, dragonfly wing, and
299 gecko skin) have been reported to exhibit bactericidal efficacy, attributed to puncturing or stretching
300 of the bacterial membrane, although the detailed mechanisms remain to be fully understood^{3,11,68-71}.
301 Hence, we suggest that the ZnO urchin surfaces could prevent bacterial growth by combining
302 synergistically the inherent chemical activity of ZnO and the spiky morphology of the urchins that
303 inhibits the bacteria to colonise on the surface of the ZnO urchins^{18,72-74}. The sharp topography of the
304 nanoneedles are particularly effective in disrupting the bacterial cell wall by imparting the localised
305 stress on the bacterial membrane⁷⁵. There are previous reports on reduced bacterial adhesion on
306 superhydrophobic surfaces⁷⁶⁻⁷⁹ where the antiwetting property of the surface plays a crucial role in
307 the interaction of the bacteria with the surface; whereas our fabricated ZnO urchin surfaces exhibited
308 bacteriophobic behaviour in a hydrophilic state (with a water contact angle $\theta \sim 0^\circ$).



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310 Figure 6: Representative SEM images of *E. coli* (false coloured Green) on different surfaces after 24 h
 311 of bacterial culture. (a) bare, unmodified silicon, (b) silicon with 20 nm thin film of ZnO and (c) & (d)
 312 ZnO urchin surface. Inset in Figure 6(c) shows the morphology of the ZnO urchin before bacterial
 313 culture on the urchin surface.

314 5. Conclusions

315 In summary, ZnO urchin structures were fabricated via a simple method involving submergence of
 316 a substrate (Si, PDMS, glass, or Cu) in an alkaline aqueous zincate ion ($\text{Zn}(\text{OH})_4^{2-}$) solution at room
 317 temperature ($\sim 20^\circ \text{C}$). On the Si substrate seeded with ZnO nanocrystals, we observed ZnO urchin
 318 structures atop vertically aligned ZnO nanoneedles. The urchins consisted of densely packed, μm -
 319 long spiky ZnO nanoneedles radiating from the urchin centre, with a tapering angle of $\sim 10^\circ$. The
 320 growth kinetics of the ZnO surface nanostructures was studied by time resolved SEM, revealing that
 321 the urchin morphology emerged at ~ 3 h reaction time with a small number of, but tapered,
 322 nanoneedles. The nanoneedle density and length then progressively increased with the reaction time.
 323 On the seeded Si substrate, a primary growth step of the vertical ZnO nanoneedles was identified
 324 preceding the secondary urchin growth. The ZnO urchin coated surface exhibited anti-reflective
 325 properties, reducing the reflectance to less than 1%. It was highly hydrophilic, with a water contact

326 angle of $\sim 0^\circ$; after silanization, it exhibited superhydrophobicity with a water contact angle of 159°
327 and hysteresis smaller than 7° . In addition, its dynamic hydrophobicity was demonstrated by
328 bouncing-off of a water drop as captured by a high-speed camera, with resident time of 15.9 ms.
329 Furthermore, the ZnO urchin surface showed bacteriophobic behaviour, as compared to the control
330 Si and ZnO-coated surfaces, with no bacterium colonization observed on the surface after 24 h
331 incubation in *E. coli*. The facile method for preparing ZnO urchins with unique morphology of
332 tapered nanoneedles, and its ready adaptability to different surfaces (including polymers), may open
333 new routes for fabrication of multifunctional ZnO nanostructured surfaces.

334

335 **Supplementary Materials:** The following are available online,

336 **Figure S1:** Process flow of the fabrication of ZnO nanostructures on seeded silicon substrates at near room
337 temperature, **Figure S2:** SEM images of a silicon substrate seeded with ZnO nanoislands in 5 mM zinc acetate
338 dihydrate solution (dipping $\times 5$ times and annealed at 300°C for 30 minutes), **Figure S3:** Nanoneedle dimension
339 measurement using ImageJ software. The values in the main text were averaged from 50 different needles, with
340 the measurements of different parameters at each single needle repeated 5 times, **Figure S4:** Size distribution
341 from ImageJ analysis of the nanoneedle tip diameter at the top D_t urchins ranging from 7-30 nm with an average
342 diameter of $D_t = 13 \pm 7$ nm, **Figure S5:** Length of the nanoneedles L in the urchin structure vs. growth time. Length
343 of the nanoneedle was found to increase with respect to synthesis time. Stars in figure shows that for time points
344 0, 0.5 and 1 h there was no formation of urchin structures, **Figure S6:** (a) the angled view after 9 h growth,
345 showing ZnO urchins on the top of highly (b) oriented ZnO nanoneedles, **Figure S7:** Representative FESEM
346 images of the seeded silicon substrate post 12 h synthesis in the zincate solution, **Figure S8:** SEM images of
347 seeded silicon substrates after growth in zincate solution at different time intervals. Density of ZnO nano urchins
348 was found to increase with the increase synthesis time, **Figure S9:** SEM images of seeded silicon substrates after
349 growth in zincate solution at different time intervals. Zn(OH)_2 crystals were observed on the substrates for
350 synthesis time $t > 3$ h, **Figure S10:** SEM images of ZnO urchins on a non-seeded silicon substrate, taken at
351 different time intervals: (a) 3 h, (b) 6 h, (c) 9 h, and (d) 12 h. (e-h) show enlarged views of the square regions as
352 labelled in (a-d), respectively, **Figure S11:** XRD of (a)&(b) seeded and (c)&(d) unseeded silicon substrates after
353 growth in the zincate solution at 20°C for 12 h, with the ZnO and $\epsilon\text{-Zn(OH)}_2$ peaks indicated by * and \blacktriangledown ,
354 respectively, and the enlarged views shown on the right-hand side, **Figure S12:** Optical image showing the loss
355 in transparency after the formation of ZnO urchin/nanoneedles on the glass substrate, **Figure S13:** 12 well culture
356 plate for pouring the bacterial suspension on different samples, **Figure S14:** (a) SEM and (b) XRD of ZnO
357 nanourchin surface after pouring the bacterial culture for 24 hours. Formation of sodium zinc phosphate hydrate
358 ($\text{NaZn-PO}_4 \cdot \text{H}_2\text{O}$) was observed on the surface due to the reaction of ZnO with PBS leading to the change in
359 morphology of the spiky ZnO nanowires, **Table S1:** Static contact angles of DI water droplet on different
360 substrates, **Table S2:** SEM of *E. coli* on different substrates, **Video S1:** Superhydrophobic ZnO urchin.

361

362 **Author Contributions:** AT and WHB conceived and designed experiments; AT and PW designed, synthesised
363 and characterised ZnO urchins; AT, SS, DN and PS performed experiments on anti-reflectivity, *E. coli*
364 colonisation, and dynamic superhydrophobicity of the urchin surfaces; PW, OB and WHB performed
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366

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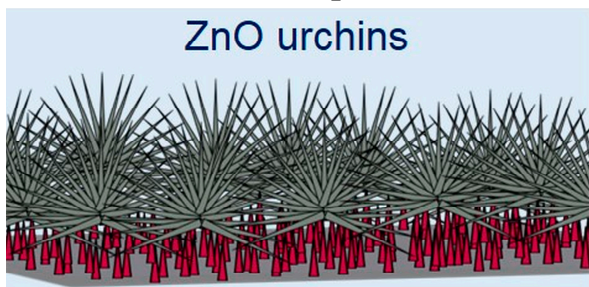
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579 Table of Contents Graphic



Dynamic superhydrophobicity



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